

# Fermi energy limits in Cu-oxides



Alternative Title:

## About the difficulties to achieve high photovoltages and conversion efficiencies with CuO and other Cu-oxides

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**LOEWE**

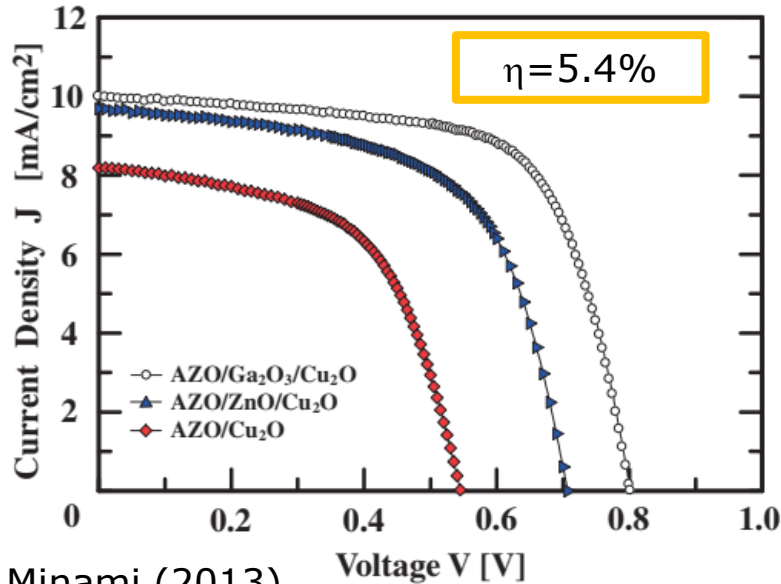
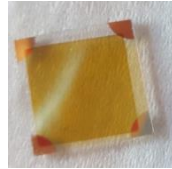
Exzellente Forschung für  
Hessens Zukunft



# Cu-oxides for solar cells

## Cu<sub>2</sub>O

direct band gap: 2.3 eV

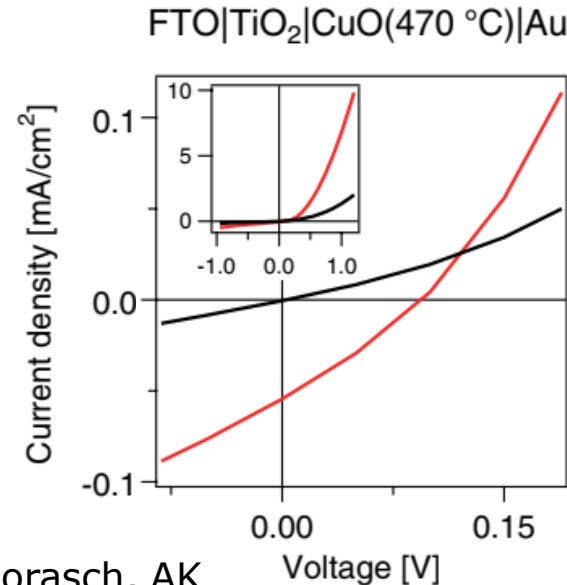
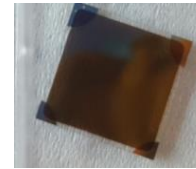


Minami (2013)

