



## Regular Article

# Construction of coherent interface between Cu<sub>2</sub>O and CeO<sub>2</sub> via electrochemical reconstruction for efficient carbon dioxide reduction to methane



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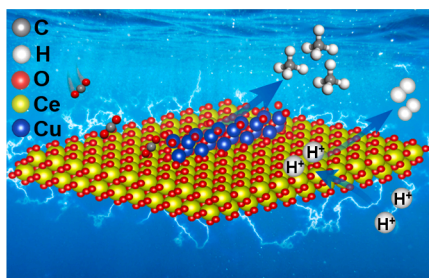
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## GRAPHICAL ABSTRACT

The *in-situ* electrochemically reconstructed Cu<sub>2</sub>O/CeO<sub>2</sub> from CuO/CeO<sub>2</sub> has been demonstrated to be effective for electrocatalytic carbon dioxide reduction, delivering a high selectivity, activity, and stability for producing CH<sub>4</sub>.



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## ABSTRACT

Developing an efficient electrocatalyst that enables the efficient electrochemical conversion from CO<sub>2</sub> to CH<sub>4</sub> across a wide potential range remains a formidable challenge. Herein, we introduce a precatalyst strategy that realizes the *in situ* electrochemical reconstruction of ultrafine Cu<sub>2</sub>O nanodomains, intricately coupled on the CeO<sub>2</sub> surface (Cu<sub>2</sub>O/CeO<sub>2</sub>), originating from the heterointerface comprised of ultrafine CuO nanodomains on the CeO<sub>2</sub> surface (CuO/CeO<sub>2</sub>). When served as the electrocatalyst for the electrochemical CO<sub>2</sub> reduction reaction, Cu<sub>2</sub>O/CeO<sub>2</sub> delivers a selectivity higher than 49 % towards CH<sub>4</sub> over a broad potential range from –1.2 V to

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–1.7 V vs. RHE, maintaining negligible activity decay for 20 h. Notably, the highest selectivity for CH<sub>4</sub> reaches an impressive 70 % at –1.5 V vs. RHE. Through the combination of comprehensive analysis including synchrotron X-ray absorption spectroscopy, spherical aberration-corrected high-angle annular dark field scanning transmission electron microscope as well as the density functional theoretical calculation, the efficient production of CH<sub>4</sub> is attributed to the coherent interface between Cu<sub>2</sub>O and CeO<sub>2</sub>, which could converted from the original CuO and CeO<sub>2</sub> interface, ensuring abundant active sites and enhanced intrinsic activity and selectivity towards CH<sub>4</sub>.

## 1. Introduction

The prospect of synthesizing chemical fuel through electrochemical CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) is highly promising, representing a robust pathway towards the realization of a carbon-neutral economy [1–4]. Through this electrochemical method, CO<sub>2</sub> is expected to undergo conversion into methane (CH<sub>4</sub>), formic acid, carbon monoxide, methanol, ethylene, ethane, ethanol, and other hydrocarbons [5–18]. Since CH<sub>4</sub> is the fundamental component of natural gas and bears the highest energy density among hydrocarbon compounds, it stands as an appealing electrocatalytic reduction product of CO<sub>2</sub>. However, the efficiency for the conversion from CO<sub>2</sub> to CH<sub>4</sub> is limited by the kinetically sluggish eight-electron transfer process [19]. In this regard, the development of electrocatalysts with high activity, selectivity, and stability is crucial for producing CH<sub>4</sub> through CO<sub>2</sub>RR.

Copper (Cu) is one of the most effective electrocatalysts for CO<sub>2</sub>RR [20–23]. However, it has relatively poor activity and selectivity in methanogenesis [2]. In contrast, the crystal plane of cuprous oxide (Cu<sub>2</sub>O) offers controllability, and its structure is considerably stable, resulting in improved CH<sub>4</sub> production performance compared to pristine Cu [24]. In order to further enhance activity, researchers have integrated Cu<sub>2</sub>O with various compounds (e.g., MOFs, CeO<sub>2</sub>, graphene, SnO<sub>2</sub>) [25–31]. This strategic combination exploits the synergistic interaction between Cu<sub>2</sub>O and the introduced compound, triggering interfacial electron interaction and consequently enhancing the selectivity and activity for CO<sub>2</sub>RR. For instance, Yi *et al.* [4] electrochemically reduced the surface Cu ion of CuHHTP to Cu<sub>2</sub>O, uniformly distributing the formed Cu<sub>2</sub>O on the remaining framework to create a highly active electrocatalyst, Cu<sub>2</sub>O@CuHHTP. The reduction of the Cu center to Cu<sub>2</sub>O released abundant uncoordinated hydroxyl groups near the active site, establishing hydrogen bonds with the intermediate and lowering the energy barrier for CH<sub>4</sub> formation. Yan *et al.* [31] modified CuO by CeO<sub>2</sub>, where the interface effect of the catalyst reduced the energy of \*CHO production. The rapid activation of water around CeO<sub>2</sub> accelerated the formation of \*CHO, which promoted the C–C coupling step and improved the faradaic efficiency (FE) of C<sub>2+</sub> products. Nevertheless, to the best of our knowledge, most of these reported composite systems still face challenges related to limited active area and structural stability, resulting in unsatisfactory efficiency for CH<sub>4</sub> production, especially across a wide potential range. Alternatively, CeO<sub>2</sub> can be used as a promising hybrid unit with Cu<sub>2</sub>O because of its several merits including potential coherent interface, excellent chemical and composition stability, and modulated morphology. Despite these advantages, there remains a gap in research exploring the electrocatalytic CO<sub>2</sub>RR through the combination of Cu<sub>2</sub>O and CeO<sub>2</sub>, which may offer potentially unlocking enhanced catalytic performance for CH<sub>4</sub>.

On the other hand, recent studies have found that electrochemical reconstruction can enable the catalyst to exhibit excellent catalytic performance, especially for the oxygen evolution reaction (OER) [32–36]. This phenomenon is primarily attributed to the capacity of electrochemical reconstruction to induce the formation of a new active phase characterized by high dispersity and defective nanodomains [32–36]. For example, Jiang *et al.* [37] synthesized a new type of OER precatalyst (Co<sub>2</sub>(OH)<sub>3</sub>Cl), with distinctive physicochemical properties, facilitating phase reconstruction during OER through lattice anion etching. This reconstruction process continuously promoted OER activity. Zhu *et al.* [38] designed CuS nanomaterials with unique atomic

arrangement sequences and certain cationic defects. After reconstruction, the formed has faster electron transport efficiency and a stronger Cu–S bond, which improves its OER performance. Interestingly, recent reports have indicated that Cu species with a lower chemical valence are formed from CuO during CO<sub>2</sub>RR [39–43], implying the possible transformation from CuO to Cu<sub>2</sub>O. However, owing to the large particle size, the phase reconstruction of CuO appears limited to the surface, resulting in an ambiguous identification of the real active structure, which deserves further attention. On the basis of this premise, one can envisage that employing ultrafine CuO particles with a size in the nanometer range in conjunction with CeO<sub>2</sub> to construct a (pre)catalyst could more readily trigger phase reconstruction, which could offer a chance to identify the real active structure. Unfortunately, there is still a lack of research on this pertinent investigation. Therefore, it is interesting to devise a strategy that allows to form an active intricately coupled Cu<sub>2</sub>O/CeO<sub>2</sub> interface via the *in-situ* reconstruction of CuO/CeO<sub>2</sub> heterophase.

