Colloidal Metal Sulfide Nanoparticles for High Performance Electrochemical Energy Storage Systems

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Abstract

Transition metal sulfides have emerged as excellent replacement candidates of traditional insertion electrode materials based on their conversion or alloying mechanisms, facilitating high specific capacity and rate ability. However, parasitic reactions such as massive volume change during the discharge/charge processes, intermediate polysulfide dissolution, and passivating solid electrolyte interface formation have led to poor cyclability, hindering their feasibility and applicability in energy storage systems. Colloidal metal sulfide nanoparticles, a special class that integrates the intrinsic chemical properties of metal sulfides and their specified structural features, have fairly enlarged their contribution due to the synergistic effect. This review highlights the latest synthetic approaches based on colloidal process. Their corresponding electrochemical outcomes will also be discussed, which are thoroughly updated along with their insight scientific standpoints.

1. Introduction

Transition metal sulfides containing S^{2-}/S_2^{2-} dimers have attracted tremendous attention for electrochemical energy storage systems (EESs) because of their unique properties of high energy density, good conductivity, excellent stability, and vital catalyst functionalization feature.[1, 2] In addition, the massive resource of metal sulfides in natural mines offers scalable, low-cost, and sustainable active materials. However, the sulfides experience problems of structural degradation during metal-ion host (Li⁺, Na⁺, K⁺) up-taking and extraction due to significant volume expansion/shrinking, causing severe capacity fading phenomena. Similar to the Li-S battery, full discharge-charge processes of lithium-ion batteries based on metal sulfides also can involve intermediate polysulfide anions. However, the effect of the dissolved polysulfide shuttling on their electrochemical performances has not been clearly identified yet.[3] Many research groups have aggressively tried to synthesize metal sulfides and conductive carbonous additives composites, which can significantly improve electrochemical performances.[4-8] In the light of this direction, functional metal sulfide materials based on colloidal routes with defined components and nanostructures are considered as excellent candidates for energy storage and conversion systems. Specifically, in the past few years, extensive effort has been reported on various forms of metals sulfide nanoparticles, such as nanospheres, nano cables, nanorods, and nanowires, to maximize the charge-storage capacity and improve the cycling performance.[3] Notably, the optimized structural property design of colloidal metal sulfide nanoparticles/nanocomposites has still been pursued with the advances in synthetic techniques. Such metal sulfide nanoparticles/nanocomposites are an ideal platform for studying the effects of the components, size, and morphology of electrode materials on their electrochemical performances.[9-11]

This mini-review aims to provide a state-of-the-art overview of the metal sulfides colloidal nanoparticles, which can work as electrode materials or high-efficiency electrocatalyst hosts for alkali rechargeable battery and supercapacitor devices within the last two years. In addition, the brief introduction of up-to-date synthetic strategies for metal sulfide@C composites with their compositions, structures, and corresponding performances are also summarized and represented herein. Along with that, the discussions of insight reaction mechanisms, side-effect issues, and prospective solutions are illuminated, targeting to actualize the use of metal sulfides in EESs technologies.

2. Design and synthesis of metal sulfides colloidal particles for electrochemical energy storage

The synthesis of metal sulfides with well-defined shapes, sizes, structures, and defect states has been intensively investigated, which is the initial step towards achieving high-performance energy storage systems. The strong dependence of high electrochemical performance on nanostructures has encouraged researchers to study the controllable synthesis of metal sulfides with tailored nanostructures. Among all methods, the colloidal method is of significant interest in both fundamental research and practical applications due to its unique features.

In general, the synthesis of metal sulfides using colloidal methods follows the dissolutionrecrystallization mechanism.[12] Solvothermal/hydrothermal, precipitation and sulfidation synthesis are some of the most preferable methods because of their superiority in controlling the morphology of nanoparticles (Figure 1). Solvothermal/hydrothermal synthesis can be defined as a method for the synthesis of single crystals that depends on the solubility of minerals in hot solvent/water under high pressure. In this kind of synthesis, precursors of metal are used to allow a reaction with sulfide source



Colloidal Metal Sulfides

Figure 1. Colloidal routes for the synthesis of metal sulfide nanoparticles and their most popular applications in electrochemical energy storage systems.

materials or simply sulfur in an autoclave, where a sequence of physicochemical reactions take place under high temperature and pressure conditions. During the process, sulfide source can release H₂S when heated, which conjugates with metal ions and results in the formation of metal sulfide nanoparticles.[13, 14] The prepared metal sulfides nanoparticles are usually post-annealed at hightemperature under an inert atmosphere to increase their crystalline quality and purity. Figure 2a-c illustrates the preparation of MoS₂/amorphous carbon (C) microtubes (MTs) by Pan *et al.* via hydrothermal method using Sb₂S₃ microrod as a sacrificial template. [10] Na₂S·9H₂O and N₂H₄CS were used as sulfide source for Sb₂S₃ and MoS₂, respectively. The produced MTs consist of heterostructured MoS₂/C nanosheets, which were applied as cathode material for high performance Na-ion batteries.

