

Roadmap for focused ion beam technologies

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ABSTRACT

The focused ion beam (FIB) is a powerful tool for fabrication, modification, and characterization of materials down to the nanoscale. Starting with the gallium FIB, which was originally intended for photomask repair in the semiconductor industry, there are now many different types

of FIB that are commercially available. These instruments use a range of ion species and are applied broadly in materials science, physics, chemistry, biology, medicine, and even archaeology. The goal of this roadmap is to provide an overview of FIB instrumentation, theory, techniques, and applications. By viewing FIB developments through the lens of various research communities, we aim to identify future pathways for ion source and instrumentation development, as well as emerging applications and opportunities for improved understanding of the complex interplay of ion–solid interactions. We intend to provide a guide for all scientists in the field that identifies common research interest and will support future fruitful interactions connecting tool development, experiment, and theory. While a comprehensive overview of the field is sought, it is not possible to cover all research related to FIB technologies in detail. We give examples of specific projects within the broader context, referencing original works and previous review articles throughout.

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NOMENCLATURE

AES	Auger electron spectroscopy
AFM	Atomic force microscopy
API	Application programming interface
APT	Atom probe tomography
ASIS	Atomic-size ion source
BCA	Binary collision approximation
BI	Backscattered ion
CAD	Computer-aided design
CMNT	Colloidal micro Newton thrusters
CMOS	Complementary metal–oxide–semiconductor
CNT	Carbon nanotube
COMB	Charge optimized many body
DAC	Digital-to-analog converter
DC	Direct current
DFT	Density functional theory
EAM	Embedded atom method
EBIT	Electron beam ion trap
EBL	Electron beam lithography
EBSD	Electron backscatter diffraction
ECR	Electron cyclotron resonance
EDS	Energy-dispersive X-ray spectroscopy
ESI	Electrospray ionization source
EELS	Electron energy loss spectroscopy
ESEM	Environmental scanning electron microscopy
ETD	Everhardt–Thornley detector
EUV	Extreme ultraviolet
ESA	Excited surface atom
FEB	Focused electron beam
FEBID	Focused electron beam induced deposition
FEBIE	Focused electron beam induced etching
FEEP	Field emission electric propulsion
FIB	Focused ion beam
FIBID	Focused ion beam induced deposition
FIBIE	Focused ion beam induced etching
FIM	Field ion microscopy
FinFET	Fin field-effect transistor
FMR	Ferromagnetic resonance
FOV	Field of view
GAP	Gaussian approximation potential
GUI	Graphical user interface
FWHM	Full width at half maximum
GFIS	Gas field-ionization source
GIS	Gas injection system
hBN	Hexagonal boron nitride
HIBL	Helium ion beam lithography

HIM	Helium ion microscope
HRTEM	High resolution transmission electron microscopy
HSQ	Hydrogen silsesquioxane
IBIC	Ion beam induced charge
IC	Integrated circuit
ICD	Image charge detector
ICP	Inductively coupled plasma
IIAES	Ion induced Auger electron spectroscopy
IL	Ionoluminescence
ILIS	Ionic liquid ion source
kMC	Kinetic Monte Carlo
LE-FIB	Low energy focused ion beam
LJ	Lennard-Jones-type
LMAIS	Liquid metal alloy ion source
LMIS	Liquid metal ion source
LoTIS	Low temperature ion source
MC	Monte Carlo
MCP	Micro channel plate
MD	Molecular dynamics
MFM	Magnetic force microscopy
ML	Machine learning
MS	Molecular statics
MEMS	Micro-electro-mechanical systems
MOTIS	Magneto-optical trap ion source
MRAM	Magnetic random access memory
NAIS	Nano-aperture ion source
NEMS	Nano-electro-mechanical systems
NIL	Nanoimprint lithography
NSOM	Near-field optical microscopy
NV	Nitrogen vacancy
PEMFC	Proton exchange membrane fuel cell
PFIB	Plasma focused ion beam
PI	Primary ion
PIXE	Particle induced X-ray emission
PMMA	Poly(methyl methacrylate)
QMS	Quadrupole mass spectrometer
RBS	Rutherford backscattering spectrometry
SDD	Silicon drift detector
SE	Secondary electron
SEM	Scanning electron microscopy
SFIM	Scanning field ion microscope
SI	Secondary ion
SII	Single ion implantation
SIMS	Secondary ion mass spectrometry
SNMS	Secondary neutral mass spectrometry
SNR	Signal to noise ratio
SPE	Single photon emitter
SPM	Scanning probe microscopy
SSPD	Superconducting single-photon detector
STEM	Scanning transmission electron microscopy
STIM	Scanning transmission ion microscopy
SQUID	Superconducting quantum interference device
TDDFT	Time-dependent density functional theory
TEM	Transmission electron microscopy
TIC	Total ion counter
TOF	Time-of-flight
UHV	Ultra-high vacuum

YBCO	Yttrium barium copper oxide
YIG	Yttrium iron garnet
YVO	Yttrium orthovanadate
YSZ	Yttrium stabilized zirconia
ZBL	Ziegler–Biersack–Littmark
ZPL	Zero-phonon line

I. INTRODUCTION

The technological origin of the focused ion beam (FIB) instruments we use today lies in outer space, or more precisely, in the application of ion beams for spacecraft propulsion. In space, thrust can only be generated by ejecting matter, the so-called reaction mass, which must be carried along with the spacecraft. In addition to chemical thrusters based on combustion, ion thrusters have emerged as an important tool for high-precision movement. The positively charged ions that are generated by field ionization or by electrospraying are accelerated by electric or magnetic fields and then neutralized before being ejected in the opposite direction to that of the intended motion. (Neutralization is important here because otherwise the spacecraft would accumulate negative charge and thus attract the ejected positive ions.) Different types of thrusters based on liquid metal ion sources are currently being tested in space for ultra-precise position control of satellites, e.g., for the LISA gravitational wave interferometer.¹ One of these thruster technologies, the electric field emission propulsion system (part of the LISA Pathfinder mission² and more recent CubeSat launches³) is in fact very similar to the heart of many of our ground-based FIB instruments.

Whereas ion thrusters enable the exertion of forces in the micronewton range for the navigation of space objects, our ground-based FIB instruments enable fabrication, modification, and characterization of micrometer- to nanometer-sized objects.

The leading example of FIB processing is still the site-selective preparation of samples for high-resolution imaging techniques, in particular, for transmission electron microscopy (TEM) and atom probe tomography (APT) and for 3D volume imaging using scanning electron microscopy (SEM). Several reviews have already been devoted to these important, well-established applications.^{4–6}

However, since focused ion beams can be used to modify any material down to the nanoscale in a variety of ways, from targeted doping to structural modification and geometric shaping, the FIB is a powerful tool in all areas from basic research to technology. Our focus is, therefore, on these novel and advanced applications of the FIB.

This document presents the state of the art of FIB research and development today and discusses future perspectives. It is organized as follows: Sec. II gives an overview of FIB instrumentation, starting with the generation and control of the focused ion beam, followed by detectors and other complementary tools and accessories. Section III summarizes the theoretical approaches that can be used to describe various aspects of ion-matter interactions, including the binary collision approximation (BCA), molecular dynamics (MD), kinetic Monte Carlo (kMC) techniques, density functional theory (DFT), and continuum modeling. The last of these enables modeling over the longest length- and time-scales, including treatment of ion- and electron-induced surface chemistry. The wide range of applications of the FIB is discussed in Sec. IV, which is organized according to the various experimental techniques (subtractive processing, defect engineering, imaging and tomography, elemental analysis, gas-assisted processing, and several other emerging directions). For each application field, a selection of examples is

discussed. As a service to the community, summary tables with references have been compiled in order to provide a more comprehensive (albeit non-exhaustive) literature review of each of these application fields. These tables can be used as a starting point to help the reader identify new FIB opportunities for their own research.

Central to this document is the roadmap in Sec. V, which highlights future perspectives for FIB research and development based on the state of the art in instrumentation, theory, and applications previously described. Here, key drivers for the FIB in different areas of science and technology have been identified and linked to FIB-specific challenges and the steps needed for future developments. We hope that this roadmap can serve as an incubator for future developments and will provide inspiration for scientific and technological breakthroughs, as well as serve as a unique resource for funding agencies and industry.

II. INSTRUMENTATION

The creation of a finely focused beam of ions presents several engineering challenges. Moreover, the beam requirements can vary widely depending on the application, sometimes with conflicting specifications. For example, for many applications, a high-current beam is desired to enable efficient milling (i.e., material removal by sputtering). However, high-current sources tend to deliver ions with a large energy spread, resulting in strong chromatic aberrations. Because of this (and other reasons, such as spherical aberrations from the lenses, as discussed later), it is challenging to build a high-current high-resolution source. Similar conflicting scenarios present themselves in many other areas of focused ion beam (FIB) instrumentation.

In the following, the various components of the FIB instrument are discussed. We start by describing the different ion sources that are used (Sec. II A). The source defines many properties of the final instrument, including the achievable spot size. Then we address beam transport via the ion optical column (Sec. II B), and subsequently detectors and analytics (Sec. II C); these elements, which are used to steer, shape, and detect the ions, must be designed in such a way as to ensure the best possible end performance of the particular source being used. In the sample chamber, there are several other components that can be incorporated, including specialized sample stages for *in situ* or *in operando* experiments, micro-/nanomanipulators, and gas injectors. These are discussed in Sec. II D. Next, we address considerations concerning experiments with radioactive samples (Sec. II E). Finally, we outline software needs and correlative approaches for beam control, automation, and multi-modal analysis (Sec. II F). An overview of these topics can also be found in the book chapter by Note.⁷

A. Ion sources

In order to achieve high spatial resolution in a FIB instrument, an ion source with high brightness is required.⁸ Analogous to the definition in electron microscopy, the brightness B of an ion source is a measure of the compactness and directionality of the ion beam, according to

$$B = \frac{I}{A\Omega}, \quad (1)$$

where the emission current I from a source area A is emitted into a solid angle Ω .^{9,10} In practice, this means that ions are suitable for microscopy and nanofabrication if they are emitted from a highly localized area into a well-defined direction. The reduced brightness is derived from the above definition by taking into account the acceleration voltages of the

TABLE I. FIB sources compared according to key parameters. Only common ion species are listed. All published or commercially available ion species can be found in Fig. 1. Superscript q marks different charge states (from 1 to 3) obtained from LMIS/LMAIS; for plasma sources even higher charge states are obtained. Subscript n for LMIS/LMAIS indicates the possible emission of polyatomic clusters comprising up to ten ions (or even more). Parameters for ionic liquid ion source (ILIS) indicated by * are estimated values from Ref. 11. For better comparison, the lateral spatial resolutions have been converted to FWHM values assuming a Gaussian beam profile resulting in an error-function-like edge profile.

Source	Common ion species	Reduced brightness (A m ⁻² sr ⁻¹ V ⁻¹)	Min. energy spread (eV)	Max. current (nA)	Lateral resolution FWHM (nm)	Key applications
LMIS/LMAIS	(Li, Si, Ga, Ge, In, Sn, Sb, Au, Pb, Bi, ...) _{1...n} ^{q+}	1 × 10 ⁶ (Ref. 12)	2–40 (Refs. 12 and 13)	100	2–2.8 (Refs. 14 and 15)	Surface patterning and cross-sectioning, volume imaging, local doping and implantation, SIMS, mask edit, ion thruster
ILIS	EMI ⁺ , BMI ⁺ , BF ₄ [−] , I [−] , ..., cluster ions	5 × 10 ⁶ * (Ref. 11)	7–10 (Refs. 16 and 17)	0.75* (Ref. 11)	50*–30 000 (Ref. 11)	Ion thruster, reactive ion etching, SIMS
PFIB	(Xe, Ar, Kr, N, O ...) ^{q+}	1 – 10 × 10 ³ (Ref. 18)	7–10 (Ref. 18)	2500 (Ref. 19)	20–100 (Refs. 18–20)	High rate sputtering, volume imaging, implantation of gaseous elements
GFIS	(He, Ne) ⁺	1 – 4 × 10 ⁹ (Refs. 21–23)	0.25–1 (Refs. 21 and 24)	0.15 (Ref. 23)	0.5/4.0 (Refs. 25–27)	High resolution imaging and nano-structuring, mask repair, implantation of gaseous elements
LoTIS/MOTIS	(Li, Cs, Cr, Rb) ⁺	1 – 240 × 10 ⁵ (Ref. 28)	0.2–2 (Refs. 29 and 30)	25 (Ref. 31)	4.9 (Ref. 28)	SIMS, implantation
NAIS	H ^{q+} , Ar ^{q+}	4 × 10 ⁵ (Ref. 32)	0.9–2.3 (Ref. 33)	0.2–20 (Ref. 32)	≈2000 (Ref. 32)	SEM as an ion source, proton beam writing

FIB instrument; representative values for the different ion sources are given in Table I. In general, a smaller (virtual) source size means a higher brightness and, consequently, a higher achievable lateral resolution. However, for the final resolution of the instrument, additional parameters such as the extractable ion current, energy spread, and of course the performance of the ion optics are also of importance.

In the field of ion sources, the first breakthroughs came in the 1970s with the development of the liquid metal ion source (LMIS) and the gas field-ionization source (GFIS). While the latter evolved from field ion microscopy (FIM) dating back to the 1950s³⁴ and was first employed for scanning transmission ion microscopy (STIM) of biological specimens,^{35,36} the LMIS was originally developed for space thruster applications.^{37–39}

The LMIS consists of a capillary or a sharp needle tip, wetted with a molten metal. Through a combination of surface tension and applied electric field, the so-called Taylor cone⁴⁰ is formed, resulting in the extraction of ions from the source apex by field evaporation. The most common form of LMIS is based on gallium due to its low melting temperature and low vapor pressure. In fact, the Ga-LMIS is still the most used source for FIB instruments, not only because of its high brightness, but also because of its high source stability.

Directly related to the LMIS is the liquid metal alloy ion source (LMAIS), which through the use of a variety of alloys enables access to

a much wider range of ion species,^{12,15,41–49} but with strongly varying source lifetimes from minutes to months. Based on the same source principle, the ionic liquid ion source (ILIS) is wetted with a compound that dissociates into molecular anions and cations such as the ionic liquids C₆N₂H₁₁-BF₄ (EMI-BF₄), C₆N₂H₁₁-GaCl₄ (EMI-GaCl₄), or C₈N₂H₁₅-I (BMI-I).^{16,50–55} The ILIS is, thus, capable of producing beams of molecular ions, both positively and negatively charged, but has not yet been implemented commercially.

The GFIS operates at low temperature, producing ions from the field ionization of adsorbed gas atoms, and is characterized by the highest brightness and hence the highest deliverable spatial resolution of all sources developed to date.^{21,25,56} This is a consequence of the atomically sharp emitter, which consists of only three atoms at its tip, the so-called trimer, whereby each atom emits a beamlet and thus forms a virtual source with a size in the Angstrom range. The trimer must be formed by the user and has a typical lifetime between a few days and a few weeks. The source gases that are widely used for high-resolution imaging and high-resolution (metal-free) milling are He and Ne. Operation of the GFIS with H₂,⁵⁷ N₂,^{58,59} Xe,⁶⁰ and Kr⁶¹ has also been demonstrated. A closely related source technology is the atomic-size ion source (ASIS) source,^{62,63} which uses field emission of adatoms (Au and Ag have been demonstrated) deposited onto a tip

made from refractory metal or another inert material with a high field ionization strength. High resolution is to be expected due to the single atomic ionization site employed in this technology. However, among other issues, it is the limited lifetime of the ASIS which has hindered its application in actual FIB instruments so far.

Another source option offering nonmetallic ions is the plasma ion source of the so-called plasma focused ion beam (PFIB) instrument. Plasma ion sources achieve high currents of up to 2 μ A and have a long lifetime, but lower brightness than LMIS sources. The plasma can be generated by electron impacts, as in a duoplasmatron,^{18,64} by inductive coupling of alternating currents in a radio frequency antenna [an inductively coupled plasma (ICP)],⁶⁵ or by microwaves in an electron cyclotron resonance (ECR) ion source.⁶⁶

In addition to the systems mentioned above, most of which are commercially available, there are other less common types of ion source. One example is the so-called cold atom ion source, of which there are two varieties: the magneto-optical trap ion source (MOTIS) and the low temperature ion source (LoTIS). These use magneto-optical trapping (in combination with laser-cooling in the case of the LoTIS) to generate a trapped cloud of atoms or an intense atomic beam with a (transverse) temperature in the microkelvin regime.^{31,67–72} The atoms in the trap or beam are then field- or photoionized to produce an isotopically pure, singly charged ion beam of high brightness and low energy spread.^{29,30,73–75} MOTIS/LoTIS achieve good resolution at low energies and are suitable for a wide variety of ions⁷⁵ but require a high degree of sophistication in their design and handling. Consequently, these sources have so far only been demonstrated for a few ion species (see Table I and Fig. 1).

Another example is the nano-aperture ion source (NAIS) that uses an electron-impact gas ion source^{76,77} and can be installed on standard scanning electron microscopy (SEM) instruments to deliver ions from all noble gases or protons. Further less common source types are Paul traps,⁷⁸ the electron beam ion trap (EBIT),^{79,80} the multicusp plasma ion sources,⁸¹ the solid electrolyte ion source (SEIS),⁸² and the electrospray ionization source (ESI).⁸³

A systematic overview of all ion species currently available for FIB instruments is displayed in the form of the periodic table of elements in Fig. 1. The most common types of ion sources that are used for FIBs, and their key parameters are compared in Table 1.

Crucial factors to be optimized for the routine use of new ion sources are source lifetime and stability. Lifetime issues are usually related to contamination and can often be handled by working in cleaner conditions. However, in other cases, lifetime and stability issues can be traced to the source itself. For example, the formation of stable beams from certain elements, such as Al and P extracted from a LMAIS, remains challenging. The exact reasons for these complications are not yet fully understood. Source stability is also of particular importance for large volume sputtering applications, as well as for focused ion beam induced deposition (FIBID) and resist-based FIB lithography. For maximum patterning fidelity, these applications require FIBs with highly stable emission currents and beam positioning. Examples of poor stability are the short term current fluctuations and long term current drift of the Ne-GFIS²⁷ and effects such as pulsation, droplet, and globule emission from the source tip in LMIS and LMAIS.⁸⁴

B. Beam transport

The main components responsible for transporting the beam in a FIB column from the source to the sample are shown in Fig. 2. The labels on the left correspond to components found in more standard instruments and the labels on the right correspond to additional components found in more specialized instruments. To summarize: after extraction from the source, the ions are formed into a beam by the condenser lens (often after passing through an entrance aperture) and then guided toward the probe current selecting aperture (often via a quadrupole lens). The current selecting aperture defines the probe current by selecting a small portion of the beam and thereby also reducing its angular dispersion. To monitor the beam current, the beam is

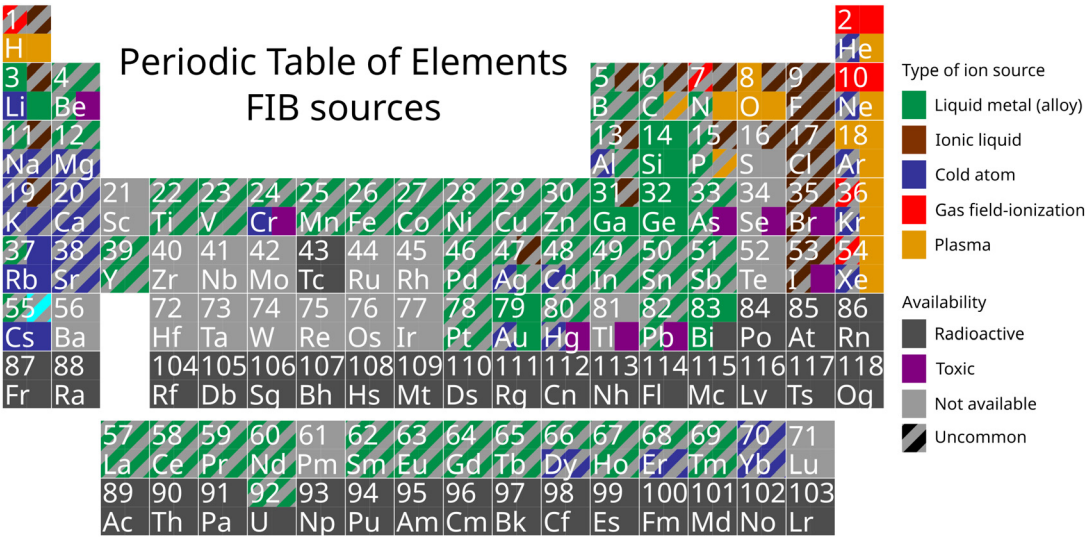


FIG. 1. Periodic table of elements of available ion species for FIBs. The color of each element shows the corresponding ion source type. Ions of elements that are not available, toxic or only available as radioactive isotopes are also indicated by color. If ions from a particular element are available using multiple types of ion source, this is shown as labeled. Diagonal stripes indicate that the ions of those elements are not widely used and/or are produced with a low yield.

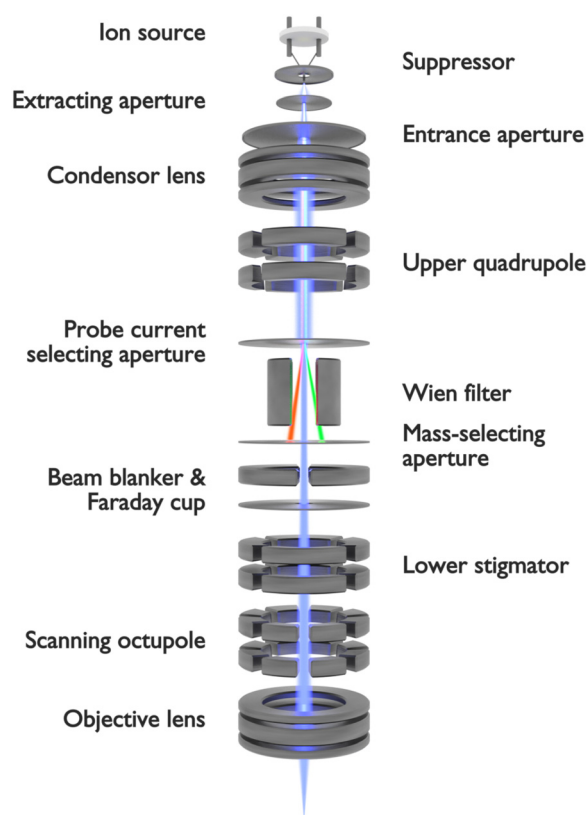


FIG. 2. Principal components for beam transport in a FIB column (not to scale). Components labeled on the left are found in standard Ga-FIB columns, whereas labels on the right indicate components usually only found in special-purpose columns.

deflected into a Faraday cup. The lower part of the column often houses a stigmator for final beam shaping, plus an octupole for beam scanning and a second electrostatic lens (objective) for beam focusing. In more specialized instruments, a Wien filter is used to select isotopes, i.e., ions of a specific mass-to-charge ratio.

The condenser and objective lenses are convergent electrostatic lenses with a cylindrical geometry and are used to steer and focus the ion beam. They are composed of several electrodes (typically three, i.e., an Einzel lens), which are electrically biased to generate electric fields that change the trajectories of the transiting ions through the electric force $\vec{F} = q\vec{E}$ (see Fig. 3). Electrostatic lenses can be operated in retarding mode where the ion energy is reduced inside the lens, or in accelerating mode where the ion energy is increased inside the lens. Figure 3 shows an example of a lens operated in retarding mode. Given the curved shape of the field lines inside the lens, the electric forces imposed on the ions have a radial component in addition to the deceleration and acceleration axial components of these forces. The radial force components result in the overall focusing effect of the lens.⁸⁴ The focal length of an electrostatic lens depends on the ion energy, the charge state of the ions, the voltage applied to the lens, and the lens geometry (and not on the ion mass, as would be the case for a magnetic lens). For a lens operated at a given voltage, the focal length

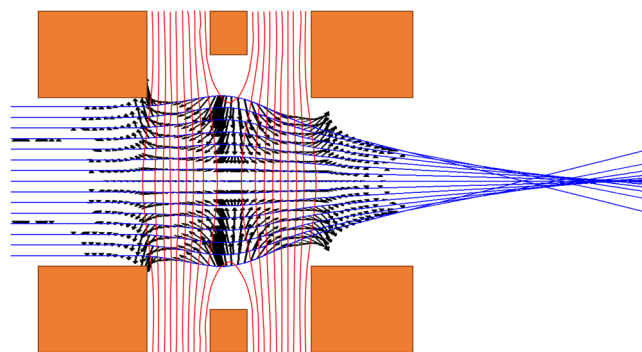


FIG. 3. Example of an electrostatic (Einzel) lens, showing three cylindrical electrodes (in orange) where the central one is biased to focus the ions onto a given position. Equipotential lines are drawn in red, ion trajectories in blue, and the black arrows indicate the direction and magnitude of the electrostatic force.

will be smaller if the retarding mode is implemented compared to the accelerating mode.

The beam blower comprises an electrostatic deflector that diverts the ion beam into a Faraday cup. This allows the user to both blank the beam to prevent ion impingement on the specimen as needed, and also to measure the probe current. Beam blanking is an important feature, not only for standard applications such as nanostructuring by FIB milling or FIBID but also for ion implantation tasks. In the most extreme case, beam blanking must be fast enough so as to allow implantation down to the single ion level from a non-deterministic ion source. In this context, fast means that for an assumed primary ion current of 1 pA, it is necessary to deliver a 10 ns ion pulse in order to achieve the correct probability, following Poisson statistics, of no more than one ion per pulse. Fast blanking is achieved using dedicated electronics and electrodes located at a specific position in the column such that the beam is blanked in a really short time without creating any beam tail artifacts. Detection on the sample must also be able to count the arrival of each ion, one by one (see Sec. II C on detectors).

Other optical elements present in nearly all FIB columns are quadrupoles and octupoles, which are used to steer the beam and are key for proper alignment of the beam with respect to the major optical elements in the column (such as the lenses and apertures). These elements are comprised of two sets of either four or eight electrostatic deflectors, which allow the user to tilt and shift the beam (by applying two counteracting deflections) onto the vertical axis of the subsequent optical element. In the lower part of the column, close to the final lens, a stigmator is often implemented, which allows the user to correct the shape of the beam by applying electric fields that compress (or expand) the beam along directions perpendicular to the optical axis. All three of these elements (the quadrupoles/octupoles and stigmator) enable some compensation for the aberrations introduced by the imperfections of the lenses as well as correction for the mechanical alignment of the column. In the FIB instruments commercially available today, chromatic or spherical aberrations are not routinely corrected. This is mostly because the magnitude of both is strongly defined by the source. However, theoretical papers have shown the possible benefits of such corrections^{85,86} and prototypes have been built using an electrostatic corrector.^{87,88}

More specialized columns are equipped with a Wien filter,^{89,90} by which perpendicular electric and magnetic fields separate the ions in the primary beam according to their mass-to-charge ratio (see Fig. 4). This is necessary when the ion source produces ions from several elements and/or ions of different charge states. The electromagnetic selection of ion species has of course no effect on any neutral atoms that may be present in the beam. To prevent neutral atoms reaching the specimen, the beam can be sent through an electrostatic chicane blanker,⁹¹ whereby all non-deflected particles (neutrals) are blocked such that only the charged particles (ions) reach the sample.

As mentioned previously, the performance of a given FIB instrument ultimately depends on the specifications of the ion source. Most of the optical elements typically used in today's FIB columns do not correct for any beam aberrations, although they might create some due to inherent/manufacturing imperfections. If the source has a poor performance, this will inevitably be propagated to the sample. Consequently, adjusting the design of the beam transport elements on a state-of-the-art FIB instrument will typically only result in small improvements in the overall FIB performance.^{84,92,93}

The main gap between the desired performance of a FIB instrument and its actual performance is currently in the area of low energy beams (<2 keV). With the ongoing trend to reduce the dimensions and increase the complexity of (3D) device and sample architectures, the interactions between the ion beam and the sample become ever more critical. Thus, the penetration depth and straggle of the ions inside the material must be minimized. The key way to reduce these effects is to decrease the energy of the beam, but the consequence is an increase in beam spot size. This increase in beam spot size is due to chromatic aberrations from the energy spread of the ions in the beam, whereby the relative energy spread and hence aberrations become more pronounced when the beam is retarded.

Finally, the performance of current FIB systems is also limited in terms of the attainable processing speeds. Using a single focused beam requires scanning with varying exposure and move/blank sequences, and is inherently serial. In contrast, template masking of a broad ion beam on the sample can provide much higher throughput, but at the expense of flexibility and spatial resolution. Therefore, it has been theoretically proposed to combine masking of a broad beam with full control of single beams in a multibeam approach that significantly improves throughput for nano-applications.⁹⁴ A working proof-of-concept demonstration of such a multibeam FIB column has been built around a controllable array of ion beamlets.⁹⁵ In brief, an aperture plate splits a broad parallel ion beam into a large number of 2.5 μm -wide beamlets. Each of these 43 000 beamlets can be individually deflected by an array of apertures with adjacent electrostatic electrodes,

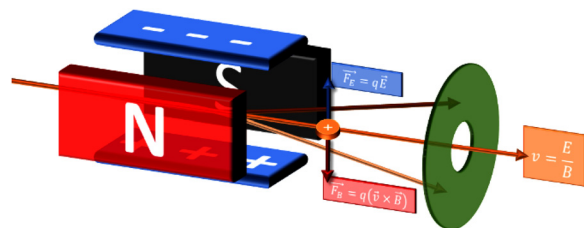


FIG. 4. Principle of a Wien filter, with orthogonal electric and magnetic fields and an aperture to select the desired ion velocity.

fabricated using complementary metal–oxide–semiconductor (CMOS) technology. All the beamlets are then passed through 200 \times reduction optics, which blocks any deflected beamlets. The remaining beamlets irradiate a user-defined pattern of pixels with a resolution of less than 20 nm. This multibeam technique has been optimized for electron irradiation⁹⁶ and is commercially available as a mask writer,⁹⁷ but further development of the concept for different ion species is highly desirable.

C. Analytical tools and detectors

Irradiating a sample with an ion beam to trigger and measure a response is a very common analytical technique, and FIB instruments that perform imaging, local irradiation, milling, etc., are typically equipped with various accessories to confer a range of analytical capabilities. For imaging purpose, the secondary electrons (SEs) emitted from the sample surface are routinely detected using an Everhardt–Thornley detector (ETD).⁹⁸ The ETD is scintillator-based, converting SE strikes to photons inside the sample chamber that then travel via a light guide to a photomultiplier outside the chamber. As a complementary imaging channel, secondary ions (SIs) can be detected using a total ion counter (TIC) (typically in positive SI detection mode using a Faraday cup or a channeltron). In addition to these imaging modes, various other analytical techniques are implemented on FIB platforms; a general overview is given in Table II.

Secondary ion mass spectrometry (SIMS) enables the mapping of elemental/chemical compositions in the form of 2D/3D images or depth profiles with high sensitivity (down to the ppm level) in combination with a high dynamic range (i.e., a given element can be measured over a concentration range of several orders of magnitude). In principle, all elements including their isotopes can be measured. However, for a truly quantitative analysis, reference samples are required since the ionization yields strongly depend on the local environment in the sample (known as the matrix effect).¹⁷⁸ Several implementations of FIB-secondary ion mass spectrometry (SIMS) systems have been explored with three main system types described in the literature for units installed on both FIB and FIB-SEM instruments:

- (1) Historically, quadrupole mass spectrometer (QMS) systems have been used on FIB platforms due to their simple design, low weight, and reduced costs.^{101,104} In a QMS, only ions with a specific mass-to-charge ratio are able to travel through the applied quadrupole field. Ions with different mass-to-charge ratios travel on unstable trajectories, thus leaving the mass spectrometer before the final exit aperture and so are not counted. QMSs, therefore, have the disadvantage of not allowing parallel detection (i.e., only one mass can be detected, the detection of several masses requiring sequential analyses and hence a duty cycle) and also have a lower performance in terms of sensitivity.
- (2) More recently, various orthogonal and linear time-of-flight (TOF)-based mass spectrometers have been introduced.^{121,122} Here, mass separation is achieved because ions with different mass-to-charge ratios will reach different velocities; measuring their time-of-flight allows the mass of the ions to be inferred. The TOF systems offer the advantage of parallel detection, but since pulsing of the primary or secondary ion beam (or both) is required, a duty cycle results.
- (3) Magnetic sector SIMS systems can offer parallel detection by using so-called continuous focal plane detectors^{107,179} and the

TABLE II. Overview of FIB based analytical methods.

Technique	Primary ion	Detected signal	Spatial resolution		References
			Lateral	Depth	
QMS-SIMS	He ⁺ , Ne ⁺ , Ga ⁺ , Cs ⁺ , Xe ⁺ , O ⁺ , Bi ⁺ , Bi ⁿ⁺ , Au ⁺ , Au ⁿ⁺ , O ⁻	Secondary ions	100 nm	<20 nm	99–106
Magnetic sector SIMS			<15 nm	<2 nm	107–115
TOF-SIMS/orthogonal TOF			<50 nm (<30 nm)	<10 nm	101,116–127
SNMS	Ga ⁺	Post-ionized neutrals	20 nm		87,128–131
STIM	He ⁺	Transmitted ions/neutrals, backside SEs	<5 nm		132–140
Backscattering spectroscopy	He ⁺	Backscattered ions/neutrals	≈50 nm		120,141–150
Backscattering yield	He ⁺	Backscattered ions/neutrals	5 nm		151,152
Ion induced SE spectroscopy	He ⁺ , Ar ⁺	Secondary electrons	Several nm		153–161
IIAES	Si ⁺ , Ne ⁺ , Ar ⁺ , Au ⁺ , Kr ⁺ , Ga ⁺	Auger electrons	10 nm		162–167
PIXE	He ⁺ , Ne ⁺ , Ga ⁺	X-rays	100 nm		168,169
IL	He ⁺	Photons	20 nm		170–177

highest sensitivity enabling high-resolution imaging applications.¹⁴⁰ In a magnetic sector mass separator, ions are forced onto circular trajectories by a perpendicular magnetic field. The radius of curvature of this trajectory depends again on the mass-to-charge ratio of the ion. Parallel detection of all masses can be achieved using focal plane detectors and the appropriate spectrometer geometry. Magnetic sector systems are operated in direct current (DC) mode, i.e., they feature a 100% duty cycle. In terms of disadvantages, these systems typically have a larger footprint and are heavier because of the integrated electromagnet.

By combining a gas injection system (GIS) with FIB-SIMS, further improvements in analytical output have been demonstrated. For example, in work combining FIB-TOF-SIMS with a GIS, ionization probabilities and hence SI signals were found to significantly increase by 2–3 orders of magnitude, thereby improving the quality of 2D and 3D chemical maps.¹²³ Furthermore, it was observed that co-injection of XeF₂ during the ion bombardment can reduce mass interference^{125,126} and invert the polarity of the negatively charged SIs to positive,¹²⁷ thus allowing the collection of more complete chemical information. Enhancements in SI yields have also been obtained with magnetic sector SIMS instruments using Cs deposition and O₂ flooding to boost the yield of negative and positive secondary ions, respectively.^{180,181}

Related to SIMS, secondary neutral mass spectrometry (SNMS)^{130,131} can obtain similar information, analyzing sputtered neutrals through laser-based post-ionization, and has the advantage of matrix-independent ion yields. However, while resonant post-ionization is an efficient process, non-resonant ionization yields are low. SNMS requires significant experimental effort and is, therefore, associated with high costs.

Scanning transmission ion microscopy (STIM) is performed using the ions of light elements. It can provide mass-thickness contrast and also crystal structure information due to ion channeling.¹³⁸ Several different STIM implementations have been investigated, most

based on the collection of secondary electrons generated from impact of the transmitted ions on a conversion plate,^{132,133,139} but also some using direct detection of the transmitted ions.^{134,135,138} STIM is a quasi-nondestructive imaging technique and for biological specimens has been shown to deliver structural contrast comparable to scanning transmission electron microscopy (STEM).¹⁴⁰ Similar to STEM, STIM demands thin samples (<100 nm). STIM can also be used to obtain high-resolution images that can be correlated with elemental/chemical maps determined by SIMS.¹¹⁰

Backscattering spectroscopy in a FIB instrument is similar to Rutherford backscattering spectrometry (RBS) but is performed at much lower energies of around several tens of keV. This results in multiple scattering events due to the relatively large nuclear scattering cross sections. Even when performing time-consuming simulations of the ion spectra, quantitative results from the raw energy spectra are difficult to obtain. However, it has been shown that when coupled with a TOF setup, backscattering spectroscopy can deliver damage-free depth-resolved elemental compositions.¹²⁰ Signal levels are good, since in addition to the SIs, the backscattering method is also sensitive to the large number of neutrals generated during the primary ion impact.

One can also analyze the yield of the backscattered primary particles, for which larger solid angles and higher signals can be achieved. This is accomplished using an annular micro channel plate (MCP) located beneath the objective lens of the microscope. While the backscatter yield approach is feasible regardless of FIB instrument type, it is most useful in a helium ion microscope (HIM). In this case, the backscatter yield of the primary He ions is high enough (in particular upon interaction with heavier elements) that damage from the primary beam can be neglected.¹²⁰ Backscatter yields depend strongly on the atomic number of the target.¹⁴¹ To a certain extent, elemental analysis is also possible but requires prior knowledge of the elements in question.¹⁴⁷ The backscatter approach has been shown to reveal elemental contrast from buried layers¹⁵¹ and has also been applied to biological specimens.^{182,183} Since backscatter yields depend strongly on ion channeling effects, this approach can be used to map crystal orientations.^{149,152,184–186} SE yields are in fact also influenced by ion

channeling and provide stronger signals, hence crystal orientation mapping based on the detection of SEs offers a simple alternative¹⁴⁹ and has been shown to enable the visualization of interfacial nanoscale dislocation networks in thin-film alloys.¹⁸⁷ The ability to infer crystallographic information from the SE yield has been extended to other ions and automated.¹⁸⁸

Ion induced SE spectroscopy (using He ions) draws on variations in SE energy to map chemical variations on a sample surface.^{155,189–191} The approach allows the user to maximize imaging contrast, thus permitting short beam exposure times, which is beneficial for beam-sensitive samples. However, a quantitative application of the SE spectroscopy method is currently limited by the complicated nature of the data obtained and the lack of a suitable reference database.

Ion induced Auger electron spectroscopy (IIAES) allows the chemical identification of surface variations, including bond structure, via the energy-resolved detection of Auger electrons emitted following an ion induced inner-shell electronic transition.^{166,192} For elements in the third row of the periodic table, ion induced Auger electron spectroscopy (IIAES) has a superior signal-to-noise ratio compared to electron beam based Auger electron spectroscopy (AES).

Ion/particle induced X-ray emission (PIXE) in a FIB instrument is essentially the equivalent of energy-dispersive X-ray spectroscopy (EDS) in the SEM, whereby characteristic X-rays (used for elemental mapping) are generated as a result of particle bombardment. However, in the FIB case, PIXE suffers from extremely low X-ray yields and, hence, has not proved practical thus far.

The final entry in Table II, ionoluminescence (IL), has been tested on various materials.¹⁷² However, the authors of these works concluded that the merits of this technique only really emerge in the case of *in situ* characterization of beam induced defects¹⁷³ and for the localization of rare earth elements.¹⁷⁵

Specialized ion detection methods for single ion implantation (SII) also deserve mention (see Sec. IV B for more detail). These methods can be categorized into pre- and post-detection techniques. In the pre-detection category, one example is the use of an image charge detector (ICD) incorporated into the FIB column that registers the passage of a single ion (or ion bunch) and is coupled with fast blanking electronics.^{193,194} Post-detection can involve standard SE detection¹⁹⁵ or more complex sample-integrated detection schemes such as ion beam induced charge (IBIC) or source-drain current measurements.^{196–199}

D. Other FIB accessories

In addition to the above-described key components, a number of other accessories have been developed that extend the capabilities of FIB instruments even further. A summary of these is given below.

The most common addition to a FIB instrument is an electron column (SEM). Vice-versa, FIB columns are often added to SEMs. The electron column enables correlative imaging, *in situ* monitoring of milling processes, FIB-SEM volumetric reconstructions, and SEM-based analytics such as EDS. The SEM addition facilitates sample navigation, since by imaging with electrons rather than ions, beam damage to the sample can be rendered negligible. Certain FIB-SEM implementations also allow automated metrology and analysis.^{200,201}

For high-resolution, large-area direct patterning applications, laser interferometer stages become critical, since the positioning accuracy and stability of standard mechanical stages are not sufficient for

these high-end applications. In conjunction with specialized sample holders, laser interferometer stages enable a sample positioning accuracy in the nanometer range over a lateral distance of several 100 mm.²⁰²

A further rather common addition is a gas injection system (GIS), which enables focused ion beam induced deposition (FIBID) and, in combination with an electron column, also focused electron beam induced deposition (FEBID).²⁰³

FIB columns are also sometimes operated in conjunction with an electron flood gun, whereby *in situ* charge neutralization is achieved by illuminating an insulating specimen with low energy electrons. In particular for HIM, this enables imaging of insulating samples at high spatial resolution without the need for conductive coatings²⁰⁴ (see Sec. IV C 2), and more generally, FIB milling of insulating samples.²⁰⁵ A low resolution approach to allow charge-free macroscopic sample navigation is to use an optical camera. The latter has been extended to enable *in situ* fluorescence²⁰⁶ and Raman spectroscopy.²⁰⁷

Various types of micro/nanomanipulators have been developed for intuitive control and mechanical manipulation of micro- to nanoscale objects, including *in situ* lift-out of lamella specimens for transmission electron microscopy (TEM),²⁰⁸ and to allow local electrical connections for various *in situ* experiments as demonstrated for SEM.^{209–211}

Through the addition of a femtosecond laser ablation system, large amounts of material can be removed very efficiently from around the final target area for subsequent finer milling with the FIB. This enables higher throughput for applications such as advanced package failure analysis and process optimization in the semiconductor industry.²¹²

Numerous add-ons for a wide range of *in situ* characterization experiments have also been developed, both by manufacturers and by researchers. These solutions include electrical probing stations, systems using Peltier elements for sample heating or cooling (not to be confused with more complex cryo-FIB add-ons), *in situ* mechanical testing, plasma-based sample cleaners, and automatic laser-based height sensing. Other often sought-after add-ons are inert gas transfer boxes to allow oxygen-free loading (and unloading) of air-sensitive samples.²¹³ Various solutions for incorporating atomic force microscopy (AFM) into FIB and FIB-SEM instruments have also been developed, e.g., Ref. 214, which can be implemented in combination with FIB-SIMS to allow an assessment of surface roughness and thus improve the accuracy of the 3D reconstructions.¹¹⁷ A recent review article summarizing state-of-the-art solutions for *in situ* characterization and micro-/nanomanipulation has been published by Shi *et al.*²¹⁰

E. FIB processing of radioactive samples

The investigation of radioactive samples using a FIB system raises a number of critical issues (legal and technical) that need to be addressed. The reason the FIB is such a useful instrument in the analysis of these samples is that it allows the preparation of small-scale specimens (by FIB milling), which by default have significantly reduced radioactivity levels. These small-scale samples can then be handled much more easily for a wide range of characterization experiments, both *in situ* and *ex situ*, including specimens for small-scale mechanical testing,²¹⁵ 3D volume imaging,^{216,217} lamellae for TEM or synchrotron investigations,^{216,218,219} and needle-shaped specimens for atom probe tomography (APT).²²⁰

The exact protocols to follow may vary widely depending on the national guidelines, individual facility, and the local safety regulations. For example, individual sub-samples prepared from a larger radioactive sample may still be subject to radiation protection guidelines even in cases where radioactivities are below a clearance limit for which the sample can be considered quasi non-radioactive. The type of material, form of radioactivity, and isotopes involved are all key factors to take into account.