Motivated by the aforementioned considerations, the heterointerface consisting of ultrafine CuO nanodomains grown on the surface of CeO<sub>2</sub> (CuO/CeO<sub>2</sub>) was initially constructed through hydrothermal treatment followed by calcination, during which a small amount of Cu ions was introduced during synthesis to prevent potential aggregation and agglomeration of CuO. Subsequently, CuO/CeO<sub>2</sub> was employed as the precatalyst for the electrocatalytic conversion of CO<sub>2</sub> to CH<sub>4</sub>. Notably, the optimized electrocatalyst (Ce-Cu-10) exhibited outstanding selectivity towards CH<sub>4</sub> (>70 %), surpassing the performance of most Cu-based catalysts. Furthermore, in a wide potential range from –1.2 V to –1.7 V vs. RHE, it delivers a CH<sub>4</sub> selectivity higher than 49 % with a total current density of current density is –23.4 mA·cm<sup>–2</sup>, showing excellent catalytic stability. Complementary X-ray absorption (XAS) and spherical aberration-corrected high-angle annular dark field scanning transmission electron microscope (SAC-HAADF-TEM) characterizations revealed a synergistic reconstruction from a coherent interface between ultrafine CuO and bulk CeO<sub>2</sub> to that between ultrafine Cu<sub>2</sub>O and bulk CeO<sub>2</sub>. Additionally, density functional theoretical (DFT) calculations elucidated the rate-determining step of CO<sub>2</sub> to CH<sub>4</sub>, which deepens the understanding of the catalytic mechanism at the molecular level. Our work also illustrates that in this unique heterostructure, the presence of CeO<sub>2</sub> is conducive to the adsorption of both CO<sub>2</sub> and proton, which contributes to the electrocatalytic conversion from CO<sub>2</sub> to CH<sub>4</sub> on the active Cu center. The findings in this work not only introduce a novel approach for designing highly efficient catalysts for the electrocatalytic reduction of CO<sub>2</sub> to CH<sub>4</sub> but also advance the fundamental research on precatalysts for CO<sub>2</sub>RR catalysis and beyond.

## 2. Experimental section

### 2.1. Materials

Cerium nitrate hexahydrate (Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O, 99.5 %) was purchased from Macklin. Nafion (~5 %) was purchased from UTEPO Technology Suzhou Co. Copper(ii) nitrate hydrate (Cu(NO<sub>3</sub>)<sub>2</sub>·3H<sub>2</sub>O, AR) was purchased from Sinopharm. Sodium hydroxide was purchased from Sodium hydroxide (NaOH, 98 %). Potassium bicarbonate (KHCO<sub>3</sub>, 99.7 %) was purchased from Aladdin.

## 2.2. Synthesis of CeO<sub>2</sub>, CuO, Ce-Cu-*x* (*x* = 6, 8, 10 and 12. *X* is the molar ratio of Ce/Cu)

A hydrothermal method was used to synthesize CeO<sub>2</sub>. Typically, 0.744 g of Ce(NO<sub>3</sub>)<sub>3</sub>·6H<sub>2</sub>O and 9.6 g of NaOH were mixed and dissolved in 40 mL of deionized (DI) water. The mixed solution was ultrasonicated for an additional 30 min at room temperature and then the inks were transferred into a 100 mL Teflon reactor. The Teflon reactor was tightly sealed and subjected to hydrothermal treatment in an oven at 350 K for 24 h. After cooling down, the white precipitate obtained was collected, washed with ultrapure water and ethanol three times, and dried overnight in a vacuum oven at 80 °C. The yellow powder obtained after drying was calcined in a muffle furnace at 450 °C for 3 h to get CeO<sub>2</sub>.

The CeO<sub>2</sub> was then used to prepare Ce-Cu-*x*. In order to prepare Ce-Cu-8, Ce-Cu-10 and Ce-Cu-12, 0.1 g CeO<sub>2</sub> was ultrasonically dispersed in 20 mL deionized water, and then 17.6 mg, 14.1 mg and 11.7 mg Cu(NO<sub>3</sub>)<sub>3</sub>·3H<sub>2</sub>O were added, respectively. After stirring for 20 min, 0.5 mol/L Na<sub>2</sub>CO<sub>3</sub> was added to adjust the pH to 9–10 of the solution. The mixed solution was centrifuged at 8000 rpm for five minutes, then it was dried overnight. The obtained powder was calcined in a muffle furnace at 500 °C for 3 h. The preparation process of CuO was the same as that of

Ce-Cu-8 except that CeO<sub>2</sub> was not added.

## 3. Results and discussion

To validate our concept, we prepared a sample with the Ce to Cu molar ratio of 10:1 (hereafter referred to as Ce-Cu-10), along with samples having Ce to Cu molar ratios of 1:0, 8:1, 12:1, and 0:1 for comparison (defined as CeO<sub>2</sub>, Ce-Cu-8, Ce-Cu-12, and CuO, respectively, and the distribution of element content in each sample is shown in Table S1). X-ray diffraction (XRD) was employed to characterize the structure of the prepared samples. As illustrated in Fig. 1a, the diffraction peak positions in the XRD patterns of CeO<sub>2</sub> and Ce-Cu-10 closely matched those of standard CeO<sub>2</sub> (PDF#81-0792). Fig. 1b represents a local enlargement of the XRD patterns depicted in Fig. 1a. It is evident that the positions of the diffraction peaks in the XRD patterns of CeO<sub>2</sub> and Ce-Cu-10 are essentially consistent, implying the copper species tend to exist in the form of copper oxide rather than as Cu-doped CeO<sub>2</sub>. To further exclude the influence of potential test factor on the diffraction peak positions, the commercial pure Fe powder was used as an internal standard, which was mixed with the prepared powder samples (50 wt %), and their XRD patterns were measured under the same conditions