Photovoltage lower than for good solar cells systems

## CuO

direct gap: 1.5-1.7 eV  
indirect gap: 1.2 eV

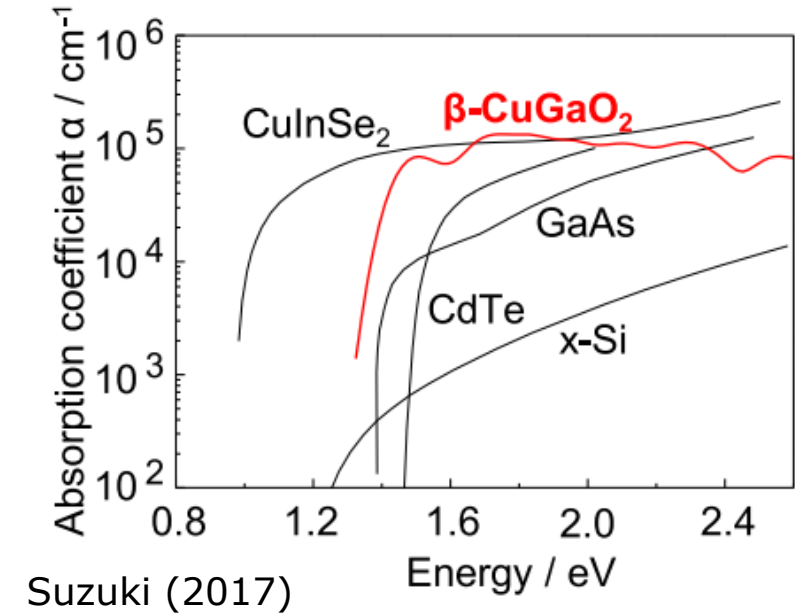


Morasch, AK (2016)

Low photocurrent and low photovoltage

## β-CuGaO<sub>2</sub>

direct gap: 1.5 eV  
absorption: >10<sup>5</sup> /cm



Suzuki (2017)

No PV properties published yet

# The Fermi energy in ionic semiconductors

The Fermi energy in a material is determined by **charge neutrality**

- Charge neutrality in covalent semiconductors:  $n + N_A^- = p + N_D^+$
- Charge neutrality in ionic semiconductors includes additional defects:

$$[h] + k[D^{k+}] + l[D_{intr}^{l+}] + [Cat_{cat}^+] + [An_{An}^+] = [e] + m[A^{m-}] + n[A_{intr}^{n-}] + [Cat_{cat}^-]$$

## Positive charges

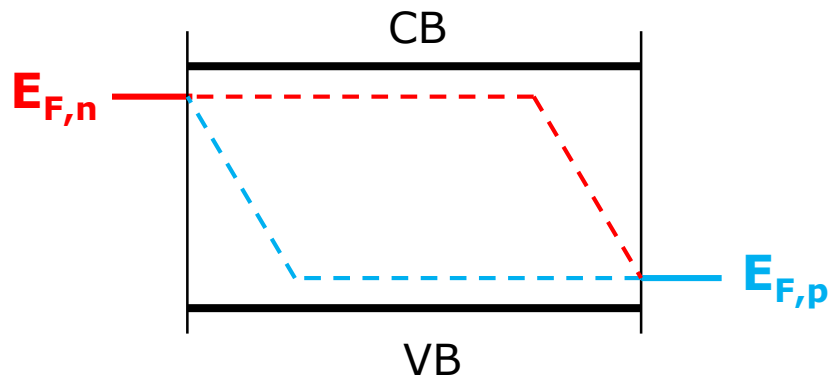
- Free holes
- Extrinsic donors
- **Intrinsic donors**  
(anion vacancies, cation interstitials)
- **Trapped holes**

## Negative charges

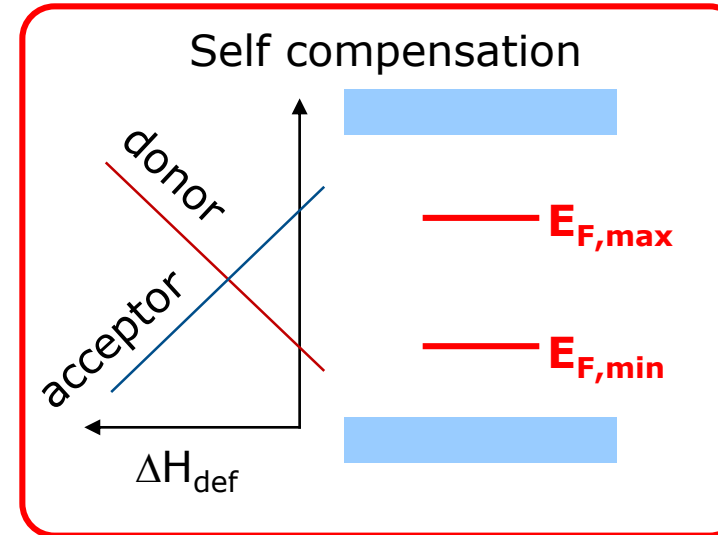
- Free electrons
- Extrinsic acceptors
- **Intrinsic acceptors**  
(cation vacancies, anion interstitials)
- **Trapped electrons**