Besides the mono-metal sulfides, hydrothermal/solvothermal method can also be used to synthesize bimetal sulfides or multi-metal sulfides. For example, Tang *et al.*[15] takes advantage of the synergistic effect of hydrothermal method to synthesize Ni₂CoS₄/graphene oxide nanoparticles using CoCl₂ • $6H_2O$ and NiCl₂ • $6H_2O$ as the metal precursor and Na₂S • $9H_2O$ as the sulfur precursor. The synthesis was taken in the Teflon-lined stainless steel autoclave at 120 ° C for 8 h.

Precipitation is another widely used method to synthesize metal sulfides nanoparticles in recent years. It is the creation of nanostructured materials as a powdered solid from a solution and allows the selection of the reaction conditions, where the nanostructured materials precipitate as a powdered solid from a



Figure 2. (a) Schematic illustration of the synthesis process and (b, c) SEM images of the MoS₂/C microtubes. Reproduced from ref. [13] with permission from American Chemical Society. *Copyright* \bigcirc 2018. (d) Synthesis procedure and (e, f) TEM images of the hollow MoS₃ nanospheres synthesized by acid precipitation method. Reproduced from ref. [19]. (g) Synthesis procedure of the FeS₂-C nanospindles, (h) SEM image of the FeOOH nanospindles, and (i) TEM image of the FeS₂-C nanospindles. Reproduced from ref. [4]

solution. In the case of precipitation, which usually undergoes at room temperature, nano-sized crystals can be prepared in a highly pure and easily separate form. The common precipitates used to prepare metal sulfides are HCl, KOH, NaOH, ammonia, urea and tetramethylammonium hydroxide. In addition to the precipitants, some reductive and oxidative agents are often added to form metal sulfides.[16] In the final process, the precipitated metal sulfides can be heat treated by calcination or annealing to form the final product. A typical example of it is the synthesis of MoS₃ nanoparticles. As an alternative to MoS₂, MoS₃ has recently been widely explored as the electrode material for energy storage systems due to its unique one-dimensional chain structure and high electronic conductivity (bandgap of <1.5 eV).[17, 18] Quan *et al.*[19] reported the synthesis of hollow MoS₃ nanospheres using spherical polyelectrolyte brushes as soft template (Figure 2d-f), which follows the acid precipitation method with the reaction as: $MoS_4^{2-} + 2 H^+ = MoS_3 \downarrow + H_2S \uparrow$, and the precipitated materials were heat-treated at 200 °C to form the final product.

Traditionally, metal sulfides can also be synthesized by the transformation from metal oxides. However, the obtained nanoparticles usually suffer from poor energy storage behavior due to the crucial transformation conditions. Recently, the sulfidation process and other methods have been widely used to convert metal oxides or hydroxides to metal sulfides, allowing the surface modification and morphology control of the final product. During the sulfidation process, sulfur powder is often used as the sulfur source. For instance, Xie et al.[4] prepared yolk-shell structured iron sulfide nanospindle (Figure 2g-i), where FeOOH acted as the iron source and provided the nanospindle structure. FeOOH was transformed to Fe_3O_4 via the first calcination under argon and then to FeS_2 during the second calcination process with sulfur powder at 400 °C. This route uses the hard templates in aqueous solutions and provides a good way to control the morphology of the final product, which can only be realized by colloidal methods. The combination of hydrolysis and sulfidation methods provides the effective route to synthesize metal sulfides with uniform size, morphology and even good colloidal stability. Besides the sulfur powder, the direct addition of H_2S or a combination of H_2S during the heating treatment can also be used for the sulfidation synthesis of metal sulfides. The former one is usually used in synthetic routes where sulfur is absent in precursors, while the other one is for the precursors, which can act as both metal and sulfur precursors. Although the reactivity of H₂S is advantageous, it also presents numerous challenges to work with. To ensure safety, single - droplet combustion experiments will have to be housed in a containment unit to be able to study these reactions.

Comparing different synthesis approaches, hydrothermal/solvothermal method is one of the best techniques to produce highly pure and nanostructured materials. It is much easier to control the properties of the synthesized materials by using different pressure and temperature conditions. However, the high temperature and pressure conditions make it very challenging to scale up the reactions. The precipitation technique is usually performed at room temperature and atmospheric pressure, which is easy to scale up the synthesis. But the surface modifications of the metal sulfides, such as carbon coating, cannot be realized at the same time by this method. Compared to them, sulfidation method is a unique technique as it transforms from the inactive metal oxide phase to an active sulfide phase, but the production process usually involves several steps, making the synthesis procedure complicated.