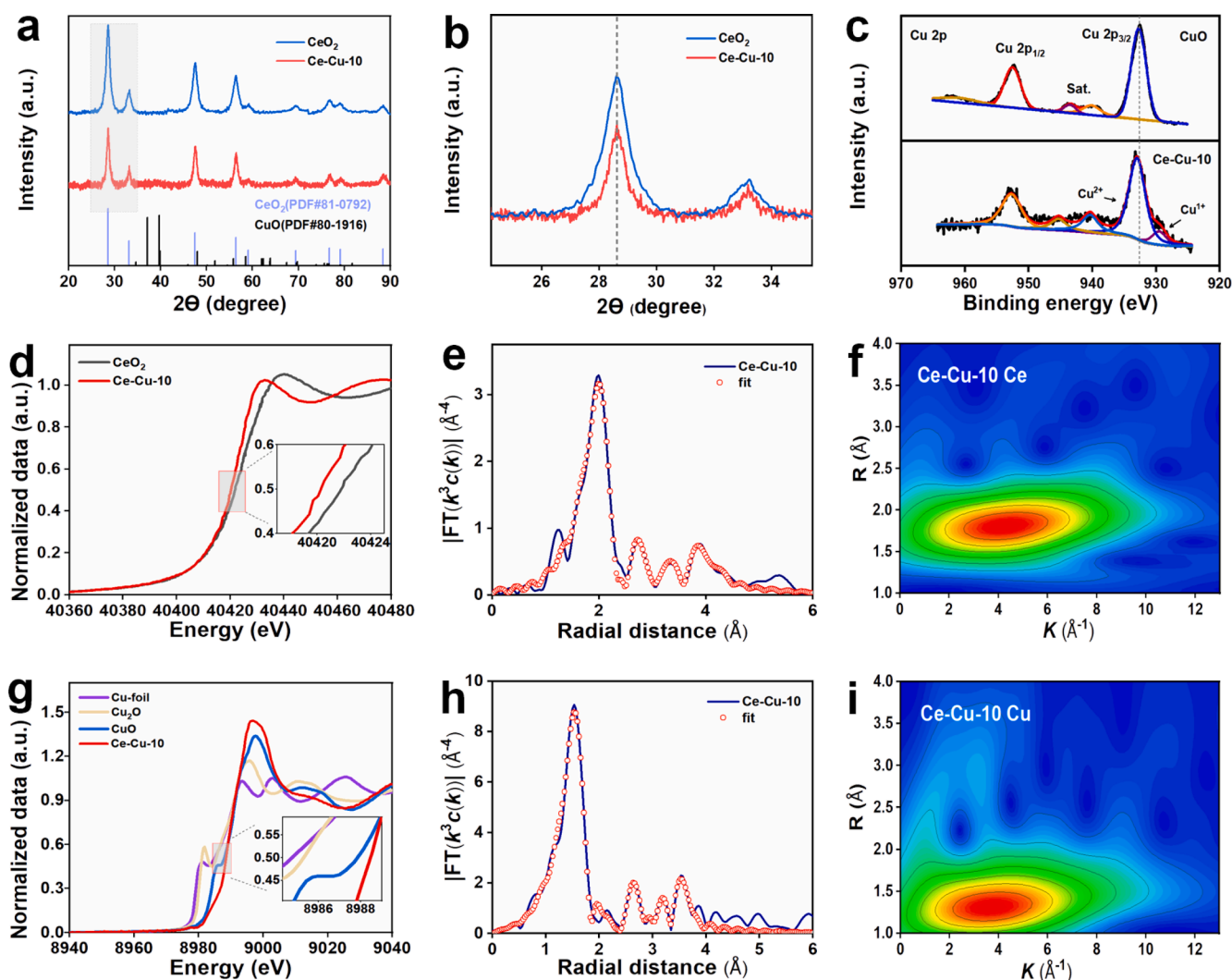


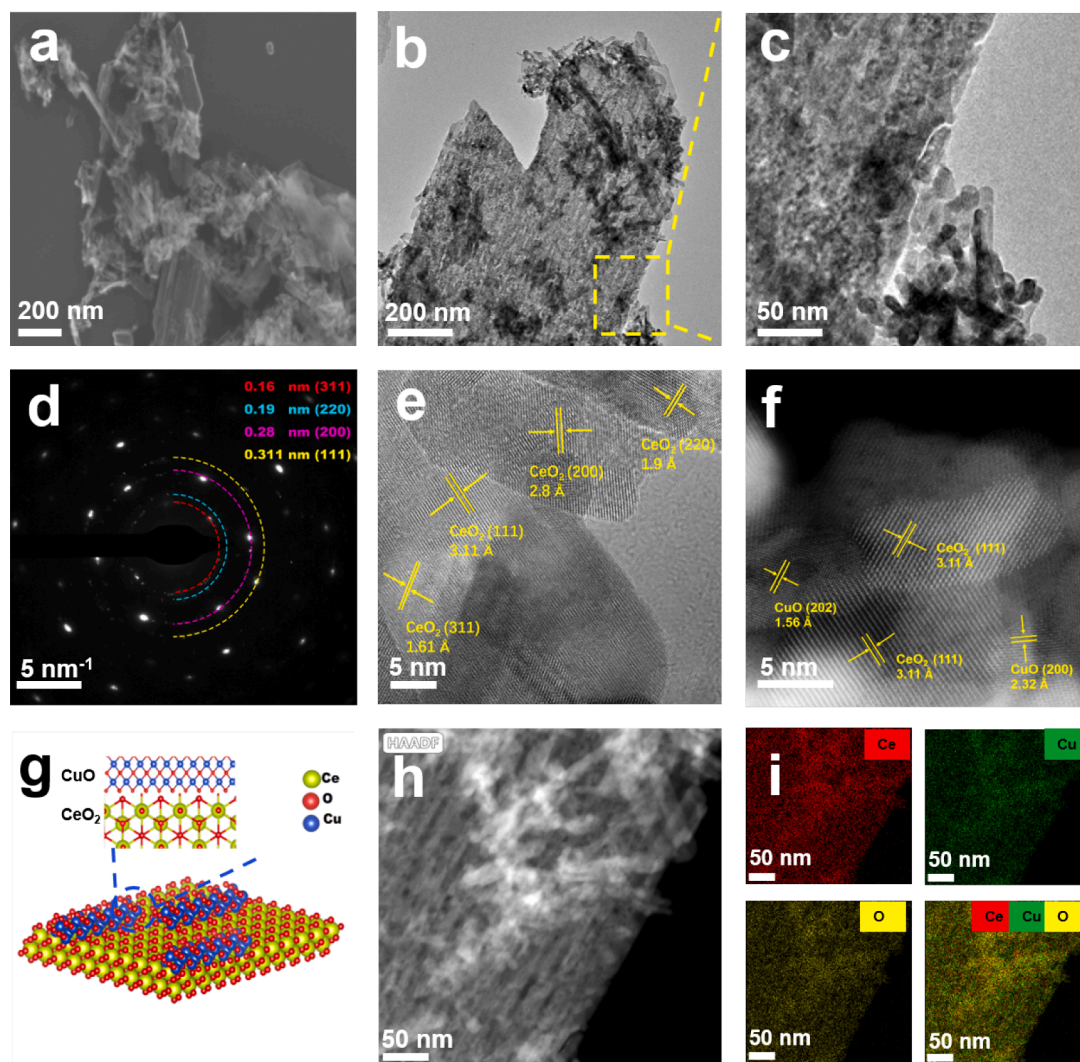
Fig. 1. (a) XRD patterns of the as-prepared CeO<sub>2</sub> and Ce-Cu-10 as well as (b) the associated magnified region in the two theta range from 25 to 35 degrees. (c) High resolution XPS spectra of Cu 2p in CuO and Ce-Cu-10. (d) Normalized XANES and the enlarged white line (the inset) spectra of Ce K-edge in Ce-Cu-10 and the standard sample CeO<sub>2</sub>. (e) EXAFS pattern for the Ce K-edge in CeO-Cu-10 fitted in R space. (f) Wavelet transformation pattern of Ce K-edge of Ce-Cu-10. (g) Normalized XANES and the enlarged white line (the inset) spectra of Cu K-edge in Ce-Cu-10, CuO, Cu<sub>2</sub>O and Cu-foil. (h) EXAFS pattern for the Cu K-edge in CeO-Cu-10 fitted in R space. (i) Wavelet transformation pattern of Cu K-edge of Ce-Cu-10.

(Fig. S1b). The results demonstrated that, compared with CeO<sub>2</sub>, there is no shift for the characteristic peaks in the XRD patterns of as-prepared Ce-Cu-8, Ce-Cu-10, and Ce-Cu-12. This implies that Cu species were not doped into the Ce sites of the CeO<sub>2</sub> lattice.