# Limitation of the Fermi energy in ionic semiconductors

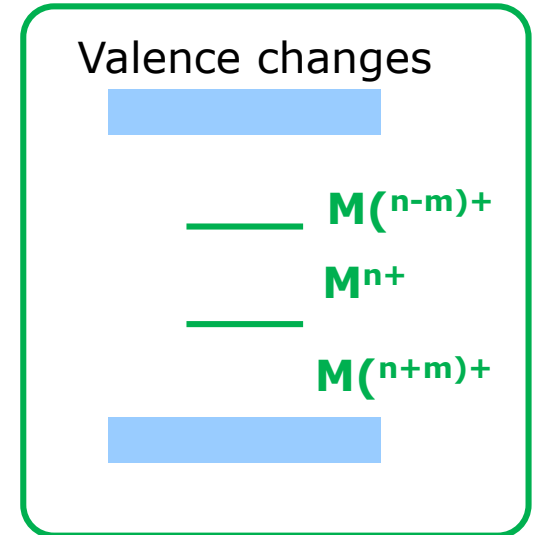
The photovoltage of a solar cell is determined by the splitting of the quasi Fermi levels under illumination



## Intrinsic defects



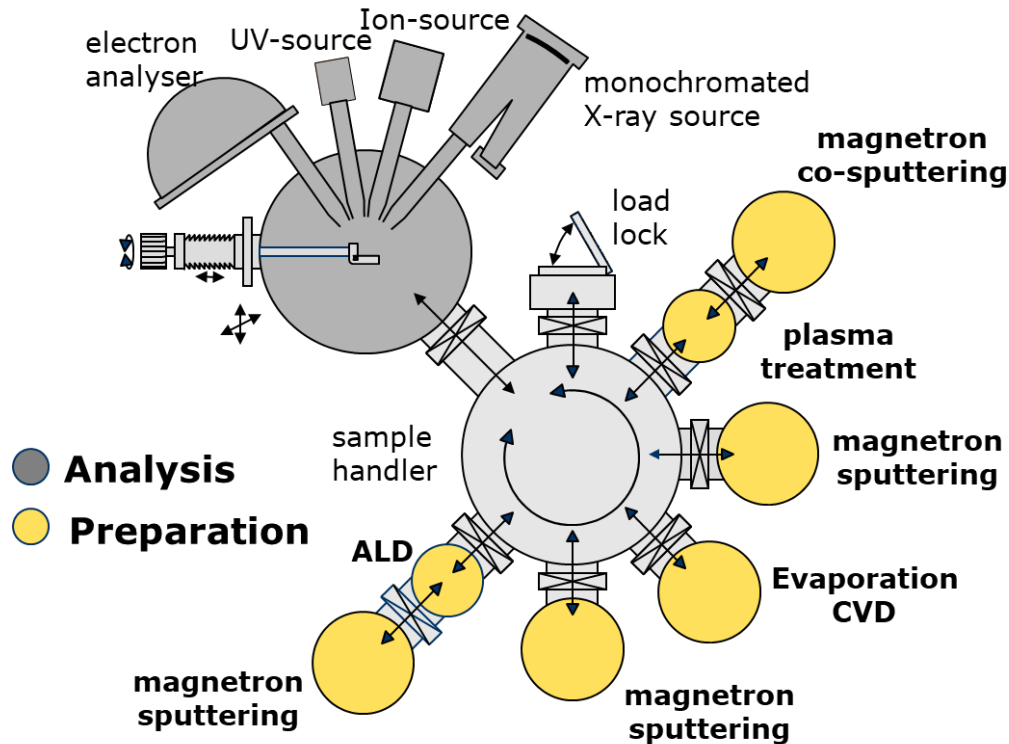
## Charge trapping



- The formation of compensating defects requires exchange of species, which can happen during synthesis at elevated  $T$ , but is typically suppressed at room temperature (operating conditions)
- On the contrary, charge trapping can occur at any temperature when charges are introduced into the valence or conduction by absorption or injection
- **Charge trapping can limit the variation (splitting) of the Fermi energy**