Up to now, a number of colloidal methods have been established to synthesize metal sulfides nanomaterials with high yield. However, some reaction mechanisms underlying the growth of different nanostructured metal sulfides, for example, nanobox, nanosphere, nanotube, remain unclear. Thus, the investigation of the reaction mechanisms behind the crystal growth route in order to search for suitable methods for different structures is of interest for future research. In addition, surface modification has been reported to enhance the properties of metal sulfides. There is still a high demand to develop facile synthesis approaches for the surface-functionalized metal sulfides with controlled size and morphology.

Table 1: Selected transition metal sulfides with detailed information on synthetic methods, morphology, and their electrochemical performances in various
battery systems.

Compound and morphology	Synthetic method	Type of batter v	Particle size	Electrode formulation/ mass loading	Performance	Rate capability (Capacity/ Current density)	Ref.
Co _{1-x} S nanofibers	Electrostatic spinning and solvothermal	LIB	5–10 nm	Freestanding	\sim 300 mAh g ⁻¹ at 2.0 A g ⁻¹ , 93% after 2000 cycles	527 mAh g^{-1} / 0.5 A g^{-1} 470 mAh g^{-1} / 1.0 A g^{-1}	[20]
Co _{1-x} S hollow spheres	Solvothermal	LIB	~500 nm	80:10:10 (active material: carbon black: PVDF*)/ ~0.8 mg cm ⁻²	~538 mAh g ⁻¹ at 2.5 A g ⁻¹ , 67% after 100 cycles	572.5 mAh.g ⁻¹ / 1.0 A g ⁻¹ 488.0 mAh g ⁻¹ / 2.0 A g ⁻¹ 371.2 mAh g ⁻¹ / 5.0 A g ⁻¹ 279.2 mAh g ⁻¹ / 7.5 A g ⁻¹	[6]
MoS ₃ Hollow nanosphere	Acid precipitation	LIB	~100 nm	70:20:10 (active material: acetylene black: PTFE*) / 2.0 mg cm ⁻²	Full cell: ~89 mAh g^{-1} at 1.0 A g^{-1} , 75% after 100 cycles	80 mAh g^{-1} / 0.5 A g^{-1} 49.7 mAh g^{-1} / 2.0 A g^{-1}	[19]
NiS@rGO nanoparticle	Hydrothermal	LIB	15~25 nm	70:15:15 (active material: carbon black: CMC/SBR*)	~1328.7 mAh g ⁻¹ at 0.1 A g ⁻¹ , 75% after 100 cycles	835.8 mAh $g^{-1}/0.5 \text{ A } g^{-1}$ 752.1 mAh $g^{-1}/1.0 \text{ A } g^{-1}$ 673.6 mAh $g^{-1}/2.0 \text{ A } g^{-1}$ 538.1 mAh $g^{-1}/5.0 \text{ A } g^{-1}$	[21]
FeS and FeS ₂ microsphere	Gas-phase vulcanization approach	LIB/S IB	400- 800 nm	80:10:10 (active material: CNT: PVDF*)/ 1.0 mg cm ⁻²	$\begin{array}{c} FeS_2@3DGF: \sim 889 \mbox{ mAh } g^{-1} \mbox{ at } 0.2 \\ A \ g^{-1}, \ 50\% \mbox{ after } 140 \ cycles \\ FeS@3DGF: \sim 1109 \mbox{ mAh } g^{-1} \mbox{ at } \\ 0.2 \ A \ g^{-1}, \ 74\% \mbox{ after } 200 \ cycles \\ \end{array}$	FeS ₂ @3DGF: 479 mAh g^{-1} / 2.0 A g^{-1} 273 mAh g^{-1} / 5.0 A g^{-1}	[22]
$Fe_7S_8@, Co_9S_8@, Ni_9S_8@$	Electro- blowing spinning and	LIB	~ 50 nm	Freestanding	Fe ₇ S ₈ @MPCNF: \sim 546 mAh g ⁻¹ at 1.0 A g ⁻¹ , 91% after 500 cycles	$\label{eq:second} \begin{array}{c} Fe_7S_8@MPCNF: 643 \mbox{ mAh } g^{-1}/1.0 \mbox{ A } g^{-1} \\ 570 \mbox{ mAh } g^{-1}/ \ 2.0 \mbox{ A } g^{-1} \end{array}$	[7]