X-ray photoelectron spectroscopy (XPS) experiments were further carried out to decouple the surface chemical state of Cu species. The high resolution XPS spectra of Cu 2p in Ce-Cu-10 and CuO are displayed and compared in Fig. 1c. It could be seen that the peak for Cu 2p (932.98 eV) in Ce-Cu-10 shifted to a higher energy than that (932.78 eV) in CuO. The peak at Cu 2p<sub>3/2</sub> (929.58 eV) of Ce-Cu-10 was assigned to Cu<sup>+</sup>, primarily resulting from oxygen vacancies in CeO<sub>2</sub> [43–46]. The XPS spectra of Ce 3d in CeO<sub>2</sub> (Fig. S2a) were deconvoluted into eight peaks, corresponding to the Ce 3d<sub>5/2</sub> (v, 882.28 eV; v', 885.28 eV; v'', 888.88 eV; v''', 898.16 eV) and Ce 3d<sub>3/2</sub> states (u, 900.78 eV; u', 902.98 eV; u'', 907.48 eV; u''', 916.48 eV). The v, v', u, and u''' peaks were assigned to Ce<sup>3+</sup> species, while the remaining four peaks were assigned to Ce<sup>4+</sup> species. Compared with Ce 3d of CeO<sub>2</sub>, Ce-Cu-10 shifted to the lower energy. The above observations strongly indicated that electrons were transferred from Cu species to the surface of CeO<sub>2</sub>, being consistent with the results of XPS spectra for Cu 2p.

On the other hand, synchrotron radiation X-ray absorption near-edge structure (XANES) and extended X-ray absorption fine structure

(EXAFS) analyses were to gain deeper insights into the chemical states and coordination environments of Ce and Cu in Ce-Cu-10 at the atomic scale (Fig. 1d–i, Figs. S3–S5). In comparison with the XANES spectra for the Ce K-edge of CeO<sub>2</sub> (Fig. 1d), Ce-Cu-10 exhibited a blue shift, signifying a slight reduction in the valence state of Ce in Ce-Cu-10. This finding was consistent with the XPS results of Ce 3d in Ce-Cu-10. Furthermore, the coordination of Ce and Cu in K and R spaces was clearly delineated using wavelet transform-EXAFS (WT-EXAFS). The Ce K-edge and Cu K-edge WT-EXAFS patterns of Ce-Cu-10 are shown in Fig. 1f and 1i, respectively. Similar to the EXAFS spectrum (Fig. 1e, h), significant WT signals related to Ce–O, Ce–Cu, and Ce–Ce bonds could be observed, while WT signals of Cu–O, Cu–Cu, and Cu–Ce were detected for Cu element. Fig. 1g displayed the XANES spectra for the Cu K-edge of Ce-Cu-10, along with those for Cu foil, commercial CuO, and commercial Cu<sub>2</sub>O as reference. Fig. 1g shows the XANES spectra of the Cu K-edge. The energy position of the pre-edge of Ce-Cu-10 is very close to the energy position of commercial CuO, indicating that the oxidation state of Cu in Ce-Cu-10 is close to +2. Fig. S4b illustrates the Fourier transform EXAFS spectra of Ce-Cu-10 and the references (Cu foil, commercial CuO, and Cu<sub>2</sub>O). It could be concluded that the EXAFS spectrum of Ce-Cu-10 was similar to that of CuO reference. In addition, the k<sup>3</sup>χ(k) oscillation curve of Ce-Cu-10 at the Cu K-edge



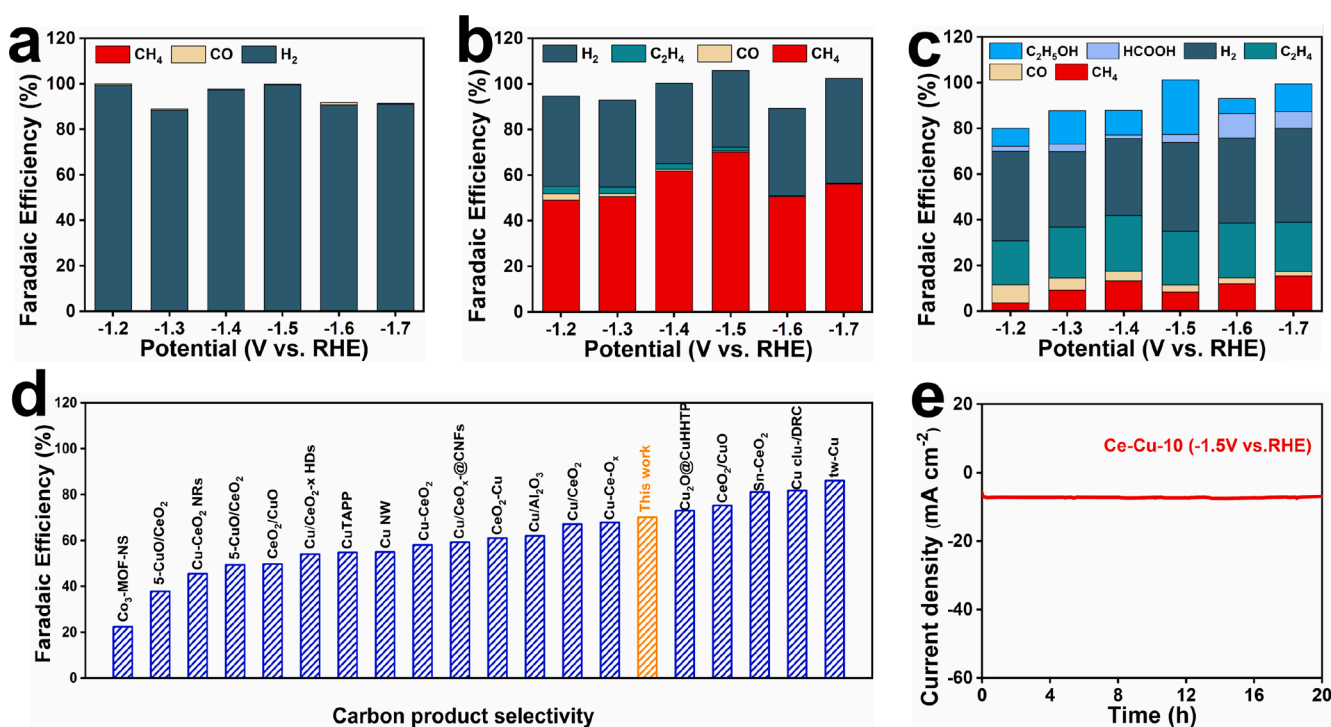
**Fig. 2.** Morphological and structural characterization of Ce-Cu-10. (a) FESEM, (b) TEM, (c) magnified TEM, (d) corresponding SAED, (e) HRTEM, (f) and SAC-HAADF-STEM images. (g) Surface structure diagram of Ce-Cu-10 (The blue, yellow, and red atoms represent Cu, Ce, and O atoms, respectively). (h) HAADF-STEM pattern of Ce-Cu-10 and its corresponding EDS mapping images of (i) Ce, Cu, and O. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

exhibited a similar trend to that of commercial CuO powder in shape and oscillation frequency, indicating that the Cu species in Ce-Cu-10 bore similar properties to those of CuO (Fig. S4a). These results collectively suggested that Cu in Ce-Cu-10 should be present in the form of CuO which had a strong charge coupling with CeO<sub>2</sub>, which could be supported by the formation of minor Ce–Cu bond observed in the EXAFS patterns for Cu K-edge and Ce K-edge in Ce-Cu-10.

