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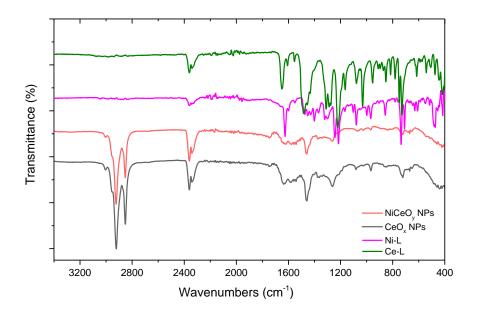


Figure S 1. FTIR spectrum of Ce (III)-L complex, Ni (II)-L complex, and as-synthesized NiCeO_y and CeO_x NPs.

Supporting information 2: XRD patterns

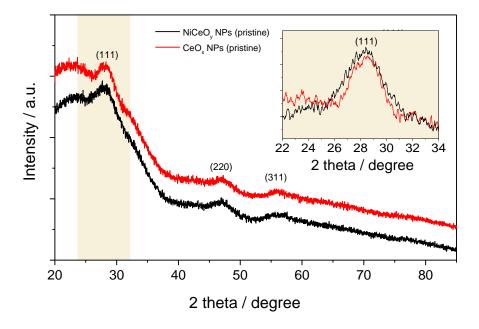


Figure S 2. XRD patterns of pristine CeO_x and NiCeO_y NPs.

Supporting information 3: TEM-EDX mapping

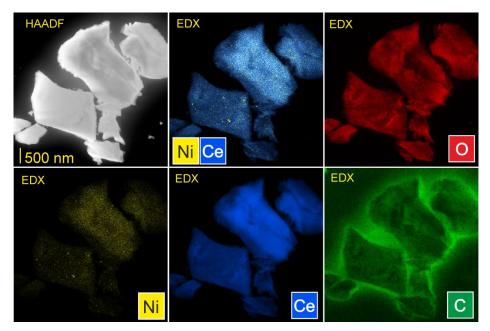


Figure S 3. HAADF-TEM and TEM-EDS elemental mapping of NiCeO $_{\rm y}$ NPs samples after calcination in O $_2$ at 400°C.

Supporting information 4: DLS results

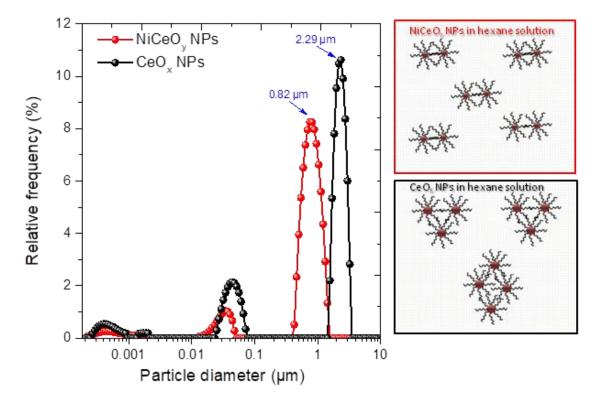


Figure S 4(a) The particle size distributions of NiCeO_y and CeO_x NPs in their organic solutions in DLS measurement. (b) Schematic representation of NiCeO_y and CeO_x NPs arrangement in their organic solutions.

Supporting information 5: Estimation of potential curves between NiCeO_y NPs and CeO_x NPs in hexane

According to DLVO theory [1–3] the total interaction forces between two particles coated by an organic protective layer V_{Total} is the sum of van der Waals attraction V_{vdW} , and total steric repulsive forces (osmotic V_{osm} and elastic V_{elas}) [1–3].

$$V_{Total} = V_{vdW} + V_{osm} + V_{elas}$$
(eq. 1)

The osmotic repulsion results from the energetic balance between solvent-ligand tail and tail-tail interactions, and the elastic repulsion originates from the entropy loss that occurs upon compression of the stabilizing ligands. The dispersion stability is essentially controlled by the osmotic term since it becomes effective as soon as the ligands start to overlap, while the elastic term does not contribute significantly to V_{Total} until the ligands are forced to compress.

The V_{vdW} can be calculated based on eq. 2:

$$V_{vdW} = -\frac{A}{6} \left[\frac{2r^2}{d^2 - 4r^2} + \frac{2r^2}{d^2} + ln\left(\frac{d^2 - 4r^2}{d^2}\right) \right] \text{ (eq. 2)}$$

A is Hamaker constant of cerium nanoparticle in the hexane $(3.5 \times 10^{-20} J)$ [4], r is nanoparticles radius, d is center-to-center separation distance between two nanoparticle cores.