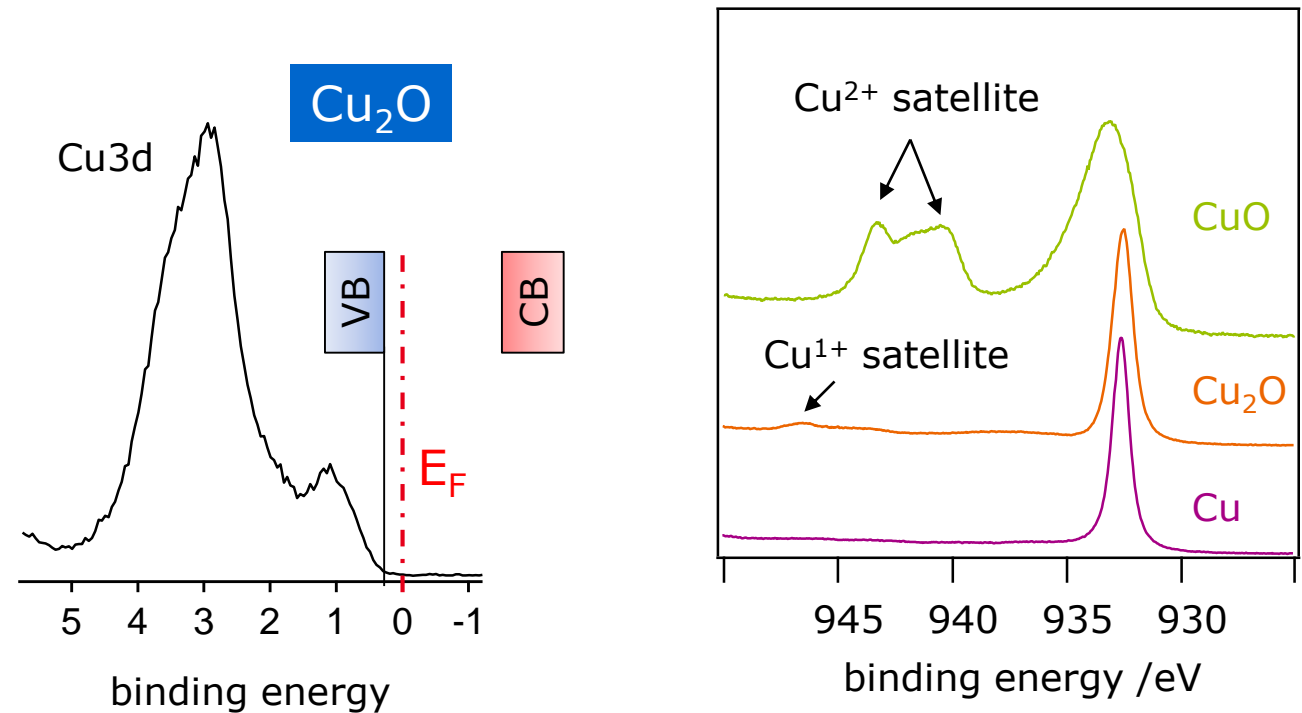
# Experimental setup

## Cluster tool DAISY-MAT



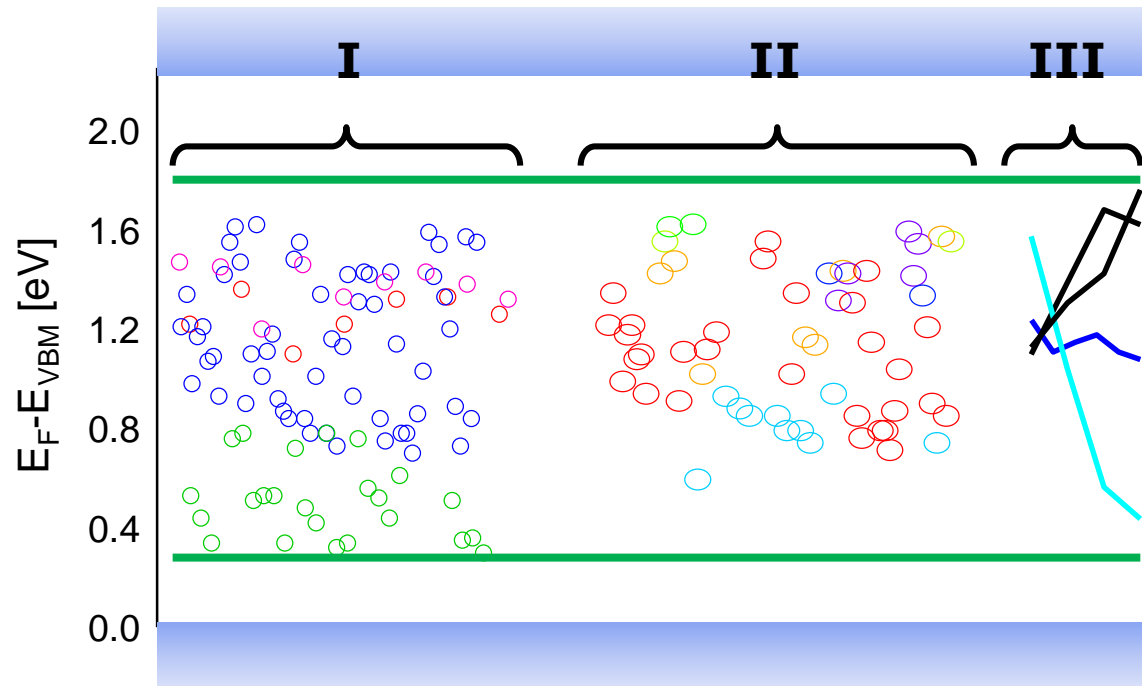
➤ **in-situ deposition and processing**

## X-ray Photoelectron spectroscopy



- **direct determination of Fermi energy**