The morphology of the as-prepared five samples was further examined by field-emission scanning electron microscopy (FESEM). The pristine CuO compound exhibited a granular shape with an average particle size of about 100 nm (Fig. S6a). In contrast, the pristine CeO<sub>2</sub> displayed a pseudo-nanorod morphology (Fig. S6b). Upon the introduction of Cu species, the morphology was largely retained, closely resembling that of CeO<sub>2</sub> (Fig. 2a, Fig. S6c, d). The microstructure of Ce-Cu-10 was further characterized by transmission electron microscopy (TEM). Consistent with the FESEM images, TEM revealed a pseudo-nanorod morphology (Fig. 2b, c). The corresponding selected area electron diffraction pattern (SAED) showed four diffraction rings that could be indexed to the (111), (200), (220), and (311) facets of the CeO<sub>2</sub> phase, respectively (Fig. 2d). The high-resolution TEM images (HRTEM) shown in Fig. 2e depicted lattice distances of 0.311, 0.28, 0.19, and 0.16 nm, corresponding to the (111), (200), (220), and (311) facets of CeO<sub>2</sub> phase, respectively [42,47]. Intriguingly, no lattice fringes belonging to CuO were observed by HRTEM. To further clarify the existence or presence of CuO in the form of ultrafine size, spherical aberration-corrected high-angle annular dark field scanning transmission electron microscope (SAC-HAADF-TEM) (Fig. 2f). As expected, in addition to the (111) facet of CeO<sub>2</sub> (0.311 nm), 0.156 and 0.172 nm of lattice distances corresponding to (202) and (200) facets of CuO were also identified. The quantum size of CuO could be a reason why it was challenging to identify its presence by XRD and HRTEM. Fig. 2g schematically displayed the coupling between the CeO<sub>2</sub> matrix and ultrafine CuO nanodots, where the former not only tuned the structure of the latter but also physically confined it. In addition, the high-angle

annular dark field scanning transmission electron microscope (HAADF-STEM) and corresponding EDS elemental mapping images for the representative Ce-Cu-10 verified the uniform distribution of Ce, O, and Cu elements in the Ce-Cu-10 composite, confirming that Cu species were uniformly loaded on CeO<sub>2</sub> (Fig. 2h, i).

The electrochemical performance of Ce-Cu-10 for the CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) was evaluated in a sealed H-cell with a CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> via the constant potential method over a potential range from –1.2 to –1.7 V vs. RHE. Gas chromatography (GC) was employed for the detection and quantification of gaseous products, while liquid products were analyzed using a nuclear magnetic resonance spectrometer (NMR) after completion of the electrochemical reaction. For a comprehensive comparison, the CO<sub>2</sub>RR activity of CeO<sub>2</sub>, synthetic CuO, and other control samples was also examined under identical test conditions. Fig. 3a–c show the CO<sub>2</sub>RR products containing both gas and liquid phases, with the gas phases being the predominant products. The FE of the gas products for Ce-Cu-10, CeO<sub>2</sub>, and CuO is displayed in Fig. 3a–c. In the case of Ce-Cu-10, even at an initial cathode potential of –1.2 V, CH<sub>4</sub> was observed (49.03 %), accompanied by a certain amount of H<sub>2</sub> gas (Fig. 3b). With an increase in cathode potential, the FE yield of CH<sub>4</sub> further increased, while the yield of H<sub>2</sub> decreased. At –1.5 V (vs. RHE), the FE for CH<sub>4</sub> formation on Ce-Cu-10 reached 70.1 % with almost no other C<sub>1+</sub> products. However, the FE of CH<sub>4</sub> for CeO<sub>2</sub> and CuO was almost zero and 8.44 % at the same potential, respectively (Fig. 3a and 3c). Compared with Cu<sub>2</sub>O, Ce-Cu-8 and Ce-Cu-12 (Figs. S7–S9), Ce-Cu-10 has the highest CH<sub>4</sub> selectivity at –1.5 V vs. RHE. Actually, except for the gas products, some liquid products by CO<sub>2</sub>RR were also identified. The liquid phase products of the samples in the electrocatalytic CO<sub>2</sub>RR were checked by NMR, which are shown in Fig. S8 and Fig. S9. Note that the total FEs for all the samples are close to 100 %, indicating that nearly all products were detected. Consequently, when compared with previously reported Cu and Ce-based catalysts for CO<sub>2</sub>RR, Ce-Cu-10 demonstrated a significantly higher CH<sub>4</sub> production (Fig. 3d, Table S2). To further demonstrate the superior catalytic activity of Ce-Cu-10, the



**Fig. 3.** Electrochemical CO<sub>2</sub>RR of CeO<sub>2</sub>, Ce-Cu-10, and CuO. Product distributions and corresponding faradaic efficiencies produced by (a) CeO<sub>2</sub> (b) Ce-Cu-10 and (c) CuO at different applied potentials in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolyte. (d) Comparison of the ethylene faradaic efficiency of Ce-Cu-10 with previously reported state-of-the-art Cu-based catalysts (the details are provided in Table S2). (e) Stability test of Ce-Cu-10 at –1.5 V (vs. RHE) in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> electrolyte.

linear sweep voltammetry (LSV) curves towards CO<sub>2</sub>RR in N<sub>2</sub>-saturated and CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> solution (Fig. S10a and 10b) were measured. It is well known that the electrode material mainly competes for HER in the N<sub>2</sub>-saturated solution, while CO<sub>2</sub>RR is more favorable in the CO<sub>2</sub>-saturated solution. Compared with the CeO<sub>2</sub> catalyst, the Ce-Cu-10 catalyst exhibited a larger positive current difference in the CO<sub>2</sub>-saturated electrolyte, indicating that it had higher catalytic activity for CO<sub>2</sub>RR. Specifically, Ce-Cu-10 had a higher current density ( $-23.43 \text{ mA}\cdot\text{cm}^{-2}$ ) than CeO<sub>2</sub> ( $-1.97 \text{ mA}\cdot\text{cm}^{-2}$ ) at  $-1.5 \text{ V}$  vs. RHE, highlighting that the introduction of ultrafine CuO significantly enhanced the catalytic activity of CO<sub>2</sub>RR (Fig. S10a).

To probe the reasons for the high activity, the electrochemical surface area (ECSA) in the potential range of 0.75–0.85 V vs. RHE in CO<sub>2</sub>-saturated 0.5 M KHCO<sub>3</sub> was evaluated for the electrocatalysts (Fig. S11a–e). ECSA was evaluated by measuring the electric double layer (C<sub>dl</sub>) value. As shown in Fig. S11f, the C<sub>dl</sub> ( $0.17 \text{ mF}\cdot\text{cm}^{-2}$ ) of the Ce-Cu-10 catalyst was larger than those of other reference catalysts, suggesting an abundance of potential active sites, thereby enhancing catalytic current density.