According to Vincent et al. [1,2,5], V_{osm} results from the energetic balance between solvent-ligand tail and tail-tail interactions, which was estimated as following:

$$V_{osm} = \frac{4\pi r k_B T}{v_{hexane}} \phi^2 \left(\frac{1}{2} - \chi\right) \left(l - \frac{d - 2r}{2}\right)^2 \quad (l < d - 2r < 2l) \quad (eq.3)$$

$$V_{osm} = \frac{4\pi r k_B T}{v_{hexane}} \phi^2 \left(\frac{1}{2} - \chi\right) \left[l^2 \left(\frac{d - 2r}{2l} - \frac{1}{4} - ln \left(\frac{d - 2r}{2}\right)\right)\right] \quad (d - 2r < l) \quad (eq.4)$$

 k_B is Boltzmann's constant, v_{hexane} is the molecular volume of the solvent hexane, \emptyset is the volume fraction profile of the organic species surrounded nanoparticle (oleylamine) ($\emptyset = 1 - \frac{r^3}{(r+l)^3}$), l is oleylamine length ~2 nm, and χ is the Flory-Huggins interaction parameter, which is written as the follows [3]:

$$\chi = \frac{v_{hexane}}{_{RT}} \left(\delta_{hexane} - \delta_{oleylamine} \right)^2 + 0.34 \quad (eq.5)$$

Here, *R* is gas constant, T is the temperature (298 K), δ_{hexane} (14.9MPa^{1/2}) and $\delta_{oleylamine}$ (16.6MPa^{1/2}) are the solubility parameters of hexane and oleylamine, respectively. v_{hexane} is calculated to be 2.17x10⁻²⁸ m³. By using eq.5 χ parameter is calculated to be 0.49.

The elastic repulsion V_{elas} originates from the entropy loss that occurs upon compression of the organic species oleylamine, namely, in the range of d - 2r < l.

$$V_{elas} = \frac{2\pi r k_B T l^2 \emptyset \rho_{oleylamine}}{M_{oleylamine}} \left\{ x ln \left[x \left(\frac{3-x}{2} \right)^2 \right] - 6ln \left(\frac{3-x}{2} \right) + 3(1-x) \right\} \quad (d-2r < l)$$
(eq.6)

$$x = \frac{d-2r}{l} \quad (eq.7)$$

where, $\rho_{oleylamine}$ and $M_{oleylamine}$ represent the oleylamine density and molecular weight, respectively.

Table S 1. The parameters for the calculation of the potential curves of $NiCeO_y$ and CeO_x NPs in organic solution.

	NiCeO _y NPs	CeO _x NPs
r (nm) ^a	1.91	2.41
<i>l</i> (nm)	2	2
Ø	0.88	0.84

^aNanoparticle radius values r are taken from XRD results of pristine cerium-based nanoparticles

Using the above values the V_{Total} , V_{vdW} , V_{osm} , and V_{elas} can be calculated based on the eq. 1-7 as shown in Figure S4. Actually, when ligands are forced to compress (the inter-particle distance $d_s < l$, with $d_s = d - 2r$)), according to the calculated result presented in the Figure S4, the repulsions including osmotic term and elastic term have small contribution, so their effect can be neglected.

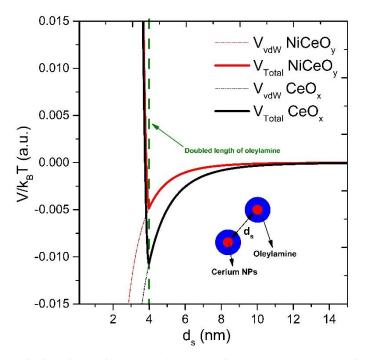


Figure S 5. The calculated total potential curves between NiCeO_y and CeO_x NPs. The inter-particle distance d_s ($d_s = d - 2r$) was defined as surface-to-surface separation distance between two nanoparticles. As shown in the figure, the repulsive forces arise at the distance, where oleylamine surrounded ceria-based nanoparticles contacts each other.