FIB instruments that are used for radioactive samples can be categorized as follows, depending on the laboratories where they are installed and on the shielding implemented:

1. Instruments classed for “low-medium” radioactivity samples. These instruments are typically also used for experiments with non-radioactive samples, and are installed in labs with radiation protection guidelines and moderate shielding.
2. “Hot cell” instruments that are highly shielded (lead wall around the whole instrument) and may even allow spent fuel samples.

Irrespective of the particular regulations, the main concerns when working with radioactive samples in FIB instruments are:

- Dose limits for the operator when loading and unloading the sample;
- Contamination of the instrument to the extent that system maintenance becomes difficult;
- Cross contamination between samples in the case of work with radioactive and non-radioactive samples;
- Lifetime of detectors and other add-ons installed on the instrument (EDS, EBSD, etc.).

Regarding contamination of the instrument itself, it has been determined that most of the milled material that does not redeposit on the sample is deposited on the pole piece and surrounding areas.^{221,222} In the case of materials that generate dust, the sample stage can also easily become contaminated. One approach to address contamination of the pole piece is to use a dedicated pole piece insert for experiments with radioactive samples. FIB-SEM instruments with environmental scanning electron microscopy (ESEM) capabilities can also be helpful here, as these provide another aperture for low-pressure mode, and the water vapor that is typically injected helps to volatilize the radioactive sputtered material. Frequent cleaning of areas prone to contamination is another effective strategy to increase the lifetime of the instrument. The inside of the chamber can also be covered to protect from contamination. Some approaches use sample shields to reduce the spread of the sputtered material by capturing most of it inside the shield. However, sooner or later, the entire FIB chamber will need to be treated as radioactively contaminated equipment and handled appropriately.

F. Software and correlative approaches

The software infrastructure of a FIB instrument needs to control various elementary subsystems, e.g., the vacuum system and sample stage, as well as the ion source and column functions (beam extraction parameters, aperture selection, column alignments, etc.). This is implemented by the manufacturer using a graphical user interface (GUI) that allows control of all FIB system components and integrated add-ons. Customized FIB setups employing, e.g., specialized, noncommercial FIB columns, or nonstandard add-ons, usually use separate control

software developed for these specific use cases, which are often difficult to set up and maintain due to closed hardware interfaces and patent issues. Regardless of the particular configuration, software control for the following applications is required:

- Imaging (including 3D volume imaging) and elemental analysis using various detectors, see Secs. [II C](#), [IV C](#), and [IV D](#);
- Milling, ion irradiation, and implantation, see Secs. [IV A](#) and [IV B](#);
- Gas-assisted processing and resist-based lithography, see Secs. [IV E](#) and [IV F](#).

All of the above demand a scan generator that controls the path of the ion beam on the sample with high precision and repeatability.

Beam scanning is realized using a digital-to-analog converter (DAC) patterning board that determines the number of addressable pixels in each direction (e.g., 65 536 for 16 bits) and sets the voltages of the beam deflection system accordingly. The beam path is then a sequence of pixels with a certain dwell time and spacing depending on the field of view. Simple line-by-line scanning routines are used for imaging and elemental analysis, and the respective detector output is synchronized to assign a signal (e.g., a gray value or a mass spectrum) to each pixel position.

For subtractive processing, deposition, resist lithography, etc., special beam paths are needed. Here, manufacturers typically offer software tools that enable the creation of geometric shapes that are rasterized according to a chosen set of parameters. Grayscale patterning offers an even easier way to realize complex shapes and the milling of 3D profiles, by loading an image in which the grayscale values of the individual pixels correspond to the local relative ion doses to be applied. The latter is an appealing plug-and-play solution for markers, text, and other non-quantitative designs, yet it is difficult to optimize systematically and is, therefore, not commonly used for actual structures. In the case of more advanced patterning needs, design files can be imported from lithographic software, assigned with parameters, and rasterized, thus enabling complex and large-scale structuring tasks.²²³ Standalone lithography systems from third-party manufacturers can also be directly integrated as accessories via the use of an external scan control unit for the FIB column. Such systems typically also have access to stage control, allowing multi-step processing with marker-based registration or stitching, the latter requiring a laser interferometer stage for the highest level of positioning accuracy.

The capabilities of these various patterning systems are sufficient for many applications, but reach their limits when more complex geometric shapes and the highest spatial resolution are required, and/or when the position of the beam must be known with certainty at all times. This is because with these systems, the user has limited control over the actual raster process and possible auxiliary routines can be hidden in the proprietary software (e.g., for stabilization of the beam position). The most robust solution is to address pixels directly on the patterning board using a point cloud, which can be encoded in so-called stream files or deflection lists. This allows arbitrary rasterized beam paths to be executed by the integrated patterning software. Various groups have developed codes to generate such beam paths and the corresponding point clouds for specific tasks.^{224,225} For arbitrary geometric shapes, an open-source Python-based solution is available, which allows the generation of patterns and geometry-adapted beam paths that can be both variably rasterized and optimized.²²⁶ For

the creation of 3D profiles, approaches based on computer-aided design (CAD) have also been developed.^{227–231} These solutions implement material removal via thin slices plane-parallel to the sample surface, which allows systematic optimization since the amount of material removed per slice remains constant (a significant advantage over the grayscale patterning mentioned above). Modeling of the milling process can also be used to take into account angle-dependent sputtering²²⁸ and redeposition,²³² or more generally, non-isotropic surface erosion.²²⁹ For more details, see Secs. III A and III D.

In addition to beam control, another software need focuses on enhancing the analytical capabilities of the instrument. Commercial FIBs are often coupled to a SEM column and/or equipped with analytical instrumentation such as SIMS. Most FIB instruments are thus furnished with electron detectors for the collection of SEs to allow for SEM-like imaging (albeit the physics of the signal formation is different).²³³ Additional detectors offer a variety of other imaging possibilities via the detection of transmitted ions,¹³⁸ backscattered ions,¹²⁰ secondary ions,¹⁴⁰ photons,^{170,171} Auger electrons,^{166,167} etc., see Table II. These FIB instruments, thus, need to be equipped with the necessary interfaces to allow for either direct control of the beam position, or synchronization between the detector signal and internal scan generator. Such interfaces are standard for the commercial FIB platforms offering these analytical modalities.

For common repetitive FIB tasks, instrument users can employ various forms of software automation. Examples of automated tasks include column alignments and focusing, trench milling (including for TEM lamella preparation), large-scale sequential lithography projects, serial sectioning, and complex analytical tasks such as 3D tomography using multiple detectors. In the past, the technical challenges of many of these processes were handled largely without automation by expert FIB operators following intensive training. However, over the last decade, this situation has been significantly relieved by the introduction of various semi and fully automated processes by the major FIB manufacturers. It must be stressed that there is still room for improvement and further automation is highly desired. Here, automation routines that are more customizable or user-programmable with well documented application programming interfaces (APIs) or scripting environments within the FIB software stack would be of great benefit. This would allow standard automation routines to be set up and executed by lab technicians after just a short training period.

Finally, the correlative microscopy and spectroscopy functionalities of modern FIB instruments (both for 2D¹¹⁰ and 3D^{234–236} analysis, see Secs. IV C and IV D) require a combination of the aforementioned software controls as well as advanced software tools for correlation, analysis, and visualization.^{237,238} In the simplest case, a correlative dataset is acquired using multiple detectors and a single scan of a surface (or in the case of 3D analysis, a stack thereof) such that multichannel data are available for every beam location (pixel). If this is not possible, datasets from separate experiments have to be combined post-acquisition under the condition that some form of alignment is possible, e.g., via fiducial markers. For the correlation of 2D images, open-source software is available, e.g., in the form of the Fiji/ImageJ image processing package²³⁹ and its plug-ins such as Correlia.^{240,241} Additional resources can be found on the webpages of the COMULIS project²⁴² and the BioImage Informatics Index.²⁴³ For the most complex cases (e.g., the correlation of X-ray and FIB tomography data),

dedicated analytical software²⁴⁴ and novel approaches for data storage and management are required, see Sec. V C 3.

III. SIMULATION APPROACHES FOR FIB PROCESSING

Simulations carried out at various levels of sophistication have the potential to provide insights into the interaction of the FIB with the target. These calculations can help rationalize the experimental results, optimize the beam parameters (ion energy, incidence angle, etc.), and guide the experimental work. Various theoretical methods have been developed to describe the interaction of the impinging ions with the target and to assess the amount and type of sample modification produced by a given ion irradiation.^{245–247} Among them, atomistic computer simulations, which describe the system as a collection of interacting atoms, have provided much insight into the behavior of materials under the impact of energetic ions. These approaches can give precise information on ion ranges and energy losses and on the types of irradiation induced defects. Defect stability and long-term evolution under ambient conditions or at elevated temperatures during annealing can also be simulated. Continuum models, on the other hand, can be computationally more efficient and eliminate the statistical noise inherent to many atomistic methods. They describe the system under investigation by continuum quantities, such as concentrations or surface contours, which conceptually require averaging over finite volumes. In practice, these quantities are often discretized on a mesh of elements with sizes larger than atomic dimensions. Continuum models also require parameters to be provided by experiments.

The choice of simulation method is dictated not only by the main goal of the simulation (e.g., to assess the electronic stopping power or to calculate ion ranges, amount and type of damage, atomic mixing, surface evolution, etc.), but also by the system size, the required level of sophistication, and the computational costs. A typical approach is to find a suitable compromise between the necessary computational resources and accuracy. Multiscale simulations are frequently used, in which more accurate but computationally demanding approaches are employed as tests, or to provide parameters for lower spatial or temporal resolution techniques.

An overview of the time and length scales accessible via the various simulation methods is given in Fig. 5. The binary collision approximation (BCA) method, discussed in Sec. III A, describes the ballistic phase of the slowing down of the ions and of the development of collision cascades. Since the ballistic phase is over in less than a picosecond, application of the BCA method makes sense only on a sub-picosecond timescale. *Ab initio* methods, in practice mostly density functional theory (DFT), are the most accurate techniques, but are restricted to picosecond and nanometer ranges because of their computational cost. Classical molecular dynamics (MD) may be used for system dimensions of roughly up to 100 ns and 100 nm. These methods are described in Sec. III B. Longer time scales are accessible by the kinetic Monte Carlo (kMC) method (Sec. III C), which describes the evolution of a system given the probabilities of events. The actual limits of kMC are determined by the event frequencies and the density of objects in the system to be studied. Finally, continuum models (Sec. III D) usually study systems beyond the single digit nanometer range, as they do not resolve materials on the atomic level.

In the following subsections, a concise review of the available computational techniques, with examples of their use, is presented. It

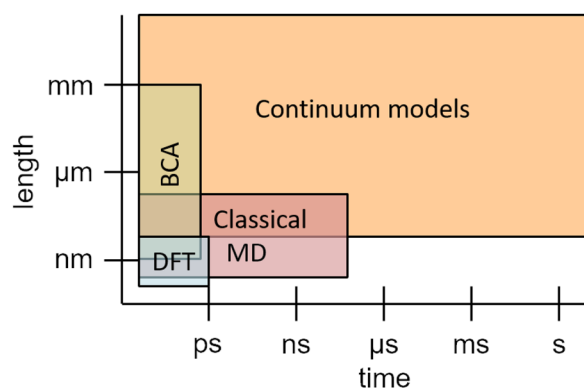


FIG. 5. Time and length scales accessible via simulation methods. The smallest time scales are accessible only using certain types of continuum models. kMC can cover the entire length-time space displayed depending on the details of the system studied and on the sub-method used, and is therefore not included in the graph. For more information, see Ref. 248.

is hoped that this overview provides a clear picture of the opportunities simulation provides for FIB processing.

A. Simulation techniques based on the binary collision approximation (BCA)

The BCA approach is the most widely used method to assess the scattering and slowing of energetic ions in matter and the effects of the associated collision cascades, i.e., sputtering, atomic mixing, and the formation of point defect damage. The motion of each energetic atom is described as a sequence of asymptotic trajectories between binary collisions with target atoms that are at rest before the collision. Detailed information on the associated algorithms is provided in the book by Eckstein.²⁴⁹

1. Principles and limitations

The binary collisions are normally treated as elastic with repulsive-only screened Coulomb interaction potentials, as, e.g., in the “universal”²⁵⁰ or “Kr-C”²⁵¹ parameterization. Energy transfer to the target electrons is included either nonlocally along the free paths between the collisions (e.g., using a universal analytical description²⁵² or specific semiempirical data²⁵³) or locally dependent on the collisional impact parameter²⁵⁴ or as a combination of both.

Simulations based on the BCA method are often denoted as Monte Carlo (MC) simulations due to the random choice of collisional parameters, such as the impact parameter and the azimuthal position of the target atoms relative to the trajectory of the moving atom. In random and amorphous media, this applies to all binary collisions during the simulation. In contrast, in crystalline media only the initial position of the incident projectile relative to the lattice atoms is treated randomly, while the subsequent slowing down, including the generation and termination of a collision cascade, is largely deterministic due to the positions of the lattice atoms being fixed except for thermal vibrations.

A prominent advantage of BCA simulations is their low computational cost and very fast performance, which allows modeling even of

large systems up to the macroscale and the treatment of incident ion energies up to the MeV regime. However, in contrast to MD simulations (see Sec. III B), the BCA model fails if a significant fraction of the collisions occurs between moving atoms. Such so-called “collisional spikes” or “elastic thermal spikes”²⁵⁵ form in dense cascades generated under heavy-ion bombardment in the energy regime centered around the maximum of the nuclear stopping force²⁵⁶ (typically around 100 keV). However, any collision cascade also dissipates into a collisional spike before its final thermalization, which sets a low-energy limit on the validity of the BCA. Depending on the material, estimates²⁴⁹ and comparisons with molecular dynamics simulations^{257,258} indicate a failure at energies below several tens of eV. However, experience shows that sputtering, which is dominated by cascade atom energies slightly above the surface binding energies of 2–8 eV,²⁵⁶ is rather well described.^{259,260} This suggests a practical lower energy limit of validity. Undoubtedly, BCA breaks down below the bulk binding energy of typically a few eV, where the purely repulsive interaction potential is no longer appropriate and many-body interaction has to be taken into account. This excludes, e.g., the derivation of local atomic configurations after thermalization.

It must also be noted that BCA simulations require predefinition of a number of options and parameters, such as the choice of interaction potential, the addition of “soft” collisions with more distant target atoms, the type of electronic interaction, and the surface binding and atomic relocation threshold energies, which are often difficult to identify and unavailable in the literature, particularly for compounds. (Note that except for the electronic interaction, these predefinitions are unnecessary in MD simulations because proper choice of the interaction potential ideally covers all bulk and surface interactions down to thermal energies.) Whereas ion ranges derived from BCA simulations are mostly reliable within 5%–10%, sputtering and defect results are significantly influenced by these preselections. As a result of parameter variations within physically sound limits, calculated sputtering yields may easily change by more than 50%.²⁶¹ For the different BCA codes, parameters are often not chosen from theoretical or experimental information, but justified based on prior experience comparing the simulation results to experimental data.

Originally, BCA codes were designed for “static” simulations during which the system is assumed not to be altered by the irradiation. This allows the prediction of range and damage distributions as well as sputtering yields with high statistical precision when a sufficiently large number of incident projectiles is employed. In the simulation of ion implantation into single-crystalline targets, “dynamic” consideration of damage buildup is required if the fluence exceeds a certain threshold,^{262,263} as crystal damage reduces or suppresses channeling. This is achieved by choosing collision partners randomly instead of from the crystal lattice, with a probability proportional to the damage. More often, the term “dynamic BCA simulation” refers to the modification of the chemical composition and geometry of the target, which occurs experimentally for a sufficiently large number of incident ions. Specific BCA codes track implanted ions as well as relocated and sputtered recoil atoms in order to continuously update the local composition in multicomponent systems and the surface position or the surface contour in 1D or 2D/3D systems, respectively. Simultaneously, the composition-dependent local atomic density must be updated by volume relaxation and/or material transport. (Again, this dynamic modification is straightforward in MD simulations, but in systems exceeding

the nanoscale and for higher ion energies it is often impeded by excessive computational cost.)

2. Specific BCA codes

Among the large variety of BCA-based simulation codes that have been described in the literature, we will in the following only address a few selected ones, in particular those that are presently in broader use and/or offer new features with respect to nanosystems, specifically for FIB applications. For a rough classification, see Table III, access information for most of the codes can be found in Ref. 264.

TRIM^{265,266} has been the most widely used BCA code for several decades. It offers a very convenient graphical user interface²⁵³ for fast generation of range and damage statistics, and rough estimates of sputtering yields in amorphous semi-infinite or thin-film systems with a flat surface. The restriction to amorphous media is often not a severe limitation, since many materials become highly damaged or even amorphized under irradiation and, depending on the conditions, most ion or recoil atom trajectories are random in nature, even in crystalline materials. TRIM partly fails in detail, such as for sputtered atom angular/energy distributions.²⁹²

Sometimes the crystal structure of the irradiated material does matter. Typical cases include low-fluence or single-ion implantation aligned with a low-index crystallographic direction²⁹³ and the sputtering of elemental metals.²⁶⁰ Codes that consider the crystal structure are Crystal-TRIM,^{262,271} which is restricted to certain materials, and IMSIL,^{263,272–274} which allows more general crystal systems and sputtering simulations.

Based on the early TRIM sputtering version TRIM.SP,²⁶¹ 1D dynamic relaxation has been implemented in TRIDYN^{267,268} and in SDTrimSP.²⁶⁹ In connection with the recent broad interest in nanostructures, BCA codes have been extended to treat 2D and 3D systems using pixel and voxel grids, respectively. For static operation only, the surface for simple bodies may be defined analytically, such as in TRI3DST²⁷⁸ and IM3D,²⁸⁰ or triangularized as in a second option of

IM3D. With a pure voxel approach, the stepped surface contour resulting from pixel or voxel grids may cause artifacts by, e.g., capturing atoms at glancing incidence (which could in fact occur experimentally). This may be overcome by a local adjustment of the near-surface nodes²⁷⁴ (IMSIL) or by local planarization of the surface²⁸⁴ (TRI3DYN). In IMSIL for 2D dynamic simulation,²⁷³ pixels that are affected by collisional transport become distorted based on an algorithm which optimizes their volumes and are subsequently projected onto the original pixel grid. In contrast, SDTrimSP-2D²⁷⁵ relaxes the pixel volumes along one dimension only toward a specified surface. 3D dynamic simulations have been demonstrated using TRI3DYN^{282–284} and EnvizION.^{285–291} For overall volume and surface relaxation, TRI3DYN makes use of material exchange with nearby voxels or the surface, whereas in EnvizION the relaxation is limited to near-surface regimes. TRI3DYN even works for macroscopic systems, while EnvizION is limited to the nanoscale, since each voxel contains only one atom. A characteristic result of a TRI3DYN simulation is presented in Fig. 6, demonstrating surface erosion and contamination during irradiation of a Au nanosphere on a Si surface.

3. BCA-based simulations of FIB processing

1D static BCA simulations have often been employed for understanding experimental findings in FIB processing such as in recent studies of FIBID.^{294–297} These have also been used to generate, e.g., angle-dependent sputtering yields and angle-energy distributions of sputtered atoms to describe erosion and re-deposition, respectively, in 2D and 3D “level set”²⁹⁸ or “segment based”^{299–301} simulation models of the surface contour development during FIB induced erosion, see Sec. III D.

A direct 2D dynamic BCA simulation of FIB induced erosion is described in Ref. 273, where the milling of trenches is exemplified. The results shown in Fig. 7 demonstrate the slowing of the milling process by atoms sputtered from the bottom and redeposited to the sidewalls. In addition, a high concentration of implanted atoms at the bottom of

TABLE III. Classification of selected BCA simulation codes. Note that the individual codes contain specific algorithms, use different parameters, and/or are designed for specific application areas, which are not detailed here.

Code	Geometry	Atomic structure	Surface	Dynamic relaxation
TRIM ^{253,265,266}	1D	Amorphous	Planar	...
TRIDYN ^{267,268}	1D	Amorphous	Planar	1D
SDTrimSP ²⁶⁹	1D	Amorphous	Planar	1D
IMINTDYN ²⁷⁰	1D	Amorphous	Planar	1D
Crystal-TRIM ^{262,271}	1D	Crystalline	Planar	...
IMSIL ^{263,272–274}	1D/2D	Amorph./cryst.	Polygons/pixeled (adjusted)	–/1D/2D
SDTrimSP-2D ²⁷⁵	2D	Amorphous	Pixeled	2D
CORTEO ^{276,277}	3D	Amorphous	Voxeled	...
TRI3DST ²⁷⁸	3D	Amorphous	Analytical	...
IRADINA ²⁷⁹	3D	Amorphous	Voxeled	...
IM3D ²⁸⁰	3D	Amorphous	Analytical/triangularized	...
SDTrimSP-3D ²⁸¹	3D	Amorphous	Voxeled	2D
TRI3DYN ^{282–284}	3D	Amorphous	Voxeled (planarized)	3D
EnvizION ^{285–291}	3D	Amorphous	Voxeled	3D, near-surface

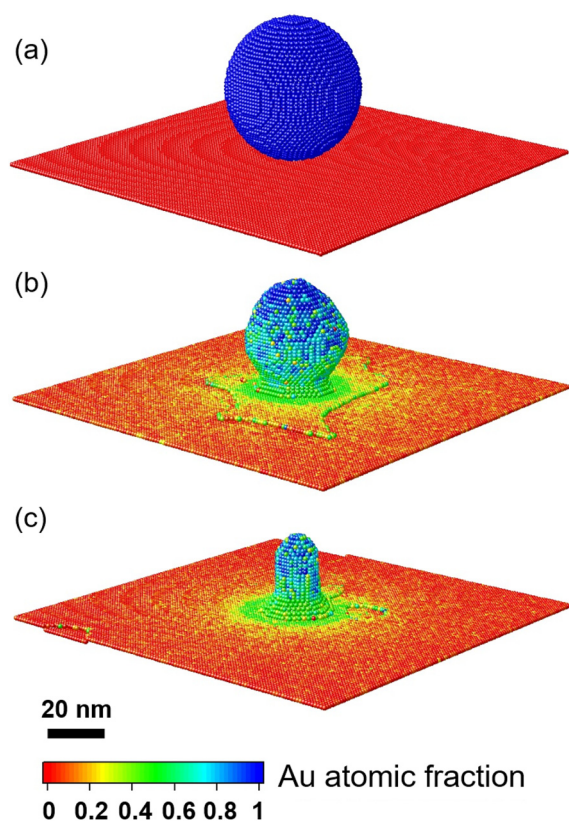


FIG. 6. Sputter shaping of Au nanosphere of diameter 50 nm on Si, (a) initially and (b) after irradiation with 30 keV Ga ions under normal incidence for fluences of 4×10^{15} and (c) $1 \times 10^{16} \text{ cm}^{-2}$. The TRI3DYN simulation (with periodic lateral boundary conditions) demonstrates size reduction and shaping by sputtering, substrate surface erosion, and contamination of the original sphere with sputtered substrate material and vice versa. The incorporation of Ga is small and not shown. For more information, see Ref. 283.

the trench emerges, which pushes material upward along the trench sidewalls via volume relaxation. The simulation of the milling of a slit in a membrane is described in Ref. 274.

With particular focus on FIB processing, Rack and co-workers have published a series of 3D dynamic results obtained with EnvizION.^{285–291} In addition to the collisional BCA simulation and relaxation algorithms used, simplified models have been implemented for reactive-gas assisted erosion and deposition, including secondary-electron mechanisms.¹⁶⁰ Selected results have recently been reviewed in Ref. 302. For an introduction to gas-assisted processes, see also Sec. III D 2.

Dynamic BCA simulations may also be employed to characterize specific FIB based nanoanalytical methods. An example is a 1D TRIDYN simulation of damage buildup during FIB milling of nanomembranes for TEM analysis.³⁰³ Another example is the 3D TRI3DYN simulation of STIM in a HIM device, as shown in Fig. 8. Simulations such as those presented in Fig. 8(a) help us to identify the detector geometry for optimum image contrast and resolution. Furthermore, Fig. 8(b) indicates that there is only minor degradation

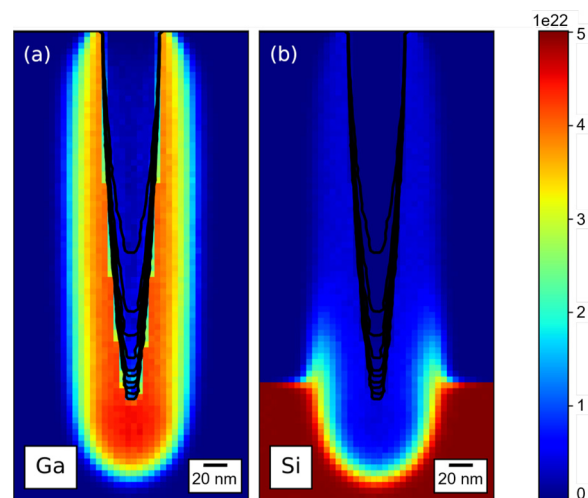


FIG. 7. Bombardment of Si target with a sharp 50 nm-wide, 50 keV Ga beam with a fluence of $5 \times 10^{18} \text{ cm}^{-2}$. Final target composition displaying concentrations (cm^{-3}) of (a) Ga atoms, and (b) Si atoms that originate from a depth greater than 30 nm in the virgin sample. Surface contours track fluence increments of $5 \times 10^{17} \text{ cm}^{-2}$. The scan area of the beam is in the center of the top boundary. The simulations have been performed with IMSIL. For more information, see Ref. 273.

of the sample by surface sputtering and recoil atom transport between the core and the shell and between the sphere and the substrate.

B. Molecular dynamics and molecular statics

Although BCA methods are able to provide insight into processes developing during the ballistic (sub-picosecond) phase of ion irradiation, they cannot predict details of the permanent modification of the target's atomic structure, such as defect clusters or dislocations including their strain fields. Particularly with heavy ions, heat spikes may develop even in keV energy cascades, and the local melting of the target affects the amount of damage formation and atomic mixing.³⁰⁴ The accuracy of the BCA approach diminishes^{257,258} at energies roughly below 100 eV, and it completely fails in the low single-digit eV energy range. Hence, MD is often the method of choice for the simulation of phenomena associated with atomic motion at eV and sub-eV kinetic energies.

1. Principles and simulation codes

Most approaches in practical use are based on the Born–Oppenheimer approximation, which assumes that the electronic subsystem relaxes instantaneously to its ground state at any time as the atomic nuclei change their positions. Consequently, the effect of the electrons can be embodied in a potential energy function that depends only on the nuclear positions.³⁰⁵ Analysis of this energy function allows determination of many material properties as well as the time evolution of the system, e.g., in an irradiation experiment. This is even true when deposition of energy into electronic degrees of freedom dominates, such as for 30 keV He ions impinging on Au clusters³⁰⁶ where defect production is still governed by ballistic energy transfer as excitations are quickly delocalized.

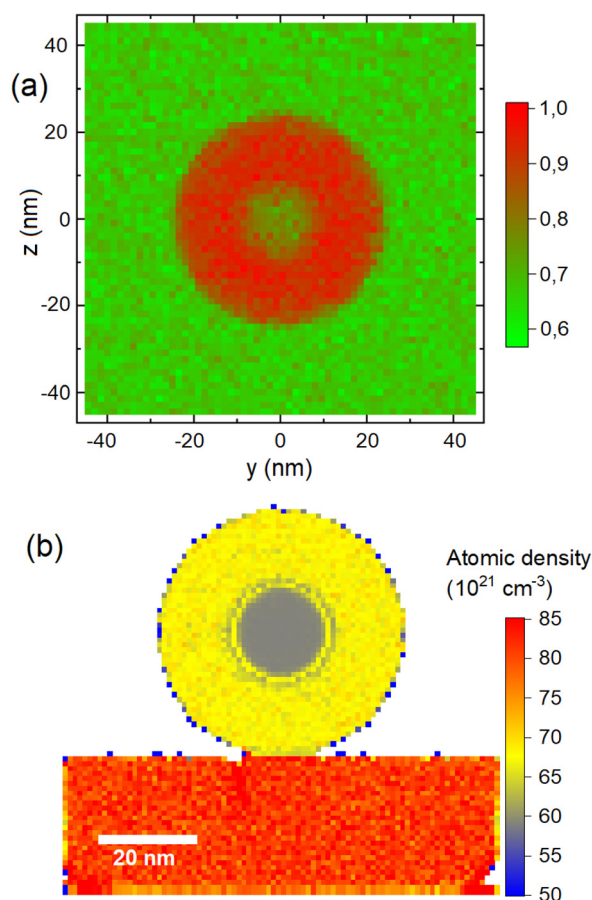


FIG. 8. TRI3DYN simulation of scanning transmission ion microscopy (STIM) in a helium ion microscope (HIM). (a) Ion transmission relative intensity image for Au (20 nm diam.)/SiO₂ (50 nm diam.) core/shell nanosphere on Si₃N₄ membrane of thickness 20 nm. A 20 keV He⁺ beam with a Gaussian beam profile of 0.5 nm full width at half maximum (FWHM) is scanned across an area of 90 nm × 90 nm up to a fluence of 2×10^{16} cm⁻², with the transmitted ions being registered in an annular dark-field detector of 0.6 and 5 cm inner and outer radius, respectively, mounted 2.5 cm behind the sample. Custom evaluation software correlates the detection of the transmitted ions with the actual y–z scan position of the beam. (b) Cross-sectional image of the total atomic density after the irradiation, averaged over a central slice of thickness 2 nm.

In the MD method, Newton's equations of motion are solved for all atoms in the system, using forces calculated from the potential energy function.^{305,307,308} This is done iteratively with time steps on the order of 1 fs with the time step being dependent on the ion/atom velocities and their masses. At the end of the simulation, when the energy is equilibrated everywhere in the system, the final structure can be analyzed. Due to the short time step required, total simulation times are limited to the nanosecond or at most microsecond range for classical MD and to picosecond for *ab initio* MD. Effects corresponding to longer timescales are sometimes mimicked by increasing the temperature in the system, which then allows observation of annealing (recombination) of defects and the relaxation of strain.³⁰⁹ Detailed descriptions of the principles of these MD methods can be found in Refs. 247 and 310.

Molecular statics (MS) methods are concerned with finding minima and saddle points in the potential energy function. The former identify (meta-) stable atomic configurations, while the latter allow the determination of energy barriers for diffusion and chemical reactions. Finding global minima, corresponding to stable as opposed to metastable configurations, can be tricky. Various methods have been developed, see Ref. 305 for details.

Simulation codes can be classified into classical and *ab initio* codes. In the first group, the potential energy and forces are calculated by evaluating mathematical expressions with empirical parameters, while in the second group, they are determined by solving approximations to the Schrödinger equation. The most widely used classical MD code is LAMMPS.^{311,312} Another general purpose MD code is DL_POLY.³¹³ Codes specifically designed for the simulation of radiation effects include PARCAS,³¹⁴ DYMOCAS,³¹⁵ and MOLDYASC.³¹⁶ GROMACS³¹⁷ and NAMD³¹⁸ mainly target biomolecular systems. Popular *ab initio* codes capable of performing MD simulations are VASP³¹⁹ and QUANTUM ESPRESSO.³²⁰

2. Models for the potential energy function

Among the *ab initio* methods, DFT³²¹ with local and semilocal exchange-correlation (XC) functionals provides a good compromise between accuracy and computational expense for the calculation of the potential energy function. Some phenomena, such as van der Waals interactions, magnetism (especially when spin-orbit coupling is taken into account) or accurate evaluation of the bandgap in semiconductors, require one to go beyond “vanilla” DFT by either using more sophisticated XC functionals³²² or post-DFT methods, such as GW/RPA (many body perturbation theory/random phase approximation).³²³ Yet otherwise, DFT simulations normally provide reliable results with respect to the atomic coordinates and energetics on the scales relevant for the modeling of ion irradiation effects. DFT is capable of predicting charge states of defects and static charge transfer,³²⁴ while time-dependent density functional theory (TDDFT) can describe charge transfer dynamics in ion collisions. The advantage of *ab initio* simulations is that they do not require any material-dependent parameters from the user. The disadvantage is their high computational cost, which scales as $n^{2...3}$, where n denotes the number of valence electrons or atoms in the simulation. This imposes a practical limit on the system size that can be handled by DFT, which using supercomputers is currently on the order of 1000 atoms.

If larger systems need to be handled, empirical interatomic potentials (sometimes called force fields) must be used to describe the potential energy function. The computational expense of these simulations scales nearly linearly with the number of atoms. On the downside, the mathematical expressions describing the interatomic potentials are only approximations of the true potential energy function, and the parameters in these models must be fitted for each combination of chemical elements in order to reproduce the physical properties of the material (e.g., lattice parameters, elastic properties, bond energies, energetics of defects, stability of sputtered fragments, etc.). Care is needed if the simulation results in atomic arrangements that are outside the range of atomic arrangements used for the fit. Moreover, the availability of force field parameters can be a limiting factor when looking for new applications, since the development of new parameter sets is a tedious task.³²⁵ To overcome this bottleneck, various methods have been developed in recent years.^{326–333}

Collections of 1000+ interatomic potentials together with their parameters for individual systems are available in the NIST Interatomic Potentials Repository³³⁴ and the Knowledgebase of Interatomic Models (OpenKIM).³³⁵ Educated choice of an appropriate interatomic potential is essential for the success of a simulation. In Table IV, the most popular options are listed together with the types of atomic interactions they can describe and their computational expense. All but the CHARMM force field, which is mainly used to describe biochemical molecules such as proteins, lipids and nucleic acids, allow for bond breaking, a prerequisite for the simulation of almost all irradiation effects. For an introductory text on interatomic potentials, see Ref. 305.

Overall fitting of a single functional form with a limited number of parameters to reproduce a vast range of material properties has frequently proven to be an unmanageable task. To overcome this limitation, machine learning (ML) potentials have recently been developed.^{348,349} Common to all flavors of ML potential is their description of the potential energy using multiparameter functions of the local environment around each atom. To determine the parameters, a large dataset of potential energies and corresponding interatomic forces for a specific set of chemical elements in different configurations is generated by DFT calculations. Hence, the flexibility of these potentials is much greater than that obtained using empirical interatomic potentials, and if carefully developed, they are able to provide results with an accuracy comparable to DFT. There have also been attempts to train ML potentials on-the-fly to handle situations that had not originally been anticipated, such as large surface reconstructions or rare chemical events.^{350,351}

As an example, Fig. 9(a) demonstrates the ability of the Gaussian approximation potential (GAP)³⁵² to reproduce the DFT results for the cohesive energy of pure W as a function of atomic volume in

TABLE IV. Some popular interatomic potentials listed with the type of interaction they are able to describe and the execution times per atom relative to the Lennard-Jones potential taken from Refs. 336 and 337 using LAMMPS. Mean values are given in cases where both of these references list values. Execution times for ML potentials vary widely, depending on which formalism they are implemented in (see main text), but still scale linearly with the number of atoms n . DFT times are usually orders of magnitude larger than those for empirical interatomic potentials. Ratios strongly depend on the system size, since DFT times scale as $n^{2...3}$, hence it is not reasonable to give a scaling factor for DFT.

Interatomic potential	Type of interaction	Execution time
Lennard-Jones ³³⁸	van der Waals	1
EAM ³³⁹	Metallic	1.8
Stillinger-Weber ³⁴⁰	Covalent	3.4
Tersoff ³⁴¹	Covalent	4.4
Coulomb-Buckingham ³⁴²	Ionic	4.7
CHARMM ³⁴³	Biomolecules	14.1
MEAM ³⁴⁴	Metallic/covalent	16.5
COMB ³⁴⁵	Chem. reactions	188
ReaxFF ³⁴⁶	Chem. reactions	197
Kieffer ³⁴⁷	Chem. reactions	...
Machine learning (ML)	All	≈10–10 000
DFT	All	≫

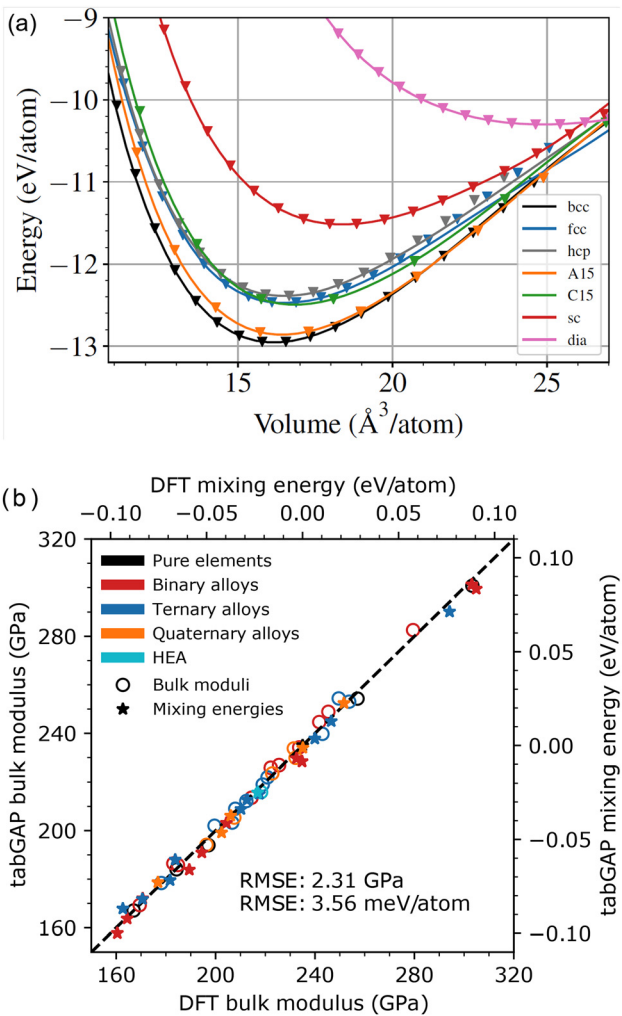


FIG. 9. Example of the agreement between DFT training data sets and machine-learned interatomic potentials using (a) GAP³⁵³ and (b) tabGAP³⁵⁴ for computing the cohesive energy of pure W in different phases, and the mixing energies and bulk moduli of various alloy combinations, respectively. The symbols are the DFT data points and the lines show the GAP and tabGAP results. (a) Reproduced with permission from Byggmästar *et al.*, Phys. Rev. B **100**, 144105 (2019). Copyright 2019 American Physical Society. (b) Reproduced with permission from Byggmästar *et al.*, Phys. Rev. B **104**, 104101 (2021). Copyright 2021 American Physical Society.

different phases.³⁵³ Similar agreement between predictions of the tabGAP potential (a more efficient version of GAP³⁵⁴) and the DFT calculations of mixing energies and bulk moduli is shown in Fig. 9(b) for multiple combinations of binary, ternary, and quaternary alloys.

The interatomic potentials discussed so far describe the potential energy function near equilibrium or a few eV above equilibrium. During irradiation simulations, much higher potential energies occur in close collisions. In this regime, purely repulsive potentials such as also used in BCA simulations, e.g., the universal Ziegler-Biersack-Littmark (ZBL) potential³⁵⁵ or the potential obtained with the all-electron DFT code DMol,³⁵⁶ provide accurate results. These potentials need to be smoothly joined with the material-specific empirical potential

at small distances.³⁵⁷ For ML potentials, it is advantageous to already include the repulsive potential in the training phase.³⁵³

3. Applications of molecular dynamics to FIB irradiation

Large-scale MD simulations have been employed to gain insight into nanostructure formation and the modification of nanostructures. Das, Freund, and Johnson³⁵⁸ investigated the Ga FIB-induced formation of nanopores in Si membranes, which are of interest for biological applications. They identified a local boiling mechanism that could be explosive at high enough ion fluxes.³⁵⁸ Moreover, they found recirculating material flow, which they explained as being driven by temperature-gradient induced surface tension gradients.³⁵⁹ Another interesting phenomenon occurs during ion irradiation of Si nanopillars. At room temperature, the pillars shorten [Fig. 10(a)], while at elevated temperature, they become thinner [Fig. 10(c)]. In the MD simulation performed at room temperature [Fig. 10(b)], Si amorphizes, which initiates viscous flow due to the ion hammering effect³⁶¹ resulting in a shortening of the nanopillar as seen in the experiment. In contrast, in the BCA simulations shown in Fig. 10(d), the nanopillar mainly shrinks in diameter, in agreement with the experimental results at elevated temperature. This is explained by the neglect of viscous flow in the BCA simulations, which in fact does not occur when the

target remains crystalline at the higher temperature. As a third example, MD simulations have examined how a balance among beam spreading, erosion, and defect-induced strain limits the thinning of FIB lamellae.³⁶²

Chemical effects in sputtering cannot be described by BCA simulations but can be using MD simulations employing reactive force fields. For instance, the sputtering of Si by low-energy O and Si ions using the Kieffer force field³⁴⁷ showed that two mechanisms are mainly responsible for sputtering: (i) emission of atoms forming clusters a few Angstroms above the surface, and (ii) ejection of clusters that are loosely bound to the surface.³⁶³ As another example, the impact of residual water molecules on sputter yields of Si bombarded with Ar has been studied (cf. Fig. 11). Here, MD simulations using the ReaxFF potential optimized for the Si-C-O-H system³⁶⁴ are compared to SDTrimSP BCA simulations. For the MD simulations, a much weaker dependence of the O and H partial sputter yields on the incidence angle is found. The difference might be explained by the random positions of the H and O atoms in the SDTrimSP simulations, presenting more efficient scattering centers than the more regular atomic configurations in the MD simulations at incidence angles between about 50° and 80°. At angles closer to perpendicular and grazing incidence, the MD results show higher sputter yields than BCA. This may be explained by the desorption of water molecules, which is not included

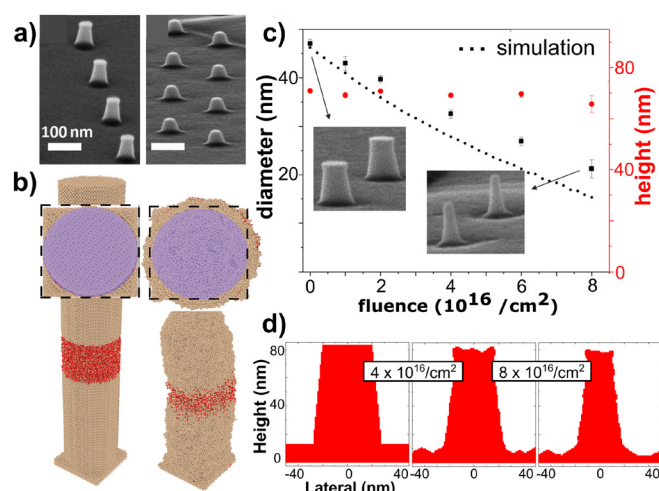


FIG. 10. Comparison of the deformation of Si nanopillars under high fluence 25 keV Ne or Si ion irradiation. (a) scanning electron microscopy (SEM) images before (left) and after (right) Ne ion fluence of $2 \times 10^{16} \text{ cm}^{-2}$ at room temperature.³⁶⁰ Note the short, but still wide shape of the pillars after irradiation. (b) Results of MD simulations for the Si/SiO₂/Si pillar ($r = 5$, $h = 50 \text{ nm}$) before and after Si fluence of $7.6 \times 10^{15} \text{ cm}^{-2}$. In the top-down views, the dashed squares outline the bottom pedestal and the purple circles show the size of the ion beam irradiation pattern used throughout the simulation.³⁶¹ (c) Experimental images of the evolution of Si nanopillar under prolonged Ne ion irradiation at 400 °C.³⁶⁰ Since the Si does not amorphize, the shape is modified only due to erosion from the top and the sides of the pillar, which is well captured by BCA simulations. (d) Corresponding nanopillar images from BCA simulation.³⁶⁰ (a), (c), and (d) Reproduced with permission from Xu *et al.*, *Semicond. Sci. Technol.* **35**, 015021 (2019). Copyright 2019 Authors, licensed under a Creative Commons Attribution (CC BY) license and (b) reproduced with permission from Fridlund *et al.*, *Phys. Rev. Mater.* **4**, 013601 (2020). Copyright 2020 Authors, licensed under a Creative Commons Attribution (CC BY) license.