- **direct determination of Cu oxidation state**

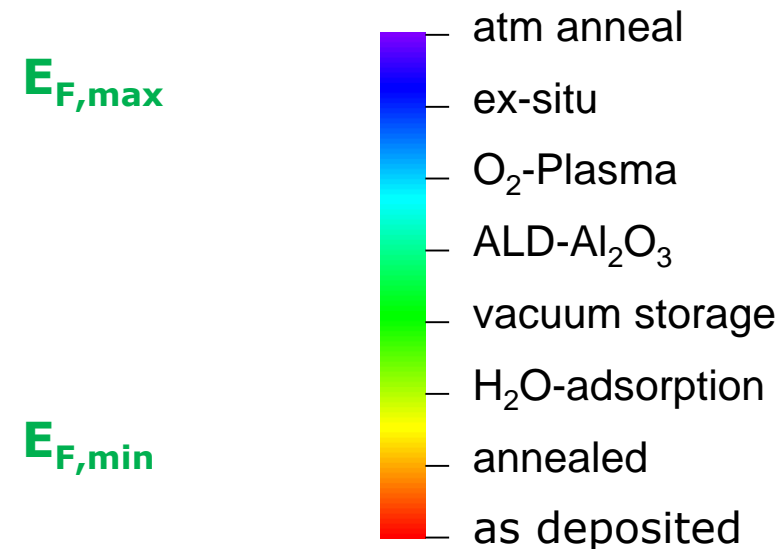
# How to find the limits of $E_F$ (e.g. $\text{Fe}_2\text{O}_3$ )



I. Doping: Mg, Si, Zr

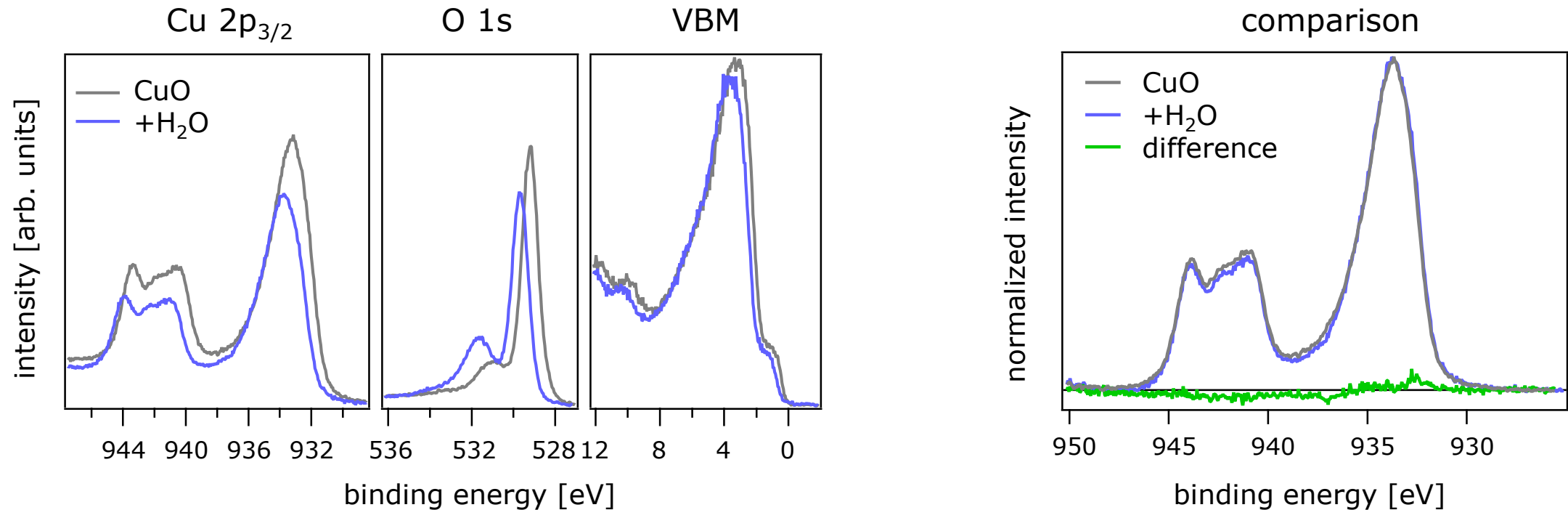
III. Interfaces:  $\text{RuO}_2$ , NiO, ITO