The electrochemical impedance spectroscopy (EIS, Fig. S10c) revealed that in the CO<sub>2</sub>RR process, Ce-Cu-10 exhibited the smallest charge transfer resistance (R<sub>ct</sub>). A smaller R<sub>ct</sub> signifies faster electron transfer from the catalyst to CO<sub>2</sub>, a pivotal factor in enhancing catalytic performance. Furthermore, we measured CO<sub>2</sub> adsorption isotherms of Ce-Cu-8, Ce-Cu-10, Ce-Cu-12, CeO<sub>2</sub>, and CuO at 25 °C (Fig. S12) and found that the CO<sub>2</sub> adsorption capacity by CuO is the lowest ( $95.83 \mu\text{mol}\cdot\text{g}^{-1}$ ). Remarkably, the adsorption capacity of CO<sub>2</sub> by Ce-Cu-10 was increased by 126 % after loading it on CeO<sub>2</sub> ( $217.02 \mu\text{mol}\cdot\text{g}^{-1}$ ). The Brunauer–Emmett–Teller (BET) surface areas of CeO<sub>2</sub>, Ce-Cu-10, and CuO are  $70.46 \pm 0.11$ ,  $55.06 \pm 0.03$ , and  $8.95 \pm 0.11 \text{ m}^2\cdot\text{g}^{-1}$ , respectively (Table S3). These results confirmed enhanced CO<sub>2</sub> adsorption in the composite material, contributing to sufficient interaction of CO<sub>2</sub> with the catalyst surface, thereby mitigating the mass transfer limitation of CO<sub>2</sub>RR at high reduction potential.

Since durability is a crucial metric for describing catalytic

performance in CO<sub>2</sub>RR, the electrochemical stability of the Ce-Cu-10 catalyst was assessed through long-term chronoamperometry at an optimal overpotential of  $-1.5 \text{ V}$  vs. RHE. As depicted in Fig. 3e, the current density remained remarkably stable throughout a continuous 20-hour test, signifying the robust and stable CO<sub>2</sub>RR activity of the Ce-Cu-10 catalyst. The post-stability of the catalyst was characterized by XRD, and the pattern was nearly identical to that of a pristine one (Fig. S13). The high resolution Cu 2p XPS spectra showed the intensified signal for Cu<sup>+</sup>, suggesting the reduction of Cu species during CO<sub>2</sub>RR, while the high resolution Cu 2p XPS spectra after CO<sub>2</sub>RR tests cycling displayed characteristics similar to the fresh catalyst (Fig. S14). Further insights into the chemical states of Cu and Ce after CO<sub>2</sub>RR were gained through XAS measurements on the Cu K-edge and Ce K-edge in Ce-Cu-10. Substantial changes were observed in the XANES and EXAFS spectra of the Ce-Cu-10 catalyst after the chronoamperometry test conducted under optimal performance conditions (Fig. 4, Figs. S3 and S4). The XANES pattern for Ce K-edge showed a slight negative shift as compared to that before CO<sub>2</sub>RR, probably due to the local structural or interfacial variation (Fig. 4a). Corresponding EXAFS and WT patterns demonstrated subtle changes in the local structure (Fig. 4b and 4c). The XANES pattern for Cu K-edge in Ce-Cu-10 catalyst post reaction showed an apparent shift to the lower energy, confirming that the valence state of Cu decreased to close to +1 after the durability test (Fig. 4d). Additionally, EXAFS patterns and WT signals revealed the presence of Cu species in the form of Cu<sub>2</sub>O (Fig. S4b, Fig. 4e, and 4f), strongly affirming the transformation of CuO to Cu<sub>2</sub>O during CO<sub>2</sub>RR. Besides, the preserved Ce–Cu bond suggested a strong coupling between CeO<sub>2</sub> and the formed Cu<sub>2</sub>O.

The morphology of the Ce-Cu-10 catalyst after CO<sub>2</sub>RR was then conducted through FESEM and TEM, which indicated that the morphology remained essentially unchanged compared to the pre-reaction state (Fig. 5a–c). The SAED and HRTEM images exhibited clear lattice spacings of 0.311, 0.28, and 0.19 nm, corresponding to (111), (200), and (220) facets of CeO<sub>2</sub>, respectively, with no identification of Cu-based phases. However, SAC-HAADF-STEM images

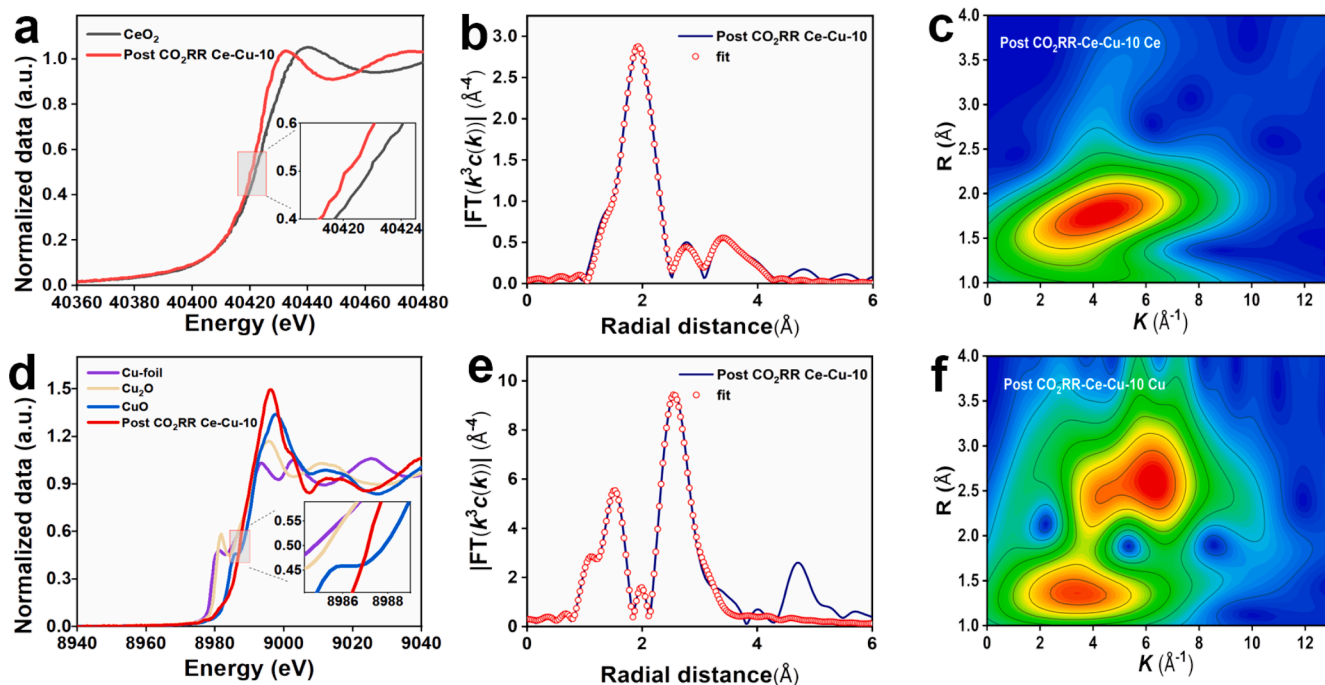
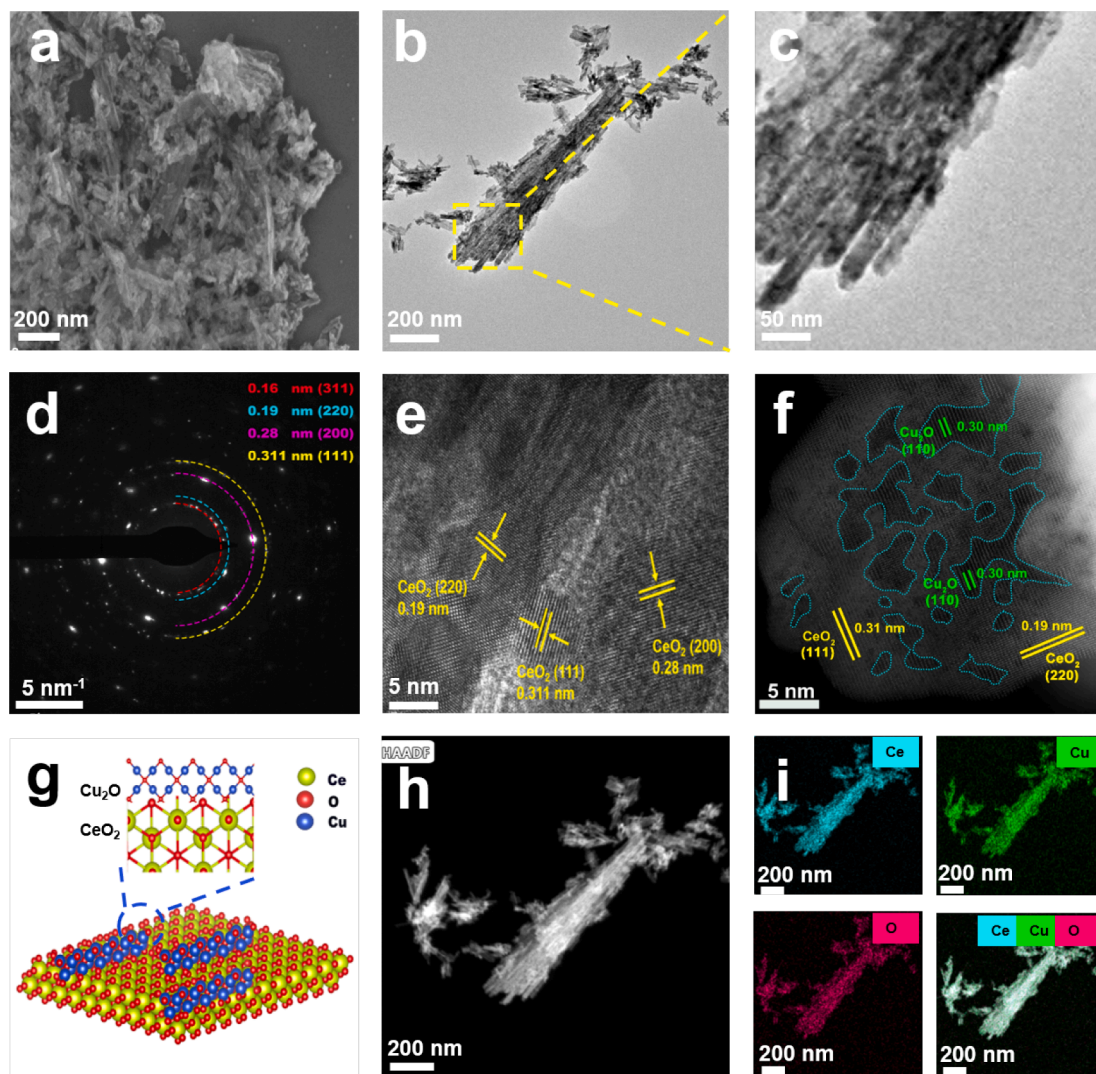