Supporting information 6: Photographs during the drying treatment

Figure S6 shows photographs of two Ni/YSZ electrodes impregnated with 50 μ L of NiCeO_y or CeO_x solution, taken at different drying stages. In the images the organic solution appears darker than the Ni/YSZ support which has a light grey color. Just after drop-casting, the support infiltrated by the NiCeO_y shows less dark areas as compared to CeO_x, indicating better spreading and pore-filling in the former case. After drying for 30 min at 80 °C, the surface of 50NiCeO_y@Ni/YSZ seems homogenous in color, while in case of 50CeO_x@Ni/YSZ spots with dark and light contrast remains even after 90 min at 80 °C. Since hexane evaporates fast at 80 °C (boiling point 68 °C) the homogenous surface color of NiCeO_y infiltrated electrode indicates that the solution was successfully entered into the pores of the Ni/YSZ electrode. On the contrary for 50CeO_x@Ni/YSZ the appearance of dark spots on the surface suggests that part of the solution remains on top of the electrode, allowing CeO_x NPs to form a layer on the surface of Ni/YSZ rather than penetrating into its pores.

Drying Time (min)	50NiCeO _y @Ni/YSZ	50CeO _x @Ni/YSZ
0	5	\$
30		
60		
90		
After calcination and reduction	1	

Figure S 6. Photographs of 50NiCeO_y@Ni/YSZ and 50CeO_x@Ni/YSZ samples during the drying treatment at 80 $^{\circ}$ C, and after calcination and reduction for 30 min in O₂ and H₂.

Supporting information 7: Laboratory XPS spectra of modified electrodes and NiCeO_y NPs

The Ni 2p and Ce 3d XPS spectra of the Ni/YSZ electrode after infiltration with NiCeO_y or CeO_x solutions and subsequent drying and calcination/reduction treatments (the spectra correspond to the samples at the bottom of figure S6). The spectrum of the reduced Ni/YSZ electrode before infiltration is also included for comparison. After calcination and reduction the Ni 2p signal increases for both electrodes, but in case of 50NiCeO_y@Ni/YSZ it is notably higher (Figure S7). Since nickel signal derives principally from Ni/YSZ while Ce 3d from ceria NPs, this difference implies that less cerium remains on the surface of 50NiCeO_y@Ni/YSZ as compared to $50CeO_2@Ni/YSZ$. Taking into account that the two impregnation solutions contain the same amount of NPs, this is indirect evidence that NiCeO_y NPs are efficiently infiltrated inside the pores of Ni/YSZ.

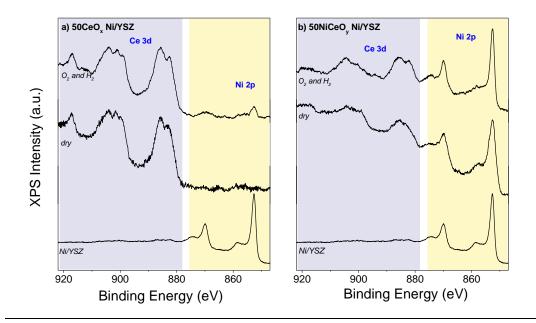


Figure S 7. The Ni 2p and Ce 3d core level XPS spectra of $50NiCeO_y@Ni/YSZ$, $50CeO_x@Ni/YSZ$ as well as pristine Ni/YSZ electrode a) after drying treatment and b) after calcination and reduction treatment in O_2 at 400 °C and H_2 at 550 °C.

The spectra below show the Ni 2p and Ce 3d XPS peak of NiCeO_y NPs measured by drop casting the hexane solution onto an Au foil. Evidently on the pristine sample the signal is highly attenuated due to the 2 nm oleylamine layer covering the NPs surface. After calcination the Ni 2p and Ce 3d peaks can be clearly distinguished however the intensity of Ni 2p is around 5 % of the overall. This indicates that when NiCeO_y NPs precipitate on the surface of Ni/YSZ the contribution of the Ni 2p signal deriving from NiCeO_y should be minor.

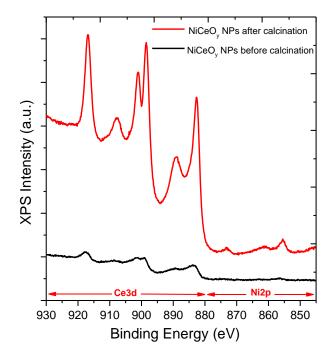


Figure S 8. Ni2p Ce3d core level XPS spectra of NiCeO_v NPs before and after calcination.