## II. Post-deposition treatments



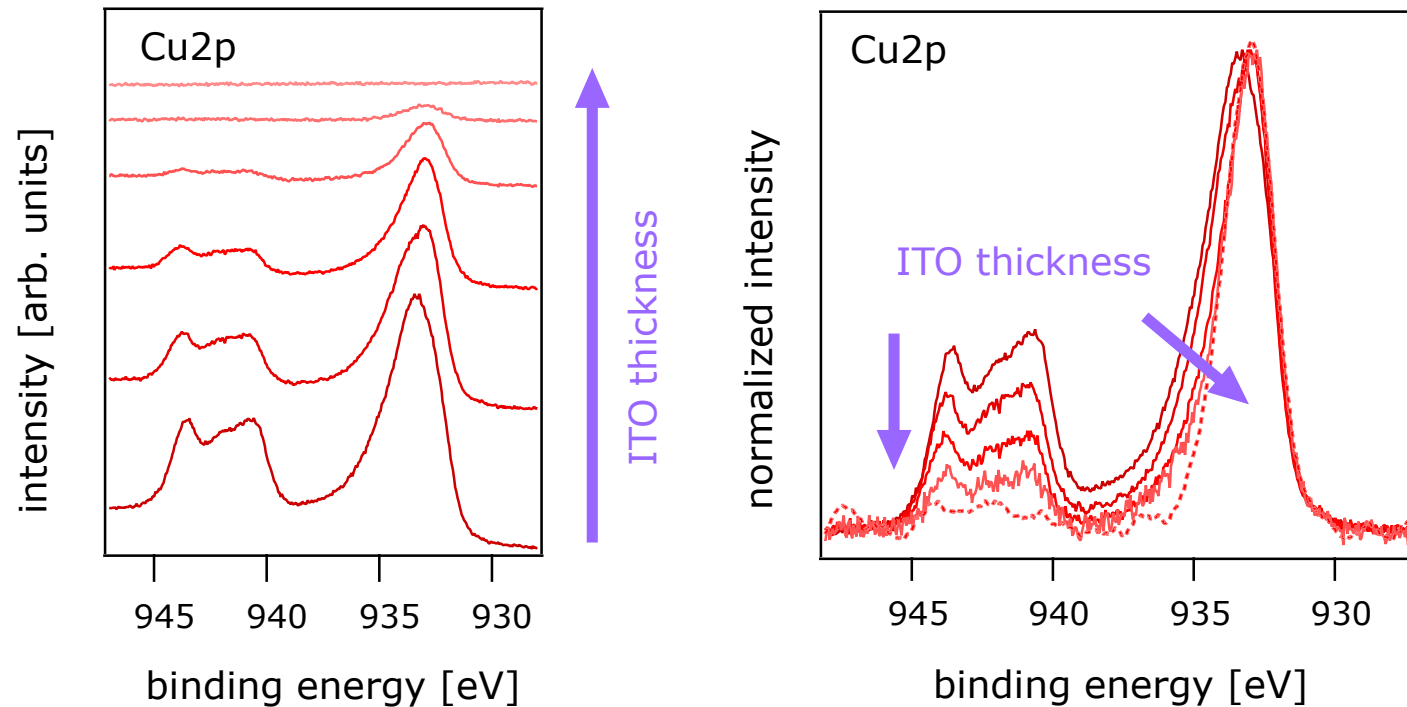
Similar limits of Fermi energy for doped and treated  $\text{Fe}_2\text{O}_3$  and at interfaces