macro-porous	one-step sulfurization	one-step furization			Co ₉ S ₈ @MPCNF: ~654 mAh g ⁻¹ at 0.2 A g ⁻¹ , 87.7% after 200 cycles Ni ₉ S ₈ @MPCNF: ~459 mAh g ⁻¹ at 0.2 A g ⁻¹ 80.1% after 200 cycles	Co ₉ S ₈ @MPCNF: 801 mAh $g^{-1}/0.1 \text{ A } g^{-1}$	
carbon nanofiber						536 mAh $g^{-1}/2.0 \text{ A } g^{-1}$	
						$Ni_9S_8@MPCNF:614 mAh g^{-1}/ 0.1 A g^{-1}$	
						$392 \text{ mAh } \text{g}^{-1}/ 2.0 \text{ A } \text{g}^{-1}$	
Fe ₃ S ₄ /Co ₉ S ₈	Hydrothermal	LIB	600 nm	70: 20: 10 (active	~945 mAh g ⁻¹ at 0.1 A g ⁻¹ , 71%	698 mAh $g^{-1}/0.5 \ A \ g^{-1}$	[23]
Hierarchical				material: Super P:	after 100 cycles	643 mAh g^{-1} / 1.0 A g^{-1}	
nanoparticles				sourum arginate)		574 mAh g^{-1} / 2.0 A g^{-1}	
Fe ₃ O ₄ /Fe ₇ S ₈ @	Rapid	LIB	100-	75:15:10 (active	~945 mAh g^{-1} at 0.1 A g^{-1} , 71%	825 mAh $g^{-1}/0.5 A g^{-1}$	[24]
C	rheological phase method	heological	2000 nm	0 material: acetylene black: PVDF*) / 1.2 mg cm ⁻²	after 100 cycles	778 mAh g^{-1} / 1.0 A g^{-1}	
nunopurtieres						717 mAh $g^{-1}/2.0 \ A \ g^{-1}$	
Fe ₃ O ₄ /Fe ₇ S ₈ /C	Hydrothermal	LIB	5–20	70:20:10 (active	~514 mAh g^{-1} at 0.5 A g^{-1} , 93%	511 mAh g^{-1} / 1.0 A g^{-1}	[25]
nanoplates			nm	material: Super P:	after 350 cycles	$358 \text{ mAh } \text{g}^{-1} / 5.0 \text{ A } \text{g}^{-1}$	
				1.5 mg cm^{-2}		283 mAh g^{-1} / 10.0 A g^{-1}	
CoS ₂ -MnS	Hydrothermal	LIB	~25 nm	70:20:10 (active	\sim 1324 mAh g ⁻¹ at 0.1 A g ⁻¹ , 88%	1130 mAh $g^{-1}/0.5 \text{ A } g^{-1}$	[26]
nanoparticles composites	, and sulfidation			material: acetylene black: PVDF*)	after 100 cycles	927 mAh $g^{-1}/1.0 \ A \ g^{-1}$	
MaS-@SpS	Undrothormal	I ID	5 nm	80.10.10 (active	$821 \text{ mAh } a^{-1} \text{ at } 2.0 \text{ A } a^{-1} 0.02\%$	$0.47 \text{ mAb } a^{-1}/0.5 \text{ A } a^{-1}$	[16]
anchored 3D	nyurotnermai	LID	5 1111	material: carbon	after 1000 cycles	747 mAn g = 70.3 A g	
carbon				black: CMC*) /		$703 \text{ mAn g}^{-7} 1.0 \text{ A g}^{-7}$	
skeleton				$1.0-1.2 \text{ mg cm}^{-2}$		591 mAh g ¹ / 2.0 A g ¹	
FeS@S and	Hydrolysis	LSB	20-30	80:10: 10 (active	FeS ₂ -C@S: \sim 761 mAh g ⁻¹ at 0.5	900 mAh $g^{-1}/0.2 \text{ A } g^{-1}$	[4]
shelled	sulfurization	urization	11111	black: PVDF*) /	$E_{0} = 0.7 \text{ / 0 arter 550 cycles}$	825 mAh $g^{-1}/0.5 A g^{-1}$	
nanospindle				$1.0-1.1 \text{ mg cm}^{-2}$	63.9 % after 350 cycles	750 mAh g^{-1} / 1.0 A g^{-1}	