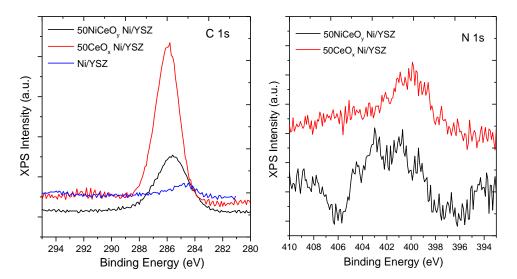


Figure S 9. C 1s and N 1s core level XPS spectra of $50NiCeO_y@Ni/YSZ$, $50CeO_x@Ni/YSZ$ and Ni/YSZ after drying treatment.

<u>Supporting information 8: Gas phase signal during calcinations of 50NiCeO_y@Ni/YSZ and 50CeO_x@Ni/YSZ samples.</u>

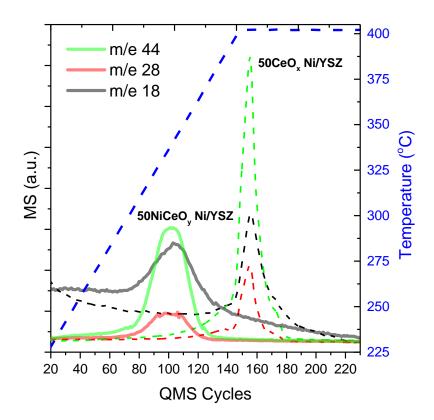


Figure S 10. Mass spectrum signal of 50NiCeO_y Ni/YSZ (full lines) and 50CeO_x Ni/YSZ (dashed lines) during calcination treatment in O_2 .

Supporting information 9: XRD of pristine and infiltrated Ni/YSZ electrodes

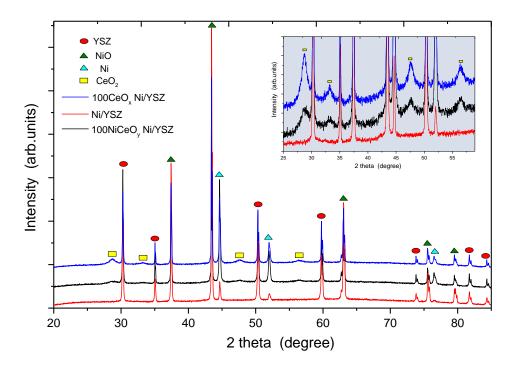


Figure S 11. XRD patterns of 100 NiCeO_y Ni/YSZ, 100CeO_x Ni/YSZ as well as Ni/YSZ after two infiltration/co-firing steps (100 μ l NiCeO_y or CeO_x solution was dropped into Ni-YSZ through two times infiltration/co-firing steps, each time for 50 μ l; Prior to the each infiltration step the electrode was reduced in H₂).

Table S 2. Crystallite size results of NiCeO_y and CeO_x NPs measured directly on the 100 NiCeO_y Ni/YSZ and 100 CeO_x Ni/YSZ samples respectively, after two infiltration/co-firing steps.

Sample	Crystallite size (nm)
100CeO _x Ni/YSZ	9.6
100NiCeO _y Ni/YSZ	6.8

Supporting information 10: Effect of the number of infiltration/co-firing steps

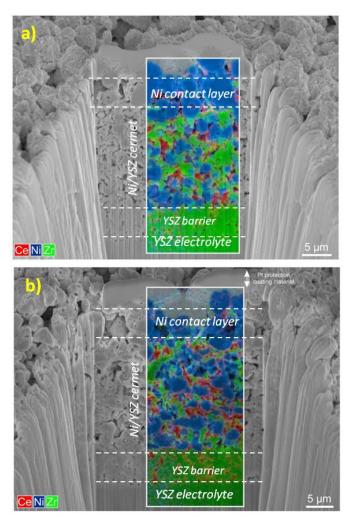


Figure S 12. Cross-sectional FIB-SEM images and the corresponding EDS elemental mapping of a) 50NiCeO_γ@Ni/YSZ prepared in 1 infiltration/co-firing step and b) 200NiCeO_γ@Ni/YSZ prepared in 4 infiltration/co-firing steps (4x50 µl each). After infiltration/co-firing step(s) the samples were reduced in 1 mbar H₂ at 550 °C for 30 min. The at.% of Ce as calculated from the EDS was 7.4±0.6 for 50NiCeO_γ@Ni/YSZ and 12.4±0.6 for 200NiCeO_γ@Ni/YSZ. Supporting information 11: XPS survey spectra of electrodes after CO₂ electrolysis tests in the <u>VPR reactor</u>