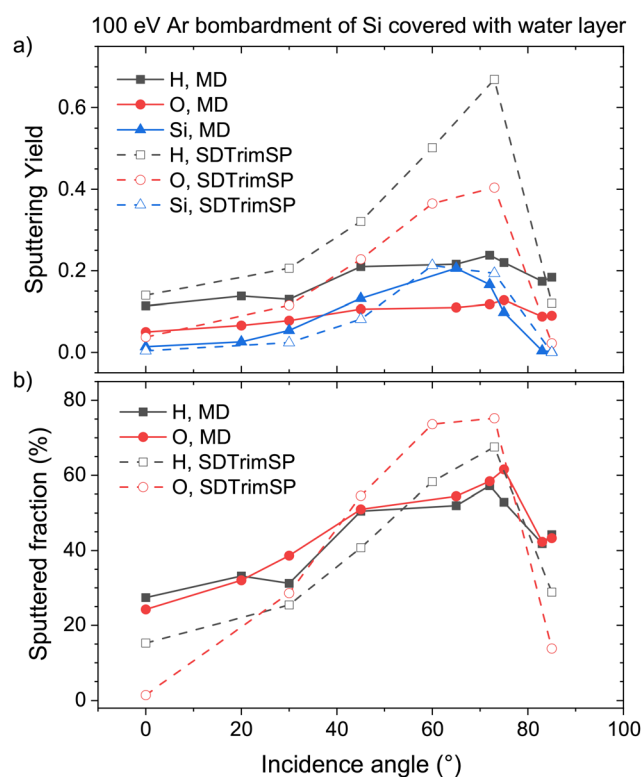


FIG. 11. Results obtained by MD and BCA (SDTrimSP) simulations after 100 eV Ar irradiation of Si covered with a monolayer of water molecules to a fluence of $5 \times 10^{15} \text{ ions/cm}^2$. For the MD Si(100) was used. The SDTrimSP sample was amorphous. The evolution of (a) the sputtering yields and (b) the fraction of sputtered H and O atoms, are shown as a function of incidence angle.

in the BCA model. Note that the maximum Si sputtering yield is found at around 65° , whereas the maxima for H and O are found between 70° and 80° . As a practical conclusion, when the mixing of contaminants into the sample should be minimized, angles closer to grazing incidence should be favored.

Defect formation in all kinds of materials has been a traditional application field for MD simulations.²⁴⁷ Although much insight has been gained, quantitative and sometimes also qualitative results are in doubt due to the inaccuracies of empirical interatomic potentials. For instance, the threshold displacement energy, the essential parameter for damage calculations in the BCA method, turned out to be substantially different for Si when calculated by DFT-MD compared to using the Tersoff or Stillinger–Weber interatomic potentials.³⁶⁵ On the other hand, when using a ML potential, essentially the same values were obtained as with DFT.³⁶⁶ Likewise, sputter yields for low-energy Ar irradiation of Si obtained with the ML potential were found to be close to experimental values, while the empirical interatomic potentials largely underestimate the yields.³⁶⁶ Moreover, it was shown that damage buildup in keV energy cascades is strongly overestimated by the empirical potentials (see Fig. 12).

MD simulations have also provided good understanding of ion induced defect formation in free-standing^{367–369} and supported^{370–372} 2D materials, which is of interest for developing ultrathin membranes to encapsulate gases and other volatile systems for ion beam analysis³⁷³ and also for the development of nanoporous membranes for advanced nanofiltration applications. As with bulk materials, new insight is gained for 2D materials using DFT-MD.^{374–376} As an example, in the study by Kretschmer *et al.*³⁷⁶ the importance of accounting for chemical interactions when evaluating the threshold displacement energy in graphene and hexagonal boron nitride (hBN) has been demonstrated.

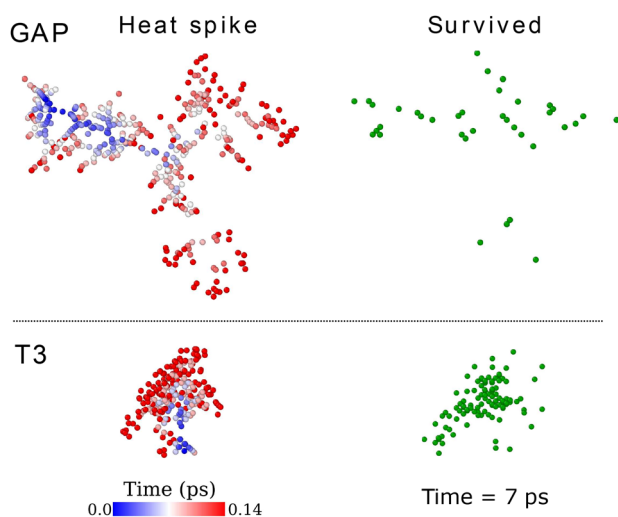


FIG. 12. Typical snapshots of 2 keV collision cascades at the heat spike phase and the resultant defects in Si. Heat spike snapshots (left panel) present the atoms with kinetic energy above 1 eV. Atoms have been color-coded based on the time of generation after firing the primary knock-on atom. Surviving defects (right panel) are acquired from Wigner–Seitz analysis of the cell after cooling to 300 K ambient temperature. The comparison visualizes the overestimated defect formation in Si predicted by the classical potential Tersoff (T3)³⁴¹ compared to Gaussian approximation potential (GAP).³⁶⁶ For more information, see Ref. 366.

4. Limitations and extensions

While in the BCA method, trajectories are only followed for atoms with energies exceeding a few eV, MD simulates all atoms in a region encompassing the collision cascade. Therefore, MD simulations are much more computationally demanding than BCA simulations. Taking into account that actual FIB processes often use rather high fluences, it is clear that the computational cost of this may be a challenge. Moreover, due to the limitation of the MD time step to values that resolve lattice vibrations, the time between collision cascades cannot be fully simulated. Thermally activated events occurring during these time spans may be covered by kMC simulations as needed (see Sec. III C). Often, however, such thermally activated events are neglected, and in addition, the system is artificially quenched to ambient temperature at the end of each cascade in order to shorten the time required to dissipate the energy deposited by the collision cascade.³⁶³ This comes at the risk of possibly missing some relevant defect annealing.

The limitations of empirical interatomic potentials have already been discussed above. Another drawback of Born–Oppenheimer MD is the absence of an inherent description of electrons in the system. Electronic stopping can be included in MD as a frictional force,³⁰⁷ and the same stopping powers may be used as in BCA simulations. Modeling of electronic stopping of channeled ions, however, cannot be directly transferred from BCA to MD. Here, attempts have been carried out on a more sophisticated level.^{377,378} In addition, there is uncertainty about the low-energy limit to electronic stopping, which has been shown to be relevant for predicting sputter yields³⁷⁹ and damage formation.³⁸⁰ A related problem is electron–phonon coupling, which describes the exchange of energy between the electronic and ionic subsystems. Electron–phonon coupling may be included in MD simulations by the so-called two-temperature model.³⁸¹ However, so far no general consensus on the model for electron–phonon coupling nor its significance at FIB-relevant energies has been reached.³⁸² Recent insights from TDDFT³⁸³ show promise for enabling a physically motivated parameterization of how electronic stopping and electron–phonon coupling relate to one another.^{384–386} Other possible effects to consider include the modification of atomic interactions due to electron excitations in the system.³⁸⁷

Born–Oppenheimer MD also fails to describe, e.g., the transitions between electronic states or the neutralization of ions in the vicinity of a 2D target. These situations may be treated by TDDFT-MD (Ehrenfest dynamics). In irradiation simulations using Ehrenfest dynamics, the nuclei are usually approximated as classical particles moving in the time-dependent average potential of the electrons, while the electron subsystem evolves according to TDDFT. Due to their complexity and high computational cost, Ehrenfest dynamics simulations using massively parallel computers are currently only feasible for systems comprising tens of atoms and for times up to picoseconds. The method has been successfully applied to electronic stopping power calculation in various targets,^{388–391} excitation mediated diffusion modeling,³⁹² and simulations of the behavior of inorganic 2D materials under electron beam irradiation,³⁹³ but its applicability is limited by the mean-field approximation and the resulting propagation of mixed states. For example, the sputtered atoms may have fractional charges that are unphysical. It is also worth noting that this approach has been demonstrated to work for describing the formation of HIM images by assessing the spatial distribution of excited electrons immediately after the ion impact.³⁹⁴

C. Kinetic Monte Carlo (kMC)

The timescale accessible by MD simulations is in the nanosecond range, which is sufficient for the system to reach thermal equilibrium after an ion impact. After this time, diffusion and interaction of defects take place by thermally activated events. Examples of such events are diffusion hops, attachment of an atom to a cluster, or recombination of two defects. The rate constant of each event may be determined by the transition state theory using the potential energy function of the system, e.g., using DFT and the nudged elastic band (NEB) method.³⁹⁵ Sometimes, this parameterization can be achieved directly from experimental data, e.g., by fitting to a known temperature-dependent diffusion coefficient. kMC is a method that evolves systems described by such events stochastically. For a basic introduction to the method, see Ref. 396. Another review is found in Ref. 397, and more recent ones with special emphasis on radiation effects and semiconductor processing can be found in Refs. 398 and 399, respectively. Other approaches for extending the timescale of MD are discussed in Ref. 400.

Several variants of the kMC method have been developed. On-the-fly kMC determines energy barriers of possible events during the course of a simulation. This technique has been used to simulate defect diffusion in Fe,⁴⁰¹ defect evolution in graphite⁴⁰² as well as deposition and sputtering.⁴⁰³ On-the-fly kMC can easily be combined with MD, since both methods can use the same potential energy function, and neither needs prior knowledge of possible events. However, the search for possible events makes the on-the-fly variant of kMC the most computationally expensive.

Setting up a catalog of events *a priori* can be approached in two ways. In lattice (or atomistic) kMC, a lattice is defined where each site can be empty or occupied by an atom. Events are hops from one lattice site to another with rates modeled based on the local environment of each site. Use cases include vacancy diffusion,⁴⁰⁴ solid-phase epitaxial regrowth,^{405,406} sponge structure formation in Ge,⁴⁰⁷ self-organized surface pattern formation,⁴⁰⁸ knock-on damage in hBN,⁴⁰⁹ and phase decomposition of substoichiometric SiO₂ into Si nanoclusters in an amorphous SiO₂ matrix.⁴¹⁰

In object kMC, in addition to vacancies and impurity atoms, other defects including clusters are treated as individual objects that are tracked as they move and/or react with each other. The atoms of the “background” material are not explicitly represented. Therefore, the system size accessible by object kMC is determined by the number of objects in the simulation, rather than by some fixed volume. The object kMC approach has mainly been used in simulations of radiation effects in nuclear materials⁴¹¹ and semiconductor processing.^{399,412}

Lattice and object kMC can be combined not only with MD^{402,403,408,411} but also with BCA.^{404,410–412} In addition, kMC schemes are occasionally also used to describe system evolution due to non-thermally activated events. For instance, in Ref. 373, the accumulation of defects in a graphene sheet is modeled based on defect production probabilities obtained by MD simulations of ion impacts.

D. Continuum modeling

Unlike the methods described in Subsections III A–III C, continuum models do not use individual atoms as the objects of simulation, but instead use quantities that are continuous functions of space and time. For example, in a continuum approach, the ion beam is described by its spatially dependent flux (number of ions per area and time increment), the adsorbed precursor molecules are described by their

concentration (number of molecules per surface area), and the boundaries of solids are described by mathematical surfaces.

1. Ion implantation and milling

In the absence of ion beam mixing and other secondary effects, the distribution of implanted ions may be obtained by the convolution of the beam flux $F_{\text{beam}}(x, y)$ and a “point response” $f(x, y, z)$. The point response is the probability density function for the stop position of an ion entering a planar target at the origin of the coordinate system. For the purpose of this discussion, we assume the ions to be incident along the negative z axis; x and y denote the other two Cartesian coordinates. If the surface is described by a height function $z = h(x, y)$, the increase in the concentration of implanted atoms $C(x, y, z)$ with time t is given by⁴¹³

$$\frac{\partial C}{\partial t} = \iint F_{\text{beam}}(\xi, \eta, t) f(x - \xi, y - \eta, z - h(\xi, \eta)) d\xi d\eta. \quad (2)$$

For practical reasons, $f(x, y, z)$ is often approximated by a Gaussian function whose parameters can be determined by a BCA simulation. For an unscanned beam, the beam flux $F_{\text{beam}}(x, y, t)$ is constant in time, for a digitally scanned beam it is piecewise constant, and for an analog scanned beam it is a continuous function of time.

Displaced atom and deposited energy distributions can be calculated in a similar way. They may be of interest for estimating the damage to the target, and as input for solving the heat transport equation, respectively.

Sputtering causes the surface to recede. The resulting surface velocity in the direction of the surface normal is given by

$$v_{\perp} = \Omega F, \quad (3)$$

where Ω denotes the atomic volume of the target atoms and F is the flux of atoms through the surface element around the point under consideration. In the absence of secondary effects, F is given by the flux of sputtered atoms, $F = F_{\text{beam}} Y(\theta) \cos \theta$, with θ being the local incidence angle of the ions and $Y(\theta)$ being the sputter yield. The sputter yield may be determined by BCA simulations or experiment.

As long as an initially flat surface, $h(x, y, t = 0) = 0$, develops only small slopes, the surface contour reflects the beam profile

$$h(x, y, t) = - \int_0^t v_{\perp} dt = -\Omega Y(0) \int_0^t F_{\text{beam}}(x, y, t) dt. \quad (4)$$

Once larger slopes occur, further prediction of the target topography usually requires iteration⁴¹⁴ or more refined numerical techniques. One option is to move surface marker points along the surface normals.⁴¹⁵ The drawback of this method is that as the nodes move, they tend to form loops which have to be removed. A simple technique to avoid loops, which works for surfaces that remain as height functions $h(x, y)$ throughout the simulation (i.e., the surface has no overhanging structures), is to interpolate the surface back to an (x, y) grid after each time step.⁴¹⁶ More general solutions are cell-based and level-set methods. The former represent the target by filled or partially filled voxels, while the latter describe the surface as the zero level-set of a distance function and solve a differential equation for this quantity.⁴¹⁵ Implementations of the marker and level-set methods for FIB milling have been presented in Refs. 417–422 and Refs. 298 and 423–426, respectively.

When the surface develops larger slopes, two additional effects come into play (see Fig. 13). First, sputtered atoms may be redeposited on other regions of the surface (dark blue arrow). Second, ions may be backscattered from their primary incidence point and impact the target somewhere else (red arrow), where they cause sputtering. The secondary sputtered atoms may then be redeposited (light blue arrow). These processes add significantly to the complexity of the simulation: the angular (and possibly energy) distributions of the sputtered and backscattered atoms (blue and red elliptical lines, respectively) have to be described in some way, and the computation of the fluxes at each marker point (D_1 and D_2) requires the summation over the contributions of all other marker points (S_1 and S_2).^{232,421,422} This summation may be substituted by randomly starting individual rays at the source points, which adds a MC aspect to the simulation.^{423,427} In contrast to a BCA simulation, however, the atom trajectories are not followed within the target, but their exit characteristics are chosen according to precalculated angular (and energy) distributions. Either way, redeposition and backscattering add contributions to the flux F of Eq. (3). The same applies to contributions from etching and deposition in the case of gas-assisted processes (see Sec. III D 2).

In principle, many physical and chemical effects may be considered in a continuum approach. However, when a new ion-material combination is considered, each model requires input from more fundamental simulations or experiments. This may be straightforward for the parameters of the most basic models, such as the mean values and standard deviations of implanted ions or the sputter yields. However, as the model complexity increases, parameter determination becomes more cumbersome.

Our discussion so far has neglected dynamic changes to the target composition. As the irradiation proceeds and ions are implanted, however, the stopping power for the ions changes, and so does the point response $f(x, y, z)$ in Eq. (2). The same applies to sputter yields and angular distributions. Consideration of dynamic effects would also require modeling of atomic mixing. This has not been attempted so far in a multidimensional continuum approach. Moreover, sputtering is a nonlocal effect, i.e., the sputtered atoms are not emitted exactly at the

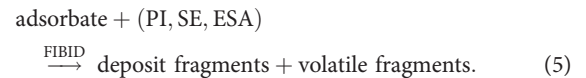
impact point of the ion, which may be important in nanofabrication. In addition, ion implantation and atom relocation may cause topography changes through volume relaxation (cf. Sec. III A). These effects are building blocks of the theory of spontaneous pattern formation⁴²⁸ but have not been considered in continuum simulations of FIB processes so far.

Another effect that can be treated within a continuum approach is beam induced heating. Temperature distributions can be calculated by solving the heat transport equation with heat sources given by the energy deposition of the collision cascades, using finite element or finite difference methods.^{429,430} The estimates are important because elevated temperatures may reduce deposition rates in FIBID (see Sec. III D 2) and may lead to heat damage in biological samples (see Sec. III D 3).

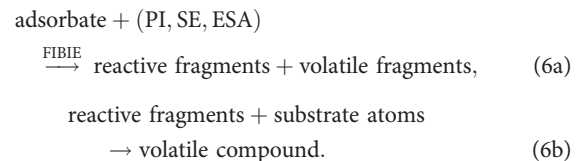
2. Gas-assisted deposition and etching

Gas assistance enables selective 3D nanoprinting and enhanced etch rates in FIB processing. Continuum models attempt to quantify the local deposition or etch rate caused by the chemical surface reaction between the volatile adsorbates and the primary ions (PIs), their locally generated SEs, and excited surface atoms (ESAs), as follows.

Deposition reaction:



Etch reactions:



FIBID typically employs volatile metalorganic compounds to deposit metal-containing nanostructures.⁴³¹ For focused ion beam induced etching (FIBIE), the adsorbed precursor needs to dissociate upon irradiation into a reactive species that removes atoms from the underlying substrate by forming a volatile compound but does not leave traces of other dissociated precursor elements.

Present FIBID and FIBIE models treat the desorption of volatile fragments in reactions (5) and (6) as instantaneous and assume that the etch reaction (6b) is fast compared to reaction (6a), see Ref. 432 for a review. Both processes share the same formalism to calculate the surface coverage Θ (normalized concentration) of adsorbates, which is defined by the rate equation (in units of inverse time)

$$\frac{\partial \Theta}{\partial t} = \nu_{\text{gas}}(1 - \Theta) - \nu_{\text{des}}\Theta - \nu_{\text{dis}}\Theta + D\nabla^2\Theta. \quad (7)$$

The terms $\nu_{\text{gas}}(1 - \Theta)$ and $\nu_{\text{dif}} = D\nabla^2\Theta$ are adsorbate source terms describing gas and surface-diffusion transport, respectively, while the other two terms are adsorbate sink terms related to the rates of desorption and dissociation, ν_{des} and ν_{dis} , respectively, as schematically depicted in Fig. 14. This approach could be extended to multiple chemical pathways, as in models developed for FEBID and focused electron beam induced etching (FEBIE).⁴³³

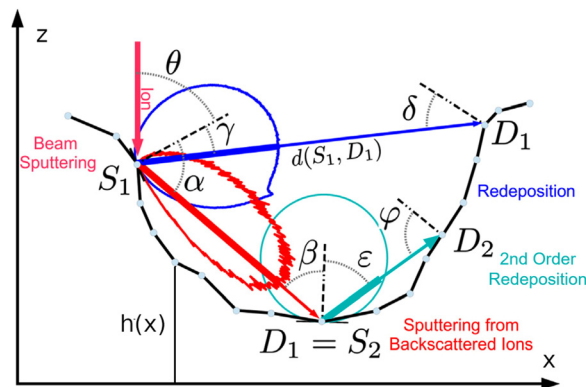


FIG. 13. Illustration of ion backscattering (red lines) and redeposition of sputtered atoms (dark and light blue lines) during FIB milling of a trench.²³² The ellipse-like curves represent angular distributions. In this case, the surface may be described by a height function $z = h(x)$. For clarity, no variation in y direction is assumed. Reproduced with permission from Lindsey *et al.*, Nucl. Instrum. Methods Phys. Res., Sect. B 341, 77–83 (2014). Copyright 2014 Elsevier.

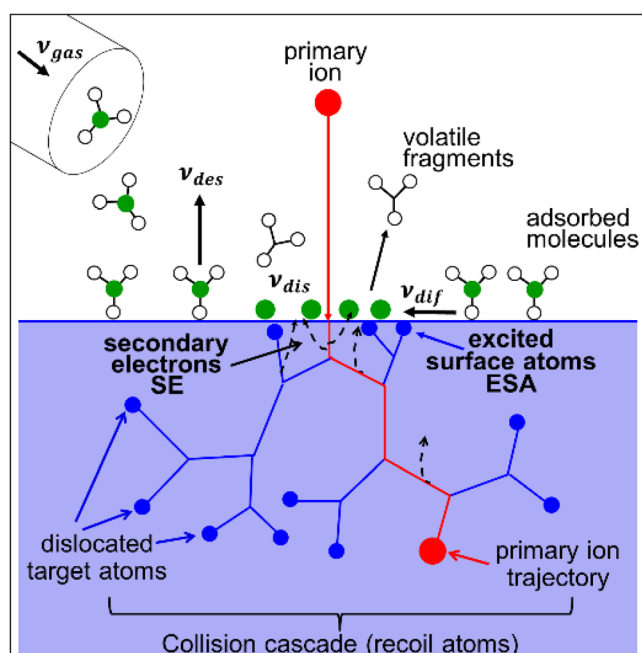


FIG. 14. Concept of selective (local) gas assisted FIBID. Volatile physisorbed molecules are dissociated by irradiation into volatile and nonvolatile fragments, the latter forming the deposit (green spheres). Primary ions typically arrive at keV energy while ESAs and SEs have an eV energy spectrum. ESAs are generated by the collision cascade (recoil atoms) and have energies below the surface binding energy (not sputtered). Rates (ν) are explained in the text. Note that ion sputtering, ligand codeposition, and incomplete adsorbate dissociation have been omitted for clarity. For more information, see Ref. 434.

PIs, SEs, and ESAs are the co-reactants. The concentration of impinging PIs is given by the ion beam spatial profile $F_{\text{beam}}(x, y)$ and the FIB exposure time. Energy and spatial distributions $f(E, \vec{x})$ of SEs and ESAs result from the interaction of the PIs and the recoils with the etched substrate or growing deposit. $f(E, \vec{x})$ must be determined from more fundamental simulations. For SEs, BCA simulations of PIs and recoils combined with MC simulations of electron transport⁴³⁵ have been performed using a simple model for SE generation. More refined theories of SE generation also exist.^{436–438} For ESAs, BCA simulations reach their low-energy limit of applicability, since atom energies of around 1 eV can already dissociate the precursors. An alternative for simulations of ESAs would be MD. Recently, first attempts of MD were undertaken for FEBID where atomistic detail on metal grain size, distribution, and composition in the deposited metal-carbon material could be obtained.^{439,440} The treatment of adsorbate dissociation by electrons relied on experimentally measured fragmentation mass spectra and fragment ion yields for structurally similar molecules, which needed smart appropriate scaling. Desorption of the adsorbate fragments was approached using an external force field. Both approaches limit the predictive power of the simulations at present but show promise if appropriately refined in the future. Modeling of FIBID would require inclusion of the ESA and milling mechanisms into the simulation.

Finally, the chemical fluxes of deposition F_{dep} or etching F_{etch} (in units of atoms per area per time) are proportional to the dissociation rate ν_{dis} and the coverage of adsorbates Θ as shown

$$F_{\text{dep,etch}} = m\nu_{\text{dis}}\Theta n_s, \quad (8)$$

where m denotes the number of atoms deposited or etched per dissociated precursor molecule and n_s is the number of adsorbate sites per unit surface area. Note that the product $\nu_{\text{dis}}\Theta$ defines the rate of deposition or etching for a single site in units of inverse time. The flux according to Eq. (8) must be added to the flux of sputtered and redeposited atoms discussed in Sec. III D 1.

Modeling of the four terms on the right-hand side of Eq. (7) is key to the successful simulation of FIBID and FIBIE. The first term $\nu_{\text{gas}}(1 - \Theta)$ represents a widely used pragmatic simulation approach. It assumes non-dissociative Langmuir adsorption, which treats adsorption as physisorption with a maximum coverage of one monolayer of adsorbates. Other adsorption schemes include dissociative adsorption,⁴⁴¹ multilayer adsorption,⁴⁴² and chemisorption.⁴⁴³ Open source MC codes are available for the calculation of the gas impingement rate ν_{gas} on the irradiated area for a given geometrical arrangement of nozzle and substrate.^{433,444,445} Typical values are in the range $0.1 \lesssim \nu_{\text{gas}} \lesssim 10^3 \text{ s}^{-1}$.

The desorption rate ν_{des} , contained in the second term on the right-hand side of Eq. (7), is the inverse of the surface residence time. It depends exponentially on the temperature as well as on the type of adsorption occurring on the growing/etched surface being irradiated. The desorption rate can cover several orders of magnitude depending on the molecule–surface interaction.

The third term, the adsorbate dissociation by irradiation, is key for selective FIBID and FIBIE as it captures the crucial non-thermal irradiation-driven chemistry of the process. Different contributions involving PIs, SEs, and ESAs are involved in adsorbate dissociation, see Fig. 14. In principle, the dissociation rate can be calculated from the energy dependent dissociation cross section $\sigma(E)$ and the energy distribution $f(E, \vec{x})$ of the SEs or ESAs via

$$\nu_{\text{dis}}(\vec{x}) = \int_0^\infty \sigma(E)f(E, \vec{x}) dE. \quad (9)$$

$\sigma(E)$ describes the efficiency of reactions (5) and (6) as functions of the energy of the PIs, SEs, and ESAs. Such energy dependent calculation is hampered by the fact that so far reliable $\sigma(E)$ only exist for electron induced dissociation reactions of molecules in the gas phase, see, for example, Ref. 446. A pragmatic approach assumes that such gas phase $\sigma(E)$ can be linearly scaled to the net deposition (etch) cross section (see below). This approach omits the quenching of certain dissociation channels (fragment production) due to the adsorbate state situation.^{432,447–449} Simulations will thus not be accurate with respect to deposit composition, but the deposit (etch hole) shape can probably be adequately modeled. The only established energy dependent cross section for deposition conditions exists for the molecule $(\text{CH}_3)_3\text{Pt-CpMe}$ under electron irradiation,⁴⁵⁰ describing the observed dissociation of one volatile CH_3 ligand from the adsorbed molecule leading to a deposit of atomic composition $\text{Pt:C} = 1:8$.

So far, the ESA induced adsorbate dissociation reactions have not been studied, probably due to experimental issues in deconvolution of the naturally involved (secondary) electron contributions. Recent condensed phase studies using Ar-ion irradiation⁴⁵¹ point to the importance of the additional mechanical sputtering effect of the PIs, which can help remove more ligand fragments than SEs alone and thus achieve a higher metal content material. The elemental species, charge,

and energy of the ions open up a wide field of fragmentation chemistry with metalorganic or “etch” adsorbates, which is still unexplored, particularly with respect to missing $\sigma(E)$. Moreover, the energy distributions $f(E, \vec{x})$ in Eq. (9) are not easily obtained in a continuum approach. Therefore, while using Eq. (9) with a MC approach for SE generation and transport is a promising approach, for the ESA mechanism, one has to resort to simpler approaches, as is also often undertaken to describe the SE mechanism.

In a simpler approach, dissociation is generically modeled via the dissociation rate $\nu_{\text{dis}}(\vec{x}) = \sigma f(\vec{x})$, where $f(\vec{x})$ is the local flux of the PIs and σ is the so-called net deposition or etching cross section, which is deduced from fitting the shape of the deposit or etch hole. Such cross sections do not distinguish between the different contributions of PIs, SEs, and ESAs, nor do they distinguish between the fragmentation reactions. The modeled shape or rate can sometimes retroactively be attributed to a certain mechanism, when the yield and/or spatial distribution of either PIs, SEs, or ESAs matches the deposit or etch hole dimensions. In this way, the ESA dissociation mechanism was identified to be dominant for FIBID using noble gas ions heavier than He^{434,452} and the SE dissociation mechanism was identified as important for He-FIBID nanopillar growth.⁴⁵³ In those cases, $f(\vec{x})$ may be interpreted as the flux of ESAs or SEs. Experimental net cross sections (yields) obtained from ion and electron induced deposition are available for a few precursor molecules,⁴³¹ and deposit compositions upon electron irradiation were recently reviewed in Refs. 454 and 441.

The fourth term in Eq. (7), the surface diffusion rate, is proportional to the surface diffusion coefficient D and the second spatial derivative $\nabla^2 \Theta$ of the adsorbate coverage Θ . Thus, it becomes largest for intense and finely focused ion beams and can result in the delivery of considerable amounts of adsorbates to non-irradiated surface areas. Increasing temperature increases the surface mobility of the adsorbates exponentially.

The generation of local heat during ion irradiation can be an important factor influencing the outcome of a deposition process. Thermal contributions can alter the nonthermal irradiative dissociation pathways and exponentially change the diffusion constant (mobility) and residence times of the adsorbates on the surface. As an example, thermal effects are thought to be responsible for the bending of lateral nanowires as they grow.⁴⁵⁵ As mentioned in Sec. III D 1, the temperature distribution needed as input for temperature dependent models of mobility and residence time can be obtained by numerically solving the heat transport equation.^{429,430}

3. Applications to FIB processing

Simulations of FIB milling may be used to investigate the influence of beam shape and scan strategy on the final topography of the target⁴¹⁷ and to better understand the milling process. For instance, Lindsey *et al.*³⁰¹ investigated microtrench formation in trench milling and clarified the roles of redeposition and backscattering. In practice, often the inverse problem of determining beam and scan parameters for a desired geometry must be solved. In the simplest case, for low aspect ratio structures, this can be achieved approximately by choosing the pixel dwell times proportional to the desired depth at the respective pixel positions.²²⁸ For higher accuracy, a system of linear equations for the pixel dwell times must be solved in order to consider the effect of beam overlap.^{414,456} For higher aspect ratio structures, when redeposition and/or backscattering need to be considered, repeated numerical

simulations of the milling process must be performed until convergence to the desired structure is achieved. In practice, such structures are fabricated by applying variable dwell times across the scan area in multiple passes. Here, one can either choose to apply the same pattern in each pass²³² or define layers to be milled in each pass and determine individual dwell time profiles.²²⁹

Finite element and finite difference simulations of heat transport have been performed by Wolff *et al.*,⁴³⁰ who used them to verify a simple model for the estimation of target heating. Schmied *et al.*⁴⁵⁷ employed a similar numerical approach to find beam scanning strategies that minimize local thermal stress in soft matter.

Gas-assisted FIB processes have been treated in a continuum approach mostly to help identify in which regime the surface reactions (5) and (6a) take place, i.e., whether they are mass-transport-limited (adsorbate-limited), or reaction-rate limited (PI, ESA, and/or SE limited), or driven by desorption or surface diffusion. These regimes strongly influence shape fidelity, homogeneity, and resolution, all of which are of utmost importance for 3D nanoprinting/nanoetching approaches. A review of both time dependent and steady-state analytical solutions of Eqs. (7) and (8), using simplifying assumptions, is found in Ref. 432. This theoretical framework closely resembles the one used for gas-assisted focused electron beam processing (FEBID/FEBIE) with the exception of physical sputtering and redeposition. Hence, analytical solutions established for the analog electron-based FIBID/FIBIE variants straightforwardly apply. These include expressions obtained for steady-state surface diffusion and flat-top focused beam profiles,^{458,459} long freestanding rods,⁴⁶⁰ 1D time dependent solutions,⁴⁶¹ and expressions for lateral resolution⁴⁶² and etch/deposition gas mixtures.^{463,464} Analytical solutions without surface diffusion can be used in the PI, ESA, SE limited processing regimes (no concentration gradient of adsorbates due to high supply from the gas phase) or approximately when adsorbates cannot diffuse into the center of the FIB spot profile due their deposition at the perimeter. These conditions translate into very small FIB current and exposure (dwell) time.⁴⁶⁵ Without the surface diffusion term, analytical solutions are more easily

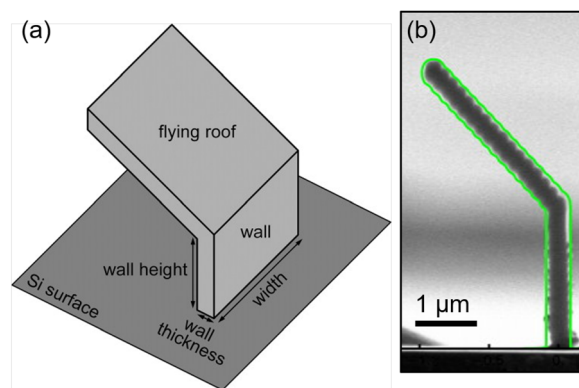


FIG. 15. Flying roof structure fabricated by FIBID.³⁰⁰ (a) Schematic illustration of the structure. (b) Side view SEM image and simulation (green line) of the structure obtained using a 30 keV Ga ion beam with a FWHM of 60 nm and pentamethylcyclotrisiloxane as the precursor. After wall formation, the ion beam was moved to the left in 16 steps of spacing 60 nm. Reproduced with permission from Ebm *et al.*, J. Vac. Sci. Technol., B 29, 011031 (2011). Copyright 2011 American Vacuum Society (AVS).

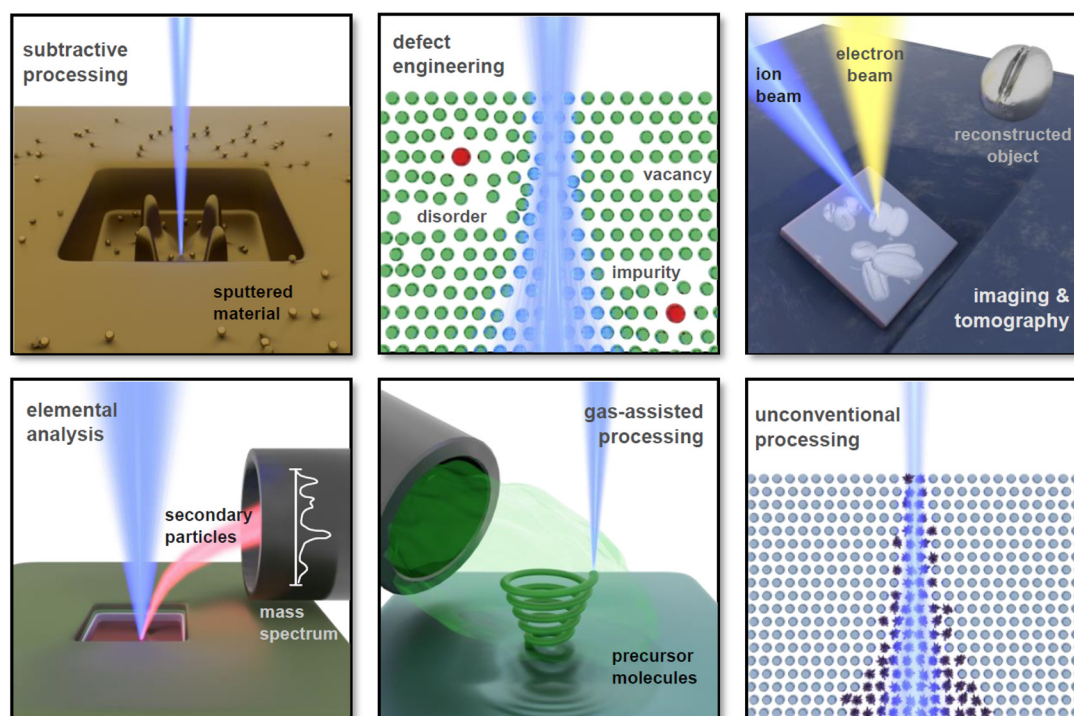


FIG. 16. Schematic overview of the different FIB techniques.

derived and thus cover a broader range of adsorbate-related dissociation mechanisms. These include the resulting deposit composition upon injection of two precursor species,⁴⁶⁶ multistep reactions involved in gas enhanced etching,^{433,467} and experimental parameter maps.⁴⁴²

Full numerical continuum simulations of a FIBID process using Eqs. (7) and (8) have been performed by Ebm *et al.*,³⁰⁰ see Fig. 15. In these simulations, an ESA-based model was used, assuming the concentration of ESAs to be proportional to the flux of sputtered atoms. The authors considered the nonlocal nature of sputtering, a prerequisite for obtaining parallel sidewalls in the lower part of the structure and the flying roof in the upper part. Simulations of pillar growth using voxel grids have been reported in Ref. 468.

Presently, simulations of FIBID/FIBIE are more advanced in the BCA-based approaches (cf. Sec. III A 3). One reason for this might be that pure continuum approaches require the modeling of the SE/ESA intensities $f(\vec{x})$ as an additional step. Also, BCA-based approaches include several collision cascade related effects, which cannot easily be modeled in a continuum approach (see Sec. III D 1). On the other hand, uncertainties in some of the assumptions of both the BCA and the continuum approaches might be more significant in practice. These include the manner in which the energy and spatial distributions of the ESAs and SEs are determined and assumptions about the fragmentation paths. Further development of either approach holds promise for model-assisted 3D nanoprinting, which has already been demonstrated for FEBID^{469,470} and has been started for FIBID.^{287,291,455}

IV. APPLICATIONS OF FOCUSED ION BEAMS

The various processes that are triggered when a beam of ions interacts with a sample form the basis of the multitude of FIB applications that are routinely used today. Accordingly, this section is intended to provide a comprehensive overview of this wide variety of experimental applications and is structured according to the different FIB processing techniques involved (see Fig. 16 for an overview).

In the following paragraphs, the breadth of FIB techniques is briefly outlined, to be highlighted in more detail in subsections IV A-F. Further reviews of these topics can be found in a number of recent articles.^{7,43,302,471,472}

Subtractive FIB processing: local material removal by a focused ion beam is based on physical sputtering. The incident projectiles not only knock out substrate atoms by momentum transfer but also penetrate the material in a collision cascade. While the ejection of atoms is desired here, all other interactions with the substrate can cause unwanted side effects such as amorphization, mixing, doping, dealloying, and others. Conventional Ga-FIB tools provide spatial resolution and sputtering yields high enough for multiple prototyping purpose, but other ion species may offer advantages in terms of increased spatial resolution or sputter yield, and reduced damage or chemical reactivity. Section IV A provides an overview of the rich variety of subtractive FIB milling applications.

Defect engineering: FIB irradiation of materials at doses significantly lower than those used for milling has been implemented in a range of applications to locally tune material properties through the introduction of defects (vacancies and implanted ions). For example,

by varying the dose, different concentrations of vacancy defects can be introduced into a crystalline material, thus modulating properties such as electrical, magnetic, and thermal behavior in the patterned regions. In the case of ion implantation, this can be directed at single ion implantation or can be used to modify the chemical and structural properties of surfaces for resistless patterning. An overview of defect engineering applications is given in Sec. IV B.

Imaging and tomography: just as is the case with the SEM, scanning the ion beam over a surface generates secondary electrons that can be detected to form an image. In this respect, imaging using a focused ion or a focused electron beam is equivalent. Even the same secondary electron detectors can be used. However, what strongly differs, is the mass of the probe. The mass of the ions is more than three orders of magnitude larger than that of the electrons, which significantly influences the contrast mechanisms and interaction volumes, and can also inflict significant damage on the sample. In this respect, the light ions of the He-FIB bring great benefit. Section IV C is dedicated to the application of the He-FIB specifically for the imaging of biological specimens. The FIB can further be used in a subtractive manner as described above to slice the sample and image the resulting cross sections by SEM. The images are then used for reconstruction of the full 3D information via so-called FIB-SEM tomography (see Subsections IV C 3 and IV C 4).

Elemental analysis: during ion beam induced physical sputtering, atoms and ions originating from the sample are ejected and can be used to study its chemical composition. In SIMS, the generated secondary ions are collected and separated by their mass in a spectrometer as detailed in Sec. IV D. Other analytical techniques, such as EDS or electron backscatter diffraction (EBSD), are of minor importance in conventional FIB systems because the generation rate of X-ray quanta is negligible, and the typical penetration depth of the conventional Ga ions is too low to generate sufficient information on lattice orientation.

Gas-assisted FIB processing: gases can also be locally introduced into the FIB chamber to take advantage of other processing opportunities. For example, a reactive gas can be introduced to enhance sputtering yields in subtractive processing by milling (see, for example, Sec. IV A 3). Alternatively, precursor gases can be selected to enable the highly localized deposition of material, which is referred to as focused ion beam induced deposition (FIBID). The flexibility of FIBID in terms of shape and spatial resolution (20–30 nm for Ga-FIBID, and 10 nm for He-FIBID) make it highly desirable for applications that require unique and precise planar or 3D designs. Top-down FIBID applications are discussed further in Sec. IV E.

Unconventional FIB processing: similar to gas-assisted processing, FIBs can also be used for the local chemical modification of lithographic resists and condensed or spin-coated metalorganic precursors. Such “unconventional” FIB processing can increase processing speeds and provide new chemical reaction pathways for ion induced deposition of high-purity metals. Focused ions can also be used to form self-assembled structures or combined with other micro-/nanofabrication techniques for advanced multiscale processing. Recent advances in the area of FIB based resist lithography and other nonstandard FIB approaches are described in Sec. IV F.

A. Applications of subtractive FIB processing

As material removal using a focused ion beam is local and direct, it offers extraordinary flexibility in terms of the target geometry to be

realized, as well as the substrate materials and their geometries. Often FIB processing can be monitored *in operando* using SEM for direct alignment and quality control. Table V provides an overview of the variety of applications based on subtractive FIB processing.

1. FIB for materials science

Arguably the most widespread application of the FIB has been the site-selective fabrication of cross section samples for high-resolution imaging using TEM.^{4,6,618–620} Due to the prevalence of Ga-FIB instruments, Ga ions are most typically employed and various strategies have been developed to mill and extract cross sections from bulk materials and to thin samples to electron transparency. This includes low-energy FIB milling under glancing incidence to minimize surface amorphization and the use of material specific capping layers and rocking stages to avoid “curtaining” artifacts. More recently, the use of other FIB species for TEM lamella preparation has been demonstrated such as the Ne-FIB⁴⁷³ and the Xe-FIB.⁴⁷⁴ The advantage of a noble gas ion species is that it does not chemically react with the sample or accumulate along grain boundaries, as can be the case with Ga.⁶²¹

In the same manner, FIB milling has emerged as an important tool for the creation of needle-shaped specimens for APT.^{5,477,622} The required needle geometry is achieved via annular milling with different currents and successively smaller milling areas. Similar considerations regarding beam damage apply as for TEM sample preparation. For example, minimizing the damage layer is important and possible indiffusion of, e.g., Ga atoms⁶²³ has to be taken into account. In this respect, a final polish with Ne-FIB has recently been shown to remove all residual Ga contamination from Ga-FIB-milled APT needle specimens.⁶²⁴ Performing the Ga-FIB milling at cryogenic temperatures has also been shown to limit Ga diffusion into the specimen.⁶²⁵

Another important application is the site-specific preparation of cross section cuts that enable *in situ* and *ex situ* imaging⁶²⁶ and/or other analysis such as EDS or EBSD (e.g., Ref. 497). By automating this process, full (destructive) tomographies can be performed by slicing the sample, extracting the desired information from each slice, and reconstructing the data in three dimensions (see, for example, Refs. 103 and 627). Tomography and its applications for the study of energy materials and, in particular, complex organic systems in the life sciences, are discussed in more detail in Sec. IV C 3.