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$1T MoS_2@S$	Solvothermal	LSB	200-	64:26:10 (active	$MoS_2@S: ~900 mAh g^{-1} at 0.2$	$MoS_2@S: 1000 \text{ mAh } g^{-1}/2.0 \text{ mA.cm}^{-2}$	[27]
and $WS_2(a)S$			300 nm	material:	mA.cm ⁻² , 72% after 100 cycles	$WS_2@S:743 \text{ mAh } g^{-1}/2.0 \text{ mA.cm}^{-2}$	
nanomoer				PVDF*)	$WS_2@S: \sim 880 \text{ mAh } g^{-1} \text{ at } 0.2$		
				IVDI)	mA.cm ^{-2} , 72% after 100 cycles		
FeS/N-C@S	Solution	LSB	500 nm	80:10:10 (active	~729 mAh g ⁻¹ at 0.5 C, 86.7 %	1149 mAh g^{-1} / 167.5 mA g^{-1}	[28]
nanocomposite	precipitation			material: Super P:	after 500 cycles	729 mAh $g^{-1}/1675$ mA g^{-1}	
	calcination			1.7 mg cm^{-2}		570 mAh $g^{-1}/3350$ mA g^{-1}	
						$335 \text{ mAh g}^{-1}/5025 \text{ mA g}^{-1}$	
	hydrothermol	SID	300	70.15.15 (active	$403 \text{ mAh} \alpha^{-1} \text{ at } 1.0 \text{ A} \alpha^{-1} 81.94$	$\frac{180.4 \text{ mAb } \text{g}^{-1}}{10.4 \text{ g}^{-1}}$	[12]
MoS ₂ /C	nydrotnermar	SID	400 nm	material: acetylene	after 500 cvcles	469.4 mAn g / 1.0 A g	[13]
tubular heterostructure				black: CMC*)		$452.9 \text{ mAh } \text{g}^{-1} / 2.0 \text{ A } \text{g}^{-1}$	
						425.1 mAh g^{-1} / 5.0 A g^{-1}	
						401.3 mAh g^{-1} / 10.0 A g^{-1}	
Fe ₇ S ₈	solvothermal	SIB	<20 nm	70:15:15 (active	~450.8 mAh g ⁻¹ at 0.2 A g ⁻¹ , 66 %	419.8 mAh $g^{-1}/0.4 A g^{-1}$	[29]
nanospindle				material: Super P: PVDF*)	after 500 cycles	$387.0 \text{ mAh } \text{g}^{-1} / 0.8 \text{ A } \text{g}^{-1}$	
						$353.4 \text{ mAh } \text{g}^{-1}$ / 1.6 A g ⁻¹	
						$327.8 \text{ mAh } \text{g}^{-1}/ 3.2 \text{ A } \text{g}^{-1}$	
NiS _x 3D	hydrothermal	SIB	1-20	80:10:10 (active	~300 mAh g ⁻¹ at 1.0 A g ⁻¹ , 58.6 %	520 mAh g^{-1} / 2.0 A g^{-1}	[8]
matrix		nm	nm	material: Super P: CMC*)/ ~0.5 mg cm ⁻²	after 800 cycles	410 mAh g^{-1} / 5.0 A g^{-1}	
						$310 \text{ mAh } \text{g}^{-1}$ / $10.0 \text{ A } \text{g}^{-1}$	
						230 mAh g^{-1} / 15.0 A g^{-1}	
NiS Film	Silica-	NIB	~150	hybrid films	~407 mAh g ⁻¹ at 0.1 A g ⁻¹ , 58.6 %	$350 \text{ mAh } \text{g}^{-1}$ / 1.0 A g $^{-1}$	[30]
composed of	mediated	liated version	nm		after 150 cycles	$300 \text{ mAh } \text{g}^{-1} / 2.0 \text{ A } \text{g}^{-1}$	
hanosheets	process					290 mAh g^{-1} / 3.0 A g^{-1}	
			1				1