Fig. 4. (a) Normalized XANES and the enlarged white line (the inset) spectra of Ce K-edge in post CO<sub>2</sub>RR Ce-Cu-10 and the standard sample CeO<sub>2</sub>. (b) Post CO<sub>2</sub>RR Ce-Cu-10 fitted in R space. (c) Wavelet transformation pattern of Ce K-edge EXAFS of post CO<sub>2</sub>RR Ce-Cu-10. (d) Normalized XANES and the enlarged white line (the inset) spectra of Cu K-edge in post CO<sub>2</sub>RR Ce-Cu-10, CuO, Cu<sub>2</sub>O and Cu-foil. (e) Post CO<sub>2</sub>RR Ce-Cu-10 fitted in R space. (f) Wavelet transformation pattern of Cu K-edge EXAFS of post CO<sub>2</sub>RR Ce-Cu-10.



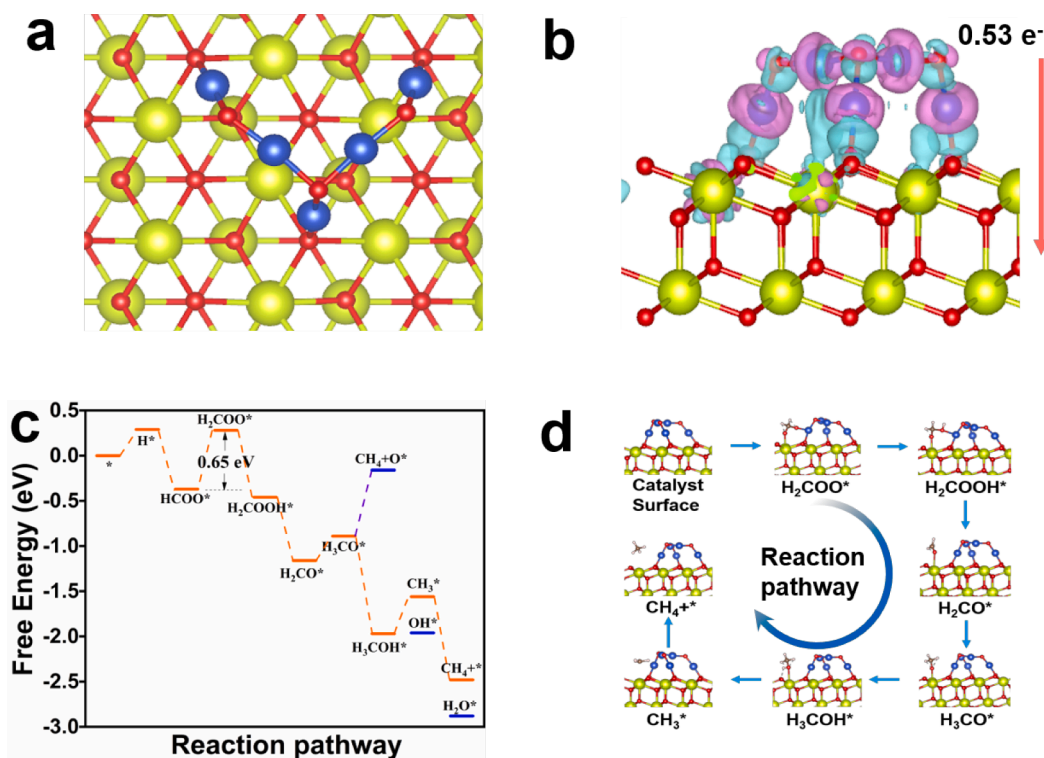
**Fig. 5.** Morphological and structural characterization of post CO<sub>2</sub>RR Ce-Cu-10. (a) FESEM, (b) TEM, (c) magnified TEM, (d) corresponding SAED, (e) HRTEM, (f) and SAC-HAADF-STEM images. (g) Surface structure diagram of Ce-Cu-10 (The blue, yellow, and red atoms represent Cu, Ce, and O atoms, respectively). (h) HAADF-STEM pattern of Ce-Cu-10 and its corresponding EDS mapping images of (i) Ce, Cu, and O. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

revealed discernible lattice fringes of 0.31 and 0.19 nm corresponding to (111) and (220) facets of CeO<sub>2</sub>, as well as lattice fringes of 0.3 nm corresponding to (110) facet of Cu<sub>2</sub>O [16], consistent with the XAS results. Further careful examination disclosed the uniform distribution of Cu<sub>2</sub>O nanodomains onto CeO<sub>2</sub> in a possibly coherent growth manner, likely owing to the lattice confinement of CeO<sub>2</sub> toward Cu<sub>2</sub>O (Fig. 5g). The STEM-HAADF and corresponding elemental mappings confirmed the even distribution of Ce, Cu, and O species on the surface of the pseudonanorod (Fig. 5h and 5i).