All of these techniques are part of the standard materials science repertoire used to study the microstructure, composition, and properties of materials and devices at specific target locations. In depth overviews on these topics can be found, for example, in Ref. 628. One specific area in materials science that particularly benefits from the 3D sculpting capabilities of FIB concerns small-scale mechanical testing and is discussed in more detail in Sec. IV A 2.

2. FIB for mechanical testing

As technology miniaturizes, the external dimensions of device components approach critical internal length scales such as dislocation interspacings. The mechanical properties of such small material volumes are still not fully understood. In 2004, Uchic *et al.*⁴⁸⁰ suggested the use of FIB milling to machine micropillars to study size dependent mechanical properties of metals, exploiting conventional nanoindentation equipment for microcompression experiments. In contrast to

TABLE V. Overview of applications of subtractive FIB processing, in which material is locally removed by FIB based physical sputtering. Applications of FIB-SEM tomography are discussed in Sec. [IV C 3](#).

Research area	Application	Ion	Material/geometry/device
Materials science	TEM sample preparation	Ne	Al alloy and Si ⁴⁷³
		Ga	Wide range of materials ⁴
		Xe	Al and Si, ⁴⁷⁴ Al alloy ⁴⁷⁵
	APT sample preparation	Ne	Al alloy and AlO _x thin-film structure ⁴⁷⁶
		Ga	Wide range of materials ⁴⁷⁷
		Xe	Al alloy, ⁴⁷⁸ Ti alloy ⁴⁷⁹
	Mechanical testing	Ga	Micro/nanopillars, ^{480–489} notches and micro-cantilevers, ^{489–497} “dogbone” tensile bars, ^{491,498,499} suspended membrane beams, ⁵⁰⁰ other geometries ^{501–503}
		Xe, Ne	Micro/nanopillars, ⁴⁸⁸ trenches ⁵⁰³
Semiconductor technology	Mask repair	He, H, N	Molybdenum silicide photomasks ^{57,504–507}
		Ga	Cr-on-glass photomask ^{508,509} including deposition for repair of clear defects ⁵¹⁰ and gas-assisted etching for opaque defects in TiN EUV masks ⁵¹¹
		Au	Cr-on-glass photomask, ⁵⁰⁸ Au-on-Si X-ray mask including deposition and gas-assisted etching ⁵¹²
	Circuit edit	He, H, N	Complementary metal-oxide-semiconductor (CMOS) circuits ^{507,513}
		Ga	Decomposition of Pd acetate for conductive lines ⁵¹⁴
Quantum and magnetic materials	Superconductivity	Ga	Unconventional superconductivity in Sr ₂ RuO ₄ , ^{515,516} Josephson-vortices in strongly layered high-T _c superconductors, ^{517,518} vortex pinning, ⁵¹⁹ directional vortex motion, ^{520–522} high-T _c superconductivity ^{523–525}
	Superconductive circuitry	Ne	Nanoconstrictions and nanoSQUIDs in Nb ^{526–528}
		Ga	Nanoconstrictions and nanoSQUIDs in Nb, ^{529–536} NbN, ⁵³⁷ and YBCO ^{538–541}
		Xe	nanoSQUIDs in Nb ⁵⁴²
	Spintronics and magnonics	Ne	Permalloy microstrip trimming, ⁵⁴³ environment dependent FMR ⁵⁴⁴
	Magnetism	Ga	Magnetic reversal, ⁵⁴⁵ spin wave phase inverter ⁵⁴⁶
		Ga	Surface time-reversal symmetry breaking in delafossites ⁵⁴⁷
	Topology	Ga	Topological semi-metals, ^{548–550} Fermi-arc surface states, ⁵⁵¹ Kagome conductors ⁵⁵²
	Strain engineering	Ga	Sr ₂ RuO ₄ , ⁵⁵³ CeIrIn ₅ (Refs. 554 and 555)
Optics	Photonic components	O	Optical bridge in diamond ⁵⁵⁶
		Ga	Aspherical microlens in Si, ⁵⁵⁷ solid immersion lenses in Si ⁵⁵⁸ and around pre-localized quantum emitters in diamond, ^{224,559,560} photonic crystal cavity around quantum emitters in diamond, ⁵⁶¹ triangular nanobeam cavities in diamond ⁵⁶² and in YVO, ^{563–565} Fresnel zone plates for X-ray microscopy, ⁵⁶⁶ Fresnel lenses milled through hard mask, ⁵⁶⁷ metasurfaces for polarization and phase engineering, ⁵⁶⁸ tapered nanoantenna for single photon emission of CdSe quantum dots ⁵⁶⁹
	Fiber-based optical components	Ga	Lab-on-a-fiber, ⁵⁷⁰ Bragg gratings and microlenses in silica nanofibers, ⁵⁷¹ superconducting single-photon detector (SSPD) on optical fiber ^{572,573}
	Plasmonic components	He	Al bowtie antennae, ⁵⁷⁴ Au dimer antennae of different shapes, ^{575–581} Au coaxial ⁵⁸² and split-ball ⁵⁸³ and nanoslit ⁵⁸⁴ resonators, pores in Au pyramidal ⁵⁸⁵ and microsphere ⁵⁸⁶ cavities, Au oligomers ²²⁶ , inverse Au trimers ⁵⁸⁷ and oligomers ^{588,589}
		Ga	Au groove waveguides ⁵⁹⁰ forming logic gates, ⁵⁹¹ Au bowtie antennae for EUV light generation, ⁵⁹² inverse Ag bowties, ⁵⁹³ ring-shaped beam patterning of Au antenna arrays, ⁵⁹⁴ Yagi-Uda-type antennae, ^{590,595} $\lambda/2$ antennae ⁵⁹⁶ asymmetric Ag oligomers, ⁵⁹⁷ two-wire waveguides in Au ^{598,599} forming spin-optical nanodevice ⁶⁰⁰
	Custom probes and sensors	He	Square nanopores in pyramidal Au nanocavities, ⁵⁸⁵ Au nanopyramid antenna ⁶⁰¹
		Ga	Circular ⁶⁰² and bowtie ⁶⁰³ nanoaperture, tip-on-aperture geometry, ^{604,605} Au bowtie ⁶⁰⁶ and nanocone antenna, ⁶⁰⁷ nanoslit grating in Au conical taper ⁶⁰⁸
		Xe	Probe shaping for near-field scanning microwave microscope ⁶⁰⁹

TABLE V. (Continued.)

Research area	Application	Ion	Material/geometry/device
	Atomic force microscopy	Ga	Nanoscale apertures for liquid and ion transport, ^{610–612} tip sharpening, ⁶¹³ modified mechanical response ⁶¹⁴
	Magnetic force microscopy	Ga	V-shaped magnetic nanostructure at probe tip ⁶¹⁵
	SQUID microscopy	Ga	SQUIDs on AFM cantilever ⁵³²
	Mechanical high-Q resonators	He	Phononic crystal ⁶¹⁶
		Ga	Graphene trampoline ⁶¹⁷

conventional cleanroom-based lithography and etching, FIB machining allows for virtually any material to be shaped into a micropillar at a specific target location. To achieve uniform pillar circumferences with minimal taper, lathe milling in combination with annular milling is often applied.⁴⁸¹ Here, lathe milling refers to a particular FIB approach in which the sample is rotated stepwise, and the ion beam performs a line scan along the rim of the pillar. While lathe milling can introduce severe levels of stress requiring careful interpretation of the micromechanical testing results,⁴⁸⁶ it is well suited for insulating or very inhomogeneous samples comprising layers with very different sputter yields.

Today, most small-scale mechanical testing experiments are performed by SEM or TEM to capture changes from the atomic to micrometer scale (e.g., formation of dislocations, stacking faults, twins, grain orientations, contraction and expansion of volume). Complex workflows include some or all of the following steps: (a) machining of micro-/nanopillars and other geometries, (b) *in situ* mechanical testing under SEM observation, (c) deposition of a regular pattern of points by FIBID (or FEBID) to follow deformations via digital image correlation algorithms, and (d) postmortem FIB slicing for 3D tomographic or 3D EBSD analysis of the deformed microstructure or the fracture pattern inside the sample.⁴⁹⁷ Popular types of geometry used for mechanical testing are shown in Fig. 17. For example, FIB machining of sharp notches with a specific orientation to the loading axis has enabled a number of fracture mechanics studies,⁶²⁹ and for *in situ* TEM tensile testing, various milling protocols have been developed to produce electron transparent specimens.⁴⁹³

To reduce ion beam induced surface damage defects, a multistep milling strategy is typically employed. In this approach, the beam voltage and current are sequentially stepped down to the last polishing step.⁴ Among the still open questions are how much damage (amorphization, implantation) is induced by FIB milling into the structures and how the damaged zones affect the mechanical behavior of the milled structure.⁴⁸⁴ Key factors are the penetration (implantation) depth of ions and the collision cascade (amorphization) they trigger (see Secs. III A 3, III B 3, and III D 1).

Once other ion sources, in particular, He, Ne, and Xe ion sources, became available, the hope was to be able to circumvent some of the Ga damage issues. Best *at al.*⁴⁹⁶ found, however, that using either He, Xe or Ga for the machining of notches affected the measured fracture strength in all cases. While He ions were shown to yield suitable sharp notches with radii below 10 nm, Xe ion beams were

resolution limited. It was unclear how the penetration depth of the different ions affects the fracture toughness. Currently, combinations of FIB milling with other technologies are becoming more and more popular. For example, the option of laser machining of large volumes followed by FIB fine sculpting has recently become available from several manufacturers in a single setup.

3. FIB in the semiconductor industry

In the field of semiconductor processing, FIBs are most relevant during the debugging of new products and for identifying production errors in the context of circuit edit and fault isolation, as well as for the repair of photomasks. Among other applications, FIBs have been used for failure analysis, fault isolation, TEM sample preparation, mask repair, and backside circuit edit.^{628,630,631} The ongoing scaling of the transistor fin, gate, and interconnect via linear and volumetric shrinking, as well as the introduction of new transistor technologies, has led to a reduction of the critical length scale from 22 nm in 2012 to the current 3 nm technology node to be introduced by all major manufacturers between 2022 and 2024.^{632–634}

FIBs have been used since the 1980s for mask repair, initially using Ga and Au ions from LMIS and LMAIS.⁵⁰⁸ However, staining from implanted ions was quickly identified as a problem limiting the application.⁵⁰⁹ Later, the application of GFIS based H⁺, H₂⁺, N₂⁺, and He⁺ for phase-shift mask and extreme ultraviolet (EUV) mask repair was investigated.^{57,505,506} For H₂⁺, the minimal repairable dimension is reported to be 11 nm, which is half of what could be achieved using electron beams.⁵⁰⁴ The application of FIBID was investigated early on for adding missing material to masks and for repairing pin holes using Ga-FIB induced carbon deposition.⁵¹⁰ This has been extended using other deposition precursors. The addition of reactive gases such as Cl₂,^{511,635} Br₂,⁵¹¹ XeF₂,^{636,637} and O₂⁶³⁷ has also been investigated for mask repair. These gases can significantly increase etch rates, remove carbon contamination, and trigger resist development. In particular, XeF₂ is now readily available in commercial FIB systems.

For circuit edit and fault isolation applications, Ga-FIB was used initially,⁵¹⁴ but later Ne ions from GFIS instruments were employed. The latter yielded minimal trench widths of 14 nm for high aspect ratio vias,⁵⁰⁷ and comparable/less collateral damage compared to Ga-FIB as measured from changes to ring oscillator timing.⁵¹³ Recently, GFIS-based H ions have been demonstrated for successful delayering.⁶³⁸ Other source technologies, e.g., Cr ions from MOTIS,⁶³⁹ also show potential for circuit edit applications.⁶⁴⁰ In general, it is found that

lower beam energies are required while still maintaining a suitably high spatial resolution.

To better understand the effects of the beam profile on the achievable mill resolution, approaches have been developed to elucidate the probe current distribution.^{641,642} This is of particular importance for via fabrication and also for the fabrication of high aspect ratio holes/pores created by milling with a stationary beam. For these applications, a good signal to noise ratio is needed for reliable end point detection, and a new figure of merit—SE yield per sputtered atom—can be defined. GFIS-based He and Ne milling of these structures promises superior spatial and temporal precision compared to Ga-based FIB processing.⁶⁴³ However, GFIS-based milling suffers from the formation of gas bubbles and the creation of defects far below the milled surface of bulk samples.⁶⁴⁴

4. FIB for quantum materials

“Quantum materials” are materials in which quantum phenomena manifest in non-trivial ways, leading to response functions that are qualitatively distinct from those of standard electronic materials such as Si or Cu. These functions generally arise from the non-locality of quantum systems, with topology, frustration, and correlation all playing a central role.^{645,646} The key goal is to refine our microscopic or phenomenological understanding of these exotic quantum states and develop predictors to find promising materials candidates within the vast space of possible compounds. A second main driver of quantum materials research is novel electronic applications that harness this exotic behavior for solving technological challenges. Most famous is quantum information technology, including quantum computation, which directly utilizes the entanglement and vast configurational space of quantum coherent systems. The search for the most effective materials platform to realize systems like qubits remains an open and very active field, i.e., the search for the “silicon” of quantum technology. Beyond coherent information processing, a large materials space has developed in sensing and data communication that utilizes quantum coherence indirectly, such as in strongly correlated oxides. The ubiquity of quantum mechanics is the greatest challenge, in that interdisciplinary efforts in materials science, chemistry, physics, engineering, and informatics are required to realize these concepts in a useful way. In this search for new physics and new quantum platforms, the FIB excels as a versatile patterning tool that is ideal for rapid processing of nonstandard materials,⁶⁴⁷ as it (a) avoids exposure to wet chemistry as compared to resist-based lithography, (b) can operate with crystallites and microparticles, i.e., nonstandard samples that do not resemble polished wafers, and (c) enables 3D structure control that is critical to many electronic 3D quantum materials. The operational goal of this research is to fabricate hybrid structures that join micro-/nanostructures of a quantum material to a Si chip of conventional architecture, both to demonstrate advanced functionalities and to probe the underlying novel physics.

Quantum systems generically respond strongly to changes in their boundary conditions, in this case, their physical shape. FIBs enable the sculpting of non-trivial boundary conditions in crystals, which has been used to study and functionalize superconductivity, e.g., investigating the unconventional superconductivity of Sr_2RuO_4 ,^{515,516} and studying Josephson vortices in strongly layered high- T_c superconductors and their excitations.^{517,518} Controlled patterns guide the motion of Abrikosov lattices by surface structuring⁵²⁰ toward non-

trivial electronic responses such as vortex diodes.^{521,522} Further toward superconducting applications on small dimensions, FIB milling also enables direct nanopatterning of Josephson junctions, nanowires, and nanoSQUIDs.^{648,649} The nanoSQUIDs are used for magnetization reversal detection of individual magnetic nanoparticles^{529,540} and for magnetic and thermal sensing in scanning SQUID microscopy.⁵³² In microwave circuits, nanoSQUIDS can be used as tunable inductors, e.g., to enable tuning of the resonance frequency of microwave resonators.⁵²⁶

Beyond superconductors, other exotic materials and properties can be probed, such as quasiparticle transport in ultra-clean metals,⁵⁴⁷ and the magnetic exchange bias in time-reversal symmetry breaking topological semi-metals.⁶⁵⁰ Reducing the sample size to the micrometer scale further enhances the contribution of surface properties to the overall physics. This is particularly important in the quest to probe the contribution of topologically protected surface states, which have been proposed to enable novel electronic and spintronic applications in topological semi-metals.^{548–551} Precise FIB machining also enables optimization of the edge quality of structures, which is a prerequisite for the fabrication of efficient superconducting single-photon detectors (SSPDs).⁵⁷² In addition, the flexibility of FIB processing on highly 3D substrates has enabled FIB milling on top of an optical fiber.⁵⁷³

Subtractive FIB processing has also been applied in the fields of spintronics and magnonics. Examples include guiding and control of propagating spin waves, which has been enabled using the Ga-FIB.⁵⁴⁶ High resolution metal free FIB processing using GFIS ion species has also enabled fine tuning of the interaction of spin waves across gaps.⁵⁴³

Not all applications of FIB aim to alter the material. In addition, there is a wide class of FIB milling experiments aimed at probing the bulk properties of novel materials that can be difficult to access when the material is in its native form, e.g., microcrystals that are too small for conventional measurements. This has allowed insights into microscopic effects on critical charge transport in Fe-based superconductors,^{523,524} the determination of topological bandstructures in microcrystals,⁵²⁵ and probing of the switchability of anti-ferromagnetic states under the high current densities achievable in FIB-machined crystals.^{651,652} The rich physical responses of quantum materials tend to be rooted in the competition between multiple electronic ground states, which leads to a strong (and potentially discontinuous) change in their properties under weak applied stimuli.⁶⁵³ One can hardly overstate the importance of residual stress in the samples, with a famous example being the enigma of the “3K phase” of superconductivity in Sr_2RuO_4 , which originated from strain mismatch.⁵⁵³ Here, a crucial advantage of FIB milling over mechanical sample preparation techniques, such as polishing or wire sawing, is its kinetic nature, which results in samples of low residual stress. Precision control over the physical shape of a sample has also enabled microscopic control over desired strain fields and their gradients, which imprint correlation landscapes into quantum materials.^{554,555}

5. FIB for photonics and plasmonics

Miniaturized on-chip photonic components are designed to guide or focus light while having a footprint as small as possible. The fabrication of these relies mostly on conventional large-scale lithographic techniques, but there are applications where a maskless direct-write strategy has significant advantages. Prime examples are localized light

sources, with the extreme case of single photon sources, where the optical component often has to be built around a pre-localized (quantum) emitter. Such quantum emitters may be defects in wide-bandgap materials (see Sec. IV B 4), where outcoupling of the generated quantum light is typically hindered by the large refractive index of the host material, e.g., diamond. Here, FIB milling either using Ga ions^{224,559,560} or, to reduce Ga-induced damage, O ions,⁵⁵⁶ has been successfully used to engrave solid immersion lenses that enhance photon collection efficiencies. In the same manner, photonic crystal cavities with small mode volumes have been milled into a suspended diamond layer around pre-localized quantum emitters and proven to enhance spontaneous emission.⁵⁶¹ An even further reduction of optical mode volume can be achieved using metallic nanostructures, since these have the ability to concentrate light below the diffraction limit by collective excitations of the free electron gas.⁶⁵⁴ Since such plasmon-polariton modes are strongly dependent on geometry, FIB processing is a powerful tool for rapid prototyping of nanoscale resonators, sub-wavelength apertures,⁶⁵⁵ and optical antennae.⁶⁵⁶

Furthermore, extreme plasmonic field confinement allows the detection of molecular species down to the single molecule level by surface enhanced Raman scattering⁶⁵⁷ (the prime example), enhanced fluorescence,⁶⁵⁸ or enhanced Förster resonant energy transfer.⁶⁵⁹ The latter two examples employ Ga-FIB milling to fabricate an antenna-in-a-box geometry, where the gap region of a dimer antenna provides the local electric field enhancement and correspondingly an enhanced local density of optical states. The remaining metallic film around the antenna blocks fluorescence from contributing to the background. In general, Ga-FIB top down patterning, especially of single-crystalline Au or Ag nanoflakes, is often employed for high-fidelity rapid prototyping of plasmonic components.^{599,660–663} When comparing the performance and material properties of plasmonic antennae, electron beam lithography (EBL) may lead to better results than FIB milling, depending on the actual geometry and processing conditions, especially in the case of Ga-FIB.⁶⁶⁴ However, electron beam lithography (EBL) fails in fabricating truly 3D profiles. In contrast, FIB milling has been used to, e.g., mill out 3D Janus plasmonic helical nanoapertures for polarization-encrypted data storage.⁶⁶⁵

A particularly beautiful example of 3D FIB-milling capabilities combines local material removal by Ga-FIB milling of a suspended gold film with low dose Ga ion irradiation at selected edges. The latter induces a stress distribution that leads to up- or downward bending of the predefined structures.⁶⁶⁶ This nano-kirigami approach has been demonstrated for various designs realizing optical chirality, polarization and phase engineering, and showing potential for optomechanical applications.⁵⁶⁸

The resolution and shape fidelity of FIB milling especially with light ions stands out compared to any other lithographic techniques^{226,471} but comes at the cost of the slow speed of the inherently serial process. To resolve this issue, lithographic large area processing by EBL can be combined with, e.g., He-FIB for the ultimate in resolution. Extremely narrow gaps between large areas of lithographically defined plasmonic Au⁵⁷⁷ and Al⁵⁷⁴ antennae have been realized using this approach. By combining Ga-FIB patterning with He-FIB patterning, the achievable spatial resolution can be significantly improved while keeping a reasonable processing speed.⁵⁷⁵ Another approach to increase the speed of top-down FIB processing relies on milling only the outlines of desired plasmonic features, which are then subsequently

obtained by peeling off the surrounding material.^{579,667,668} This “sketch and peel technique,” combining Ga-FIB patterning for the outlines with He-FIB for the definition of ultra-narrow gaps, was used to reproducibly demonstrate strong coupling in plasmonic dimer antennae.⁵⁸⁰

6. FIB fabrication of custom probes

Since FIB milling is not restricted to planar sample geometries, a vast body of research exists on the modification of optical fibers,⁵⁷⁰ nanofibers,⁵⁷¹ and nano-probes. The comprehensive review by Sloyan *et al.*⁵⁷⁰ presents examples for optical near-field imaging, plasmonics, beam shaping, fiber-based photonic cavities, fiber-chip coupling, and many more. He-FIB manufacturing was reviewed in detail by Allen.⁴⁷¹ Direct control over precise nanostructure geometries by FIB milling is especially interesting for near-field optical microscopy (NSOM), where a metal coated glass fiber tip localizes the excitation and/or collection of light by making use of a nanoaperture or antenna for plasmonic field concentration. In applications concerned with NSOM, FIB milling has been employed to fabricate high-definition plasmonic circular nanoapertures at the apex of a metal coated glass fiber tip.⁶⁰² Further signal enhancement and fluorescent background reduction can be achieved by adding a tip at the aperture rim, combining the concepts of scattering and aperture NSOM.^{669,670} While the tip was grown by FEBID in these works and metalized afterward, later work engraved the tip-on-aperture geometry into the metallic (Al) fiber coating by Ga-FIB milling and tuned the tip resonance to behave as a monopole antenna, thus achieving single molecule imaging⁶⁰⁴ and imaging of proteins on cell membranes.⁶⁰⁵ Additional geometries realized in this

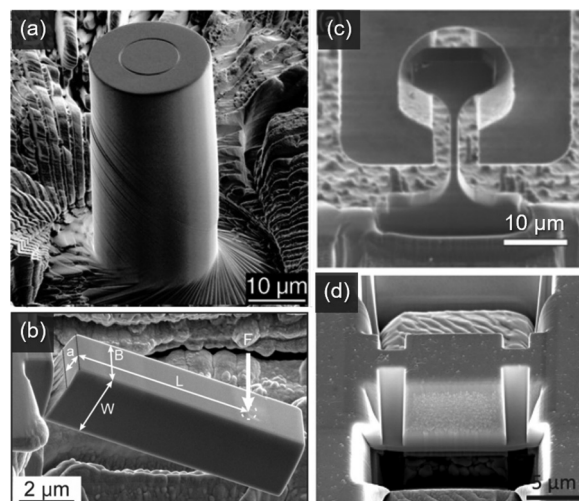


FIG. 17. Popular FIB-machined sample geometries for micromechanical testing: (a) micropillar⁴⁸⁰ reproduced with permission from Uchic *et al.*, *Science* **305**, 986–989 (2004). Copyright 2004 The American Association for the Advancement of Science. (b) Notched micro cantilever.⁶²⁹ Reproduced with permission from Ast *et al.*, *Mater. Des.* **173**, 107762 (2019). Copyright 2019 Authors, licensed under a Creative Commons Attribution (CC BY) license). (c) Micro-tensile bar⁴⁹⁹ reproduced with permission from Casari *et al.*, *J. Mater. Res.* **34**, 2517–2534 (2019). Copyright 2021 Authors, licensed under a Creative Commons Attribution (CC BY) license and (d) micro-shear test structure⁵⁰² reproduced with permission from Heyer *et al.*, *Acta Mater.* **62**, 225–238 (2014). Copyright 2014 Elsevier.

way are bowtie nanoapertures on the end of metal-coated fiber tips⁶⁰³ and square plasmonic nanoapertures in pyramidal cavity structures,⁵⁸⁵ with the latter example employing He-FIB milling.

Further specific geometries of metallic tips and metal-coated fiber tips have been fabricated to control the plasmonic response for other purpose. These include nanocones for resonance and radiation pattern tuning by Ga-FIB milling,⁶⁰⁷ nanopillars for optimizing local electric near-fields by He-FIB milling,⁶⁰¹ and nanoslit gratings into the conical taper of metallic tips for nanofocusing upon far-field illumination, again by Ga-FIB milling.⁶⁰⁸ FIB milling has also been employed to modify AFM probes for specialized applications. In one example, FIB-milling was used to partially remove the magnetic layer from a commercial magnetic force microscopy (MFM)-probe. The V-shaped nanostructure remaining on one side of the triangular pyramid results in enhanced magnetic phase contrast without compromising spatial resolution.⁶¹⁵ Another example employs nanoscale apertures milled into hollow pyramidal AFM tips that can be loaded with liquid to dispense attoliter volumes and create droplet arrays.⁶¹⁰ Such tips can also be used for the manipulation and analysis of single cells via so-called fluidic AFM.⁶¹¹ Tips can be sharpened using FIB milling and can also be detached from their initial substrate and mounted onto a cantilever of choice with the help of a micro-manipulator. This technique has, for example, been used to produce customized tips with high aspect ratios.⁶¹³ In other works, the AFM cantilevers themselves have been fashioned with FIB-milled apertures to enable ion implantation through those apertures aided by scanning probe alignment.⁶¹² FIB milling of larger patterns into piezoresistive cantilevers has been used to increase deflection sensitivity to optimize cantilever performance.⁶¹⁴ Recently, Ga-FIB milling has also been used for shaping AFM tips to produce mesa structures on the cantilever onto which Nb nanoSQUIDs have been fabricated. Such structures simultaneously enable topographic, magnetic, and thermal imaging.⁵³²

7. Outlook

As an all-in-one processing tool, the FIB provides a versatile platform for flexible and fast top-down fabrication and prototyping of nanostructures and devices.

To further advance these developments, FIB technology needs improvement not only in spatial resolution but also in the *in situ* control of the material removal process. Critical issues include material sputtering selectivity and increasing the signal to noise ratio of SE monitoring during the milling of high aspect ratio structures. However, FIB top-down nanofabrication is intrinsically limited by the fact that the energetic particles and the chemical species involved cause structural and chemical damage or changes, such as amorphization or poisoning. While traditional Ga-FIB still represents the FIB standard, some of the aforementioned fundamental limitations may be resolved by using other ion species, such as those offered by GFIS, MOTIS, LoTIS, PFIB, or LMAIS (see, e.g., Sec. II A).

Additionally, scale-up of FIB based top-down nanofabrication will only be achieved when increased throughput and reproducibility are realized. Here, advances in software and hardware tools for, e.g., modeling-informed automated operation will be beneficial, also allowing in-line integration of FIB processing into larger (nano)-fabrication pipelines. Theoretical knowledge, modeling, and simulation tools that provide precise descriptions of FIB modifications to materials, from the atomic to the microscopic scale, are a prerequisite for predictive

design and for the realization of tailored properties and functionalities of nanostructured surfaces and operating devices.

There are also opportunities for coordinated FIB system development based on specific experimental needs. For example, strong interest in emerging fields such as quantum technology, as well as novel miniaturized platforms in the fields of microfluidics for lab-on-a-chip or organ-on-a-chip developments, would benefit from synergies with theory groups and tool makers in the FIB application community.

B. Engineering materials properties through defects

In addition to the subtractive nanofabrication applications described above, site-specific and dose-controlled irradiation of samples with the FIB is used to tune material properties in spatially defined regions down to the nanoscale and even down to the single-atom level, without surface erosion or deposition. These applications are based on the local introduction of varying concentrations of defects (vacancies, dopants, and dopant–vacancy complexes), controlled by the dose.

1. Basics of irradiation induced defect creation

A common technique used to create defects in crystals relies on bombardment with high energy ions or electrons,^{246,671,672} and in some cases photons, protons, or neutrons.⁶⁷³ For example, tuning the fluorescence properties or the refractive index of a material for optical applications by bombardment with ions is a well-established technique.^{674,675} Ion implanters are often employed for such applications. Since these deliver broad beams with spot sizes in the millimeter regime, a lithographically defined mask is needed when higher spatial resolution is required. However, the large collision cascades that are inherent to both high energy ion beams and light ion bombardment make mask design rather challenging. This is where FIB processing comes into play.

As described in Sec. III, the incident ion induces a collision cascade in the solid with a size that is determined by the target material and the ion's kinetic energy, as defined by the primary beam voltage. In a simplified picture, when a recoil is generated, it leaves behind a vacancy and forms an interstitial when it comes to rest.⁶⁷⁶ In reality, defect clusters and amorphous pockets may form depending on the target material⁶⁷⁷ and on the ion mass and energy.⁶⁷⁸ Larger defect structures such as these are usually undesirable for defect engineering, as their properties are less well defined than those of simple defects. They are best avoided by using light ions, which cause more dilute collision cascades than heavier ions due to the lower nuclear stopping power. The recoils in these cases are also less energetic because of the unequal masses of the ion and target atoms. In addition, light ions cause less sputtering, which is generally an unwanted effect in defect engineering. Regardless, the applied ion doses and beam parameters still have to be carefully adjusted. Considering the example of He ion irradiation of Si and Cu substrates, Livengood *et al.*⁶⁴⁴ differentiated the various defect regimes according to dose, ranging from individual point defects at low fluence, to extended subsurface microbubbles at very high fluence. While the picture will vary somewhat for other ion–substrate combinations, the same overall trends apply.

Table VI gives the reader a flavor of the breadth of FIB based defect engineering applications, categorized according to the specific materials property engineered (electrical/electronic, magnetic, optical, quantum, chemical, thermal, and mechanical). Further examples can be found in the review articles by Gierak,⁶⁷⁹ Orús *et al.*,⁶⁸⁰ and Allen.⁴⁷¹

In the remainder of this section, we focus on those applications in Table VI that are concerned with engineering the electronic, magnetic, and optical/quantum properties of materials, with additional detail on property engineering of low-dimensional materials (in particular, thin films and 2D materials). The study of irradiation damage mechanisms (e.g., the formation of gas nanobubbles) is also discussed.

2. Engineering electronic properties

From the outset, FIBs have been used to fabricate electronic devices at the nanoscale. Early examples, using Ga-FIB milling, are the fabrication of GaAs quantum wires⁶⁹³ and a quantum wire transistor.⁶⁹⁵ More recently, however, Ga-FIB irradiation at lower dose was employed for the fabrication of a lateral diode based on a 2D heterostructure of MoSe₂ and graphene.⁶⁹⁹ Here, the preferential sputtering of Se locally transformed the top MoSe₂ layer into a quasimetallic state, generating a pn junction with rectification performance comparable to that of a vertical diode. These examples illustrate a development from the prevalence of subtractive materials modification by FIB milling to increased interest in engineering the electrical and electronic properties of materials at lower dose through defects.

Here, light (and often preferably inert) ion beams are of relevance due to their high resolution defect engineering capabilities. Ion beams of electronically active elements (e.g., P, Sb, As) also have the potential for interesting applications such as local doping with high spatial resolution or the generation of qubits by SII. For applications in superconductivity, Josephson junctions, i.e., weak links between two superconducting electrodes, are key elements for superconducting electronic devices. For Josephson junctions based on conventional metallic superconductors, such as Nb or Al, thin-film devices with grown insulating (I) or normal conducting (N) barriers, sandwiched between superconducting (S) electrodes, are well established. However, in the case of the high- T_C cuprate superconductors this as-grown approach does not produce well-defined SIS or SNS junctions, due to the complex nature of these materials and their strong sensitivity to defects on the atomic scale. As an alternative approach, the local FIB-enabled modification of superconducting thin films in order to “write” thin N or I layers as Josephson barriers in prepatterned thin film structures has been successfully demonstrated. Here, the high spatial resolution offered by He-FIB has been shown to be of great importance. Prime examples are the fabrication of insulating or normal-conducting barriers in yttrium barium copper oxide (YBCO) or MgB₂ thin films by He-FIB irradiation that locally suppress superconductivity on the nanometer scale,^{705,707,712} and the creation of He-FIB-induced nanoconstrictions in Bi₂Sr₂CaCu₂O₈ (BSCCO) single crystal flakes. The latter form the basis of two-dimensional cuprate nanodetectors with single telecom photonic sensitivity.^{713,714} Moreover, He-FIB irradiation of YBCO films has produced artificial pinning arrays with unprecedented lattice constants down to 30 nm,^{717,718} which opens a new regime for motion control of Abrikosov vortices in superconducting fluxonic devices.

Due to the nanoscale beam size that is achievable with FIBs, in particular with GFIS-based FIBs, these are ideal tools for the fabrication of single quantum dots, tunneling devices and single electron transistors. Examples include the fabrication of single electron transistors in Si/SiO₂ stacks⁴¹⁰ and the fabrication of single quantum dots in carbon nanotubes (CNTs).⁷¹⁵

3. Engineering magnetic properties

Ion irradiation, in general, has been used for many decades for the modification of magnetic materials.⁸⁰¹ To achieve this sort of modification, complex and expensive masking processes have to be used unless one employs FIBs, which can modify the magnetic properties locally with nanometer precision. In 1989, Ishii *et al.*⁷³³ used Si and Ge ions from a LMAIS-FIB to create nucleation sites for controlled switching of the magnetization in Bi- and Ga-substituted yttrium iron garnet (YIG) films. Ga-FIB processing has also been used for creating read/write heads⁸⁰² for magnetic hard disks, and for modern approaches such as magnetic random access memories (MRAMs) composed of magnetic multilayer structures.⁸⁰³ FIB-induced local changes in magnetic anisotropy have been used to create isolated tracks of single domain islands in recording media⁸⁰⁴ and to control the injection location and field dependency of magnetic domain walls in multilayers with perpendicular magnetic anisotropy (using Ga- and He-FIB).⁷²⁶ In the case of Co nanowires fabricated by FEBID, irradiation with Ga-FIB has allowed modification of the propagation speed of the magnetic domain wall.⁸⁰⁵ In some materials, FIBs can be used to generate ferromagnetism in selected regions such as in paramagnetic FeAl alloys. For example, the spatially resolved creation of disorder-induced ferromagnetism using Ga-FIB modification has been demonstrated by Menéndez *et al.*,⁷⁴⁵ achieving a spatial resolution of below 100 nm with the possibility of erasing the pattern by annealing. Later, Fe₆₀Al₄₀ was studied in depth for order-disorder phase transitions and ferromagnetic resonance (FMR) in magnetically written single structures induced by Ne-FIB irradiation.^{544,742} Paramagnetic regions in non-magnetic materials can also be created by FIB irradiation, as has been done for InGaAs using Ga-FIB to create a spin filter device.⁸⁰⁶

The continued quest for an increase in storage density in combination with novel data processing approaches (i.e., neuromorphic computing) has triggered the development of advanced patterning of complex magnetic alloys and multilayer systems.^{807–809} In these systems, various interactions due to the intrinsic anisotropy, or lattice and system geometry, may lead to topologically protected spin configurations such as skyrmions or the controlled formation and transport of spin waves. Here, the FIB stands out, as it enables tuning of magnetic anisotropy landscapes without any topographical changes to the devices. This was demonstrated for the nucleation control of skyrmions in Co/Pt multilayers using Ga-FIB⁷³⁹ and He-FIB,⁷³⁸ and through the patterning of skyrmion racetracks in a Pt/Co/MgO thin film using the He-FIB.⁷³¹ The patterning of dots as nucleation sites within a low-dose racetrack using He-FIB realized both the deterministic generation and movement of skyrmions with nanometric precision.⁷³² Finally, spin-wave optical components such as graded-index lenses for applications in magnonics can be created by ion induced modification of the local magnetization in YIG, which corresponds to an effective refractive index for spin wave propagation. Here, 50 keV Ga-FIB was employed and the penetration depth of about 100 nm allowed for an effective refractive index of up to 1.8, comparable to light-based optics. Since this depth was only one third of the actual device thickness, the much lighter He ions are expected to significantly improve the optical device performance.⁷⁴⁵

4. Engineering optical/quantum properties

Typical applications of ion irradiation in optics include the local change of refractive index and the creation of fluorescence.⁸¹⁰ Since

TABLE VI. Overview of FIB based property engineering applications, in which fine control over the ion dose, beam energy, and irradiated regions enables local tuning of material properties.

Engineered property	Experiment	Ion	Material/device
Electrical and electronic	Conductivity tuning	He	Graphene, ^{681–686} few-layer/monolayer MoS ₂ , ^{687–689} epitaxial MoS ₂ , ⁶⁹⁰ few-layer/monolayer WSe ₂ , ^{688,691} monolayer WSe ₂ , ⁶⁸⁸ InGaZnO thin film ⁶⁹²
		Ne	Si nanocrystals (self-assembly of) in buried silica layer ⁴¹⁰
		Ga	AlGaAs/GaAs quantum wires, ^{693–696} GaAs quantum constrictions, ⁶⁹⁷ GaN nanowires, ⁶⁹⁸ 2D MoSe ₂ /graphene heterostructure, ⁶⁹⁹ MoSi superconducting thin film, ⁷⁰⁰ In ₂ O ₃ thin film, ⁷⁰¹ Al ₂ O ₃ film, ⁷⁰² ZnO and VO ₂ ⁷⁰³
		Si	AlGaAs/GaAs quantum wires ⁷⁰⁴
		He	Superconducting electronic circuitry including Josephson junctions and nanoSQUIDS in YBCO ^{705–711,1198} and MgB ₂ , ⁷¹² nanoconstrictions in BSCCO ^{713,714}
	Formation of barriers and nanoconstrictions	Ga	quantum dots in single-walled CNT ⁷¹⁵
	Tuning thermoelectric properties	He	MoSe ₂ thin films ⁷¹⁶
	Tuning superconductivity	He	Periodic arrays of artificial pinning sites in YBCO ^{717,718}
	Pinning ferroelectric domains	He	PbZr _{0.2} Ti _{0.8} O ₃ thin film, ⁷¹⁹ multilayer MoTe ₂ flake ⁷²⁰
	Ferroelectric phase change	He	Induced ferroelectricity in HfO ₂ , ⁷²¹ conversion to paraelectric phase ⁷²²
		Ga	HfO ₂ thin film, ⁷²³ damage in ferroelectric capacitors ^{724,725}
Magnetic	Tuning magnetic anisotropy	He	Co/Pt multilayers, ⁷²⁶ Ir ₁₇ Mn ₈₃ /Co ₇₀ Fe ₃₀ bilayer, ⁷²⁷ Co-based thin-film stack, ^{728,729} Co/Pt/Ru multilayer system ⁷³⁰
		Si/Ge	skyrmions in Pt/Co/MgO thin film, ⁷³¹ and in Pt/Co ₆₀ Fe ₂₅ B ₁₅ /MgO thin film ⁷³²
		Ga	Bi and Ga substituted YIG ⁷³³
		Ga	CoPt multilayers, ⁷³⁴ Fe ₇₈ Ni ₂₂ thin film, ⁷³⁵ Ir/Co/Pt multilayers ⁷³⁶
	Tuning metamagnetic transition temperature	He	skyrmions in FeGe thin plate, ⁷³⁷ and in Co/Pt multilayers ^{738–740}
	Inducing ferromagnetism	He	FeRh thin film ⁷⁴¹
	Magnonics	Ne	Fe ₆₀ Al ₄₀ thin film ^{544,742}
		Ga	Fe ₆₀ Al ₄₀ alloy ⁷⁴³ and Fe ₇₈ Ni ₂₂ on copper ⁷⁴⁴
Optical and quantum	Tuning photoluminescence	Ga	Patterned Fe ₇₈ Ni ₂₂ films, ⁷³⁵ spin wave optical elements in YIG ⁷⁴⁵
		He	SiN _x membrane, ⁷⁴⁶ monolayer MoSe ₂ , ⁷⁴⁷ InGaAs/GaAs quantum well, ⁷⁴⁸ reducing parasitic fluorescence in hBN ⁷⁴⁹
	Tuning of passive optical components	Ga	AlGaAs/GaAs and GaInAsP/InP quantum wells ⁷⁵⁰
		He	Plasmonic resonance in graphene nanostructures, ⁷⁵¹ Bragg grating in silica nanofiber ⁷⁵²
		Li	Si microdisc nanophotonic resonator ⁷⁵³
	Creation of photoluminescence, quantum emitters and qubits	Ga	ZnSe quantum dot micropillar cavities, ⁷⁵⁴ periodic poling of LiNb waveguide ⁷⁵⁵
		He	NV centers in diamond with substitutional nitrogen, ^{756,757} Si vacancy centers in SiC, ^{758,759} luminescent defects in monolayer MoS ₂ (Refs. 760 and 761)
		Li	Si vacancy centers in SiC ⁷⁶²
		Si	Si vacancy centers in diamond ⁷⁶³ and diamond nanostructures ⁷⁶⁴ , W and G centers in silicon ⁷⁶⁵ , isotopic enrichment of ²⁸ Si to increase spin qubit coherence times ⁷⁶⁶
		Ar	NV centers in diamond ⁷⁶⁷
		P	Deterministic single-atom implantation of ³¹ P into Si ²⁹³
		Ga	Luminescent defects in hBN, ^{768,769} monolayer WSe ₂ , ⁷⁷⁰ MoS ₂ and WS ₂ ⁷⁷¹
		Ge	Ge vacancy centers in diamond ⁷⁷²

TABLE VI. (Continued.)

Engineered property	Experiment	Ion	Material/device
		Sb, Si	Deterministic single-atom implantation into Si ⁷⁷³
		Xe	Recoil implantation of color centers in diamond, ⁷⁷⁴ spin defects in hexagonal boron nitride (hBN), ⁷⁷⁵ water assisted fabrication of extended defects in hexagonal boron nitride (hBN) ⁷⁷⁶
		Bi	Qubits in silicon and germanium ⁷⁷⁷
Chemical	Enhancing chemical etch rates	He	Si ₃ N ₄ thin film, ⁷⁷⁸ SiO ₂ thin film, ⁷⁷⁹ mono-/bilayer MoS ₂ ⁷⁸⁰
	Enhancing catalytic activity	He	MoS ₂ and MoSe ₂ flakes ⁷⁸¹
	Nucleation of growth	He	Epitaxial graphene on SiC ⁷⁸²
		Co	CoSi ₂ nanowires in Co-saturated Si ⁷⁸³
		Ga	InAs dots on GaAs, ⁷⁸⁴ Ge dots on Si ^{785,786}
	Functionalization	Ga	Graphene (fluorination using XeF ₂) ⁷⁸⁷
	Tuning surface wettability	Ga	Molecular self-assembled monolayer ⁷⁸⁸
Thermal	Reducing thermal conductivity	He	Si nanowires, ⁷⁸⁹ Si membrane, ⁷⁹⁰ VO ₂ nanowires, ⁷⁹¹ Bi ₂ Te ₃ nanoribbons ⁷⁹²
Mechanical	Tuning Young's modulus	He	Monolayer MoSe ₂ ⁷⁴⁷
	Gas bubble effects	Ga	Polymer, ⁷⁹³ hydrogels ⁷⁹⁴
		He	Nanobubble superlattice in Cu, ⁷⁹⁵ nanobubbles in Si, ⁷⁹⁶ diamond, ⁷⁹⁷ Eurofer steel, ⁷⁹⁸ W, ⁷⁹⁹ ultrafine-grained W and W-Cu nanocomposite ⁸⁰⁰

FIB processing is serial and the penetration depth is limited, the use of the FIB in this area is mostly of interest in cases where a small number of ions needs to be placed with high spatial accuracy. A particularly beautiful example is the tuning and imaging of resonant optical modes in a silicon disk resonator using Li-FIB.⁷⁵³ A low beam current of 1 pA at 3.9 kV acceleration voltage was used to radially raster scan in pulses of ≈ 3000 ions while the transmitted light passing through a critically coupled waveguide was monitored. The optical shifts observed could be attributed to ion damage and a rapid thermal (and thus reversible) response. Knowledge of this time-varying response was used to enable imaging of different optical modes with minimum device degradation.