# Water exposure to CuO

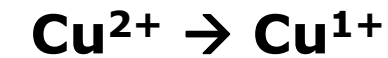


- Water adsorption leads to an upward shift of the Fermi energy  
→ also observed for SnO<sub>2</sub>, ZnO, NiO, Fe<sub>2</sub>O<sub>3</sub>, BiVO<sub>4</sub>, CuFeO<sub>2</sub>, BiFeO<sub>3</sub>, ...
- The intensity of the Cu<sup>2+</sup> satellite intensity is slightly reduced → partial reduction of Cu

# CuO/ITO interface formation



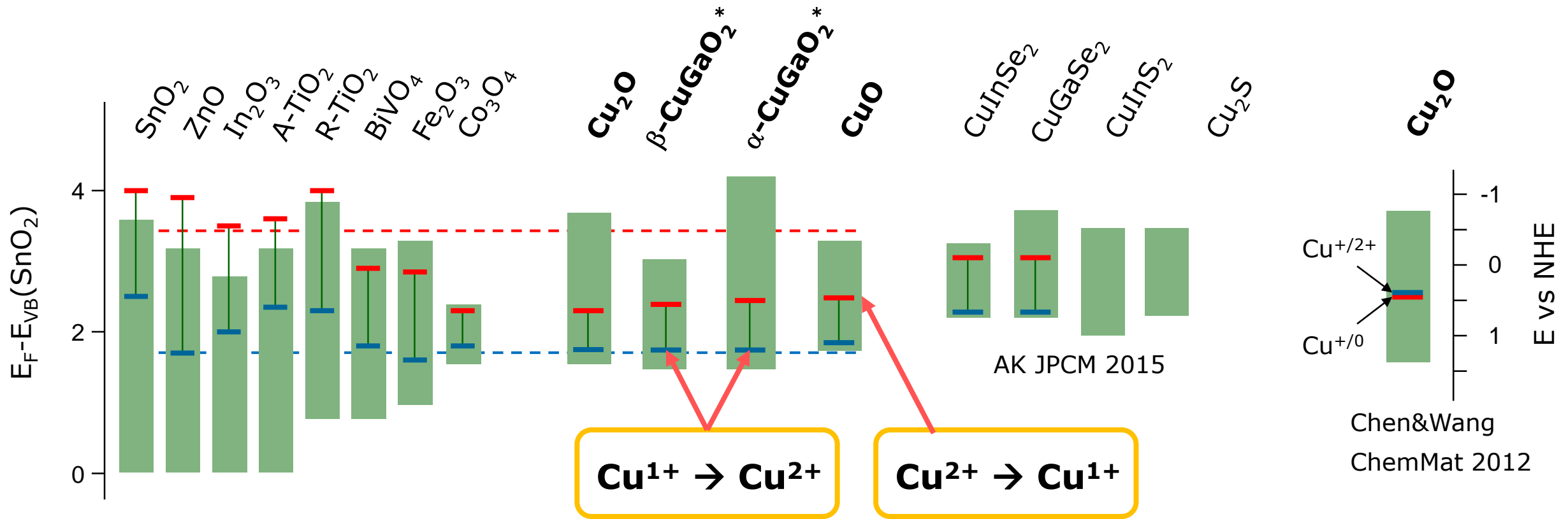
- upward shift of the Fermi energy
- narrowing of the Cu 2p emission
- reduction of Cu<sup>2+</sup> satellite intensity



- **Upper limit of Fermi energy by electrochemical reduction of Cu**
- **Corresponds to electron trapping at Cu<sup>2+</sup> sites**



# Summary



**The variation of the Fermi energy in Cu-oxides is limited by the oxidation and reduction of Cu to  $\sim 0.6$  eV**