CoS2 Multi- shelled Nanoboxes	Ion- conversion- exchange strategy	NIB	~800 nm	70:20:10 (active material: Super P: CMC*)/ 1.0 mg cm ⁻²	~509 mAh g ⁻¹ at 0.2 A g ⁻¹ , 95 % after 100 cycles	$\begin{array}{c} 422 \text{ mAh } \mathrm{g}^{-1} / \ 1.0 \ \mathrm{A} \ \mathrm{g}^{-1} \\ 395 \ \mathrm{mAh} \ \mathrm{g}^{-1} / \ 2.0 \ \mathrm{A} \ \mathrm{g}^{-1} \\ 346 \ \mathrm{mAh} \ \mathrm{g}^{-1} / \ 5.0 \mathrm{A} \ \mathrm{g}^{-1} \end{array}$	[31]
Cu ₂ MoS ₄ Hollow Nanospheres	Solvothermal	NIB	1-2 μm	80:10:10 (active material: carbon black: CMC*)/ ~1.0 mg cm ⁻²	~205.7 mAh g ⁻¹ at 0.5 A g ⁻¹ , 95.6 % after 2000 cycles	259.2 mAh g^{-1} / 0.05A g^{-1} 40.3 mAh g^{-1} / 5.0 A g^{-1}	[32]
SnS- SnSe2@3DC heterostructure s	Spray-drying	NIB	200 nm	80:10 :10 (active material: carbon black: CMC*)/ ~ 0.8 mg cm ⁻²	580/100/100 ~580 mAh g ⁻¹ at 0.1 A g ⁻¹ , 90.2 % after 100 cycles	$\begin{array}{c} 420 \text{ mAh } \mathrm{g}^{-1} / \ 1.0 \text{ A } \mathrm{g}^{-1} \\ 328 \text{ mAh } \mathrm{g}^{-1} / \ 5.0 \text{ A } \mathrm{g}^{-1} \\ 267 \text{ mAh } \mathrm{g}^{-1} / \ 10.0 \text{ A } \mathrm{g}^{-1} \end{array}$	[33]
SnS2@rGO Nanosheets	Solution precipitation	KIB	~ 100 nm	80:10:10 (active material: Super P: PAA*)/ 1.0–2.0 mg cm ⁻²	$\sim 387 \text{ mAh g}^{-1} \text{ at } 0.05 \text{ A g}^{-1}, 87\%$ after 100 cycles	$\begin{array}{c} 355 \text{ mAh } \mathrm{g}^{-1} / \ 0.2 \text{ A } \mathrm{g}^{-1} \\ 289 \text{ mAh } \mathrm{g}^{-1} / \ 0.5 \text{ A } \mathrm{g}^{-1} \\ 247 \text{ mAh } \mathrm{g}^{-1} / \ 1.0 \text{ A } \mathrm{g}^{-1} \end{array}$	[34]
NiCo _{2.5} S ₄ @rG O microrods	Sulfurization process	KIB	~ 4 nm	80:10:10 (active material: Super P: CMC*)/ 1.0 mg cm ⁻²	~495 mAh g ⁻¹ at 0.2 A g ⁻¹ , ~82% after 1900 cycles	$507 \text{ mAh } \text{g}^{-1} / \text{ 0.2 A } \text{g}^{-1}$ $472 \text{ mAh } \text{g}^{-1} / \text{ 0.5 A } \text{g}^{-1}$ $442 \text{ mAh } \text{g}^{-1} / \text{ 1.0 A } \text{g}^{-1}$	[35]
ZnS dendrites embedded nanorods	Solvothermal and sulfurization	KIB	~ 120 nm	90:10 (active material: PVDF*)/ 0.8-1.2 mg cm ⁻²	~330 mAh g ⁻¹ at 0.05 A g ⁻¹ , ~32% after 100 cycles	295 mAh $g^{-1}/0.1 \text{ A } g^{-1}$ 223 mAh $g^{-1}/0.2 \text{ A } g^{-1}$ 162 mAh $g^{-1}/0.5 \text{ A } g^{-1}$	[36]
CuS@GO Nanosheet	Precipitation	KIB	~ 20– 40 nm	80:10:10 (active material: carbon black: PVDF*)/ 0.8 mg cm ⁻²	~290.5 mAh g ⁻¹ at 0.1 A g ⁻¹ , ~71% after 100 cycles	291.4 mAh $g^{-1}/0.3$ A g^{-1} 256.2 mAh $g^{-1}/0.5$ A g^{-1} 196.5 mAh $g^{-1}/1.0$ A g^{-1}	[37]

Note: Lithium-ion battery (LIB), Lithium-Sulfur battery (LSB), Sodium-ion battery (NIB), Potassium-ion battery (KIB)

PVDF: poly(vinylidene fluoride); PTFE: poly(tetra fluoroethylene); CMC: carboxymethyl cellulose; SBR: Styrene-Butadiene; PAA: polyacrylic acid

Compound and morphology	Synthetic method	Particle size	Specific Surface area	Electrode formulation/ mass loading	Performance	Ref.
VS ₄ 3D sandwich structure	hydrothermal	50-70 nm	1890.1 m ² /g	80:10:10 (active material: carbon black: PVDF*)/ 4 mg cm ⁻²	497 F/g at 1 A/g, 64.7% after 5000 cycles at 1 A/g	[38]
CoS Flower-like hierarchitectures	hydrothermal	~2 µm	423.4 m ² /g	80:10:10 (active material: acetylene black: PTFE*) / 2 mg cm ⁻²	697 F/g at 1 A/g, 85.9% after 10000 cycles at 10 A/g	[39]
NiCoS nanoplate	co-precipitation and in situ sulfuration process	nanometers thickness	114.9 m ² /g	70:20:10 (active material: acetylene black: PVDF*)/2.1 mg cm ⁻²	758.9 F/g at 1 A/g, 63.6% after 10000 cycles at 10 A/g	[40]
Zn-Co-S microsphere	hydrothermal	$\sim 8 \ \mu m$	$106.8 m^2/g$	Binder free/ 4.2 mg cm ⁻²	~149 mAh/g at 1 A/g, 81.4% after 6000 cycles	[41]
2D SnS ₂ Nanorods	hydrothermal	an average length of about ~2-4 μm with a diameter of 20-60 nm	72.4 m ² /g	80:10:10 (active material: carbon black: PTFE*)/ 2 mg cm ⁻²	165.2 F/g at 3 A/g, 93% after 4000 cycles at 100 mV/s	[42]
Mesoporous NiCoMn-S	solvothermal	~100 nm	41.4 m²/g	80:10:10 (active material: acetylene black: PTFE*)/ 4 mg cm ⁻²	661 F/g at 1 A/g, 86.5% after 1000 cycles	[43]

Table 2. Selected transition metal sulfides on their synthetic methods, morphology, and electrochemical performances in supercapacitor application

*: PVDF: poly(vinylidene fluoride) PTFE: poly(tetra fluoroethylene)

3. Transition metal sulfide colloidal particles in electrochemical energy storage systems 3.1. Colloidal metal sulfides for high-performance metal-ion batteries