The spatial accuracy of defect creation using the mask-less FIB approach becomes especially beneficial in the field of quantum technology. Here, a defect in an otherwise ideally perfect crystal can take the role of an artificial atom emitting quantum light. Furthermore, the defect may exhibit electronic and nuclear spin degrees of freedom that may be coherently coupled to the emitted quantum light, turning the defect into a solid state quantum bit (qubit). The most popular example of a defect-based quantum emitter, or qubit, is the nitrogen vacancy (NV) center in diamond. Diamond NV centers are routinely fabricated using ion implanters,⁸¹¹ but have also been realized using He-FIB processing.^{756,757} FIB implantation of Si ions has been used to realize nearly lifetime-limited single Si vacancy (SiV) quantum emitters in diamond nanostructures.⁷⁶⁴ This maskless approach achieved ≈ 32 nm lateral precision and <50 nm positioning accuracy and is thus promising for the development of scalable solid-state quantum

information processors. Focused Si ions at 35 keV with beam diameters of 5–10 nm have also been used to fabricate negatively charged silicon vacancies (V_{Si}^-) in 4H-SiC with single emitters observed starting from 40 ions per irradiated spot.⁸¹² A comparative study by the same author using H, He, and C ions showed that He ions produced the highest defect concentration, while the lowest brightness emitters were observed for C ions.⁸¹³ In addition, Li-FIB has been employed for the creation of Si vacancies in SiC.⁷⁶² Recently, focused Ar⁸⁺ ions from an EBIT were also used for NV-creation in diamond.⁷⁶⁷ A particularly interesting study made use of Xe-FIB for recoil implantation of targeted group IV dopants in a deposited surface layer.⁷⁷⁴ In this work, Pb, Sn, Ge, and Si color centers in diamond were fabricated with a spatial accuracy of better than 50 nm in combination with an ultra-shallow doping profile. So-called G and W centers in Si have recently been demonstrated using Si-FIB to create spatially resolved single defect centers of this type for application as quantum emitters in the telecom band.⁷⁶⁵ In a very recent approach, a dedicated FIB system equipped with a ultrahigh vacuum (UHV) sample chamber⁸¹⁴ was used to suppress unwanted isotopes in Si that would otherwise limit coherence times and therewith the performance of solid-state spin qubits. This was achieved using a LMAIS source to implant ²⁸Si⁺⁺ into natural Si, lowering the level of unwanted ²⁹Si to the single-digit ppm range without introducing contamination from the FIB processing.⁷⁶⁶ The deterministic placement of single atoms (e.g., Sb or P) into Si to act as qubits is another challenge where progress has recently been made.^{293,773} While the two cited works are a clear demonstration

of the placement accuracy of single (or few) atoms using a FIB, the approaches rely on implantation into a host material that can also double as an active detector of the ion impacts.

In parallel, 2D materials have been heavily explored as hosts for usable quantum emitters, e.g., sources of single photons.⁸¹⁵ Here, ion irradiation can be used to both tailor the optical properties of the 2D system⁸¹⁶ as well as to produce defects that introduce novel optical activity.^{817,818} He-FIB processing, in particular, has been used for the site-selective creation of defects in, e.g., 2D MoS₂.⁷⁶¹ While the exact type of defects being introduced is still under investigation (true for most luminescent defects in 2D materials), they are expected to cause highly localized trapping potentials for exciton emission. Encapsulation with hBN leads to very narrow linewidths of the emission lines in the visible range, 100–220 meV below the neutral exciton.⁷⁶¹ He irradiation has also been shown to result in a fivefold reduction in background and parasitic fluorescence, enabling the isolation of bright and stable quantum emitters in hBN.⁷⁴⁹ Recent results using Ga-FIB patterning indicate that the formation of luminescent defects is positively correlated with material swelling rather than material removal by milling.⁷⁶⁹ This helps to explain earlier results on the generation of defects by Ga-FIB, where it was not clear whether the single photon emission was actually generated inside the milled holes, at their edges, or in the damaged region near the edges.⁷⁶⁸

5. Engineering of low-dimensional materials

2D materials such as graphene and atomically thin transition-metal dichalcogenides are expected to play an important role in the realization of future electronic and optoelectronic devices, and there exists a significant body of work in which FIB based defect engineering has been used to tune the unique properties of these materials. In fact, as can be seen in Table VI, the majority of materials that are used in FIB based defect engineering work are indeed low-dimensional (thin films, 2D materials, and nanowires). The simple reason for this is that typical FIB instruments deliver beams with energies of several tens of keV, which limits the ion penetration depth. For example, in most materials the maximum penetration depth of the ubiquitous 30 keV Ga ion beam is around 60 nm, which severely limits possibilities for property engineering of bulk materials. While lighter ions of the same energy penetrate more deeply, for many defect engineering applications implantation of the ion into the material is not actually desired. Rather, it is the local damage created by the ion in transit that is leveraged. Therefore, low-dimensional materials are often the ideal testbed material for defect engineering using the FIB. Notably, an increase in ion energy typically causes a decrease in defect production in 2D materials,⁸¹⁹ such that lower beam energies are in fact better suited for efficient materials modification in these cases.

An example of FIB based defect engineering of a low-dimensional material is the use of the He-FIB to tune the electrical resistivity of 2D MoS₂.⁶⁸⁷ In the cited work, the electrical resistivity of the MoS₂ monolayer was tuned from semiconducting to insulating depending on the vacancy defect concentration, as controlled by the irradiation dose. Above a certain threshold dose, total amorphization (i.e., disordering) of the material occurs. By irradiating samples locally in specific patterns with fluences below the total amorphization threshold, the FIB can be used to tune the defect density and hence the electrical conductivity in spatially defined regions at the nanoscale. For example, using the He-FIB and a line irradiation strategy on 2D MoS₂, a neuromorphic memtransistor device has been demonstrated.⁶⁸⁹ Similarly, Ga-FIB has been used for defect creation in supported graphene with MD simulations

providing insight into the effect of irradiation angle on the lateral damage distribution.⁸²⁰ He-FIB line irradiation of graphene has also been used to create graphene nanoribbons,⁶⁸⁶ which opens up a bandgap in graphene due to quantum confinement. Tuning of the ribbon width and, hence, the size of the bandgap is a key step toward the use of graphene in electronic applications. As discussed in Sec. III B 3, for supported graphene, the minimum achievable nanoribbon width is limited by the influence of the backscattered ions from the substrate.³⁷⁰

Other defect engineering applications concerning 2D materials include the use of Ga-FIB in combination with the simultaneous flow of XeF₂ gas for the localized fluorination of graphene⁷⁸⁷ and the use of the He-FIB to create localized defects in graphene to serve as nucleation points for the epitaxial growth of hBN.⁷⁸² An example of defect engineering of a 1D structure is the employment of localized He-FIB irradiation to induce regions of disorder in a silicon nanowire in order to tune the thermal conductivity along its length.⁷⁸⁹

6. Study of irradiation damage mechanisms

Fundamental curiosity and the need for experimentally straightforward replacements for long-duration reactor-based irradiation experiments have resulted in FIB instruments being used for the systematic study of irradiation effects on structural materials. In order to simulate irradiation damage caused by α particles, for example, He⁺ ions are a convenient species to use. A further advantage of using an inert species such as He is that fundamental studies of irradiation effects can be made while avoiding alloying effects and grain boundary embrittlement, as can occur with a metallic ion species such as gallium.⁶²¹

A phenomenon that was discovered several decades ago is the formation of He nanobubbles upon He ion irradiation of various metal targets. These nanobubbles form as a result of vacancy defects and He interstitials that diffuse and recombine to produce gas clusters. However, fundamental experimental studies of the resulting changes in the mechanical properties of the target material were for a long time confined to the use of large-scale ion accelerators and plasma devices. The introduction of the He-FIB changed this, enabling the first systematic dose-controlled irradiation studies. Using the He-FIB, gas bubble phenomena that had been observed in large-scale experiments have been reproduced with much greater control over the experimental conditions. Initial He-FIB studies in this area used Si targets, mostly driven by the need of the semiconductor industry to evaluate the invasiveness of He-FIB processing.⁶⁴⁴ Later studies included the investigation of He gas-bubble superlattices, using single-crystal Cu as a simple model system and probing the effect of the superlattice on the mechanical properties of the host material using *in situ* TEM nanoindentation.⁷⁹⁵ Since samples are irradiated with the He-FIB on the nanoscale, individual grains and grain boundaries can be targeted in polycrystalline samples.⁸²¹ By increasing the dose, the coalescence of nanobubbles forming crack planes that ultimately result in blistering and delamination can also be studied with a level of control that was previously unattainable.⁷⁹⁹ Furthermore, irradiation induced mechanical property changes of a number of materials that are directly relevant for nuclear applications have been investigated using the He-FIB combined with nanoindentation, e.g., in studies of Eurofer steel,⁷⁹⁸ W,⁸²² ultrafine-grained W, and a W–Cu nanocomposite.⁸⁰⁰

Finally, He gas bubble effects have also been employed for nanofabrication tasks using patterning with the He-FIB at high dose for localized subsurface swelling to deform surfaces and thereby create 3D nanostructures such as nanopillars and nanohemispheres.^{823–826}

7. Outlook

The use of the FIB for property engineering down to the nanoscale is now well established. Recently, strong efforts have been under way to increase the spatial resolution and control of the technique even further with the ultimate goal of creating single defect centers or implanting single impurity atoms for applications in quantum technology. Here, a significant challenge is posed by the intrinsically stochastic nature of both the production of ions from most FIB sources and of the ion–solid interaction itself. The deterministic placement of single ions either into the bulk or at a surface thus pushes the FIB technique to its fundamental limits.

There are two distinct approaches that are being pursued for realizing deterministic SII using the FIB. One is to count the ions as they travel to the sample and the other is to detect individual ion impact events, ideally in an universal manner. The first approach can be realized by using an Paul ion trap as a deterministic source of single ions.^{827,828} While the presence of the ion can be unambiguously proven, ejecting the ion and re-loading the trap is time-consuming and the spatial localization of the eventual impact event using ion-optical elements causes significant losses. Given that the probabilities for the creation of specific defects are usually low and highly material dependent (see also Sec. III B), this deterministic ion source approach is appealing, but technologically extremely demanding. Alternatively, ion columns can be equipped with image-charge detectors that count ions produced from different types of ion sources. The beam is directed toward the sample only if an ion is detected. In this way, imperfect detection efficiencies can be mitigated since all non-detected ions are automatically blanked.¹⁹³ The drawback of this technique is the large charge state (at least 10^+) that is necessary for detection of the image charge, requiring ion sources such as an EBIT. Here, Ar^{18+} ions created by an EBIT have been successfully used.¹⁹⁴ Light ions are excluded from this technique, unless the sensitivity of the image-charge detectors can be significantly increased, and this method also presents significant technological challenges for column design.

The second approach, i.e., the detection of individual ion impacts on the sample, could in principle work for any ion source. Promising attempts combine an on-chip IBIC detector^{199,829} with a plasma ion source and a near-surface aperture in an AFM cantilever.²⁹³ Here, scanning of the aperture realizes the spatial localization^{830–832} and thus allows use of various types of ion source, including broad beam implanters. While this is an important step for all quantum devices that tolerate a nearby pn-junction and amplifier, other device concepts, e.g., in photonic quantum technology, require truly agnostic sample detector concepts. Conventional SE detectors (Everhart–Thornley, see also Sec. II C) or novel detector types (e.g., channeltrons⁷⁷⁷) may work as such, but suffer from their strong dependence on the SE yield, which is unknown for most materials and is very surface dependent. There is also a need for a better understanding and prediction of SE generation and expected detection yields, and for the development of advanced SE detection schemes with detection efficiencies approaching 100%.^{777,833} Instead of detecting all ion impact events (that may each come with a small probability of creating the target defect), one may instead detect device function *in situ* while delivering ions to the target locations. A combination of short dwell times at extremely low current results in a Poissonian probability of either one or zero ions per shot, and the process may be stopped once the desired device operation is achieved. Here, the technological challenge of these implementations involves the co-integration of an extremely sensitive fluorescence detection setup.^{834,835}

In conclusion, several promising development routes for SII are currently being pursued that are showing first promising results. Whether modifying the ion supply toward being deterministic or enhancing the detection capabilities toward single-ion sensitivity (or a combination of both) will finally lead to success is still an open question.

C. FIB-based imaging and tomography

Multiscale imaging of complex systems is becoming increasingly important. For example, to understand data transmission in the brain or chemical energy storage in a battery, one needs to capture the “big picture,” ideally in three dimensions, as well as resolve features down to individual atoms. This is where FIB based techniques play a central role, by direct imaging using light ions such as He, by 3D volume imaging combining FIB slicing with SEM, and by preparing thinned cross sections and specialized geometries for TEM and APT.

1. A brief history of FIB-based imaging and analysis

The enormous potential of electron microscopy for imaging the structure of condensed matter became clear with the invention of the TEM in 1932⁸³⁶ and soon thereafter first images of bacteria^{837,838} and virus particles⁸³⁹ proved the same for the life sciences. The introduction of ion microscopy for imaging, however, took much longer. Pioneering work was carried out in the 1970s by Levi-Setti and co-workers, who experimented with a scanning transmission ion microscope using up to 65 keV H_2^+ ions from a GFIS and published images of myofibrils from rabbit muscle.³⁶ However, since the achievable lateral resolution of ion microscopy had always been inferior to that of electron microscopy, ion microscopy was not used for nanoscale imaging until the early 2000s with the invention of the HIM^{21,25} (see Sec. IV C 2).

The breakthrough in the use of ion microscopy for sub-cellular imaging in biology, or “the revolution in ultra-microscopy” as stated by Ballerini *et al.*,⁸⁴⁰ occurred once Ga ion beams from LMIS-based sources could be focused down to better than 10 nm.⁸⁴¹ It was recognized that the ion beam could be used “like a tiny scalpel which slices away layers of the sample (and that one could) exploit this destruction to gain high-resolution three-dimensional structural, morphological, and chemical information.”⁸⁴⁰ Thus, the FIB based microscopy and machining that had until the turn of the century mainly been used in the semiconductor industry was brought to the life sciences.^{840,842} Various other fields also adopted the technique. For example, one of the first reports of FIB tomography was presented in 2001 by Inkson *et al.*⁸⁴³ on the 3D characterization of a metallic nanocomposite, and FIB based tomography also became increasingly common in, e.g., metallurgy,^{103,626,844} energy research,⁸⁴⁵ and geology.⁸⁴⁶ Recently, FIB tomography has even been applied to paleontology samples.⁸⁴⁷ While early work sometimes used the Ga-FIB for the imaging as well as the sectioning, the most common implementation of FIB tomography uses the dual beam FIB-SEM, comprising a Ga-FIB column for the milling coupled with a SEM for high-resolution surface imaging of the exposed faces. Significant progress in 3D image segmentation and reconstruction⁸⁴⁸ was also key to enabling 3D structural imaging of complex specimens by the FIB-SEM technique. Furthermore, the development of cryo-workflows for cryo-FIB-SEM tomography was a major step forward allowing preservation of cell ultra-structure.^{849,850} Such cryo workflows are now used for other beam-sensitive samples as well (see Sec. IV C 4).

TABLE VII. Overview of FIB based imaging and analysis in the life sciences.

	Application	Technique	Experiment
Molecules	Localization of proteins	cryo-FIB-SEM	Protein detection in vitrified HeLa-cells using Au-conjugated antibodies against RNA polymerase II ⁸⁵³
		HIM and fluorescence microscopy	Detection of quantum-dot labeled proteins ⁸⁵⁴
	Metabolic imaging	ToF-SIMS OrbiSIMS	Detection of proteins on DNA bio-sensor chip ⁸⁵⁵ Imaging of the distribution of neurotransmitters in mouse hippocampus ⁸⁵⁶
Viruses and nanoparticles	Virus imaging	HIM	3D metabolic imaging of <i>Paramecium caudatum</i> using orbitrap-MS coupled to TOF-SIMS ⁸⁵⁷ Imaging of bacteriophage infection, ^{858,859} and transport by bacteria ⁸⁶⁰
	Nanoparticle-cell interactions	FIB-SEM HIM	Imaging of SARS-CoV-2 infected Vero cells ^{861,862} Investigations into SARS-CoV-2 viral density at cell-cell contacts ⁸⁶³
		HIM-SIMS ToF-SIMS	Investigation of biogenic nanoparticles by HIM-SIMS ⁸⁶⁴ 3D-imaging of nanoparticles in an algal biofilm ⁸⁶⁵
Cell-level	Imaging of microbial cells	HIM	Imaging of nitrate-reducing bacteria with spikey minerals, ^{866,867} archaea ⁸⁶⁸ , predatory bacteria ⁸⁶⁹
	Imaging of biofilms	HIM	<i>Geobacter sulfurreducens</i> , ⁸⁷⁰ microbial mats from Himalayan hot springs, ⁸⁵⁹ chlorella micro-algae ⁸⁷¹
	Imaging of mammalian cells and tissues	HIM	Imaging of mitotic HeLa cells, ⁸⁶⁶ visualization of lipid nanodomains in human neural stem-cells, ⁸⁷² imaging of kidney ⁸⁷³
	3D imaging and remodeling of microbial cells	(cryo-)FIB-SEM	FIB-SEM of yeast cells, ^{840,849} 3D remodeling of <i>Acantharia</i> -microalgae ⁸⁷⁴
	3D imaging of cells and tissues	(cryo-)FIB-SEM	Imaging of FIB-sectioned gland cells, ⁸⁷⁵ lymphoid tumor tissue, ⁸⁴⁹ carbon nanotubes (CNTs) in mouse lung tissue, ⁸⁷⁶ tobacco petal epidermis, ⁸⁵⁰ human bone ⁸⁷⁷
	Sub-cellular elemental mapping	nanoSIMS, ToF-SIMS	Element distribution in <i>E.coli</i> exposed to natural antibacterial clay, ⁸⁷⁸ mapping of N/P ratio in a photosymbiotic <i>Acantharia</i> -microalgae system, ⁸⁷⁴ identification of TiO ₂ in an algal biofilm ⁸⁶⁵
		PIXE	Elemental analysis of hair, skin, nervous tissue ⁸⁷⁹
	Metabolic activity measurements	nanoSIMS	Identification of Ni-U-phosphate deposits in bacteria ⁸⁸⁰ nanoSIMS and stable isotope labeling to measure metabolic activity of microbial cells ^{881,882} and heterogeneity in isogenic microbial populations ^{883,884}
	Biomarker imaging of single cells	ToF-SIMS	Biomarker imaging of diatom cells in a microbial mat ⁸⁸⁵
	Chemical analysis of biofilms	ToF-SIMS	Localization of P in micro-algae biofilms ⁸⁸⁶
	Bacterial predators	HIM	Imaging of the different stages of the life-cycle of <i>Bdellovibrio bacteriovorus</i> ⁸⁶⁹
	Geomicrobiology	HIM	Imaging of iron-oxidizing bacteria, ^{866,867} structure of <i>Geobacter sulfurreducens</i> biofilm ⁸⁷⁰
		ToF-SIMS	Microaerophilic Fe(II) oxidizing <i>Zetaproteobacteria</i> , ⁸⁸⁷ iron corrosion by methanogens ⁸⁸⁸
	Anti-microbial properties of surfaces	HIM milling	Investigation of the anti-bacterial properties of the nanostructures on dragonfly wings ⁸⁸⁹

An overview of the various FIB based imaging and analysis modalities used in the life sciences is given in Table VII. In all of the examples given, the imaging and analysis was primarily performed *in situ*, i.e., inside the FIB-(SEM) instrument. Key works on cryo-FIB milling for the site-selective preparation of electron transparent biological specimens for analysis by cryo-TEM can be found in Refs. 851 and 852.

2. HIM—A tool for imaging of the tiniest living objects

SEM is a standard method for imaging biological specimens, in which samples are typically coated with a conductive layer to prevent charging under the beam. SEM imaging of uncoated samples without charging becomes possible if the accelerating voltage of the electron beam is reduced below 1 kV, but at the expense of increased chromatic aberration and a reduced depth of field.⁸⁶⁶ In contrast, with HIM, *in situ* charge compensation with an electron flood gun allows the investigation of non-conductive biological samples without conductive coatings while maintaining good spatial resolution. This is important, as metallic coatings can alter and conceal nanoscopic samples and features, such as virus particles or the cell membrane topography. Furthermore, compared to SEM, HIM delivers a larger depth of field, enhanced topographic contrast, and greater surface sensitivity.^{187,890} Typically, HIM imaging relies on the detection of SEs, but backscattered ions (BIs) can also be detected, which enhances material contrast. (This is comparable to SE vs backscattered electron detection in SEM.) Since backscatter detection in HIM is also sensitive to the neutrals that are generated, detection yields are relatively high meaning that images can be collected at lower dose (see also Sec. II C). Bidlack *et al.*¹⁸³ pointed out that BI imaging could improve the visualization of immuno-gold labels in cells and Ma *et al.*¹⁸² used BI imaging to detect different concentrations of colloidal nanoparticles in cells. However, in general, the use of BI detection for HIM in the life sciences is still underexplored.

a. Preparation of the specimen. Since the sample requirements for room-temperature HIM (dry and vacuum-compatible) are almost identical to those of electron microscopy, established protocols from the SEM community provide a good basis for HIM sample preparation. In general, the sample preparation steps involve:

1. chemical fixation with aldehydes (e.g., formaldehyde or glutaraldehyde),
2. optional post-fixation with strong oxidizing agents (e.g., osmium tetroxide),
3. dehydration in a graded series of an organic solvent (e.g., methanol, ethanol or acetone),
4. drying (e.g., critical point drying).

Instead of the final dehydration and drying of the sample, the application of ionic liquids has been proposed as an alternative⁸⁹¹ and successfully applied for imaging bacterial biofilms using HIM.⁸⁹² In all cases, no final metallization of the sample is required (due to the *in situ* charge compensation mentioned previously); hence, samples can be visualized in a more pristine state.

b. Viruses. Sharma *et al.*⁸⁵⁹ used HIM in investigations of the interactions of viruses and their hosts in sediment and microbial mat samples. In the same manner, HIM has been shown to be a powerful tool to investigate bacteriophages (i.e., viruses that attack bacteria) and their

interactions with their host cells. Both Leppänen *et al.*⁸⁵⁸ and You *et al.*⁸⁶⁰ published high-resolution HIM images of T4-phage infected *E. coli* bacteria, and Vinner *et al.*⁸⁹³ visualized phage Felix O1 infected Salmonella bacteria. The contributions that HIM and other microscopies have made to our understanding of bacteriophages have been reviewed in more detail by Almeida *et al.*⁸⁹⁴ Furthermore, Frese *et al.*⁸⁶¹ and Barreto-Vieira⁸⁹⁵ recently reported the first visualization of SARS-CoV-2 virus particles by HIM, demonstrating significant topographical differences compared to previously published SEM images of sputter-coated samples. Examples of HIM imaging of virus are shown in Fig. 18. In the future, HIM could be used, for example, to examine the internal structure of phage-infected cells by milling cross sections. In addition, HIM could be used to better understand the spread of viral particles through other organisms, since it is possible to remove lightly attached phages during a washing step in the sample preparation and image what remains.

c. Single-celled organisms. HIM is an excellent tool for viewing single-celled organisms such as bacteria, archaea, yeast, micro-algae, protozoa, etc., at a resolution approaching that of TEM without needing to embed and section the samples. Here, it is important to note that the HIM technique primarily enables imaging of the cell surface. In their 2013 paper, Joens *et al.*⁸⁶⁶ employed HIM for the first time for the imaging of microbes, namely, iron-oxidizing bacteria. The ability to view highly textured biological specimens on non-conductive surfaces with strong topographic contrast, charge-compensation, and large depth-of-field renders the HIM an excellent tool for this work. The first HIM images of a culture of archaea were published by Chen *et al.*,⁸⁶⁸ where HIM proved particularly useful for investigation of the alleged cell budding of *Ca. A. ethanivorans*. Furthermore, HIM has been utilized by Said *et al.*⁸⁶⁹ to image bacterial infections by the bacterial predator *Bdellovibrio bacteriovorus*. HIM of bacterial biofilms (also often highly textured) was published by LeTourneau *et al.*⁸⁹⁶ (rhizobacteria), Belianinov *et al.*⁸⁷⁰ (*Geobacter sulfurreducens*), and Schmidt *et al.*⁸⁹² (*Pseudomonas putida*). Further HIM studies of biofilms include imaging of microbial mats from hot springs in the Indian Himalayas by Sharma *et al.*⁸⁵⁹ and investigations of micro-algal biofilms with regard to architecture and P distribution by Moreno Osorio *et al.*⁸⁷¹ Other potential applications of HIM imaging of microbiological objects

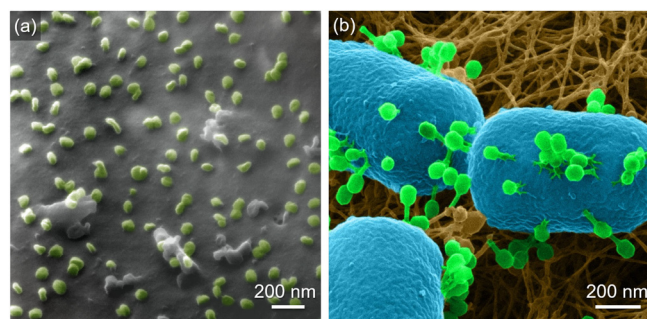


FIG. 18. (a) SARS-CoV-2 virus particles on the cell membrane of a Vero E6 cell,⁸⁶¹ reproduced with permission from Frese *et al.*, Beilstein J. Nanotechnol. **12**, 172–179 (2021). Copyright 2021 Authors, licensed under a Creative Commons Attribution (CC BY) license. (b) T4-phage infected *E. coli* bacteria on a fibrous agar surface.⁸⁵⁸ Reproduced with permission from Leppänen *et al.*, Adv. Biosyst. **1**, 1700070 (2017). Copyright 2017 John Wiley and Sons.

include investigations of biofilms in different environments, e.g., on medical implants or in bio-reactors. HIM also holds great promise for studying phage-infection or predation by protists or bacteria in environmental bacterial biofilms. Furthermore, one can conceive of HIM investigations into the complex interplay of microbes with plants or fungi. A recent example in this area is given by You *et al.*,⁸⁶⁰ where HIM was used to analyze phage co-transport on hyphal-riding bacteria.

d. Cells, tissues, immuno-gold labeling. The HIM has been used in many studies to investigate whole cells of multicellular organisms, including plant,⁸⁷¹ animal,⁸⁹⁷ and human cells.^{898,899} In the study of nanoscopic cell structures, which are usually distorted or obscured by metallic coatings, charge compensation allows visualization of topographical structures that would not be observable by SEM. For example, Schürmann *et al.*⁸⁷² visualized lipid nanodomains in the cell membranes of neurons, which were demonstrated to otherwise be concealed under a 10 nm metallic coating. Sato *et al.*⁹⁰⁰ have shown that immuno-gold-tagged antibodies can be resolved within cells to identify specific organelles, demonstrating the detection of ionoluminescence from ZnO nanoparticles in biological specimens using HIM. When imaging animal nanostructures such as insect wings, the superior depth of field of the HIM has proven advantageous.^{889,901} In addition, Joens *et al.*⁸⁶⁶ demonstrated that the mouth cavity of a nematode can be dissected using Ne-FIB milling with subsequent imaging by HIM, with minimal thermal damage during the milling process.

e. HIM in correlative workflows. In addition to structural analysis, the ability to investigate chemical compositions on the subcellular level is also needed. In order to achieve this with a HIM-based platform, detectors such as for SIMS (cf. Subsection II C) can be installed and used for *in situ* chemical investigations, as has been shown for *E. coli* cells incubated with TiO₂ nanoparticles.¹⁴⁰ Alternatively, HIM imaging can be integrated into correlative workflows *ex situ* with SIMS, EDS, or fluorescence microscopy using fluorescent markers (see also Sec. II F). A recently published example shows how HIM in correlation with the aforementioned techniques was used to analyze a resin-embedded soil sample with regard to roots, sediments, soil-organic matter, and bacteria.⁹⁰² Further correlative studies that make use of the unique charge-free high-resolution imaging capabilities of HIM can be expected, e.g., to analyze biomineralization processes, and to address ecological questions in environmental microbiology, plant and soil science, etc.

3. FIB tomography in the life sciences

Three-dimensional nano-imaging of biological samples has become increasingly important, e.g., for investigations of cell architecture, the ultra-structure of cell components, cell–cell interactions, structural analysis of tissues and biofilms, etc.⁹⁰³ Until the early 2000s, investigations into the 3D ultra-structure of samples such as resin-embedded tissue or cells were carried out by serial block-face SEM⁹⁰⁴ or by serial sectioning for TEM.⁹⁰⁵ While a very high lateral resolution can be achieved by these approaches, depth resolution is limited by the achievable thickness of the slices. Alternatively, serial block-face SEMs with an ultramicrotome fitted into the analysis chamber can be used for tomographic imaging of resin-embedded samples. However, the depth resolution is still limited by the minimum amount of material that can be removed in each slice.

The advent of FIBs (principally the Ga-FIB) propelled the field of biological micro-tomography, allowing much better control and much finer removal of material.⁹⁰⁶ In one of the first examples of FIB based tomography in the life sciences, Ballerini *et al.*⁸⁴⁰ investigated the internal structure of yeast cells using Ga-FIB sectioning followed by a sample tilt to allow FIB based imaging of the cross section via secondary electrons and (negatively charged) secondary ions. Soon thereafter, the image resolution and efficiency of the technique were improved using dual-beam FIB-SEM systems where the specimen is Ga-FIB-sectioned and imaged by SEM. The use of oxygen ions from a PFIB appears to reduce curtaining and other artifacts, allowing faster preparation of smoother surfaces for the imaging step.⁹⁰⁷ The basic workflow for FIB-SEM analysis of a resin-embedded biological specimen consists of repeated FIB-milling and SEM imaging of the exposed face followed by 2D image-stack registration and finally 3D reconstruction, as summarized in Fig. 19.

In order to obtain sufficient contrast in the SEM images, the sample is usually stained with heavy metals. Pioneering work was carried out by Drobne *et al.*,⁸⁷⁵ who investigated the digestive system of a crustacean by FIB-SEM and compared imaging modes. In order to achieve better contrast, Leser *et al.*⁹⁰⁸ systematically investigated different OsO₄ and uranylacetate staining techniques and concluded that for plastic resin embedded samples, the OTOTO method (OsO₄/thiocarbohydrazide/OsO₄/thiocarbohydrazide/OsO₄) in combination with backscattered electron imaging provides best results. Other protocols involve a combination of potassium ferrocyanide, OsO₄, and uranylacetate, and were successfully applied to brain⁹⁰⁹ and bone tissue,⁹¹⁰ as well as for the investigation of the interaction of CNTs with lung tissue.⁸⁷⁶ A detailed description of standard protocols for FIB-SEM of dried as well as plastic resin embedded (tissue) samples can be found in the work of Drobne.⁹¹¹ An alternative preparation method for FIB-SEM investigation of cells on surfaces uses thin layer plastification of the samples in place of resin embedding and full impregnation.⁹¹⁰

It is well known that dehydration and embedding in plastic resins introduces artifacts and may destroy the ultra-structure of cell components. These effects can be partially avoided by high-pressure or plunge freezing of the sample followed by cryo-substitution of water and slow infiltration with an epoxy-resin. However, FIB-SEM imaging of the embedded sample is then done at room temperature, as described, e.g., by Schmid *et al.*⁹¹² for samples of the nitrate-reducing and Fe-oxidizing bacteria BoFeN1. (This is the same cell culture as used by Joens *et al.*,⁸⁶⁶ see Sec. IV C 2 c.) By the early 2000s, it became clear that the ultra-structure of cells could be preserved even better if full cryo-workflows are employed, where a plunge- or high-pressure frozen specimen (still hydrated) is imaged at cryogenic temperatures. Marko *et al.*⁹¹³ explored the possibilities of thinning vitrified biological specimens by cryo-FIB milling plunge-frozen water and concluded that (a) cryo FIB-milling of vitreous ice does not induce heating sufficient to cause devitrification, and (b) subsequent experiments should be performed below −135 °C to maintain a vitreous state. In 2006, Heymann *et al.*⁸⁴⁹ published SEM data of plunge-frozen and FIB-milled yeast cells, as well as SEM and TEM images of FIB-milled lamellae of plunge-frozen tumor tissue. Although the frozen samples were not stained, sufficient contrast for SEM imaging with SEs was achieved through a combination of sublimation and metal deposition on the exposed surface. Similar work was performed by Heyles *et al.*,⁸⁵⁰ who used a GIS system to deposit platinum onto frozen yeast cells and onto a petal epidermis sample of *Nicotinia tabaccum*. A

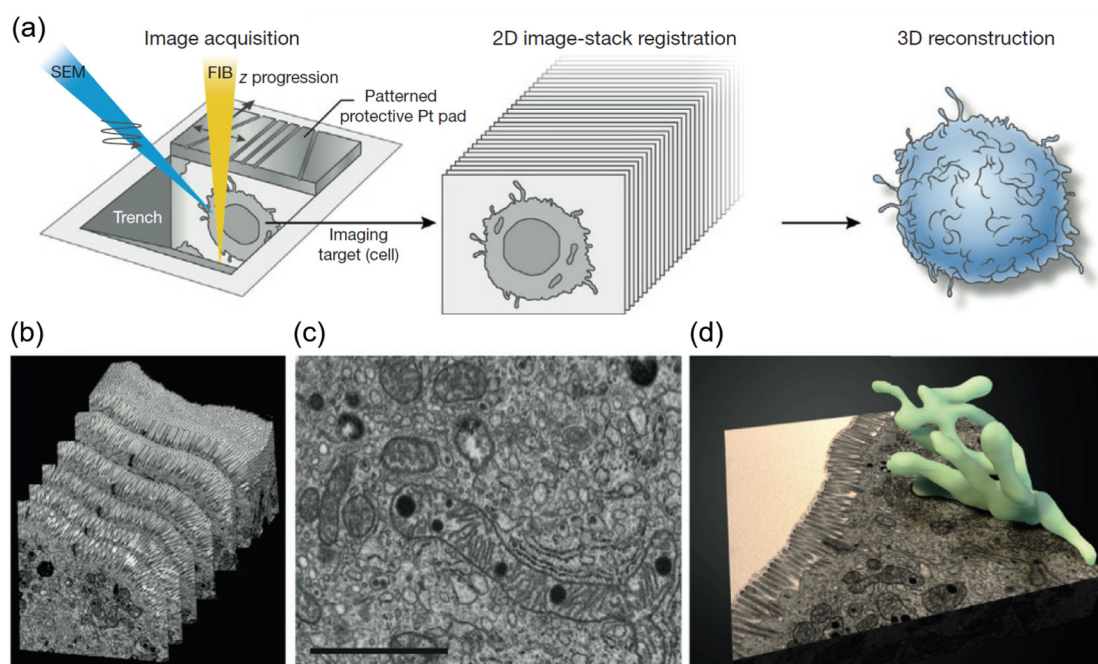


FIG. 19. Illustration of the basic workflow for volume imaging by FIB-SEM tomography.⁹⁰⁸ (a) Repeated FIB milling followed by SEM imaging, image-stack registration and reconstruction. (b)–(d) Example of a mouse intestine sample. Reproduced with permission from Narayan and Subramaniam, *Nat. Methods* **12**, 1021–1031 (2015). Copyright 2016 Springer Nature America.

recent overview of cryo-FIB-SEM techniques for frozen, hydrated biological samples is given by Hayles and De Winter.⁹¹⁴

Another leap forward was the development of correlative confocal laser scanning and FIB-SEM tomography using fluorescent dyes, high-pressure freezing, and cryo-substitution embedding.⁹¹⁵ For example, Gorelick *et al.*⁹¹⁶ developed a photon-ion-electron microscope (PIE-scope), which employs correlative cryogenic fluorescent light microscopy and FIB-SEM enabling the rapid and precise location of protein complexes in cells. Another example is the recently published (non-cryo) 3D-correlative workflow combining super-resolution light microscopy and FIB-SEM using reactive oxygen ion PFIB milling and a rotating sample stage to study embedded cultured cells.⁹¹⁷

Driven by questions in cell biology and connectomics, strategies have also been developed to enable FIB-SEM analysis of large volumes at high spatial resolution. Pivotal work in this area by Xu *et al.*,⁹¹⁸ involving automated recovery from beam/system malfunctions, enabled enhanced large-volume 3D FIB-SEM analysis of one third of a fruit fly brain at a depth resolution of 8 nm. In order to analyze large volumes of hard materials such as bone, FIB-SEM using high-current beams of Xe ions available from PFIB sources has been demonstrated.⁸⁷⁷

In the context of the COVID-19 pandemic, FIB-SEM tomography proved valuable for gaining insight into the microscopic mechanisms of the disease. Baena *et al.*⁸⁶³ employed FIB-SEM to analyze how SARS-CoV-2 spreads from infected cells to new hosts, as shown

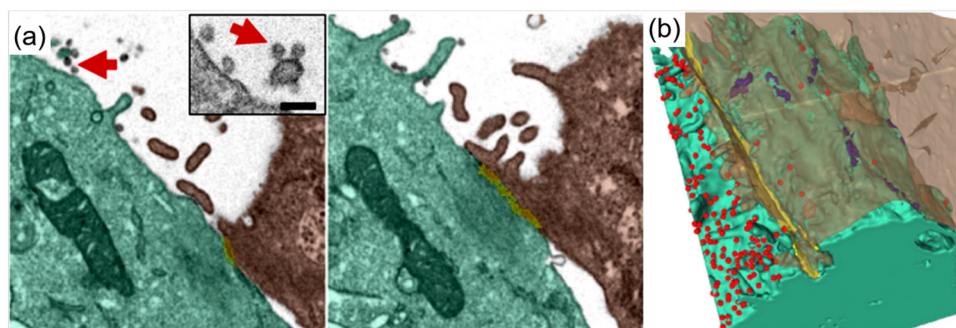


FIG. 20. (a) FIB-SEM images of two 8 nm slices showing two contacted Vero E6 cells (green, brown) and SARS-CoV-2 particles in the extracellular space (arrows).⁸⁶³ Inset scale bar: 200 nm. (b) 3D reconstruction of the plasma membranes of the two cells shown in (a). The junction-mediated contact between the cells is colored in yellow. The virus particles are depicted in red and show a different density on either side of the junction. Reproduced with permission from Baena *et al.*, *Viruses* **13**, 611 (2021). Copyright 2021 Authors, licensed under a Creative Commons Attribution (CC BY) license.

in Fig. 20. Using a combination of TEM-tomography and FIB-SEM, Cortese *et al.*⁹¹⁹ were able to visualize morphological changes in SARS-CoV-2-infected lung cells. In a recent study on the frequently observed infection of the human kidney by SARS-CoV-2, FIB-SEM was also used to image virus particles in vacuoles.⁹²⁰

4. FIB tomography of energy materials

One of the very first applications of FIB-SEM tomography in energy materials research was presented by Wilson *et al.*⁹²¹ in 2006. The authors investigated a solid oxide fuel cell anode based on yttrium stabilized zirconia (YSZ) and used 3D reconstruction to differentiate the YSZ, Ni, and pore phases. This enabled quantification of important structural features, such as volume fractions, surfaces, and even information on the so-called three-phase boundary.

In 2010, Ostadi *et al.*⁸⁴⁵ investigated a microporous layer used in a hydrogen-powered proton exchange membrane fuel cell (PEMFC). FIB tomography was used to complement the X-ray tomography investigations of the fuel cell electrode structure in order to capture the nanostructure of the microporous layer, which cannot be resolved using standard X-ray imaging equipment due to the limited spatial resolution of $\approx 1\ \mu\text{m}$ compared to the FIB-SEM tomography resolution of $\approx 10\ \text{nm}$. The porous structure was reconstructed in 3D and the porosity and pore size distribution analyzed.

In the same year, Zils *et al.*⁹²² demonstrated how FIB tomography can be used to study the effects of manufacturing parameters on the 3D structure and resulting properties of the fabricated material. For various catalyst electrode materials intended for use in PEMFCs, the authors compared porosity and pore size distributions made with air-brushing and a fast spray coating technique, and related the 3D structural features to the performance of the materials. Schulenburg *et al.*⁹²³ used FIB tomography to study degradation effects in the 3D structure of carbon-based fuel cell electrodes after many start-stop cycles and discovered breakdown of the entire nanoporous structure leading to a significant reduction in the supply of reaction gases to the catalyst. In combination with other (X-ray based) tomography techniques and structural modeling, FIB tomography can also be used for correlative tomographic imaging⁹²⁴ to fill in missing information at the nanometer scale and to produce multiscale 3D data covering structural features over a broad length scale. For example, Zielke *et al.*⁹²⁵ have shown how FIB tomography can be used to supplement the information missing from synchrotron tomography data due to the limited spatial resolution. Here, high-resolution 3D data obtained using the FIB method was used to create a structural model at the nanometer scale, to enhance the synchrotron dataset with nanometer detail. In a recent study, Paulisch *et al.*⁹²⁶ used FIB tomography to analyze the complex 3D path network inside Ag-based gas diffusion electrodes (Fig. 21). 3D data acquired with FIB tomography has also been used for mathematical modeling of the 3D structure of energy materials,⁹²⁷ whereby the computer-generated 3D structures can later be used for the development of digital twins and for future virtual materials design.

FIB tomography is thus now well established for high-resolution 3D structural studies of materials for energy-technology research, including battery materials,^{928–938} fuel cell materials,^{939–942} electrolyzers^{943–946} and catalytic materials.^{947–950} In the case of beam- and air-sensitive materials, cryogenic FIB-SEM implementations are becoming increasingly popular, since radiation damage can be significantly reduced at low temperature and cryotransfer workflows are by nature

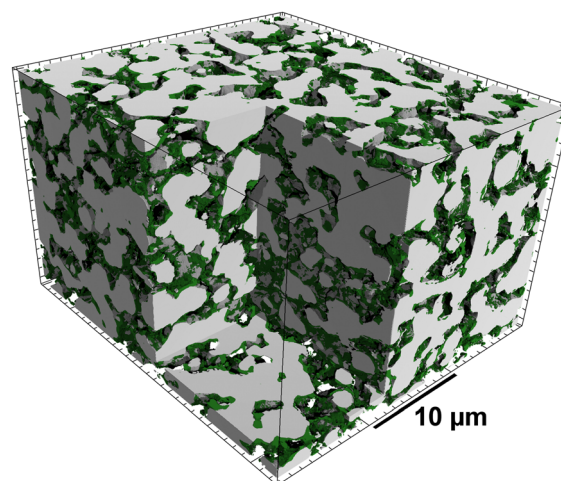


FIG. 21. FIB-SEM tomography of Ag-based gas diffusion electrode. Gray: Ag grains, green: binder/PTFE within the pore system. Only the 3D image can reveal the complex pathways inside the pore system. For more information, see Ref. 926.

air free. For example, cryo FIB-SEM techniques have enabled 3D nanoscale insight into Li dendrite formation and the so-called solid-electrolyte interface.^{951–956}

5. Data processing of FIB tomographies

With the growing volume of data, the analysis of large tensors containing mostly grayscale information is becoming increasingly challenging. Sophisticated analysis requires data classification, i.e., assigning a class to each point. This is particularly challenging for FIB based tomography, where a measurement contains a large number of classes, but in addition, artifacts that need to be detected and addressed. Here, ML approaches, especially easy-to-use mainly random forest based tools such as Ilastik⁹⁵⁷ or the trainable Weka segmentation,⁹⁵⁸ are becoming increasingly established for image annotation and classification.