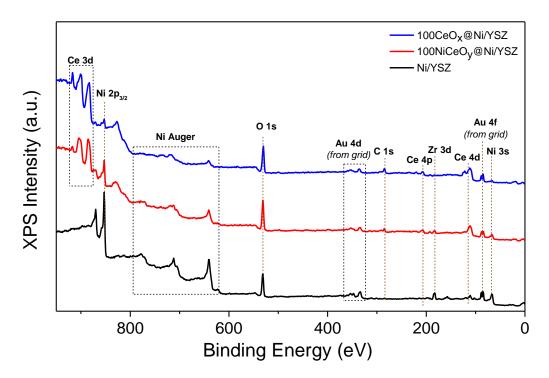


Figure S 13. XPS survey spectra of 100NiCeO_y@Ni/YSZ, 100CeO_x@Ni/YSZ and Ni/YSZ after CO_2/H_2 (96/4) electrochemical tests shown in Figure 9.

Supporting information 12: CO gas phase signal monitored by *on line* gas spectrometry analysis of the NAP-XPS chamber

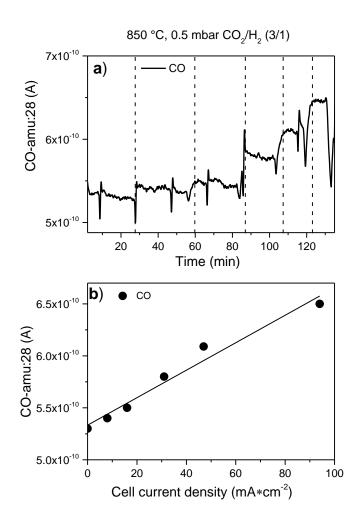


Figure S 14. The evolution CO gas phase signal as a function of **a**) time and **b**) the applied current density. The experiment was performed in the NAP-XPS chamber at 850°C in 0.5 mbar CO_2/H_2 (75/25).

Supporting information 13: NAP-XPS Ni 2p spectra of pristine Ni/YSZ electrodes under CO₂ electrolysis

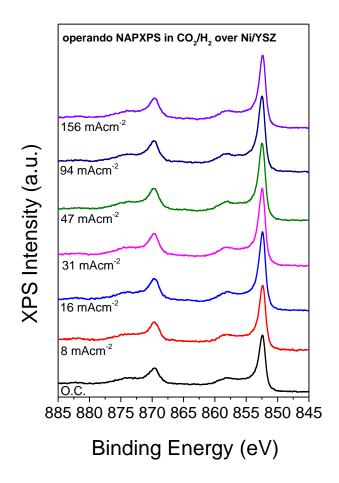


Figure S 15. NAP-XPS Ni 2p spectra of the pristine Ni/YSZ cathode recorded at 850 °C in CO_2/H_2 75/25 (0.5 mbar) under galvanostatic operation at several current densities. The spectra were recorded with hv= 1065 eV photon energy.

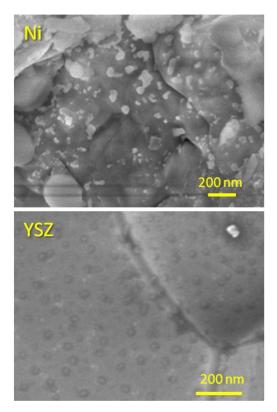


Figure S 16. SEM images of the 100NiCeO_y@Ni/YSZ electrodes, focusing in specific areas dominated by Ni (top) and YSZ (bottom) particles.

References

- [1] P.S. Shah, J.D. Holmes, K.P. Johnston, B.A. Korgel, J. Phys. Chem. B 106 (2002) 2545–2551.
- [2] T. Arita, J. Yoo, Y. Ueda, T. Adschiri, Nanoscale 2 (2010) 689–693.
- S. Usune, M. Ando, M. Kubo, T. Tsukada, K.I. Sugioka, O. Koike, R. Tatsumi, M. Fujita, S. Takami, T. Adschiri, J. Chem. Eng. Japan 51 (2018) 492–500.
- B. Faure, G. Salazar-Alvarez, A. Ahniyaz, I. Villaluenga, G. Berriozabal, Y.R. De Miguel, L. Bergström, Sci. Technol. Adv. Mater. 14 (2013).
- [5] B. Vincent, J. Edwards, S. Emmett, A. Jones, Colloids and Surfaces 18 (1986) 261–281.