DFT calculations were carried out to gain in-depth insights into the electrocatalytic CO<sub>2</sub> reduction on the Cu<sub>2</sub>O (110)-CeO<sub>2</sub> (111) surface. First of all, we constructed a structural model Cu<sub>2</sub>O (110) cluster loaded onto the (111) surface of CeO<sub>2</sub> (Fig. 6a). Bader charge analysis revealed a positive charge of +0.53 e<sup>-</sup> on CeO<sub>2</sub> (111). This observation implied that electrons were transferred from Cu<sub>2</sub>O to CeO<sub>2</sub> at the Cu<sub>2</sub>O/CeO<sub>2</sub> interface, favoring the progression of the CO<sub>2</sub>RR. This electron transfer trend was further visualized through charge density differences (Fig. 6b). Subsequently, we calculated the adsorption free energy of intermediates at the Cu<sub>2</sub>O/CeO<sub>2</sub> interface and explored the selective hydrogenation steps in the catalytic CO<sub>2</sub> reduction to hydrocarbon fuels (CH<sub>4</sub>). As shown in Fig. 6c, the maximum Gibbs energies ( $\Delta G_{\max}$ ) for converting CO<sub>2</sub> to CH<sub>4</sub> on the Cu<sub>2</sub>O/CeO<sub>2</sub> was HCOO\* → H<sub>2</sub>COO\* with a

much higher free energy change of 0.65 eV, which was considered as the potential determining step (PDS).

Fig. S15 illustrates all the intermediate species involved in the process of CO<sub>2</sub> conversion to CH<sub>4</sub> for the two different reaction pathways analyzed in this study. Firstly, H was easily adsorbed on Cu at the Cu<sub>2</sub>O/CeO<sub>2</sub> interface with a  $\Delta G$  of 0.29 eV, and subsequently capturing a CO<sub>2</sub> molecule to form HCOO\* intermediate ( $\Delta G = -0.66$  eV). The formation of H<sub>2</sub>COO\* needed a  $\Delta G$  of 0.65 eV. Following this, the H<sup>+</sup>/e<sup>-</sup> pair tends to adsorb onto the unreacted O atom of CO<sub>2</sub> to generate the H<sub>2</sub>COOH\* intermediate. H<sub>2</sub>CO\* was more likely to be obtained because of the aforementioned intermediate formation. The  $\Delta G$  for the formation of CH<sub>3</sub>O\* was 0.27 eV. The hydrogenation of CH<sub>3</sub>O\* occurred on the C atom to form CH<sub>4</sub> + O\* or on the O atom to form CH<sub>3</sub>OH\* with the  $\Delta G$  of 0.73 eV and -1.08 eV, respectively. Forming CH<sub>3</sub>OH\* was more energetically favorable than CH<sub>4</sub> + O\*. CH<sub>3</sub>OH\* hydrogenated into CH<sub>3</sub>\* with the  $\Delta G$  of 0.42 eV. The final step in the hydrogenation process to form methane was an exothermic process of ( $\Delta G = -0.92$  eV). Corresponding geometric structures of intermediates along the optimal reaction pathway on Cu<sub>2</sub>O/CeO<sub>2</sub> are shown in Fig. 6d. In summary, the orange line in Fig. 6c delineates the most favorable pathway in our study, encompassing the intermediate species of H\* → HCOO\* → H<sub>2</sub>COO\* → H<sub>2</sub>COOH\* → H<sub>2</sub>CO\* → H<sub>3</sub>CO\* → H<sub>3</sub>COH\* → CH<sub>3</sub>\* → CH<sub>4</sub>.



**Fig. 6.** (a) Top view of the slab model used for the computational study of the Cu<sub>2</sub>O-CeO<sub>2</sub> (111) surface; (b) differential charge densities of the Cu<sub>2</sub>O-CeO<sub>2</sub> (111) surface. The charge density represented in blue and purple shows the positive and negative phases, respectively. The isosurface was set to 0.006 e/Å<sup>3</sup>. (c) The Gibbs free energy diagrams for the CO<sub>2</sub>RR on (a) Cu<sub>2</sub>O-CeO<sub>2</sub> at the applied potential. The blue, yellow, and red atoms are Cu, Ce, and O, respectively. The asterisk (\*) represents a surface site for adsorption, and the maximum Δ*G* of the pathways is highlighted in orange. (d) The formation path of CH<sub>4</sub> on the catalyst surface was proposed. (For interpretation of the references to colour in this figure legend, the reader is referred to the web version of this article.)

This signifies that the Cu<sub>2</sub>O/CeO<sub>2</sub> can effectively convert CO<sub>2</sub> to CH<sub>4</sub>.

#### 4. Conclusion

In summary, we introduced an ultralow amount of Cu species into the CeO<sub>2</sub> matrix to form a composite system comprising ultrafine CuO loaded on CeO for electrocatalytic CO<sub>2</sub> reduction to produce methane. A series of comprehensive investigations involving synchrotron radiation spectroscopy, spherical aberration-corrected TEM, and theoretical DFT calculations, unequivocally established that the composite system, featuring Cu<sub>2</sub>O nanodomains loaded on the surface of CeO<sub>2</sub>, undergoes reconstruction during CO<sub>2</sub>RR, which not only exposed a large number of Cu<sup>+</sup> sites with high intrinsic catalytic activity, but also effectively promoted the structural stability by the coupling and confinement effect of CeO<sub>2</sub> unit towards Cu<sub>2</sub>O. Consequently, this catalyst exhibited a notable high selectivity for CH<sub>4</sub> in a wide voltage range, and achieved a peak selectivity of 70.1 %, effectively suppressing the hydrogen evolution reaction. Moreover, when assembled as the electrode in H-cell configuration, its Faraday efficiency surpasses 70 % sustaining methane production over 20 h. This innovative approach, involving ultra-low loading of CuO on a CeO<sub>2</sub> support as a precatalyst, not only sheds light on the evolution mechanism of high-valence Cu in electrocatalysis but also provides novel insights and strategies for designing highly efficient electrocatalytic CO<sub>2</sub> reduction catalysts.

#### 5. Experimental section

A detailed description of all applied methods can be found in the [Supporting Information](#).

#### CRediT authorship contribution statement

**Xiong Yan:** Writing – original draft, Methodology, Data curation. **Shuo Wang:** Software, Data curation. **Ziliang Chen:** Validation, Conceptualization. **Yunjie Zhou:** Investigation, Data curation. **Hui Huang:** Writing – review & editing, Validation, Methodology. **Jie Wu:** Investigation. **Tiwei He:** Methodology. **Hongyuan Yang:** Data curation. **Likai Yan:** Writing – review & editing, Validation, Software. **Kaili Bao:** Methodology, Investigation. **Prashanth W. Menezes:** Writing – review & editing, Supervision, Conceptualization. **Zhenhui Kang:** Writing – review & editing, Validation, Supervision.

#### Declaration of competing interest

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

#### Data availability

Data will be made available on request.

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## Appendix A. Supplementary material

Supplementary data to this article can be found online at <https://doi.org/10.1016/j.jcis.2024.05.212>.

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