One of the major issues in tomographic imaging of porous materials is the problem of separating the pore foreground from the background. In order to address this issue, infiltration of the pores with an appropriate resin may be carried out. Silicone-based resins are frequently used to infiltrate the pore system of energy materials due to the generally high carbon content of the latter.⁹⁵⁹ However, due to sample alteration upon this treatment, or inaccessibility of the pore system, pore infiltration is often either impossible or undesirable. Thus, when the FIB mills through a pore, the SEM image will also collect signal from the back of the pore. In such cases, the 3D reconstruction becomes much more complex, because only the foreground around the pore edges represents the actual material distribution corresponding to the exposed surface. To address this issue, various optimized segmentation algorithms have been developed,^{960–963} as well as some ML tools.^{964–966} Recently, Osenberg *et al.*⁹⁶⁷ demonstrated a technique combining two different SE signals (from an in-lens detector and from a detector sensitive to SEs from backscattering electrons) together with a ML algorithm to improve results for the separation of pore foreground and background in the case of a lithium battery

cathode material. However, despite much progress, the general porosity problem is still not completely solved for all material structures.

6. Outlook

Over the past 20 years, FIB-SEM has become a mainstream technique for investigating the 3D structure of a range of complex specimens, often with fully established cryo-workflows.

Further improvements involving optimized milling strategies to reduce local heating effects due to the FIB (in non-cryo implementations) would be beneficial for nanoscale milling to reveal the internal structure of biological samples and to reduce artifacts when milling energy materials, as these may also involve heat-sensitive components such as polymers.⁹⁶⁸

In parallel, the ever growing and complex data volumes demand many more automated analysis pipelines. To make this possible, standardized data and metadata formats are needed.⁹⁶⁹ The combination of standardized data sets and high quality individual image classifications generated based on, e.g., random forest models, would then allow for the compilation of diverse training sets to train deep learning models. With sufficient training data, neural networks such as the U-Nets⁹⁷⁰ can be trained and applied to data that is very different from the original training set.⁸⁴⁷

On the instrument side, much progress has been made in increasing the sample volumes accessible to FIB based tomographies through the use of high-current PFIB sources for heavy noble gas ions, especially Xe. Due also to advances in automated data acquisition and analysis, FIB tomography of cubic millimeter volumes with a spatial resolution on the order of tens of nanometers is now possible. Complex data from different experimental sources with different lateral resolutions can be made available to the research community in large multiscale atlases. This is celebrated particularly in the life sciences, where Ga-FIB-SEM was used to map and reconstruct the connectome of the *Drosophila* fly brain.^{918,971} In terms of open data, life science offers prime examples: the virtual fly brain⁹⁷² and the fly brain atlas,⁹⁷³ a digital twin of *Drosophila*'s brain combining huge amounts of data from different sources and methods in an easily accessible format. Creating and analyzing such massive data in an automated way has relied on the rapid development of ML, which will be one of the key factors for the future evolution of FIB technologies.

D. Elemental analysis using FIB-SIMS

While many correlative FIB imaging and analysis approaches exist (mainly using FIB-SEM instruments), FIB-SIMS is by far the leading method for achieving ion beam based analysis of elemental distributions. Dedicated SIMS instruments often use one or two FIB columns in combination with a highly optimized mass spectrometer to provide high mass resolution and sensitivity,^{974,975} but a dedicated FIB instrument equipped with a SIMS add-on can provide greater versatility. So far, GFIS-based FIB instruments provide the highest spatial resolution for SIMS.^{121,976} Recently, LoTIS-based FIB-SIMS has also been introduced.⁹⁷⁷

1. Overview

SIMS is based on irradiating the sample with a focused beam of ions (the so-called primary ions) and extracting, mass-filtering and

detecting the ions sputtered from the sample surface (the so-called secondary ions). It is a powerful technique for analyzing surfaces owing in particular, to its ability to detect all elements from hydrogen (H) to uranium (U), its excellent sensitivity to low concentrations, its high dynamic range and its ability to differentiate between isotopes. SIMS offers four main analysis modes, namely, mass spectrum acquisition, depth profiling, 2D imaging, and 3D imaging. In addition, SIMS does not require any complex sample preparation (in contrast to TEM based analysis techniques such as EDS or electron energy loss spectroscopy (EELS), or APT).

Therefore, equipping a FIB instrument with a SIMS system adds unique analytical capabilities for 3D chemical characterization. Furthermore, this add-on enables unique workflows for *in situ* multi-modal analytics by correlating the SIMS data with that obtained from other imaging/analytical techniques available using the same FIB platform (e.g., SE-imaging, EDS, EBSD, etc.).

2. FIB-SIMS in materials science

SIMS on FIB(-SEM) instruments has been applied to a number of topics in materials science, in particular for the nano-characterization of complex 3D architectures and compositions. Microelectronic devices are a typical example, where 2D and 3D SIMS imaging provides analysis of elemental distributions with high lateral and depth resolution.⁹⁷⁸ Figure 22 shows an example of the 3D analysis of a fin field-effect transistor (FinFET) using FIB-SIMS.¹¹⁵ These data were obtained using a magnetic sector SIMS system installed on a HIM instrument, achieving a spatial resolution of better than 15 nm. 3D maps such as these can be used for end pointing and circuit edit applications.

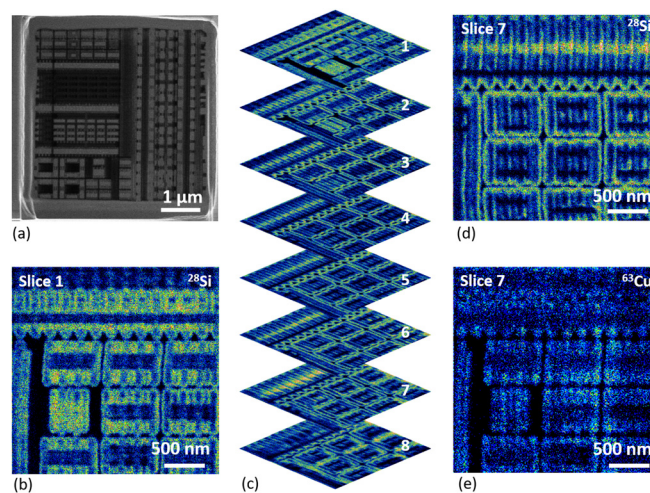


FIG. 22. SE and SIMS images of a fin field-effect transistor (FinFET) device obtained using a HIM instrument equipped with a magnetic sector SIMS system: (a) SE image acquired using 25 keV He⁺ beam. (b)–(e) SIMS images for ²⁸Si and ⁶³Cu acquired using 20 keV Ne⁺ beam corresponding to a zoomed in region in the SE image. (c) Stack of eight successive SIMS images recorded for the same area [field of view (FOV) 3 × 3 μm²] allowing determination of the elemental composition in 3D. (d) and (e) Individual ²⁸Si and ⁶³Cu SIMS maps corresponding to the seventh slice. For more information, see Ref. 115.

There are also a number of examples where FIB based SIMS has been a major asset in the development of photovoltaic and battery materials. The latest advances in photovoltaic conversion technologies are based on thin films, whether second generation, such as copper indium gallium selenide (CIGS), or third generation dye-based, such as lead halide perovskites. Their development requires a very comprehensive understanding of composition in order to correlate structure and chemical composition with performance and ageing. FIB based SIMS has allowed investigation of many questions related to the chemical heterogeneity of grains⁹⁷⁹ and grain boundaries,^{122,980} precipitation of secondary phases (e.g., PbI_2), and chemical/structural modification at the interfaces with charge-collecting electrodes.^{981–983} Optimizations based on these results have helped realize new photovoltaic cells with efficiencies approaching 25%.

In battery materials, SIMS technology allows the detection of Li (in addition to any other element of interest) with a very high sensitivity, which is not possible using, e.g., EDS. Therefore, with FIB-SIMS, the Li concentration gradient can be mapped over several orders of magnitude at high spatial resolution across different surfaces and interfaces.^{122,127,984–988}

Further examples of the application of FIB-SIMS in materials science include studies of petroleum-based polyolefin plastics,¹¹⁴ polyester and melamine resins,⁹⁸⁹ the investigation of alloys and superalloys,^{990–992} the study of nanoparticles,^{993–996} the investigation of chemical inhomogeneities in grain sizes between 50 and 200 nm in Delafossite materials,⁹⁹⁷ oxygen diffusion in grain boundaries⁹⁹⁸ the distribution of rare earths in optical fibers,^{999,1000} and the characterization of thin films.^{126,1001}

3. FIB-SIMS in the life sciences

Ion beam analysis has been used for chemical micro-analytics of biological specimens since the 1970s when high-energy PIXE was found to be valuable for determining trace elements in bulk tissue sections.¹⁰⁰² By the beginning of the 1990s accelerator-based proton beams could be focused down to $1\ \mu\text{m}$ and could thus be employed for the spatial mapping of trace elements in biological samples.⁸⁷⁹ The development of high-brightness LMIS in the second-half of the 1970s³⁷ also facilitated the introduction of medium-energy (<100 keV) ion beam microprobe techniques for bio-imaging and micro-analysis, in particular in the field of SIMS.^{1003,1004} Already at that time it turned out that SIMS is advantageous for biological applications because of its low detection limits but also its sensitivity to different isotopes. Hence in the life sciences, numerous studies based on imaging with SIMS exist.¹⁰⁰⁵ For example, FIB-SIMS has been used to spatially resolve chemical signatures from the surface and interior of mammalian individual cells¹⁰⁰⁶ and bone tissue.¹⁰⁰⁷ In this area, the Cameca nanoSIMS (a sector-field mass spectrometer with about 50 nm lateral resolution) has become an important tool.¹⁰⁰⁸ It was designed to achieve a mass resolution high enough to resolve the stable isotopes of elements such as carbon and nitrogen (^{13}C and ^{15}N), which are commonly used for isotope-labeling experiments.⁸⁸¹ In addition, TOF-SIMS approaches for 2D and 3D chemical imaging in the life sciences are becoming increasingly important.^{871,1005} For example, metabolic imaging on a sub-cellular level with a mass resolution of 240 000 was recently achieved by combining TOF-SIMS with an orbitrap mass analyzer.⁸⁵⁶

FIB-SIMS at the highest spatial resolution has been enabled using the GFIS source, which can produce sub-nanometer spot sizes. Here, HIM instruments have been equipped with SIMS spectrometers and used for high-resolution chemical analysis of a range of life science samples including micro-organisms, viruses, cells, small animals, plants, etc.⁸⁹² Recent specific HIM-SIMS applications include the localization and identification of silver nanoparticles and nanowires in intestinal or alveolar cell cultures^{994,1009} and the characterization of biogenic nanoparticles produced by bacteria.⁸⁶⁴ Finally, correlative HIM based SE imaging, STIM¹³⁸ and SIMS (all in the same instrument) has been used in investigations of thin tissue sections of *Daphnia magna* exposed to TiO_2 nanoparticles (see Fig. 23).¹⁴⁰ In this correlative nano-toxicological test study, the nanoparticles were shown to accumulate in the gut without penetration of the epithelial barrier.

Beyond materials science and life science applications, FIB-SIMS is also proving valuable for high-resolution structural and chemical analysis in other fields, such as soil science,¹⁰¹⁰ geology,¹¹³ and cosmochemistry,^{1011,1012} and even for the analysis of cultural heritage objects.^{1013–1015}

4. Outlook

In order to further increase the spatial resolution of FIB-SIMS, we must now look beyond reducing the beam probe size. In HIM-SIMS,

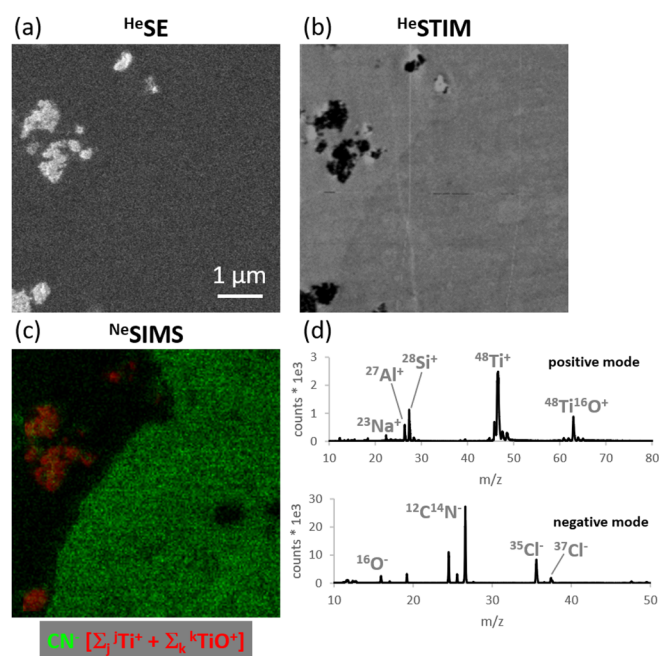


FIG. 23. npSCOPE HIM-STIM-SIMS correlative high-resolution imaging and nano-analytics of keratinocyte cell cultures exposed to 30 nm Si-Al-TiO₂ nanoparticles: (a) He-FIB SE image. (b) He-FIB STIM bright field image. (c) Ne-FIB-SIMS false color image of CN (green) and Ti species (red). (d) Ne-FIB-SIMS spectra. He-FIB SE image: FOV $15 \times 15\ \mu\text{m}^2$, (1024×1024) pixel², dwell time $1\ \mu\text{s}$; He-FIB STIM: FOV $8 \times 8\ \mu\text{m}^2$, beam current 2 pA, dwell time $400\ \mu\text{s}$, (512×512) pixel². Ne-FIB-SIMS negative mode: FOV $8 \times 8\ \mu\text{m}^2$, magnetic field 300 mT, dwell time $9000\ \mu\text{s}$, (256×256) pixel². Ne-FIB-SIMS positive mode: FOV $8 \times 8\ \mu\text{m}^2$, magnetic field 325 mT, dwell time $4000\ \mu\text{s}$, (256×256) pixel².

the lateral resolution is in fact mainly limited by the low signal yield and destructive nature of SIMS. In order to achieve the smallest possible detection limits, as much signal as possible must be extracted from the sputtered volume. One possibility here is to use other primary ions that produce a higher useful yield of sputtered ions. However, other primary ion species typically only achieve a higher yield of either positive or negative ions (not both). Higher yields of positive ions, which are particularly suitable for the detection of metal elements, are achieved using primary ions with high electronegativities, such as oxygen or halogen ions. Enhancement of positive ion yields could thus be achieved using smaller spot sizes from PFIBs or through the use of ILIS. Higher yields of negative ions are achieved by using primary ions that lower the work function of the sample surface, which is where developments using LoTIS can bring benefit by, e.g., providing Cs ions.⁹⁷⁷ Alternatively, the use of precursor gases can also increase the ion yields (see Sec. II C).

Another approach that could be explored to increase yields and therefore also spatial resolution involves further development of SNMS schemes for FIBs, in which sputtered neutrals are analyzed. Through the use of lasers for post-ionization of these neutrals, the dependence of the ion yields on the surrounding material in the sample (the so-called matrix effect) is reduced. However, since high laser intensity is required, SNMS is only possible in pulsed operation, which inevitably requires pulsing of the primary ion beam. Here, the use of TOF spectrometers becomes an obvious choice. Post-ionization with focused electron beams is also conceivable, although preliminary experimental work in this area is scarce.

A further issue in FIB-SIMS that needs to be addressed is material intermixing induced by the primary ions. In order to achieve the smallest beam probe sizes, comparatively high beam energies in the range of 20–50 keV are typically used. However, the disadvantage is the resulting long range of the ions in the target (compared, e.g., to that of the low energy Ar ions commonly used for SIMS depth profiling). Long ion ranges can quickly lead to atom mixing of deeper layers and therefore to a decrease in depth resolution with increasing sputter depth.¹⁰¹⁶ In the future, primary ions with high atomic number and/or small kinetic energy per nucleon could be used. For example, Cs ions from LoTIS or Xe ions from GFIS could provide both small beam sizes and good depth resolutions. Alternatively, the energy per nucleon can also be reduced by using cluster ions, e.g., Bi or Au clusters from LMAIS (see Table I).

Small cluster ions,¹⁰¹⁷ reactive primary ions, e.g., Cs,^{31,46} or laser post-ionization¹³⁰ can also reduce the fragmentation of molecules, which is another aim for future developments (see references in Table II). Molecule fragmentation needs to be understood both theoretically and experimentally in order to deduce the molecular composition of the sample from the measured spectra. In this regard, the creation of molecular fragmentation databases for typically used ion energies and types would enable fingerprinting in future work.

Continued spectrometer development is also needed to enable complete detection of all sputtered ions. Here, the TOF spectrometers and magnetic sector spectrometers with large area MCPs lead the way. Development efforts should also be invested into spectrometers that allow higher mass resolution in order to enable the separation of isobars. In this area, results obtained using Orbitraps are very promising.⁸⁵⁶

In the perfect FIB-SIMS instrument, all the key parameters (spatial resolution, mass resolution, detection limit, measurement time)

would be optimized simultaneously. In reality one has to compromise between different priorities. Furthermore, *in situ* combinations of FIB-SIMS with other analytical modes, e.g., SEM based analysis, are highly desirable. The integration of an electron flood gun for *in situ* charge neutralization (as is implemented in HIM) would be universally beneficial for the analysis of insulating sample surfaces, and cryo-FIB-SIMS would be of great interest for life science and other beam-sensitive samples. *In situ* AFM would allow correlation of the SIMS data with the actual sputtering depths enabling more accurate 3D reconstructions.

Finally, most elemental analysis methods, including SIMS, benefit from vacuum levels that are better than the usual high vacuum (HV) conditions of typical FIB chambers in order to avoid background signals originating from residual gas. FIB instruments with UHV or near UHV chambers have been built by researchers in the past^{140,1018} and the benefits for normal imaging as well as analysis have been demonstrated.¹⁴⁹ Manufacturers should similarly consider the merits of UHV in future instrument development.

E. Applications of gas-assisted FIB processing

Gas-assisted bottom-up prototyping of nanostructures using the FIB is known as focused ion beam induced deposition (FIBID). In this technique, the focused ion beam induces the partial decomposition of precursor molecules, dissociating them into volatile and nonvolatile components. The volatile products are pumped away, while the non-volatile products are deposited on the surface, thus creating a nanostructure in a single-step process. This direct writing of nanostructures using the FIB is powerful, flexible, and conceptually simple, but at the same time extremely complex in the chemical and physical processes involved. In addition to the surface processes and electron-driven chemistry that also occur in direct writing with electron beams, competition with physical sputtering, and thermal or ion induced structural changes, contribute here to the shape and microstructure of the deposit. Hence, FIBID strongly depends on multiple growth conditions, such as ion species, acceleration voltage, ion beam current, ion dose, beam dwell time, scanning strategy, substrate type (insulating or conductive), GIS nozzle position, precursor molecule (vapor pressure, sticking coefficient to the surface, residence time, dissociation cross section), precursor temperature, base and process pressure in the vacuum chamber, among others. This makes FIBID of functional nanostructures highly dependent on the experience of the particular user and calls for better theoretical understanding and empirical modeling tools that generate patterns based on few calibration routines, thus also enabling less experienced users to fabricate complex nanostructures. Table VIII provides an overview of the fields where FIBID processing has been employed so far.

1. Basics of ion induced chemistry

The ion–molecule reactions that have been studied in the context of FIBID involve decomposition of the precursor on the substrate surface during ion bombardment. In FIBID of metal nanostructures from organometallic precursors, the deposited material is composed of the metal(s) and contamination from nonvolatile ligand-derived decomposition byproducts. Ion induced chemistry during FIBID is only partially understood, but has been postulated to involve ion–molecule collisions with the precursor, reactions with low energy SEs produced

TABLE VIII. Overview of applications of gas-assisted FIB processing, in which gaseous precursor compounds are locally dissociated by the ion beam to grow nanostructures.

Property	Target application	Ion	Deposit geometry and precursor
Electrical and electronic	Electrical connections	He	Semiconducting nanowires from CpPtMe ₃ ¹⁰¹⁹ and metallic Co lines from Co ₂ (CO) ₈ ¹⁰²⁰
		Ne	Metallic nanowires from CpPtMe ₃ ¹⁰¹⁹
		Ga	Conducting nanowires from CpPtMe ₃ , ^{1021,1022} MeCpPtMe ₃ , ^{1023–1025} Au (hfac)Me ₂ , ^{1026,1027} Co(CO) ₃ NO, ¹⁰²⁸ Cu(hfac)(VTMS), ¹⁰²⁹ W(CO) ₆ contacts to Co and Cu nanowires, ^{1030–1032} Pt-based sidewall contacts using MeCpPtMe ₃ to MoSi layer stack ¹⁰³³
	Insulating barriers	Si	Thin layer from TMOS [Si(OCH ₃) ₄] ¹⁰³⁴
	Superconductivity	Ga	Thin layer from TMOS ^{1035,1036} and PMCPS [(CH ₃ SiHO) ₅] ¹⁰³⁷
		He	Planar and 3D superconducting wires from W(CO) ₆ ^{1038–1041}
		Ga	Planar nanowires from W(CO) ₆ ^{1042–1052} and Nb(NMe ₂) ₃ (N- <i>t</i> -Bu), ^{1053,1054} arrays of Josephson junctions from Nb(NMe ₂) ₃ (N- <i>t</i> -Bu), ¹⁰⁵⁵ 3D SQUID from W(CO) ₆ , ¹⁰⁵⁶ nanoSQUID from W(CO) ₆ ¹⁰⁵⁷
Magnetic	Nanomagnets	He	Nanowires from Co ₂ (CO) ₈ ¹⁰²⁰
		Ga	Nanoparticles from Co ₂ (CO) ₈ , ¹⁰⁵⁸ nanowires from Co(CO) ₃ NO ¹⁰²⁸ and Co ₂ (CO) ₈ ¹⁰⁵⁹
	Nanomagnetic probes	Ga	Filling FIB milled trenches with W, ¹⁰⁶⁰ magnetic force microscopy (MFM) tips via deposition of thin layer by milling Co ₇₁ Cr ₁₇ Pt ₁₂ target, ¹⁰⁶¹ Hall-probes from Co ₂ (CO) ₈ ¹⁰⁶²
Optical	Photonic components	Ga	Deposition of defects in etch mask for photonic crystal, ¹⁰⁶³ deposition of the central Pt beamstop in Fresnel zone plates, ⁵⁶⁶ chiral nano-antennae, i.e., helices from MeCpPtMe ₃ ^{1064–1066} and from phenanthrene ¹⁰⁶⁷
	Plasmonic components	Ga	Chiral nano-antennae, i.e., helices from phenanthrene covered with Au ^{1068–1070}
	Chiral biosensing	Ga	Helix arrays from MeCpPtMe ₃ coated with poly o-phenylenediamine ¹⁰⁷¹
Mechanical	Atomic force microscopy (AFM) tips	He	Hammerhead probes from MeCpPtMe ₃ ¹⁰⁷²
	Tunable elastic moduli	Ga	DLC (diamond-like carbon) core-shell pillars and springs from phenanthrene, ^{1073,1074} metal containing pillars and springs from phenanthrene and ferrocene, ¹⁰⁷⁵ MeCpPtMe ₃ , ¹⁰⁷⁶ W(CO) ₆ , ¹⁰⁷⁷ TMCTS [(HSiCH ₃ O) ₄], and O ₂ ¹⁰⁷⁸

by impact of ions with the substrate, and thermal processes driven by dissipation of the collision energy (see also Fig. 14).^{431,1079} An additional feature of FIBID is the concomitant sputtering of material from the surface that competes with deposition. Although preferential sputtering of ligand-derived atoms can purify the material under appropriate conditions,^{451,1080} etching of the metal will also occur and the choice of reaction conditions is critical to obtaining deposits with high metal content.

Broad application of FIBID will require development of custom precursors that undergo well-understood and efficient ion induced decomposition to metal atoms and volatile byproducts. In addition to the standard physical properties common to all deposition methods that involve gas phase delivery of reactants to the substrate (volatility, thermal stability, ready availability, etc.), mechanism-based design^{1079,1081} of FIBID precursors will require a fundamental understanding of ion–molecule reactions on the surface. Much of the current understanding of these processes comes from model studies of FIBID

carried out by ion bombardment of a few monolayers of precursor molecules on a substrate in UHV.^{1079,1081} Although there are differences between FIBID and its UHV models (the UHV environment is cleaner than standard HV FIBID deposition conditions, UHV studies are carried out at lower temperatures, and the ion source is more diffuse), UHV studies can be used to identify ligands that dissociate from the complex and ultimately desorb cleanly from the surface during ion–molecule reactions. Purification of deposits by sputtering away ligand-derived contaminants can also be modeled in UHV. Additional mechanistic information has been obtained in gas phase studies of ion–molecule collisions between organometallic precursors and ions of different masses and energies.¹⁰⁸²

As an example, a study of the reactions of Ru(CO)₄I₂ with 860 eV Ar⁺ ions on a surface in UHV demonstrated that the initial reaction was loss of all four CO ligands to form adsorbed RuI₂.⁴⁵¹ Additional ion exposure resulted in sputtering of iodine from the surface at a rate five times faster than sputtering of Ru. These results suggest that

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proper optimization of FIBID conditions could result in deposition of high metal content Ru from this precursor. This study also highlights the difference in the charged-particle induced reactivity between FIBID and FEBID, as modeling of the FEBID process in UHV with 500 eV electrons results in loss of the CO ligands, but not the iodide, to afford RuI_2 as the final product.¹⁰⁸³ Rapid loss of all of the carbonyl ligands was also observed upon ion bombardment (860 eV) in a UHV model study of the heterobimetallic precursor $(\eta^5\text{-C}_5\text{H}_5)\text{Fe}(\text{CO})_2\text{Re}(\text{CO})_5$.¹⁰⁸⁰ In contrast to the electron-induced chemistry of this complex under the same conditions, the ion beam slowly removed the remaining carbon from the cyclopentadienyl ligand by sputtering, leaving behind non-stoichiometric Fe/Re material.

Additional information on ion–molecule reactions was obtained during studies of FIBID of Au from $(\text{Me})_2\text{Au}(\text{hfac})$ (hfac = hexafluoroacetylacetonate) with He^+ , Ne^+ , Ar^+ , Kr^+ , or Xe^+ at energies of 2–100 keV.^{452,1084} The decomposition yield from the precursor increased with increasing ion mass and energy, consistent with a BCA model for decomposition. The authors concluded that the FIBID process was driven by energy deposition from ion–surface collisions. These results have implications for ion source choice in FIBID, because the variation in ion mass impacts the deposition chemistry through differences in SE yields, sputter yields and ion implantation yields. Ar^+ and Ne^+ generate sufficient SEs at low beam energies (30 eV) but have a shallower beam penetration depth than He^+ , leading to lower ion–substrate interactions. Heavier ions have higher sputtering yields, which can remove light atom impurities but at the cost of deposition rates, as the target material will also be sputtered away.^{56,1085}

Gas phase studies of reactions of $\text{Fe}(\text{CO})_5$ with He, Ne, Ar, and Kr ions of varying charge and energy¹⁰⁸² also demonstrated facile loss of CO ligands upon ion–molecule collision. The distribution of fragment ions from CO loss $\text{Fe}(\text{CO})_{5-x}$, where $x=1-5$, was dependent on the ion mass and the energy transfer in the collision. Under these conditions, light ions such as He^+ resulted in electronic excitation with low fragmentation efficiency. Heavier ions yielded more extensive CO dissociation from energy transfer and nuclear stopping. The highest extent of CO loss occurred upon collision with Ne^+ , where both electron transfer and energy transfer were significant. It was pointed out that the gas phase results may not accurately predict behavior on a surface during FIBID due to the heat sink behavior of the substrate. However, more efficient fragmentation upon impact with heavier ions is to be expected. Although there is not yet an extensive set of privileged ligands for FIBID, some preliminary conclusions about ion–molecule chemistry of organometallic precursors on substrate surfaces can be drawn from the mechanistic studies described above,¹⁰⁷⁹ namely:

1. To minimize ligand-derived contamination in deposits, ligands should be small and few in number;
2. CO ligands can be removed quite easily by ions;
3. Halide ligands can be removed by sputtering, but more slowly than CO;
4. Carbon derived from anionic π -facial ligands such as cyclopentadienyl and allyl can be removed by sputtering, but this is also slow and competes with removal of metal.

2. FIBID nanostructures for electronics

The ease to directly write functional structures with different conductivities renders FIBID a powerful tool in electronic applications,

with at least three different electrical conductivity regimes reported: semiconducting, metallic, and superconducting at low temperature (see Table IX).

Key parameters here are the carbon content of the deposit as well as its microstructure, which depend on the growth conditions (as well as on the precursor selected). Modulating the growth conditions thus offers an efficient way to tune electrical and electronic behavior. Examples include Pt¹⁰²⁴ and Au^{1026,1027} deposition using Ga-FIBID, for which the conductivity increases with the thickness of the as-grown material. This dependence is the result of a gradient in metal content through the film thickness, with a higher carbon content in the first deposited layers vs at the bottom. Pt-based deposits below 50 nm in thickness behave as semiconductors, but display metallic behavior¹⁰²³ for thicknesses above 150 nm.¹⁰²⁴ The metal content of these deposits is less than 33 at. %, consisting of fcc Pt nanocrystals with a diameter of 5–10 nm embedded in an amorphous carbon matrix.¹⁰²⁵ The observed conductivity regimes are expected to be strongly influenced by the disorder and doping introduced by Ga ion impact.¹⁰²⁵ In the case of Cu deposits (copper nanocrystals embedded in amorphous carbon), it has been found that the ion beam current also directly controls the metal content and thus the deposit conductivity.¹⁰²⁹ Furthermore, the ion species used for the deposition can have a great effect. For example, using Ne-FIBID, Pt deposits exhibit metallic behavior, whereas they behave as semiconductors using He-FIBID.¹⁰¹⁹ In the case of Ga-FIBID of Au-based deposits, metallic behavior is easily obtained, with significant Au contents of 75 at. % and resistivities exceeding the bulk resistivity of Au by two orders of magnitude.¹⁰²⁷

A number of works have investigated FIBID of Co deposits due to their magnetic properties, for example, as-grown^{1058,1086} and annealed¹⁰²⁸ Ga-FIBID Co deposits (containing 50 at. % Co) exhibit ferromagnetic characteristics and resistivity values at room temperature that exceed the resistivity of bulk Co by several orders of magnitude. Using He-FIBID, nanostructures can be fabricated with higher lateral resolution, such as metallic Co lines of 10 nm in width displaying room temperature resistivities comparable to those grown using Ga-FIBID.¹⁰²⁰

Intriguing FIBID-grown materials that exhibit metallic behavior at room temperature and transition into the superconducting state at low temperature include W and Nb based deposits, which become superconducting below 4.2–5.1 K (Refs. 1042, 1043, 1047, and 1050) and 5 K,¹⁰⁵³ respectively. In the case of freestanding 3D NbC nanowires, a broadened superconducting transition between 4 and 11 K has been reported.¹⁰⁵³

The WC nanostructures, in particular, exhibit remarkable superconducting properties, such as upper critical magnetic field values of $B_{c2}(2\text{K}) = 7\text{--}8.5\text{ T}$ (Refs. 1048 and 1050) and critical current density values of $J_c(2\text{K}) = 0.01\text{--}0.1\text{ MA cm}^{-2}$.^{1042,1044–1046,1050} For very narrow nanostructures ($\leq 50\text{ nm}$ in width), finite size effects that are key to their application in superconducting circuits have been studied. This includes phenomena such as a magnetic field-driven re-entrance of the superconductivity,¹⁰⁴⁹ the occurrence of non-local voltages,^{1051,1052} and electric field-induced control of superconductivity.¹⁰⁴¹

Furthermore, He-FIBID has facilitated the fabrication of sophisticated 3D WC nanosuperconductors with additional outstanding characteristics. These include a critical temperature ranging from $T_c = 6.2\text{--}7.1\text{ K}$ and an upper critical magnetic field value extending from $B_{c2}(0) = 12\text{--}15\text{ T}$. Noteworthy among these structures are hollow

TABLE IX. Overview of various precursor ion combinations and the resulting electrical properties. Room temperature resistivity values ρ_{RT} are given for the respective deposits and for the pure bulk metal from literature.

Material	Precursor compound	Ion	Shape	Composition (at. %)	Microstructure	Electrical behavior	ρ_{RT} ($\mu\Omega$ cm)	ρ_{RT} bulk ($\mu\Omega$ cm)	References
Platinum	MeCpPtMe ₃	Ga ⁺	Planar	Pt:C:Ga:O 46:24:28:2	Amorphous	Metal	70–700	10.4	1023
	MeCpPtMe ₃	Ga ⁺	Planar	Pt:C:Ga:O 22:60:11:7	5–10 nm nanocrystals	Metal	700		1024
	MeCpPtMe ₃	Ga ⁺	Planar	Pt:C:Ga 17:72:11	≈3 nm nanocrystals	Metal	800		1025
	MeCpPtMe ₃	Ne ⁺	Planar	Pt:C 17:83	≈4.5 nm nanocrystals	Metal	600		1019
	MeCpPtMe ₃	He ⁺	Planar	Pt:C 16:84	≈3 nm nanocrystals	Semiconductor	180000		1019
Gold	Au(hfac)Me ₂	Ga ⁺	Planar	Au:C:Ga 50:35:15	–	Metal	500–1500	2.4	1026
	Au(hfac)Me ₂	Ga ⁺	Planar	Au:C:Ga:O 75:<5:20:<5	–	Metal	500–1300		1027
Cobalt	Co ₂ (CO) ₈	Ga ⁺	Planar	–	–	Metal	189	6	1058
	Co ₂ (CO) ₈	Ga ⁺	Planar	–	Amorphous	Metal	19–38		1059
	Co(CO) ₃ NO	Ga ⁺	Planar	Co:C:Ga:N:O 54:7:9:13:17	>20 nm nanocrystals	Metal	189		1028
Copper	Co ₂ (CO) ₈	He ⁺	Planar	–	6 nm nanocrystals	Metal	64–116		1020
	Cu(hfac)(VTMS)	Ga ⁺	Planar	Cu:C 55:45	20 nm Nanocrystals	Metal	50	1.7	1029
Tungsten	W(CO) ₆	Ga ⁺	Planar	W:C:Ga 40:40:20	Amorphous	Superconductor below 5.2 K	200	5.6	1042,1043
	W(CO) ₆	Ga ⁺	Planar	W:C:Ga:O 40:43:10:7	1–2 nm nanocrystals	Superconductor below 4.2–5.1 K	275		1050
	W(CO) ₆	Ga ⁺	Planar	W:C:Ga:O 53:34:11:2	≈1 nm nanocrystals	Superconductor below 5–5.5 K	100–350		1047
	W(CO) ₆	He ⁺	3D	W:C:O 72:20:8	20–30 nm nanocrystals	Superconductor below 6.2–7.1 K	398		1038–1040
Niobium	Nb(NMe ₂) ₃ (N-t-Bu)	Ga ⁺	Planar	Nb:C:Ga:N 29:43:15:13	15–20 nm nanocrystals	Superconductor below 5.0 K	550	15	1053
	Nb(NMe ₂) ₃ (N-t-Bu)	Ga ⁺	3D	–	15–20 nm nanocrystals	Superconductor below 8.1 K	380		1053

nanowires with an outer diameter of 32 nm and aspect ratios of about ≈ 200 ,^{1038,1040} as well as nano-helices with a diameter of 100 nm and an aspect ratio of about ≈ 65 ¹⁰³⁹ (see Fig. 24). These nanostructures exhibit a metallic content of 70 at. % W and consist of 20–30 nm-sized fcc WC_{1–x} nanocrystals.

3. FIBID nanostructures for mechanics

One of the main advantages of FIBID over standard lithographic techniques is the capability to directly write high aspect ratio structures onto any substrate topography. When it comes to mechanical applications, this allows, e.g., placement of dedicated tips onto AFM cantilevers and spring structures into prefabricated mechanical devices. A recent review by Utke *et al.*¹⁰⁸⁷ summarizes the mechanical properties

of FIBID pillars and springs. All structures were deposited using Ga-FIBID (ion energies mainly 10–30 keV, beam currents 0.5–9 pA). An interesting finding was that the Ga-FIBID pillars tended to grow as core-shell structures. Using the precursor phenanthrene (C₁₄H₁₀), elastic moduli of the hydrogenated carbonaceous material in the core and shell were ≈ 300 GPa (stiff core with Ga implantation, low H content) and ≈ 30 GPa (soft shell, presumably high H content).^{1073,1074} Depending on the inner diameter of the core and the thickness of the shell, the elastic modulus of the structure ranges between these boundary values. Metallic pillars with metal contents <40 at. % (i.e., below the percolation threshold) have mechanical properties dominated by the co-deposited carbonaceous matrix.¹⁰⁸⁷ Increasing metal contents yield elastic moduli as follows (metal at. %/elastic modulus normalized to that of pure metal): Fe (21 at. %/0.4),¹⁰⁷⁵ Pt (60 at. %/0.8),¹⁰⁷⁶ W

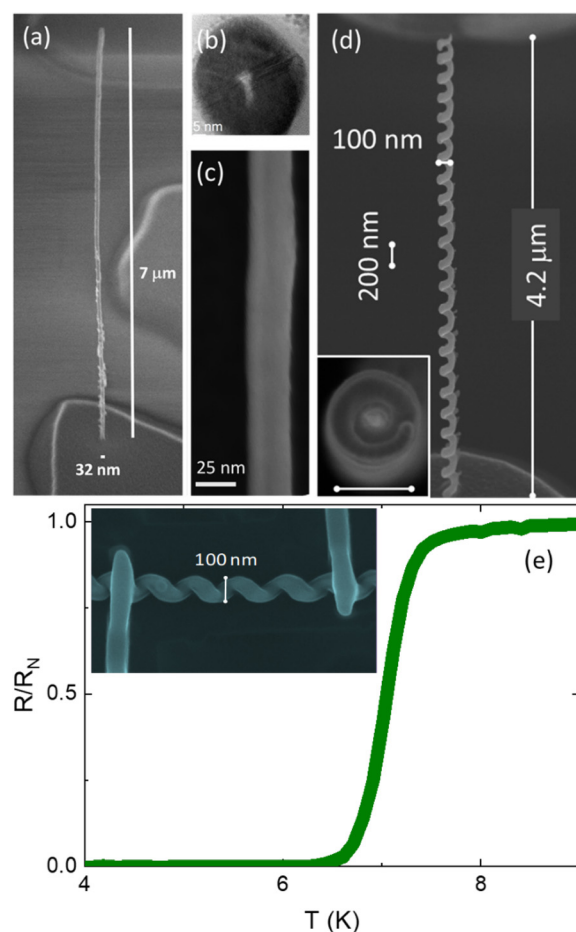


FIG. 24. 3D FIBID examples. (a) SEM image of a hollow crystalline WC superconducting nanopillar grown by He-FIBID.¹⁰³⁸ (b) Cross-sectional TEM view of the hollow nanowire. (c) Higher magnification SEM image of (a). Reproduced with permission from Córdoba *et al.*, *Nano Lett.* **18**, 1379–1386 (2018). Copyright 2018 American Chemical Society. (d) Superconducting WC nano-helix grown by He-FIBID.¹⁰³⁹ Inset shows its top view. (e) Superconducting transition of the nano-helix. Inset shows the nano-helix contacted for the electrical transport measurements. Reproduced with permission from Córdoba *et al.*, *Nano Lett.* **19**, 8597–8604 (2019). Copyright 2019 American Chemical Society.

(80 at. %/0.4),¹⁰⁷⁷ and SiO₂ (95 at. %/0.8).¹⁰⁷⁸ The non-correlation between the metal content and elastic modulus of the pillars may be due to Ga implantation, which softens the metals. Furthermore, the carbonaceous matrix stiffens or softens the pillar depending on its sp² vs sp³ hybridization ratio and its hydrogen content.

So far, FIBID is typically not used to fabricate high aspect ratio scanning probe tips, in contrast to FEBID; see review by Plank *et al.*¹⁰⁸⁸ This can be attributed to the strong tendency of the Ga ions to mill the cantilever substrate rather than induce tip (pillar) growth. Implantation of Ga into the cantilever can also be a problem. Here, the enhancement of deposition yields using pulsed irradiation can bring benefit.⁴³¹ Since the sputter yield of He ions is much lower, He-FIBID also shows promise in this area and has been used for the fabrication of 3D hammerhead AFM probes.¹⁰⁷² In the case of strain sensors

based on piezoresistive granular material, FEBID is also still the method of choice.¹⁰⁸⁹ However, here too changing to He or another light ion species for FIBID could prove fruitful.⁴³⁴

4. FIBID nanostructures for photonics and plasmonics

FIBID is also a powerful nanofabrication tool for the exploration of novel optical properties thanks to its unique design flexibility.^{302,1090,1091} The site specific and local nature of FIBID allows the growth of photonic structures on different material substrates,¹⁰⁹² providing a path to the fabrication of a number of optical elements, such as chiral optical antennae,^{1064,1069} Fresnel lenses,⁵⁶⁶ and nanosensors.¹⁰⁷¹ Combined with large area laser interference lithography to fabricate a large area photonic crystal, FIBID has been used to define defects to form a multimode waveguide.¹⁰⁶³

Under optimized parameter conditions, it is possible to evolve planar structures along the third dimension, thus manufacturing free-standing nanostructures with more exotic optical properties. For example, 3D chiral nano-helices, prototyped by FIBID, have shown tunable chiroptical response at optical frequencies^{1065,1066} [Fig. 25(a)], been used as subwavelength circular polarization optics [Fig. 25(b)]¹⁰⁶⁹ exploited in compact and miniaturized high-sensitivity biosensors,¹⁰⁷¹ and integrated onto the apex of background-free and broadband chiroptical probes.¹⁰⁶⁸

The wide range of available gas precursors enables the deposition of metal and dielectric materials that can be applied in the emerging field of photonic resonant nanostructures.⁴³¹ Pure and homogeneous compositions are, however, intrinsically hindered due to various aspects of the deposition process, such as incomplete dissociation of the precursor molecules and the residual contamination from hydrocarbons present in the vacuum chamber, both of which result in carbon content in the nanostructure, and the inevitability of implanted ions, which cause absorption losses, and in some cases, material damage. As a result, FIBID materials exhibit either a nanocrystalline composite structure^{1023,1025,1093} or a core/shell architecture, depending on the precursor properties and on the FIB irradiation conditions.¹⁰⁹⁴ Apart from purification approaches, which have mostly been used for FEBID (and which might alter the nanostructure shape and dimensions),^{1095–1098} or reduced implantation effects due to enhanced deposition yields (which require the employment of cryo-FIBID^{294,295,1051}) other promising strategies to control the final optical response are post-processing coating methods. In the case of complex 3D shapes, conformal coverage can be obtained by combining FIBID with glancing angle deposition of metals,^{1068,1069} cyclic voltammetry coating of polymers¹⁰⁷¹ and atomic layer deposition of oxides.¹⁰⁹⁹ Also, the core-shell architecture (resulting from the scattering of Ga into a lower molecular weight matrix) can be controlled to engineer specific optical effects.^{1100,1101} Other approaches may emerge from the exploitation of other ion beam sources for FIBID, such as LMAIS.^{47,1102}

5. Outlook

Focused ion beams are a flexible and versatile tool for gas-assisted processing, but so far have been much less used than focused electron beams for this purpose. The reason for this may be the dominance of Ga-FIB, whereby Ga ions are not only implanted into the deposits but can also cause significant physical sputtering and trigger undesirable chemical and physical side effects. In addition, there is still no

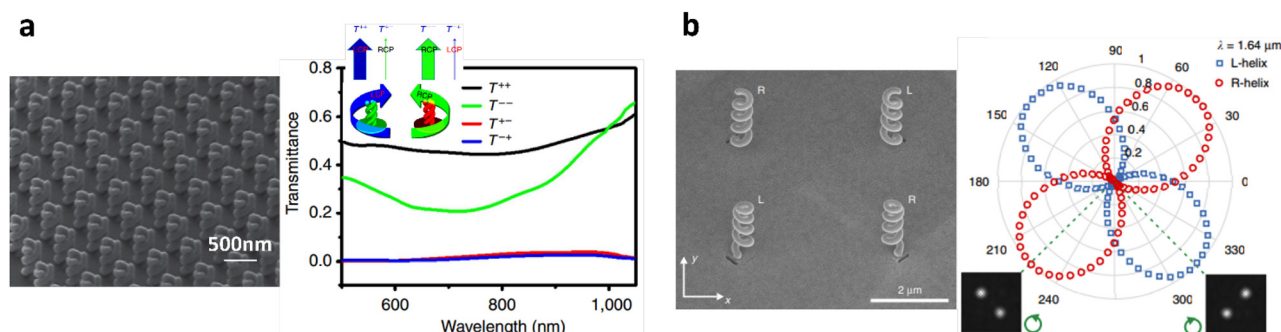


FIG. 25. Examples of photonic nanostructures realized using FIBID and their related optical properties. (a) SEM image of an array of intertwined nano-helices and their transmission spectra T_{++} , T_{--} , T_{+-} , T_{-+} for right and left circularly polarized incident light (RCP and LCP, respectively).¹⁰⁶⁶ The inset in the plot presents a scheme of the handedness of incident and transmitted circular polarization. Reproduced with permission from Esposito *et al.*, Nat. Commun. **6**, 6484 (2015). Copyright 2015 Springer Nature Limited. (b) SEM image of subwavelength optical device made of helices with opposite handedness (L and R) coupled to orthogonal apertures.¹⁰⁶⁹ This system enables conversion of an incoming light beam into four beams of tunable polarization and intensity. The plot shows helicity analysis of the device, recording the transmitted power after placing a rotating quarter-wave plate and a fixed polarizer in front of the detector. The insets show the far-field optical images of the device when the quarter-wave plate rotates by $\pm 45^\circ$. Reproduced with permission from Wang *et al.*, Light **8**, 76 (2019). Copyright 2019, Authors, licensed under a Creative Commons Attribution (CC BY) license.

precursor development specifically for FIBID. Deposition with ions can in fact deliver purer and denser materials due to their much higher mass compared to electrons. In this regard, ion beams from noble gases, which are chemically inert, are of particular interest. While several studies have already presented promising results for FIBID using He and Ne ions (cf. Table VIII), FIBID using beams of Ar, Kr, or Xe ions from PFIBs has not yet been reported and should be explored.