On the roadmap of developing better energy/power densities and excellent rate capability EESs, metal sulfides coupling with various doping compounds leads to novel nano-sized morphologies, significantly boosting the alkali metal host (Li⁺/Na⁺/K⁺) diffusion sites and electronic conductivity.[44] Table 1 represents the updated works utilizing metal sulfides derived by various precursors and chemical synthesis processes. The details of nanoparticle size, electrode preparation, and their electrochemical outcomes are also included. It is worth pointing out that most systems could operate at high current rates, delivering excellent specific capacity and stability. These promising results are facilitated by the unique layered structure of the systems such as MoS₂, WS₂, ZrS₂, VS₄, or SnS₂, containing the covalent bond between two-dimensional sandwiched S-Metal-S layers and held by the weak Van der Waals force.[45-47] The metal host ions can, therefore, be easily accommodated during the insertion process.[45] For example, molybdenum oxide-based materials (MoS₂ or MoS₃) are considered as one of the most promising candidates owning to their interesting electrochemical properties. It has been noted that the morphologies and particle size of them have significantly influenced the performances. High surface area MoS₃ hollow spheres facilitate ideal ion diffusion pathway and large electroactive sites suitable for battery/supercapacitor hybrid systems, [19] while MoS₂ anchored 3D carbon skeleton or tubular heterostructure provide robust insertion/exertion channels for long cycle life and high stability.[13, 16] In fact, the formation of host up-taking products Li/Na/K_x-MS_y is generally unstable, and they quickly decompose to metallic Li/Na/K and polysulfide species. These mechanisms lead to the significant expansion of layers, especially in the bulk-phase pristine metal sulfide particles.

Therefore, various structure/shapes (plate, sphere, mulberry, fiber, etc.) modification in nanoscale and proper doping/compositing with carbonaceous compounds could remarkably improve and reinforce diffusion pathways, which strongly enhances their electronic/ionic conductivities and structural stability of the composite, delivering superior high-rate performances.[25] Moreover, these new synthetic approaches could also reinforce interface kinetics during electrochemical processes, supporting remarkable rate and stable cyclability.[22] For instance, Fe₃O₄/Fe₇S₈/C nanoplates,[25] MoS₂ /C nanosheets,[13] or NiS_x 3D matrix[8] can operate at >10 A g⁻¹ current densities, retaining high discharge capacity in lithium, sodium, and potassium -ion batteries. Moreover, several studies created freestanding or self-supporting films with carbon nanofibers by colloidal physical/chemical processes.[7, 20] These composites demonstrate encouraging capacity retention after long cycling due to solid binding between nano-sized metal sulfide and carbon matrix. The composite/hybrid systems with polysulfide absorbers are also considered a promising approach to utilize each component feature. For example, Fe₃S₄/Co₉S₈,[23] Fe₃O₄/Fe₇S₈@C,[25] CoS₂-MnS,[26] and MoS₂@SnS[16] hybrid systems have achieved superior cycling ability and capacity retention.

Noted that the irreversible transformation of polysulfide and metallic compounds derives a new reaction of $A_2S + M/A_x \leftrightarrow S + M + A_{x+2}$ (A is alkali metal, M is transition metal) upon the subsequence cycles.[48] Unfortunately, these polysulfides are soluble in liquid electrolyte mediums, causing mass loss phenomena in the active electrode side and passivating layer formation in the reference electrode surface. Although recent reports have revealed stunning results of colloidal metal sulfide particles in EESs, effective strategies to compensate for these issues should be further addressed. Modifying the solid electrolyte interface (SEI) by different electrolytes, which could facilitate high compatibility to sulfide systems, may play a critical role in maintaining high cycle ability.

3.2. Catalytic metal sulfide for Metal-S batteries

Utilizing elemental sulfur as cathode and active metal (e.g., Li, Na, Mg, and Al) as anode, the rechargeable metal-sulfur batteries (RMSBs) enable a leap in specific energy owing to the high

theoretical specific capacity of both the metallic anode and the sulfur cathode.[49, 50] However, the shuttle effect led from highly soluble polysulfide intermediates and the sluggish kinetics of polysulfide conversion hinder the pragmatic application of RMSBs via rendering unstable cycling performances.[51] In this regard, metal sulfide nanoparticles designed and generated from colloidal routes stand out as efficient sulfur host due to the following advantages: (i) the sulfurphilic surface of metal sulfides enables a strong, but not exclusive binding to polysulfides, which can suppress the shuttle effect and preserve the intact configuration of polysulfides; (ii) metal sulfides show higher electronic conductivities than their oxide counterparts, some of which even exhibit half-metallic or metallic bonding; (iii) metal sulfides can serve as a catalyst for the conversion of polysulfides, which remarkably accelerates the redox reaction kinetics. Specifically, metal sulfide nanoparticles are far less limited by their ionic or electronic conductivities when compared to the bulk materials due to the very short internal diffusion paths. They can also withstand much greater mechanical deformation during charge/discharge processes, leading to prolonged cycling life.