F. Novel and unconventional approaches in FIB processing

Apart from removing or depositing material, introducing dopants or creating defects, the interaction between ions and solids can be used to change materials chemically or morphologically. Such “nonstandard” approaches include FIB-induced cross-linking or dissociation in different types of organic and metal–organic surface layers and also the triggering of self-organization processes. Furthermore, “standard” FIB procedures can be combined with other nano- and microfabrication techniques to mitigate specific limitations and broaden applicability.

1. Resist patterning by FIB

In resist based lithography, the goal is to induce a chemical change in the exposed resist layer to render it either soluble (positive tone resist) or insoluble (negative tone resist) to the subsequent development step. Resist based nanopatterning techniques using photons (optical lithography) and electrons (EBL) are common in industry, with the main properties being, respectively, high throughput and high resolution.¹¹⁰³ With the aim of improving the throughput of EBL while maintaining a high resolution, pioneering studies demonstrated the use of Ga-FIB to expose resists.^{39,1104–1106} Ions have the advantage over electrons that due to their mass they do not induce backscattering from the underlying substrate and, therefore, reduce unintended exposure of neighboring pixels (known as the proximity effect). For example, line widths of less than 100 nm and a sensitivity of 8×10^{11} ions/cm² were found using a commercial negative resist and 30 kV Ga-FIB

irradiation.¹¹⁰⁵ Line widths of less than 30 nm and a sensitivity of 1×10^{13} ions/cm² were achieved using the positive poly(methyl methacrylate) (PMMA) resist and 50 kV Ga-FIB irradiation.¹¹⁰⁴ Furthermore, using 50 kV Ga-FIB irradiation of the negative poly(phenyl silsesquioxane) (PPSQ) resist, sub-100 nm tunneling tip structures were fabricated for a sensing device.¹¹⁰⁶ Yet the drawbacks of Ga-FIB include sputtering of the resist and a relatively large beam spot size (several nanometers) with significant beam tails. Compared to the significant developments of EBL over the following years, these drawbacks hampered further use of FIBs for resist exposure until the advent of the HIM. Helium ion beam lithography (HIBL) combines a smaller probe size with a low sputter yield, high resist sensitivity, and a reduced proximity effect.¹¹⁰⁷

First studies published in 2009 already highlighted the potential of HIBL for sub-10 nm patterning.^{1108,1109} In both cases, hydrogen silsesquioxane (HSQ) negative resist was used. An ion dose of only $31 \mu\text{C cm}^{-2}$ was required for complete exposure¹¹⁰⁹ [see Fig. 26(c)]. Compared to EBL using the same resist, the dose required for HIBL is lower by a factor of four. HIBL can, therefore, simultaneously provide a similar resolution to EBL with a better throughput. HIBL with HSQ has also been employed to fabricate a hard mask for subsequent metal etching¹¹¹⁰ [Fig. 26(a)] and as a mold for nanoimprint¹¹¹¹ [Fig. 26(b)].

The 3D nanofabrication of complex crosslinked HSQ nanostructures, such as embedded nanochannels and suspended grids, was demonstrated using different exposure configurations.¹¹¹² Furthermore, due to the large depth of field of HIBL, large-area dense patterns have been achieved on HSQ as well as exposures on tilted surfaces.¹¹¹³ The use of the Ne-FIB for HSQ exposure has proven interesting, providing a trade-off between high resolution and dose efficiency.¹¹¹⁴

HIBL using other resists has also been investigated. For example, sub-10 nm resolution has been obtained on a fullerene based resist using a dose of $40 \mu\text{C cm}^{-2}$, which is three orders of magnitude lower than that of a similar process using EBL.¹¹¹⁵ Likewise, sub-10 nm resolution was achieved on an alumina-based resist, and 10 nm lines (with an aspect ratio of 1:10) were then transferred to Si by means of low-bias reactive ion etching.¹¹¹⁶ HIBL using a new metalorganic negative resist, $\text{Cr}_8\text{F}_8(\text{pivalate})_{16}$, has been shown to enable sub-10 nm

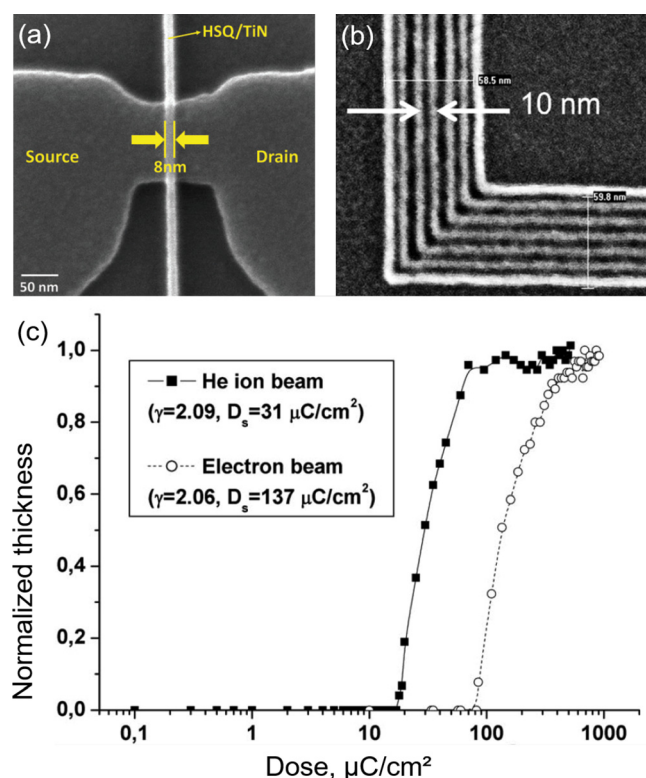


FIG. 26. (a) 8 nm-wide hydrogen silsesquioxane (HSQ) lines patterned by helium ion beam lithography (HIBL) and used as hard masks for subsequent etching of TiN gate.¹¹¹⁰ Reproduced with permission from J. Mater. Chem. C **9**, 8285–8293 (2021). Copyright 2021 Royal Society of Chemistry. (b) High-resolution dense nano-patterns in HSQ written using HIBL. Reproduced with permission from Li *et al.*, J. Vac. Sci. Technol., B **30**, 06F304 (2012). Copyright 2012 AIP Publishing. (c) Comparison of the charge dose required for HIBL vs electron beam lithography (EBL) using HSQ resist.¹¹⁰⁹ Reproduced with permission from Sidorkin *et al.*, J. Vac. Sci. Technol., B **27**, L18–L20 (2009). Copyright 2009 AIP Publishing.

resolution patterning using a line irradiation dose of 20 pC/cm (three orders of magnitude lower compared to EBL), demonstrated for Si and W substrates with good resistance to subsequent dry etching.¹¹¹⁷ New hybrid organic-inorganic resists based on Ni have also been demonstrated to perform well in HIBL, requiring an irradiation dose of 22 $\mu\text{C}/\text{cm}^2$, producing sub-10 nm resolution with low line edge/width roughness.¹¹¹⁸ Notably, HIBL using a new resist composed of a radiation-sensitive monomer (MAPDST) with an organic functionality (TIPMA) only requires a very low exposure dose of 6 $\mu\text{C}/\text{cm}^2$.¹¹¹⁹ The exposure of PMMA to He and heavier ions has also been investigated.¹¹²⁰

Additionally, HIBL has shown good performance in combination with liftoff processes, in particular with Au, which has been used to fabricate plasmonic nanoantennae.¹¹²¹ A different approach consists of the direct He- and Ne-FIB milling of HSQ or PMMA resists that have been previously patterned using standard EBL.¹¹²² This can, for instance, be used for pattern repair. HIBL has also been proposed as a method for the (pre-) screening of EUV resists to help evaluate their performance.¹¹²³

In general, it can be stated that HIBL is very promising for high-resolution resist based nanopatterning due to the low proximity effect and the lower ion dose required in comparison with EBL. Thus, broader use of HIBL in the coming years can be anticipated. More detailed discussions on this topic can be found in previous review articles.^{471,1107,1124}

2. Cryo-FIBID

In the past, it was found that electron or ion irradiation of Sn-based precursors condensed on a substrate at 120 K produced deposits with metallic behavior.¹¹²⁵ The topic was recently revived through the use of Ga-FIB to dissociate condensed $\text{W}(\text{CO})_6$ precursor at -100°C , giving rise to WC nanodeposits that exhibit metallic resistivity.¹⁰⁵¹ The particular advantage of the cryo method is the low ion dose required (600 times lower) compared to the standard Ga-FIBID process at room temperature. The new technique, coined cryo-FIBID, requires that the thickness of the condensed precursor layer is comparable to the ion penetration depth, so that the deposit does not lift off when heated back to room temperature.^{294,295} Cryo-FIBID has also been carried out using the precursor MeCpPtMe_3 ,²⁹⁶ although the resulting PtC deposits were less conductive than their WC counterparts.

So far, the best cryo-FIBID results have been achieved using the precursor $\text{Co}_2(\text{CO})_8$.²⁹⁷ In this case, the Ga ion dose required is only 15 $\mu\text{C}/\text{cm}^2$, and the obtained metallic resistivity is low (200 $\mu\Omega/\text{cm}$). Using cryo-FIBID it is thus possible to write electrical contacts with greater efficiency directly onto a micro-/nanostructure, as outlined in Figs. 27(a)–27(d), and applied to grow electrical contacts onto materials such as graphene, as shown in Fig. 27(e).

Conveniently, the need for a cryogenic module in order to decrease the substrate temperature for the cryo-FIBID experiment can be circumvented by means of a cheap, vibration-free and compact thermoelectric plate. Recent process optimization along these lines has enabled thermoelectric cooling to -60°C for cryo-FIBID with $\text{W}(\text{CO})_6$.¹¹²⁶ The resistivity of the deposits (620 $\mu\Omega/\text{cm}$) is even lower than that obtained for cryo-FIBID performed at -100°C . Cooling via the thermoelectric plate method thus provides a promising route for the widespread use of cryo-FIBID in any FIB chamber. Further cryo-FIBID studies will benefit from ongoing process optimization of the thermoelectric cooling method and from the use of alternative and novel precursors for metallic deposits in condensed form.^{1080,1127} Future application of cryo-FIBID to directly and efficiently create electrical contacts onto 2D²⁹⁷ and other ion-sensitive materials is to be expected.

3. FIB irradiation of metalorganic spin-coated layers

Although spin-coated layers such as resists are typically used as sacrificial material in a multi-step patterning process, it had also been proposed that metalorganic spin-coated layers could be directly patterned using ion irradiation, producing metallic structures based on Pd,^{514,1128} Ir,¹¹²⁹ and Au.¹¹³⁰ However, post-annealing treatments were required to achieve the desired metallic resistivity values, which limited interest in this approach. Similarly, electron irradiation of Pd-based spin-coated layers has been used to create nanopatterned structures, but there too, a post-annealing step is required to achieve metallic resistivity.¹¹³¹ However, recent process optimization using Pd based spin-coated layers in combination with either focused electron¹¹³²

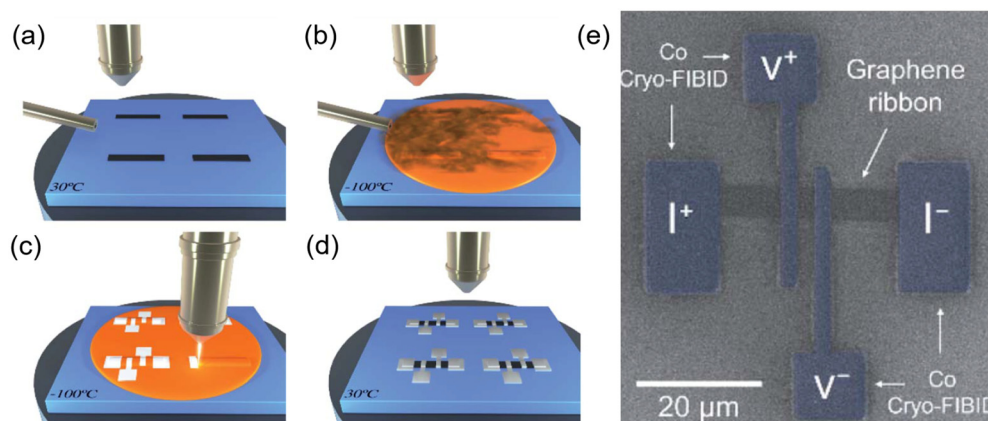


FIG. 27. Cryo-FIBID process used to grow electrical contacts on 2D materials (e.g., graphene):²⁹⁷ (a) Graphene ribbons on Si/SiO₂ substrate loaded into chamber. (b) Substrate is cooled and precursor from gas injection system (GIS) delivered giving rise to a condensed precursor layer. (c) FIB irradiation of the condensed layer using predefined pattern. (d) Substrate heated up to 30 °C thus sublimating the non-irradiated condensed layer. (e) False color SEM image of a graphene ribbon with four electrical contacts grown by cryo-FIBID using Co precursor. Reproduced with permission from Salvador-Porroche *et al.*, *Nanoscale Adv.* **3**, 5656–5662 (2021). Copyright 2021, Authors, licensed under a Creative Commons Attribution (CC BY) license.

or focused ion irradiation¹¹³³ has led to the development of an efficient process that allows the direct patterning of metallic micro- and nanostructures at the wafer level without the need for annealing. Due to the simplicity of the process, together with its high resolution and scalability, it is tempting to foresee its wide application for the patterning of electrical contacts or other conductive structures in nanoelectronics.

4. FIB induced mass transport

Many effects known from broad beam irradiation studies are also observed during FIB processing. This includes the observation of ion hammering and the viscous flow of nanostructures under FIB irradiation.^{360,1134} Spatially resolved ion beam mixing using FIB with subsequent rapid thermal annealing has been used to create nanometer sized Si crystals at freely defined positions for single-electron transistor applications.⁴¹⁰ Furthermore, localized ion beam induced growth of nanostructures is often observed and can result in epitaxial overgrowth⁴⁰⁶ or the selective growth of nanopillars.¹¹³⁴ An overview of these and other effects related to mass transport along surfaces or between layers, including variations such as atomic mixing, can be found in Ref. 471.

A further example of such irradiation effects is the formation of self-organized patterns (such as “ripples”) on the surface of semiconductors, depending on the angle of incidence and the energy of the ions. The broad beam process using noble gas ions is qualitatively understood with mass redistribution playing an important role.^{1135–1137} FIB induced self-organized pattern formation has mostly been studied for other (mostly metallic) ion species.^{1138–1145} Here, the underlying physics is more complex due to the unavoidable change in stoichiometry of the sample as a result of the high fluence of primary ions. The situation could be simplified by using inert gas ions from GFIS or PFIB. In combination with spatially resolved surface structuring, these mass transport effects have the potential for interesting applications, e.g., controlling the wettability of FIB milled trenches for

microfluidic applications. Here, understanding and control of the wavelength of the surface ripples will be key.^{1146,1147}

5. Hybrid fabrication and heterogeneous integration

Despite the flexibility of FIB based processing, the intrinsically low throughput of the serial technique means that large-scale patterning remains elusive. A way to overcome this limitation is to combine the FIB step with other lithography methods, using the so-called ‘mix and match’ approach. For example, in a method combining FIB patterning with nanoimprint lithography (NIL), HSQ resist was patterned by He-FIB and used as a nanoimprint template after development.¹¹¹¹ Remarkably, this allowed the transfer of lines with a half-pitch of 4 nm into the UV-curable nanoimprint resist. Similarly, miniaturized fluidic channels for single molecule analysis consisting of complex, multilevel, multiscale fluidic circuits have been patterned using FIB and photolithography to make a silicon master stamp followed by UV-NIL to fabricate a negative replica that was used as the working stamp to imprint the fluidic devices quickly and with high fidelity.¹¹⁴⁸ The fabrication of arrays of complex plasmonic nanostructures has been achieved by combining EBL and Au-FIB, demonstrating both planar and 3D nanostructures with high patterning accuracy over large areas.¹¹⁰² In further examples, optical lithography has been combined with FIB milling to allow the fabrication of nanochannels in Si-glass microfluidic chips for studies of ion and molecule transport^{1149,1150} and for the separation of nanoparticles by size exclusion.¹¹⁵¹ FIB techniques have also been combined with other processing methods, offering improved versatility in terms of the material to be processed and reducing total processing times. For example, FIB patterning of molds into Si was applied for subsequent hot-filament CVD growth of diamond nanocone arrays on a free-standing diamond membrane formed after silicon removal.¹¹⁵² Similarly, direct patterning of inorganic hardmask layers has been demonstrated using FIB milling and FIBID combined with reactive ion etching, comparing favorably with the much more complex and

material-limited resist based lithography techniques that are conventionally used in this area.¹¹⁵³ Instead of using the hard mask for later etching steps, it can also serve as a sacrificial layer that enables the use of high ion beam currents. For example, milling through a chromium oxide mask about 63 nm thick provided the same spatial resolution with 100-fold increase in ion beam current (2.6 nA instead of 26 pA). This approach has been used to fabricate Fresnel lenses from silicon dioxide, increasing throughput by a factor of 75 for functionally identical structures.⁵⁶⁷

A further promising application is the use of the FIB as a post-processing tool in CMOS chip manufacture. This extends the semiconductor industry's conventional use of the FIB as a debugging and inspection tool, to the top-down FIB based integration of novel semiconductor circuitry, such as nanometer-scale mechanical devices.^{1154,1155} In connection with new (nano)-electronic materials, such as single walled CNTs, it has also been shown that there is low, amendable damage to electrical circuits when a FIB patterning process is implemented after fabrication of the basic structure.^{1156,1157}

6. Outlook

In terms of applications, machining using FIBs goes far beyond a purely top-down approach, since it can also be used to trigger various chemical and physical processes, and combined with various other micro-/nanofabrication techniques. Device fabrication based on unconventional processing steps, as well as unique combinations or alternative advanced sequences, may further gain in importance since these methods can mitigate typical limitations in either FIB processing or the other fabrication techniques. It is hard to foresee in which way such unconventional processing techniques will develop, but they will certainly benefit from progress made in many other areas, e.g., new insights into ion induced chemical processes, novel precursor materials, as well as novel ion species and newly accessible energy ranges.

V. ROADMAP

Focused ion beams can be employed as an all-in-one tool for studying and modifying matter down to the nanoscale. A showcase example of these diverse applications is given by the He-FIB, which enables imaging of corona virus particles and the fabrication of Josephson tunnel junctions (and much in between), all using the same instrument.

Building on Secs. II–IV of this article describing the current state of the art of FIB instrumentation, theory, and applications, we can now ask where to go next with FIB technology. There are many avenues to pursue, so the answer to this question is multifaceted and complex. This section is devoted to trying to answer this question in some detail.

Figure 28 provides an overview of the fields that drive various FIB techniques and the resulting challenges and demands for the further development of instrumentation, hardware/software interfaces, accessories, process modeling, and fundamental understanding. As the figure outlines, all of these required developments are driven by established or newly emerging research fields that use the FIB as a tool for any or all of the following: localized and comprehensive analytics, material modification, and advanced device fabrication.

To account for the complexity of the situation without getting buried in detail, this section is organized around the important future developments that we have identified as necessary in order to meet the

needs of the drivers. Each driver requires different FIB techniques, brings its own challenges, and demands different actions to move forward. In the following, a short description of these drivers is given and an evaluation of the needs of the user community is then presented. The latter is based on the results of a survey completed by over 50 scientists from across all FIB research fields. The developments necessary for future progress are then derived and described in terms of their challenges and overall potential.

A. Drivers for future applications

The drivers for FIB technologies span a wide range of fields in science and technology. Currently, the semiconductor industry is the largest driver, demanding process automation, modeling, and the analysis of big data.

In general, the drivers can be separated into two interrelated areas. In the first, the FIB is mainly used for analytical purpose, providing access to a wealth of structural information and physical properties. In the second, the FIB enables rapid, flexible device prototyping and stimulates novel device development. In both of these areas, material modification using the FIB is key, be it to enable a particular form or analysis, or to fabricate novel structures and devices.

A leading example of FIB based analysis is in the life sciences, where it often involves large-volume 3D tomographic analysis of cells and tissues under cryogenic conditions (see Sec. IV C3). This drives the need for increased processing speed, advanced cryo capabilities and automation. One potential way to increase throughput is through cryo plasma-FIB milling, which enables faster removal of bulk material, as discussed in Sec. IV D. However, cryogenic processing is not straightforward and is further complicated by insufficient vacuum levels in the sample chamber of current dual beam instruments. A better vacuum, as has been demonstrated in the past, can reduce contamination, and in the case of cryo-FIB, can also lead to less water condensation on the target structure.

In both the life sciences and materials science, as well as in environmental research, FIB techniques such as cross sectioning, tomography, and sample fabrication (e.g., for TEM and APT studies, or *ex situ* characterization of physical properties) have become established standards that would now benefit from powerful process automation. Materials science in particular, can benefit from other ion types, such as ions from noble gas elements that (a) may not accumulate in the material, since they can either pass right through if the sample is thin enough or leave by out-diffusion, and (b) will not undergo chemical reactions. This also applies to radiation hardness testing of materials, where FIB is used as a means to locally mimic the effects of a plasma or solar radiation. Increasingly, cryo-FIB techniques are also being applied to beam-sensitive samples, such as novel battery materials that are relevant for energy-technology applications, which also calls for the cryo-related developments mentioned above. In the case of multi-modal imaging approaches and the identification of target areas for, e.g., TEM lamella preparation, workflows are currently hindered by the very different contrast mechanisms in, e.g., the fluorescence light microscope vs SEM. Addressing this will require improved correlative techniques as well as more advanced data management and analysis (cf. Sec. IIF).

In a broader context, FIB research into quantum and 2D materials, as well as other novel materials, can largely be subsumed under materials science, but should be recognized as drivers of “non-

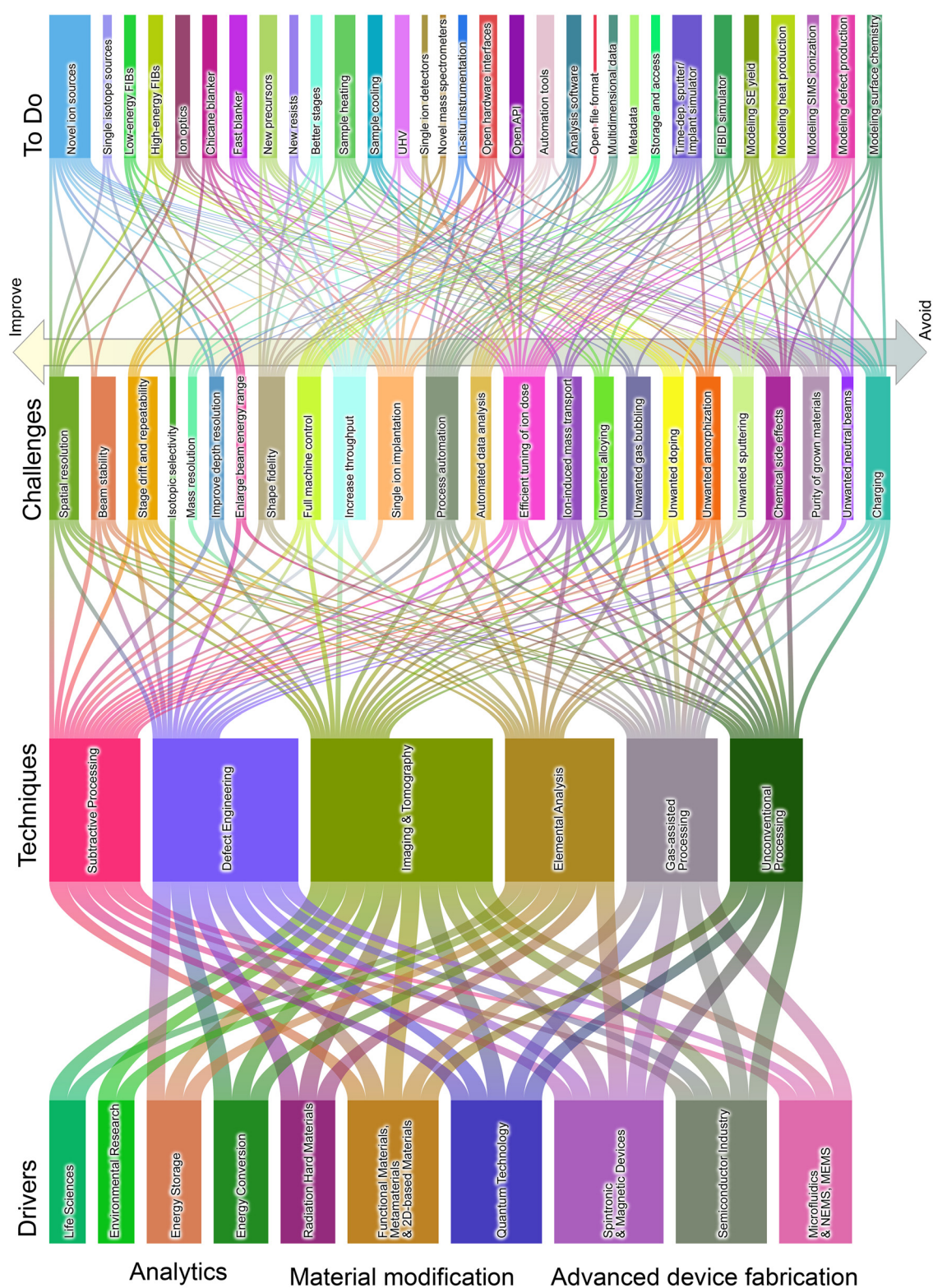


FIG. 28. Sankey diagram showing an overview of the interrelated drivers, techniques, challenges and to-do items for future development of FIB science and technology.

conventional” FIB techniques in their own right. For example, the growing interest in quantum materials is driving the need for rapid and high-precision patterning of large 3D structures from novel materials. This could be achieved by combining process modeling with an easy-to-use interface for 3D shape definition and calculation of the required beam and patterning parameters. In addition, chemical functionalities can be integrated into patterning through the development of novel ion sources; oxygen, in particular, is an important tuning parameter for the electronic structure of many quantum materials.

At a more fundamental level, the maskless approach of FIB processing renders it an ideal technique for the precise creation of defects at the exact location where they are needed. This is driven by applications in quantum technology, where defects in an otherwise perfect crystal can act as quantum emitters or, if they carry spin information, as qubits.

On the theoretical side, the creation of specific defects and especially single ion implantation (SII) require a deeper understanding of the mechanisms of defect formation and the evolution of the defect’s local environment, which in turn determines the physical properties. Such challenges can be addressed by modeling at different levels of sophistication, as described in Sec. III.

In terms of instrumentation, a key technological challenge for quantum applications is to be able to place individual ions in a (quasi-) deterministic manner and detect them with very high efficiency. Certain defect types may require the implantation of a specific ion for which stable ion sources still need to be developed. The implantation depth is controlled by the mass and kinetic energy of the incident ion, and in order to reduce dephasing, quantum emitters in photonic chip architectures are usually desired to have a depth of 10–100 nm beneath the surface. As such, the ion energy ranges of tens of keV that are currently available are mostly well suited. However, in the case of spatially resolved heavy element implantation, a higher primary beam energy would be required, which poses challenges for the power supplies and for electrical isolation in the ion optical column. If instead non-native atoms are to be implanted into 2D materials, e.g., for high resolution flexible area doping, lower energies are needed. In this context, low energy means less than 100 eV to ensure a significant number of replacement collisions in which the ion exchanges position with a target atom and the latter is removed from the 2D material.³⁷⁶ Methods to reduce the primary energy of the beam include using deceleration lenses in the final stage of the beam focusing or implementing reverse sample biasing (see also Sec. V C). For the detection of the transiting

primary ion itself or the resulting interaction with the substrate, various approaches are under development (see the Outlook of Sec. IV B).

Another strong driver for FIB technologies are micro-electro-mechanical systems (MEMS)/nano-electro-mechanical systems (NEMS) and microfluidic devices, which are ubiquitous in our modern world, be it in sensors, displays, printers, transducers or microphones. Such devices often combine various active and passive components consisting of electronic and moving parts, and therefore typically rely on a whole chain of thin film lithographic, deposition, and etching techniques. However, these fabrication techniques require that the devices are built up layer by layer. As such, true 3D geometries are difficult to achieve.

FIB processing by milling and deposition, on the other hand, enables the flexible fabrication of complex multi-level geometric structures and smooth 3D shapes, as it permits direct top-down and bottom-up structuring in the same instrument. What has limited the default implementation of FIB technologies in device manufacturing thus far is the slow processing speed of the inherently serial process. Here, the availability of a multibeam system comprising thousands of individually addressable focused beams would represent a major leap forward (see also Sec. II B).

B. The generic needs of the community

FIB systems have become a standard in scientific research and industrial processing, but the market is still heavily dominated by the Ga-FIB. While Ga represents a good compromise between spatial resolution and sputtering yield for many material systems, other applications require different beam parameters and/or suffer from Ga contamination. In our survey, the participants were therefore asked in which areas they conduct research and upon which material systems and/or devices they focus. The corresponding results and percentages, thus, reflect the perspectives of this particular group of users. Figure 29 shows that materials science, physics, and engineering are the most prominent research areas in the FIB community surveyed. Fields such as biology, medicine, and earth science may be underrepresented in these statistics, since fewer scientists from those areas participated in the survey. It would be interesting to repeat the survey with a larger number of participants from a wider range of fields in order to be more statistically representative. The material systems most commonly studied by the respondents are 2D materials and semiconductors, followed by various other systems including biological samples, ceramics,

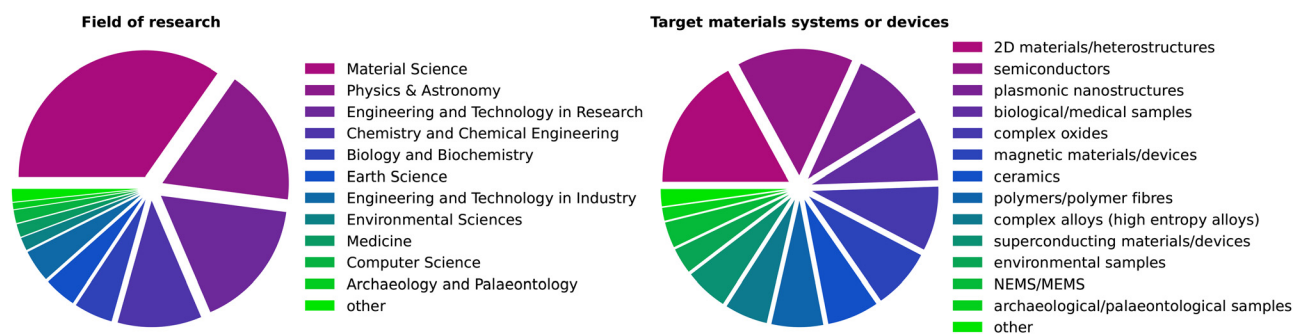


FIG. 29. Results of a survey of the fields of research (left) and the target materials systems or devices (right) being investigated or addressed.

polymers and environmental samples. This reflects current research trends, but also underscores the universality of the FIB as a tool for modifying matter at the nanoscale.

The typical problems encountered by the FIB users participating in our survey (see Fig. 30) show a clear trend, with surface damage, unwanted implantation/doping and contamination being the most dominant problems. This last point on contamination is mostly related to the typical vacuum operating range of around 10^{-6} – 10^{-7} mbar and the associated residual gases in the vacuum chamber. In this pressure range, a monolayer of molecules still forms on the sample roughly every second, and with a typical water content of around 60% in the residual gas, most surfaces will also be covered with a double layer of water molecules. The hydrocarbons from the residual gas are also decomposed by the ion beam and, depending on the ratio of the sputtering to deposition yield, can form a carbonaceous layer on the sample. Especially in the case of FIB processing with light ions, even a small amount of contamination can result in greatly reduced net sputtering rates or in the extreme, net contamination deposition. Generally, both water and hydrocarbons can add various, mostly unwanted, chemical reaction channels (cf. Sec. IV E 1). Here, a significant improvement of the vacuum level toward UHV would be an important development. Since some of the contamination is introduced with the sample itself, a dedicated load lock for plasma cleaning of samples prior to FIB processing would be an ideal complement. Other unwanted chemical side-effects may occur when the ion interacts inside the sample and can be mitigated using inert Ion species. However, unwanted physical side-effects such as interlayer mixing or

dealloying are difficult to avoid as this would require a significant reduction in primary ion energies; in the shorter term these effects can be better addressed through process modeling (cf. Secs. III A and III B).

Problems with long patterning times are either due to beam currents not being high enough or to stability issues of the system. PFIB columns address the first issue by providing higher beam currents, but at the expense of a larger probe size. One especially intriguing way of increasing processing speed and thereby reducing patterning times would be the implementation of a multibeam system. Thousands of parallel beams could in principle be available while maintaining small probe sizes (cf. Sec. II B). System stability is mainly limited by the use of mechanical stages, which suffer from drift over time and from inaccuracies in positioning. In principle, laser interferometric stages solve this problem. However, since most FIB systems use a tilted column, the sample must also be tilted in order to allow FIB processing under normal incidence. Thus, novel system designs with a vertical FIB column are of great value. Finally, instabilities in the beam current itself (more of a problem for some of the novel ion sources) can also limit patterning performance, so the development of novel and stable emitters is needed (cf. Sec. II A).

Interestingly, the survey answers to questions about desired spatial resolution, milling depths, deposition thicknesses, and beam energy ranges more or less correspond to what is currently technologically possible. This makes sense, in that users often aim for what is feasible with their instruments. Only a few users were interested in depositing

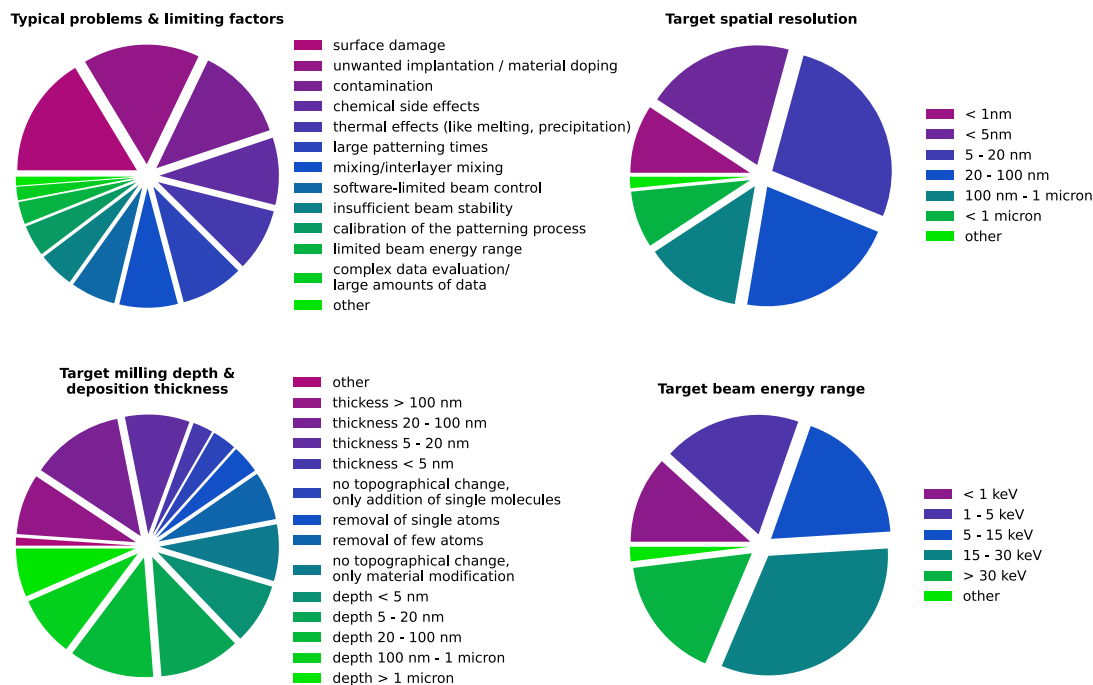


FIG. 30. Results of a survey of the typical problems and limiting factors in FIB processing (top left), the target spatial resolution (top right), the target milling depth and deposition thickness (bottom left), and the target ion beam energy range (bottom right).

single molecules or removing/implanting single atoms. Currently, such approaches are beyond the technological capabilities of commercial systems and would require substantial developments. However, with these developments the impact on areas such as quantum technology research would be substantial.

C. Important future developments

Having discussed the needs and wants of the FIB research community in terms of future applications and user experiences, we now turn to specific technologies that need to be developed. For this purpose, Fig. 28 has been divided into subplots based on different groups of to-do items. This allows us to trace the impact that certain developments will have on specific challenges, to explore the techniques that are then enabled, and ultimately link back to the drivers.

1. Ion beam generation and transport

Figure 31 displays the required developments (i.e., to-do items) and the associated challenges, techniques and drivers in the area of next-generation FIB sources and ion beam transport. Specific needs for novel ion sources, ion optics for both low and high energy beams, and fast beam control/blanking have been identified.

a. Novel source types including single isotope sources. A key area requiring development concerns novel ion sources, in particular, high brightness and light element sources, which are crucial for high-resolution imaging and nanostructuring applications. Here, further development of GFIS based sources to operate with gases other than He, Ne, and N₂ is desired. The first steps in this direction have already been taken by various researchers testing GFIS based emission of H,¹¹⁵⁸ Xe,⁶⁰ and Kr.⁶¹ However, the heavier noble gas elements have

numerous isotopes, sometimes in nearly equal proportions; if the ion optical column is not well shielded from magnetic fields, this can affect spatial resolution, since magnetic fields present in the column can split the ion beam according to the mass of the isotopes. Solutions to this problem, other than shielding, include the addition of a Wien filter. Again, these developments will require new ion optical columns that can reduce or correct the aberration induced by the Wien filter. Alternatively, isotopically pure source gases can be used, similar to the already successful implementation of isotopically pure source materials by manufacturers of Ga-LMIS and AuGeSi-LMAIS. Further LMAIS single isotope sources might also be of interest in the context of SII,^{194,293,829,835} local isotope enrichment⁷⁶⁶ and for a wide range of other applications, e.g., SIMS, also in combination with reactive gas flooding.^{123,125,992,1001} The GaBiLi-source,^{46,1159} for example, provides heavy Bi atoms and small Bi clusters for milling, and a reactive Li ion beam for analysis by SIMS. With the LMAIS, switching between ion beams for these different species is fast (only takes a few seconds) and can be automated. For the future, new alloy concepts that include heavier reactive species are desired; to date only Au-Si, Au-Ge, Au-Ge-Si, and Ga-Bi-Li LMAIS are commercially available.

Ion sources for heavy ions are desirable for applications such as SIMS in order to increase sputter yields and also for device debugging in the semiconductor industry, where the shorter penetration depth of the heavier ion is an advantage. Furthermore, ions from group 1 and group 17 in the periodic table are of interest due to their chemical reactivity, which enhances sputter yields. While successful attempts have been made in the past with group 1 elements,^{29,31,46,1160} efforts for the development of sources based on group 17 elements such as fluorine or iodine are more limited. A possible approach for reactive ion sources using group 17 elements could be the ILIS if implementing the appropriate salts.⁵⁴ Such ILIS prototypes have been developed, but

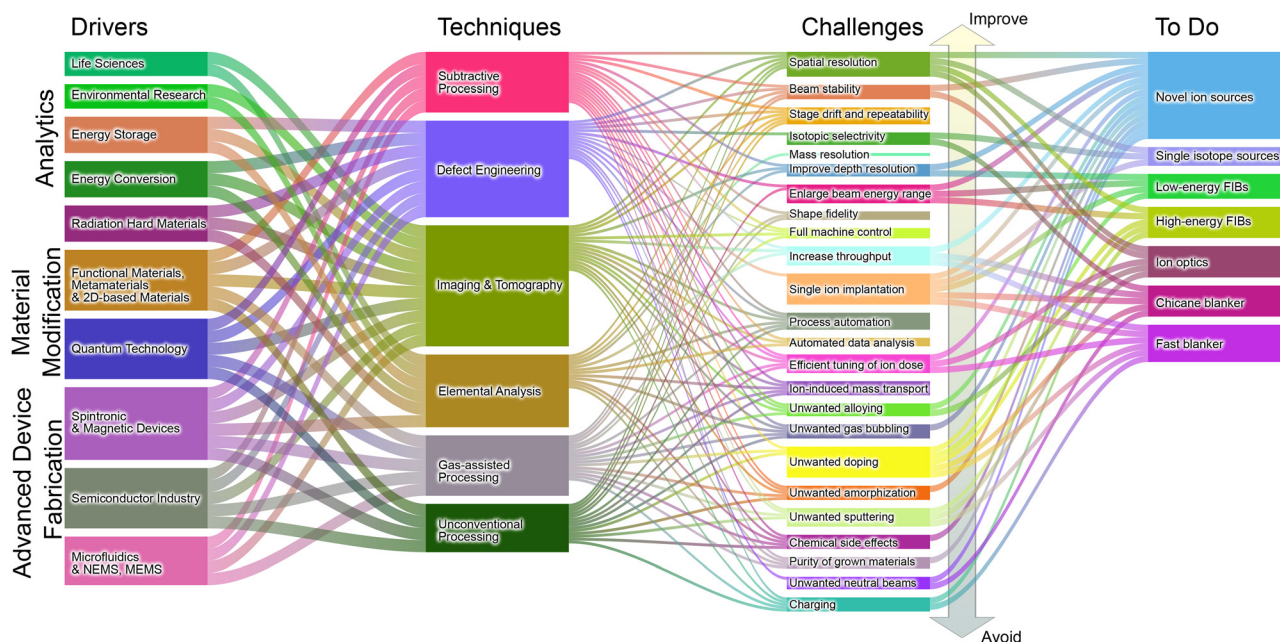


FIG. 31. Developments that are required with respect to the generation and transport of focused ion beams.

they did not yet enter the market. One of the intrinsic issues when operating an ILIS is the need to switch the polarity of the extracted ion beam in order to keep the source composition in balance. While this is usually considered a nuisance it can become an advantage, since by using clever chemistry, ion beams suitable for performing SIMS with both polarities could be extracted. However, collateral damage by the reactive ions to the ion optical elements needs to be considered and avoided as much as possible. We note that the first commercial FIB systems combining reactive ion species with SIMS are currently entering the market.^{977,1161}

With regard to source brightness, the recently developed MOTIS/LoTIS have been shown to achieve high brightness for a number of ion types.^{29,31,1160} Furthermore, fundamentally new concepts based on, e.g., insulator-on-conductor ion sources¹¹⁶² and other ideas that currently have a very low technology readiness level (TRL) may also have potential for application as high brightness sources, since they partially exploit similar concepts to the GFIS technology.