Overall, these favorable effects significantly enlarge the variety of metal sulfide nanomaterials for RMSBs applications, especially in cathode development. Some representative works are listed in Table 1.[4, 22, 27, 28] In this direction, the strong adsorption of polysulfide intermediate is perceived as the major function of metal sulfide nanomaterials due to their sulfurphilic properties. For instance, Cui *et al.* systematically investigated the polysulfide adsorption capabilities of 7 candidate metal sulfides, revealing that TiS₂, FeS and MoS₂ exhibit greater polysulfide adsorption capabilities than Cu₂S, CuS, TiN, ZnS and CoS.[52] In addition, the catalyzing oxidation capability of metal sulfides is critical in reducing the energy barrier, leading to accelerated surface-mediated redox reaction and controlled Li₂S precipitation, therefore, contributes to the remarkably improved battery performance.[53] Particularly, the metallic 1T (octahedral) phase of metal sulfides exhibits a high electronic conductivity that is six orders of magnitude higher than that of 2H (hexagonal) phase, which can further catalyze the polysulfide conversion reactions.[54] In addition to the intrinsic advantages of metal sulfides, creating "sulfur deficiencies" is an innovative approach to suppress the shuttle effect.[55, 56] Taking MoS_{2-x} as an example, the sulfur deficiency results in larger charge densities on MoS_{2-x} (001) surface, finally promotes the corresponding thermodynamic and kinetic processes in the polysulfide conversion.[57]

Nevertheless, current studies of metal sulfide nanoparticles for RMSBs mostly focus on monovalent Li-S, K-S and Na-S batteries. The multivalent metal-sulfur systems, Mg-S, Al-S, and Ca-S, offer promising energy storage capabilities and better safety features as well. The research and development on these batteries is far behind the Li–S system due to many critical challenges. Future efforts and directions for both the fundamental and practical research are in great demand.

3.2. Supercapacitor

Metal sulfides, including layered and nonlayered types, are expected to meet the requirements for supercapacitors due to their rich redox reactivity than metal oxide counterparts.[58] The layered metal sulfides, such as MoS₂ and WS₂, are composed of three atom layers (S-M-S) bound by covalent bonds where the individual layers are held together by van der Waals forces, while the nonlayered ones, such as FeS₂ and ZnS, have chemical covalent bonds in three dimensions.[12] Both two types have attracted intensive research for supercapacitors in recent years due to their outstanding properties, such as good electrical conductivity, low electronegativity, high specific capacitance, high redox activity, and unique crystal structures.[59-61] The layered metal sulfides nanomaterials usually have high specific surface area due to their unique crystal structures, resulting in the high electrical double layer capacitance. Besides it, there is always intercalation followed by conversion process, which leads to high specific capacitance while undesirable cycling performance. Compared to it, only conversion process exists in nonlayered metal sulfides nanomaterials. For both of them, the relatively low electrical conductivity

and the large volume change during cycling are the most crucial problems that restrict their commercial use. Based on it, the combination of metal sulfides with conductive materials, or the growth of metal sulfides on conductive materials (carbon matrix) is an effective way to resolve the problems and has drawn great attention. For example, Maron *et al.*[60] fabricated a reduced graphene oxide/ multi-walled carbon nanotube-ZnS (rGO/MWCNTf/ZnS) hybrid 3D nanocomposite for supercapacitors. It was found that the addition of rGO improved the conductivity of the material, thus achieved high capacitance retention of up to 85% after 4000 cycles in symmetric two-electrode supercapacitors with 6 M KOH electrolyte.

Despite the efforts, the electrochemical performances of the combination of individual metal sulfides with carbon are still not satisfactory, which is ascribed to their structural damage in the electrochemical procedures resulting in fast capacitance fading. Therefore, many researchers are studying the bi- or multi-metal sulfides for supercapacitors. The synergetic effects between individual components will offset their respective drawbacks.

4. Conclusion and outlook

In summary, we provide an overview of the recent developments on the rational design and synthesis of metal sulfide nanoparticles as high-performance electrode materials for EESs. Great progress has been achieved in developing various types of metal sulfides from colloidal processes, forming desired compositions, morphologies, and properties. The advantageous effects of metal sulfide nanoparticles for different EESs involving metal-ion batteries, metal sulfur batteries, and supercapacitors are comprehensively discussed. Despite the encouraging progress in applying colloidal metal sulfides for advanced EESs, many challenges still exist due to the underlying complex electrochemical mechanism, which is still unclear and needs to be optimized. With continuous studies on metal sulfide nanoparticles and growing interest from battery communities, we anticipate that metal sulfide nanoparticles will play a vital role in energy storage applications in view of the unique structural and compositional features.

Declaration of interest: none

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