An aspect that is becoming more and more important for users is instrument uptime, which can be increased by minimizing source maintenance needs. For example, LMIS and LMAIS lifetimes could be increased by increasing the reservoir size of the liquid metal/alloy. The lifetimes of the apertures could be increased by adjusting the material and thickness of the aperture strip, as well as the number of apertures incorporated.

b. Ion optics for low and high energy ion beams. Access to low energy ion beams (<1 keV) is becoming increasingly important for a range of FIB applications in which the penetration depth and ion straggle in the material must be minimized. However, the transport of a low-energy beam without sacrificing on resolution is a challenge. This is mainly due to chromatic aberrations (see also Sec. II B). To address this issue, specialized beam optics such as energy filters and chromatic aberration correctors are needed, which are, in general, complicated to produce, align, and tune,¹¹⁶³ thus posing extra challenges for system automation. Low-energy beams also pose challenges in terms of the stability and precision of the associated electronics.

Key areas where low-energy high-resolution FIBs will bring benefit are in FIB based debugging and mask repair in the semiconductor industry,⁵⁰⁷ where minimized ion penetration and thus minimal damage to underlying layers and structures is required. If these low-energy FIBs employ heavy ions, the beam-sample interactions can be confined even more closely to the surface. A further application that requires ultra-low energy beams (down to a few tens of eV) is the controlled and spatially resolved doping of 2D materials or the ion-beam induced modification of surfaces (see Sec. III A). For the latter, the energy spread of the ions emitted from the source must be exceptionally small to begin with, smaller even than the LoTIS and GFIS can deliver. Indeed, in order to reduce chromatic aberrations for all of the above, it makes sense to start with new types of ion sources that have a reduced energy spread and then employ subsequent beam elements for chromatic aberration correction accordingly.

For other applications, higher energy beams are required, such as for the deep placement of (heavy) atoms for quantum technology applications, and for applications requiring even higher spatial resolution. Since the ion collision cascade expands the FIB beam in the material and most damage occurs at the end (cf. again Sec. III A), it may be useful to use high energies for devices based on (potentially suspended)

thin films. The target defects are then generated close to the surface with high spatial precision, but low yield, and the beam damage occurs spatially separated from this in the substrate (or not at all in the case of suspended membranes). While the technological challenges for increasing the beam energy are moderate, the increased size of the required electrical isolation and power supplies will actually limit the achievable resolution, since they may act as additional noise sources.

c. Fast beam control and blanking. Standard beam blankers installed on FIB instruments typically achieve blanking times (known as rise times) on the order of 20–100 ns. However, for a number of applications, rise times below 10 ns are desirable and have been demonstrated in isolated cases.¹²⁰ Applications requiring faster blanking times include resist patterning, high-end FIBID and SII, in all cases to avoid unintentional exposure.

Another need for specialized beam blanking arises from the fact that ion beam sources typically also create a certain fraction of neutrals in the beam, which is generally unwanted. These neutrals result from charge exchange collisions between energetic ions and neutral gas atoms. The latter are generally present due to insufficient vacuum at the source and in the ion optical column, and in the case of the GFIS, as a direct result of the gas based source technology. In the case of the GFIS, 0.29% He⁰ have been reported under conditions used for medium resolution imaging and for TOF-SIMS (6.3×10^{-6} mbar He pressure, 20 μ m aperture).¹¹⁶⁴ However, these numbers depend heavily on the gas pressure and aperture used. Since neutral species are not influenced by electric or magnetic fields and thus cannot be deflected away using ion optical elements or the standard electrostatic blankers, they can result in broad background exposure, especially in the case of long FIB processing times. This is detrimental in applications concerned with few or single-ion impacts at specific target locations, which often also involve high-dose irradiation (and hence even more unwanted neutrals reaching the sample) to make fiducial markers for subsequent correlative analysis. In order to eliminate the neutral beam, a physical obstacle can be introduced into the central portion of the column, which only the ions can bypass by means of a series of electrostatic deflectors. This is known as a chicane blanker. Similar solutions, such as a curved column, could also be explored but come with their own challenges.

2. Accessories and complementary instrumentation

In Fig. 32, the required developments related to new accessories (add-ons) for FIBs and complementary instrumentation are outlined. These to-do items can be broken down into categories focused on ion induced chemistry, adaptable sample environments, and analytical add-ons.

a. Ion induced chemistry. Applications using FIBID, gas-assisted etching, and resist-based lithography could all be greatly enhanced by the development of more advanced modeling tools, new gaseous precursors and new resists.

In the case of FIBID, new precursor chemistries that enable higher deposit purity (e.g., for plasmonic applications) and superior resistivity (e.g., for circuit edit applications) are needed. The prerequisite for the development of these precursors is a deeper understanding of the ion-molecule reactions that occur during the deposition process.⁴³¹ For high-purity metallic deposits, the precursor molecule will ideally be fully decomposed so that only the desired metal component

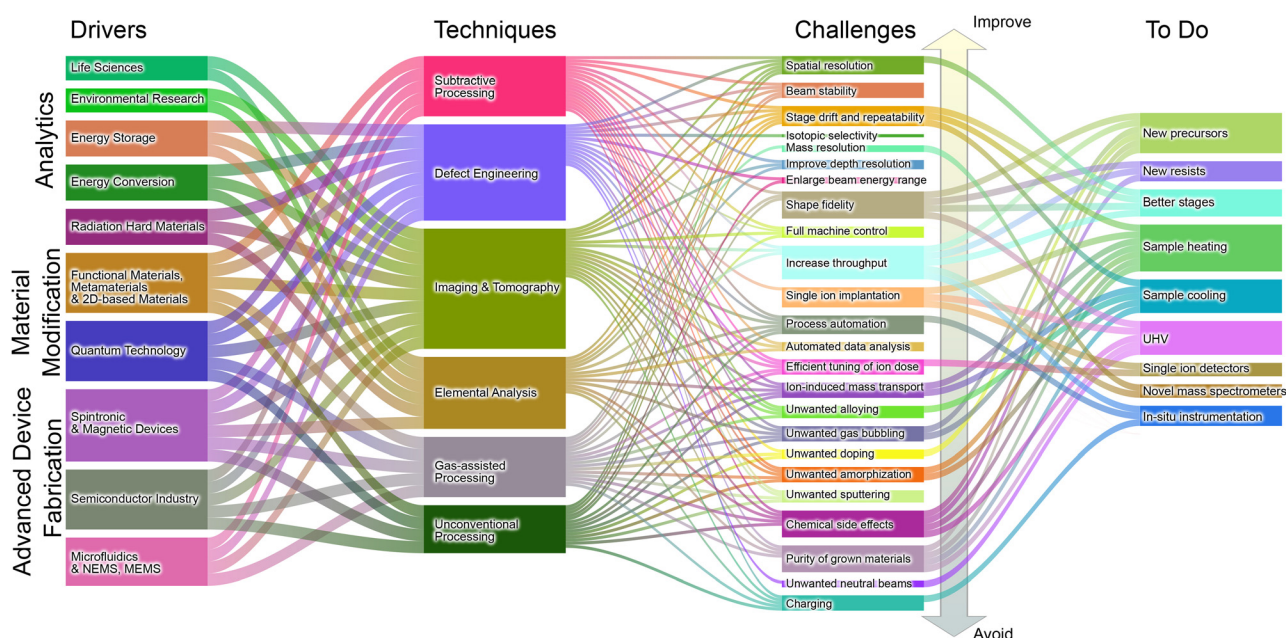


FIG. 32. Developments that are required with respect to accessories and complementary instrumentation.

is deposited and the ligand fragments are pumped away. Similarly, in the case of gas-assisted etching, the precursor molecules should undergo ligand loss to generate empty coordination sites that bind with the substrate to enhance the rate of removal of substrate atoms without forming a residue of nonvolatile reaction products. Once the fundamentals of beam induced chemical reactions are better understood (including aspects such as the effect of beam induced heating), researchers will be better poised to develop simulation tools to enable the prediction and design of more complex deposition and etching scenarios. The development of these simulation tools will also rely on input from systematic parameter optimization, where automated beam control and data analysis will play a key role.

A common but generally undesired effect from ion induced chemistry is the deposition of a thin carbon film over ion-irradiated regions, arising due to residual hydrocarbon species present on the sample surface and in the microscope chamber.⁸⁹⁰ This effect can be minimized by, e.g., plasma cleaning samples before loading into the microscope or working in UHV. Other unwanted beam induced effects that can occur during intentional deposition processes include implantation of the ions themselves into the deposit during the growth process, or concurrent milling of the deposit. To minimize these effects, it is desirable to increase the deposition rate, so that a lower total ion dose is needed for the process. Recently, it has been shown that cryogenic cooling of the substrate is one way to increase deposition rates, which could be investigated further.¹⁰⁵¹ However, it is worth mentioning at least one application where a balance between deposition and concurrent milling by the primary ion beam is actually desired, namely, the formation of high-aspect-ratio hollow nanopillar deposits with unique electronic properties using He-FIBID.¹⁰³⁸ Overall, fine control over the rate of deposition should prove highly beneficial for future applications.

For resist based lithography, further development of resists specifically for lithography using ion beams is needed. It has already been shown that resists tend to be much more sensitive to ions compared to electrons, but so far most resist development has still been for EBL. In order to optimize resists for lithography using different ion species, a deeper understanding of the ion-resist interaction process is required.¹¹²⁰ Consequently, the full potential of FIB resist lithography has yet to be realized, even though HIBL has already demonstrated high resolution patterning with dramatically reduced proximity effects compared with EBL.^{1108,1109}

b. Adapted sample environments. In electron microscopy, sample environments are now routinely adapted through the use of liquid cells that separate, e.g., chemical reagents, from the surrounding vacuum in the chamber. This principle does not readily translate to the FIB, since the typically used Ga ions can easily damage the cell membrane by sputtering and also suffer from limited penetration depths. Accordingly, only He-FIB has been used in a prototype liquid cell,¹¹⁶⁵ because He ions combine negligible sputtering of the cell membrane with a sufficient penetration depth to reach the reaction volume. In contrast, heating of samples during FIB processing has been used for decades, including more recently for the reduction of ion beam induced amorphization and blistering during He- and Ne-FIB processing. Quasi-local heating with an infrared laser¹¹⁶⁶ and homogeneous heating using a heating stage³⁶⁰ have both been tested. However, the latter approach, in particular, suffers from sample drift induced by the high thermal mass of the sample heater. Low drift implementations are thus needed that also have lower power requirements in order to protect the rest of the system (e.g., the piezo motors in the stage and ion optical components in the pole piece) from overheating and to allow faster heating ramps.¹¹⁶⁷ FIBID-related applications could also

benefit from new simple heating stages for exploring *in situ* annealing of the deposited structures.

Sample cooling can also be of interest to reduce beam damage, de-alloying, and surface microstructure formation, and for applications using FIBID or HIBL. For example, often temperatures slightly below 0 °C are sufficient to prevent the formation of eutectic phases or phase separation. In the same manner, molecular desorption rates may be tuned by either decreasing (or increasing) the temperature by only a few tens of Kelvin. Hence, for many experiments with a relatively small temperature range to bridge, Peltier based solutions become possible.¹¹²⁶

A pressing requirement for substantial cooling comes from biological applications, where cryo-stages and cryo-transfer units have been developed to enable complete cryo workflows (see Sec. IV C for examples). However, due to the stage cooling mechanism and shielding, these systems can suffer from reduced imaging performance as well as limitations regarding sample geometry and compatibility with certain accessories in the FIB chamber. Further development of specialized FIB instruments dedicated to the handling of biological and other sensitive samples is expected. These systems require special load locks for vacuum and/or cryo transfer, and better vacuum conditions, in general, will result in less condensation on the specimens.

By adding fluorescence light microscopy to a FIB-SEM system, a dedicated and time-efficient platform for the correlative preparation of TEM lamellae for biology, medicine, and related fields becomes possible, since well-established fluorescence light microscopy protocols can be used to locate regions of interest for FIB milling *in situ*. Indeed, a system including fluorescence light microscopy has recently been demonstrated in a cryo triple-beam configuration.²⁰⁶ Other implementations for fluorescent light detection either employ an ellipsoidal mirror in conventional Ga and plasma FIB-SEM systems for fluorescence spectroscopy of the atoms sputtered by the ion beam,⁸³⁴ or a full confocal microscope setup integrated into the vacuum chamber of a 100 kV LMAIS-FIB with a high-precision stage to move between working positions.⁸³⁵

The latter system demonstrated the *in situ* detection of fluorescent light from arrays of optically active defects created by the FIB, thus representing an important step toward realizing the deterministic creation of quantum emitters (cf. Outlook of Sec. IV B). For this application, it is of utmost importance to reduce all kinds of residual contamination (fluorescent and chemically active), both on the sample and in the vacuum chamber using UHV chambers and ideally inert sample transfer. Dedicated UHV FIB systems for quantum technology do not yet belong to the commercial standard, but have recently been demonstrated.^{91,814}

c. Novel detectors and analytical add-ons. Quantum technology and therewith the deterministic implantation of ions is one of the main drivers in science and technology today, and correspondingly, there is an urgent need for single ion detection. As outlined in the Outlook of Sec. IV B, specialized detectors^{193,194} or specific types of ion sources like a Paul trap,^{827,828} can either count the ions that impinge on the sample or measure the ion impact event. The latter can be realized using an on-chip IBIC measurement,^{199,829} which can be implemented for many CMOS compatible quantum devices that are not affected if there is an amplifier nearby. For all other quantum devices, truly agnostic SII detectors that detect, e.g., SEs will constitute an

especially important development. In addition to very high detection efficiencies approaching 100%,^{777,833} appropriate theory support is needed to calculate the expected SE yields. If such detectors are combined with a complementary system that detects device function, e.g., fluorescence⁸³⁵ associated with single-photon emission from the defects generated, truly deterministic implantation can be realized for an industry-ready commercial platform.

Several types of detection systems exist, which can often be combined to enable the acquisition of unique multimodal data sets. The general trend is toward better spatial resolution in 3D, higher sensitivity (and hence lower detection limits), and better quantification of results. A promising approach in this area is FIB-SIMS. Resolution and sensitivity are directly linked, which must be considered in efforts to improve performance. For example, in the case of GFIS and LoTIS based SIMS, simultaneous improvement in both resolution and sensitivity is not possible, since further reducing the sampling volume renders the number of target species in that volume statistically insignificant. Another analytical approach that is complementary to SIMS uses fluorescence spectroscopy to detect and identify atoms sputtered by the FIB and has been shown to be feasible for certain applications.⁸³⁴

Various other “classic” ion beam analytical methods have been implemented in FIB instruments or are interesting candidates for future implementations. These include IL¹⁷⁵ and RBS,¹⁴³ which have been explored by a number of researchers but are not available as commercial solutions. In IL, the luminescence of excited target atoms, defects or recoiled ions is detected using parabolic mirrors and a combination of gratings and lenses to obtain spectral information that allows identification of the emission source. IL was recently demonstrated for semiconductor heterostructures and Li-ion batteries⁸³⁴ and also has potential for various insulating samples in fields such as geology, mineralogy, biology, and medicine, including nanoparticle research such as nanotoxicology. In contrast, RBS measures the energy and scattering angle of backscattered ions and allows standard-free and depth-resolved elemental analysis. However, RBS in FIB systems has largely been abandoned due to fundamental problems related to the high number of multiple scattering events. Implementing a low energy ion scattering (LEIS) configuration may be a good alternative, as it only measures the energy of the charged particles that are primarily scattered at or close to the sample surface. Nevertheless, due to the low yield of charged scattered particles, high fluences are required to obtain good statistics.

There has been some debate around the potential to use X-rays as a signal for ion induced processes. Some reports indicate that this could indeed be possible.¹⁶⁹ However, in that particular case, the observed X-rays were not generated by the interaction of the primary ion beam with the sample. Instead, non-conductive samples became positively charged under ion impact and backscattered ions caused the emission of SEs from components in the chamber. These SEs were then accelerated toward the positively charged sample surface and generated X-ray peaks due to random interactions with the chamber components. It turns out that because the inelastic interactions that are responsible for X-ray generation depend on velocity and not on energy, typical primary beam energies in FIB systems are not sufficient for X-ray production. As stated by Joy *et al.*,¹¹⁶⁸ a beam energy of about 53 MeV for He ions, for example, would be required in order to excite the same X-ray lines as can be obtained using a 6 keV electron beam. Such high beam

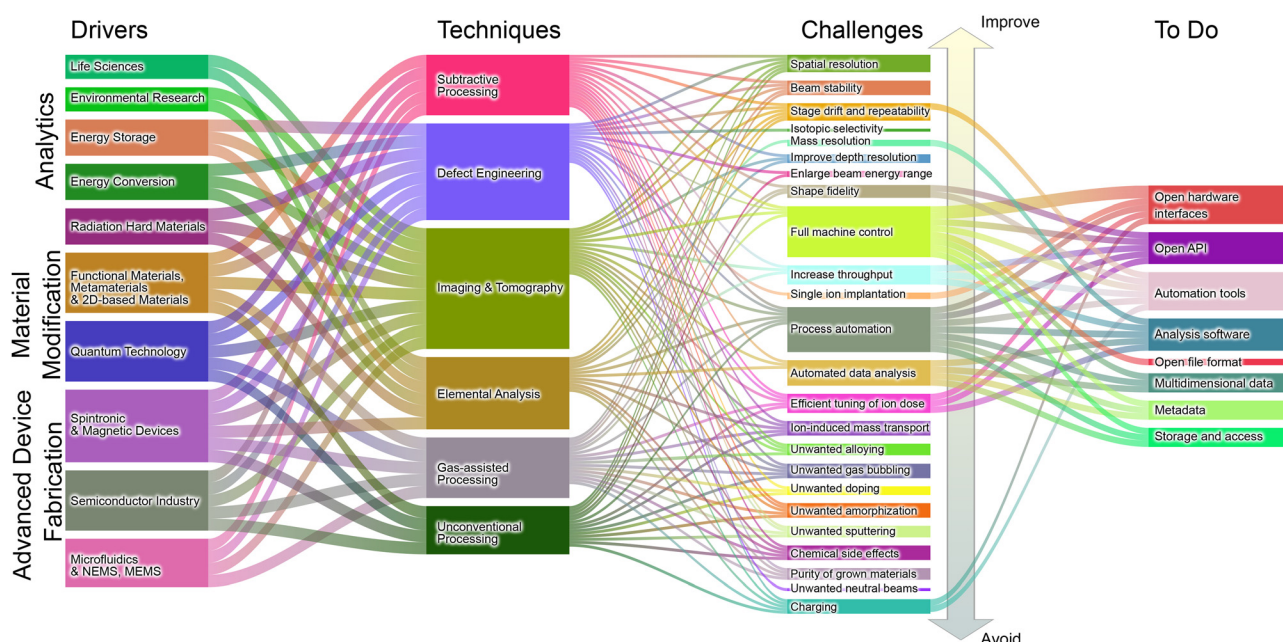


FIG. 33. Developments that are required with respect to automation and data management.

energies are out of reach for conventional FIB systems and thus the high-energy PIXE technique remains elusive to FIB.

In contrast to X-ray spectroscopy, the generation and detection of Auger electrons is indeed possible using the beam energies of current FIB systems. Auger electrons are generated due to the creation of core-holes in both the projectile ions and the substrate atoms, thus Auger signals emanate from both colliding particles, resulting in spectra with rich features that can be quantitatively evaluated.¹⁶⁶

Another analytical approach is based on *in situ* monitoring of ion beam-induced property changes, such as *in situ* electrical measurements to test nanostructured samples or to probe, e.g., magnetization via the measurement of a Hall voltage. Various other analytical techniques can also be incorporated into the FIB chamber, including AFM for *in situ* surface characterization or a nanoindenter for *in situ* mechanical testing. The benefit of performing these techniques *in situ* is not only control of the sample environment (e.g., no exposure to air), but also the possibility of real-time imaging (usually with the SEM) enabling immediate quality control of the measurement. For improved implementation of all these approaches, open hard- and software APIs are required.

Further developments that will facilitate work with analytical add-ons are related to the continued integration of sample cleaning options based on UV light or remote plasma, as well as the combination of FIB processing with direct laser milling.¹⁶⁹ Systems integrating femtosecond laser milling with FIB have recently become available and the associated workflows and possibilities are currently being explored.

3. Automation and data management

Figure 33 shows the technological developments required in terms of automation and data management. Both of these areas are

rapidly growing in importance due to the sheer volume of samples and data. There is enormous potential to facilitate the daily work of technologists and researchers using recent developments in ML and artificial intelligence. However, the black-box nature of such routines requires awareness and regular cross-checking of both the training data used and the results obtained. In the following, open interfaces for hardware and software, and then needs related to data access and management, are discussed in more detail.

a. Open hardware and software interfaces. In recent decades, science has increasingly moved toward openness and transparency. Today, this is generally a prerequisite for receiving taxpayer money for scientific research.¹⁷⁰ The idea of sharing not only raw data (as discussed in Subsection VC3b) but also task-specific developments in terms of machine operation and data analysis holds great potential. Instead of developing isolated solutions specific to particular research groups, which often become obsolete or are lost when the original developers move on, a larger community can maintain and further develop methods much more efficiently. In addition, the overall speed of development is significantly increased, as existing data and tools can be reviewed much more easily before “reinventing the wheel.”

Open access to nonproprietary development is already standard in many scientific fields, but the more technologically sophisticated and commercially valuable scientific work becomes, the more conflicts over intellectual property, licenses and potential patents can arise. Nevertheless, the major FIB instrument manufacturers have recognized the benefits of opening up their machines for modification by the users. Everyday users seek the ultimate in machine performance. Therefore, a key resource for driving future developments comes from enabling those users to make the most efficient use of instrument time and gain the most insight from the data generated. The larger the

active user community is, the more far-reaching the benefit, which in turn translates to increased sales for the manufacturers even if new technological developments do not have immediate impact on a patent or a price tag. What users need are open hardware interfaces to give them real-time access to detector signals and to enable them to add their own custom components. Even more important is a full-fledged API that allows control of all system parameters accessible in the GUI and ideally all others in developer mode. In this respect, software development on the manufacturer side must be a compromise between the needs of those who require easy-to-use software yet with more limited access, and those who require full access.

For process automation, certain tools are already included in some FIB systems. For example, routines that have already been executed can be stored and modified in a quasi-graphical approach, removing the hurdle of programming in a text-based interface. A key area needing a high degree of automation is in the efficient acquisition of tomographic data sets with imaging based on SE detection, as well as EDS and EBSD. FIB based tomography is particularly used in the life sciences and in environmental research, and increasingly in all branches of materials science, including for battery research and energy conversion. In all cases, high precision and repeatability are required.

In the future, automation tools should be complemented by ML algorithms that help with experimental control and real-time data analysis. Integration of an active learning user interface (human in the loop)¹¹⁷¹ for *in operando* improvement of the supporting ML algorithms would then allow easy continuous training and thus automatic control and/or optimization of processing based on predefined quality parameters or figures of merit. Such feedback control would also be beneficial for advanced patterning tasks, where modeling-informed pattern generation and optimization could be coupled with FIB-processing in a fully automatic way.

Open software interfaces are also required for data analysis, which varies greatly depending on the application typically requiring a tailored approach. ML-based solutions are similarly very diverse, with application to problems including simple data noise reduction, the broad field of feature recognition and data reduction, super-resolution imaging, and full quantification. The integration or linking of different methods and data streams thus needs open interfaces. Therefore, in place of closed all-in-one tools for data analysis and automation, manufacturers should instead provide the infrastructures needed for users to efficiently develop solutions for the broader community.

b. Data access and management. Likewise, all data generated should be open and ideally FAIR. The acronym FAIR, coined by Wilkinson *et al.*¹¹⁷² in 2016, stands for Findable, Accessible, Interoperable, and Reusable. The FAIR principles have led to global initiatives in all scientific disciplines.^{1173–1176} In particular, research areas that generate large amounts of data, such as the life sciences and materials science, are driving the development of data harmonization including a comprehensive set of metadata, ontologies, and workflows that are accessible along with the data.⁹⁶⁹ For experimentally derived data, this includes sample tagging, e.g., with QR codes linked to each resulting dataset, along with the corresponding experimental workflows documented in electronic lab journals. One of the biggest challenges with open and FAIR data are the diversity of data sets from different sources. The Novel Materials Discovery (NOMAD)

laboratory^{1177,1178} can serve as a blueprint here, as it maintains a collection of data repositories (called oases) for data from virtually all codes in computational materials science.

Several factors are crucial for the successful implementation of such data repositories:

- The data must be easy to upload and integrate into the repository.
- The data must be presented in a common format that allows easy access and efficient searching.
- Ideally, additional toolkits such as metadata crawlers or ML tools for data analysis or prediction should be offered.
- There should be an embargo period option to restrict access to collaborators and/or industry partners before publication or patent.
- The data sets themselves, as well as scripts used for data processing and analysis, should be given DOIs to make them citable.

With respect to data from (FIB-related) microscopy and spectroscopy, proprietary file formats are used by various manufacturers. While this can be useful when providing additional functionality or easier access to certain data analysis routines, it poses a major problem if there is no way to access the raw unprocessed data to analyze it and make it available. In order to integrate such data into a FAIR infrastructure, a standardized and multidimensional open file format specifically for correlative image and spectroscopy data are highly desirable. HDF5 would be such an option (where HDF stands for hierarchical data format), allowing raw data to be stored in image or spectroscopy file formats (such as tiff or binary files) together with metadata and all other types of information in a structured way.¹¹⁷⁹ However, this would require that instrument manufacturers agree on such a uniform standard, possibly following the lead of the neutron, x-ray and muon research communities where the HDF5-based data format neXus is used.¹¹⁸⁰

In general, open and FAIR data enables platform- and tool-independent analysis as well as correlation with data from other sources. Furthermore, image processing tools originally developed for other purpose (which may be based on completely different data generation platforms) can be applied and further developed for FIB-specific analysis. A prime example is the open source and platform-independent image processing software ImageJ and its derivatives,^{239,1181,1182} which has already been extended through a variety of community-developed plug-ins as well as further adapted and integrated into other programs via its API. Here again, ML can be leveraged as an extremely powerful tool for automated data evaluation, especially for large and complex data sets from correlative measurements or tomography. For example, external programming libraries for ML-based image classification and analysis such as scikit-learn,¹¹⁸³ Tensorflow,¹¹⁸⁴ PyTorch,¹¹⁸⁵ dragonfly,¹¹⁸⁶ and Ilastik⁹⁵⁷ are already available, actively developed and are becoming easier to use.

Ultimately, protocols will need to be established for the pre-processing of raw data into data packages for storage while others parts are to be discarded. In this regard, bottom-up harmonization of FAIR data including hierarchical and structured data repositories with full metadata can help with data management and sorting in order to provide “useful” raw data for further processing both internally and externally, while avoiding or at least minimizing the typical data graveyards.

4. Process modeling

The required developments concerning process modeling are outlined in Fig. 34. Process modeling plays a key role in technology development for integrated circuits (ICs)^{1187,1188} and together with other types of simulation, is considered one of the drivers of Moore's law. It is also heavily used beyond traditional device scaling ("Beyond CMOS" and "More than Moore"¹¹⁸⁹). However, in spite of FIB processes sharing much physics and chemistry with IC processes, process modeling is much less used in the FIB community. This is due to the hefty price tag for commercial simulation tools from the IC business and the much smaller industry behind FIB activities. Nevertheless, the success of process modeling for IC development demonstrates its potential for the further development of related FIB processes.

In the following, process modeling needs for FIB applications are discussed in terms of new BCA simulation tools, continuum modeling and approaches for modeling FIBID, as well as further areas where more advanced computational techniques are critical for progress.

a. BCA simulations beyond TRIM. The simulation program most widely used by the FIB community is TRIM,²⁵³ which allows the calculation of range and damage statistics, in addition to rough estimates of sputtering yields in 1D amorphous targets (see Sec. III A 2). Its success is based on the fact that it provides reasonable estimates for any ion-target combination, and the fact that it incorporates a convenient user interface. Other BCA simulation tools that in addition allow consideration of dynamic changes to the target and other target types (2D and 3D, and crystalline targets) would certainly benefit from more intuitive user interfaces. Work toward this goal has recently been reported for one particular code¹¹⁹⁰ and could probably be translated to other codes

with moderate effort. This would promote adoption of a wider range of BCA simulation codes by the community.

A further desired feature is the implementation of SE generation in BCA codes. This will allow more accurate interpretation of FIB images, will foster model development for precursor dissociation in FIBID processes, and may also aid SII when detecting single ion impacts using an SE detector.

b. Continuum modeling and approaches toward FIBID modeling. Continuum simulations of surface evolution are hampered by the need for various input parameters and the fact that they neglect processes taking place beneath the surface (see Secs. III D 1 and III D 3). However, these simulations are still useful for qualitative investigations. Wider applicability is expected from coupling BCA with continuum simulations, which could enable modeling of surface diffusion and reactions, heat transport and/or relaxation of density changes. This could pave the way for accurate and general process simulation of FIB processes, although taking into account physical and chemical parameters will still be an issue. Here, input from more fundamental methods, or strategies for experimental parameter determination, will be required. The former is already standard in IC process simulation, while an example of the latter in the context of FEBID can be found in the use of tilted nanowires on a pedestal for model calibration.⁴⁶⁹

As an alternative to physico-chemical modeling, ML has opened up new possibilities in many fields of research. In FIB technology, an example of this is a neural network that has been trained on a set of binary design patterns, the corresponding SEM images, and the ion beam parameters used in their production.¹¹⁹¹ The trained network was then able to accurately predict the patterning outcome for previously unseen designs in nearly real time. Such a neural network

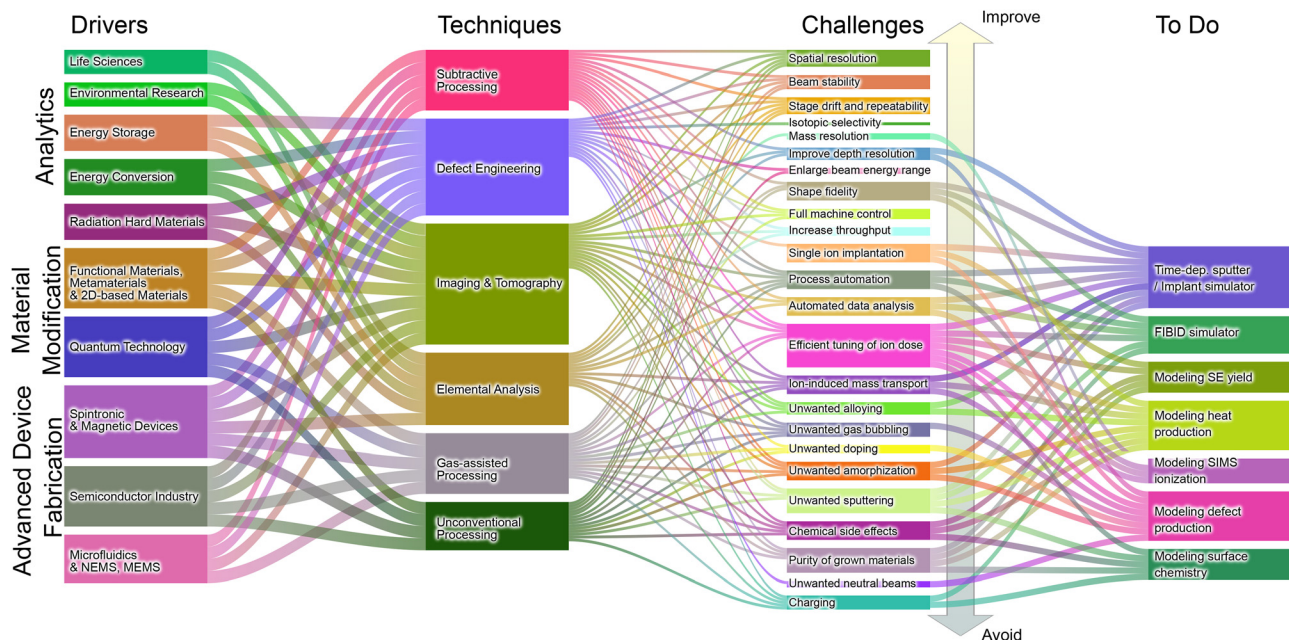


FIG. 34. Developments that are required with respect to process modeling.

approach holds promise for process control and, in a more general setup, for reducing the number of trial and error iterations in the fabrication of nanostructures.

c. Addressing the gap in understanding. The most important fundamental issues which cannot be addressed at the BCA or continuum level are:

- Accurate description of the energy loss of energetic ions and recoils accounting for their charge state and the electronic structure of the target.
- Production of defects in the irradiated material accounting for electronic excitations.
- Evolution of the atomic structure of the target over macroscopic time scales at room or elevated temperatures during annealing.
- Effects of defects and impurities on the mechanical, electronic, optical and magnetic properties of the irradiated system.

The simulations needed here can be divided into three groups: modeling of the collision, evolution of the system after the ion impact, and assessment of the influence of defects on materials properties.

Future progress in these fields is expected simply due to increasingly powerful computers, which allow for the simulation of larger systems and longer times. These advances will lead to more widespread use of MD and kMC simulations for specific FIB applications and to explore the physics of ion–target interaction in general. Currently these research areas are only accessible using BCA-type simulations. In addition, new developments in computational techniques widen the range of applicability of these advanced methods. For example, MD simulations with ML-derived interatomic potentials trained by DFT calculations (see Sec. III B 2) raise the accuracy and reliability of MD predictions to DFT level while being applicable to much larger systems.

Combinations of various techniques in a single run (e.g., first-principles approaches combined with ML-potential MD, or analytical-potential MD combined with BCA methods) can also help elucidate irradiation effects in various situations, such as in strongly non-uniform systems when the main focus is on the description of damage production in one of the subsystems, or over longer time scales. Examples are the response of 2D materials on substrates to ion bombardment^{370,1192} or the evolution of the target under high-dose FIB irradiation.

Computers with massively parallel architectures and recent development of Ehrenfest dynamics allow prediction of the energy transfer from the ion to the target through ballistic collisions and electronic excitations, and of the energy dissipation in the target involving defect formation without any empirical parameters. In particular, the direct juxtaposition of experimental data and corresponding simulation results concerning ion transit through atomically thin targets (e.g., graphene and other 2D materials) should shed light on the mechanisms of ion neutralization and energy transfer. Indeed, if the initial/final charge states of the ion and its kinetic energy are known, along with the scattering angle, this information can be correlated with impact parameter-dependent results of atomistic simulations carried out using TDDFT-based Ehrenfest dynamics with the assumption that the nuclei behave as classical particles. TDDFT-based Ehrenfest dynamics can also improve parameterization of two-temperature models, bridging the gap between different approaches in multiscale atomistic

simulations. Though very useful, their application will remain limited to small systems for quite some time.

While significant progress in process modeling has been made, there are still challenges for the development of more advanced methods and their application to irradiation simulations. One of the challenges is simulation of more complex scenarios of ions passing through 2D materials (touched on above), and interpretation of the theoretical results. After passing through the target, the ion may be in a mixed charge state because the nuclei move in an average potential created by the electron subsystem, which can be in an excited state. For example, in the experiment, a hydrogen atom (proton) which has passed through the target can be in charge state zero or plus one, while non-integral charge localized around the atom can be found in the simulations. This problem is related to the fundamental question of the connection between quantum-mechanical averages and observable quantities, and requires careful analysis in the context of ion–target interaction.

There are also challenges in the methodology to be overcome. As an example, the common computational schemes of DFT codes fail at small separations between the nuclei, when atom cores overlap. Even for full-electron codes, numerical instabilities in the eigenvalue solvers are expected at small interatomic distances. There are also other challenges related to spin-polarization energies of atoms, as recent calculations have indicated.³⁷⁶

As for the long-term evolution of the system after irradiation (e.g., during post-treatments such as annealing at temperatures of several hundred Kelvin over hours), techniques other than MD with DFT or ML/analytical potentials are required. The bottleneck is the large number of computational steps to be performed. Parallelization can resolve the issue of system size (ultimately each processor calculates the forces acting on one specific atom) and thus reduce the cost of the time step down to a certain minimum. However, the time step in MD simulations should still be at the atomic scale (normally a few femtoseconds), which indicates that $\sim 10^{15}$ steps must still be performed in order to simulate one second of the physical time. Even if performing each time step takes $\sim 10^{-4}$ seconds (a typical value for a system comprised of less than 100 atoms), the total CPU time will be several years. In some cases (e.g., migration of point defects and their annealing in crystalline solids), kMC simulations may provide insights into the evolution of the system on a macroscopic timescale (see Sec. III C). However, kMC can hardly be applied to systems with a high degree of amorphization or when the atomic structure of the system changes substantially during annealing.

Assessment of the effects of defects on the properties of materials is in a much better situation. DFT-based approaches make it possible to calculate the electronic and magnetic properties of the defective system and directly juxtapose the results of calculations with the experimental data. In particular, insights into the optical properties and exciton binding energies of defective materials have recently been obtained,¹¹⁹³ and photoluminescence spectra of materials with defects such as single-photon emitters have been calculated.^{818,1194–1197}

VI. EXECUTIVE SUMMARY AND PERSPECTIVE

A focused ion beam (FIB) instrument employs a finely focused beam of ions with an energy of typically 2–30 keV that is scanned across a sample to effect modifications down to the nanometer scale. The technique relies on the transfer of energy from the energetic primary ion to target atoms, resulting in the displacement of the substrate atoms and potentially their removal from the sample. As such the FIB instrument

is a universal tool for all areas of research enabling comprehensive analyses, maskless local material changes and rapid prototyping of devices.

The aim of this document is to provide an overview of the current state of the art of FIB technology, its applications and important tool developments, all of which require attention by researchers and technologists developing new FIB based workflows and instrumentation. This document can thus serve as an important reference work for students, FIB users, academic and commercial developers of related technologies, and funding agencies. It includes overview tables, providing a bird's eye view of the relevant works in this field. New developments in the various driving fields of research pose new challenges for FIB techniques, for which we propose targeted solutions and provide a list of technical developments that will be required.

From an instrumentation point of view, the ubiquitous gallium FIB is an excellent tool for the preparation of both transmission electron microscopy (TEM) samples and functional nanostructures. However, FIB sources producing ions of other elements are becoming increasingly common and provide additional degrees of freedom for the fabrication of new functional materials. In fact, liquid metal alloy ion sources (LMAISs) and gas field-ionization sources (GFISs) were used even before the gallium liquid metal ion source (LMIS) started to control the market. Newer developments include the plasma focused ion beam (PFIB), which has quickly gained relevance due to its ability to remove large volumes of material while maintaining a lateral resolution sufficient for many applications. Combined with the very recent addition of the magneto-optical trap ion source (MOTIS), about two thirds of the elements of the periodic table can now be accessed for FIB applications (see Fig. 1 and Table I).

New applications related to, e.g., life and environmental science, quantum technology, energy conversion and storage, present new challenges for existing techniques that will require innovations in ion sources, beam transport and detectors. Low energy FIBs (<1 keV), for example, are particularly relevant for the semiconductor industry, quantum technology, and applications based on low dimensional materials. New developments related to heavy and unconventional ion beams are important for single ion implantation (SII) and analytical applications such as secondary ion mass spectrometry (SIMS).

The theory of ion–solid interactions and all related processes that may occur in FIB processing must bridge many orders of magnitude in terms of time and length scales (see Fig. 5). Therefore, a trade-off must always be found between accuracy and the size of the modeled system. The binary collision approximation (BCA) provides insight into a range of FIB-relevant quantities that are mainly driven by the ballistic phase of the collision cascade, such as sputtering yield, ion penetration depth, and degree of amorphization (see Table III for an overview of available BCA codes). Other techniques such as molecular dynamics (MD), kinetic Monte Carlo (kMC), density functional theory (DFT), and continuum modeling can address further important aspects. These include short-term defect evolution (MD), long-term system evolution at elevated temperatures (kMC), charge state of defects and charge transfer (DFT), and ion induced chemistry as well as surface processes (continuum modeling).

An important FIB application that needs support from theory and simulation in this respect is gas-assisted processing, usually referred to as focused ion beam induced deposition (FIBID). This is the nanometer-scale counterpart to industrial and consumer-based 3D printing. The variety of precursor gases (see Table IX) and the

patterning flexibility of the FIB enable the fabrication of nanoscale devices with unique properties and geometries for emerging applications (see Table VIII). Here, an in-depth understanding of the complex processes related to the non-uniform gas flow, the surface kinetics of all involved molecular species, and the secondary particle generation, together with the related energy-dependent dissociation cross sections of molecular bonds, is needed for accurate process simulation.

Process modeling is also relevant for other FIB applications, such as subtractive processing (see Table V) and defect engineering (see Table VI). Applicability, however, will also require an extension of the existing knowledge on ion–solid interactions into the energy range relevant for FIB applications. This knowledge transfer would be particularly helpful for calculations of the production rate of charged secondary particles, especially secondary electrons (SEs). The yield of SEs is relevant for several applications, including SII and also general imaging and analysis. For analysis applications, robust models for the prediction of secondary ion (SI) yields are needed. In addition to addressing these fundamental questions, efforts are required in the areas of empirical process modeling and shape prediction for additive and subtractive patterning. Here, open application programming interfaces (APIs) and open source software in combination with machine learning (ML) will enable better control over the focused ion beams, which will in turn allow higher precision, resulting in faster and more reproducible device fabrication.

New software and computing technologies enable the collection and evaluation of large data sets, e.g., for the FIB based analytical methods of FIB-SIMS and serial sectioning (see Table VII). ML algorithms can assist with the acquisition, alignment, and segmentation of these typically multidimensional data sets, which should ideally be made available in a FAIR data infrastructure. Currently, biological applications are driving this field by using data sets from various analytical modalities, including FIB based tomography, to generate nanoscale 3D representations of biological specimens that provide new insight into microscopic structures and how they function. Other drivers are energy storage and conversion, meta- and functional materials, and the semiconductor industry.

As the applications of FIB processing are extremely diverse, it will not be a single technological development that brings a breakthrough. Instead, numerous advances will be needed in order to address the challenges that have been identified. The resulting developments will often impact several scientific fields, as can be seen from the overview graph in Fig. 28. Therefore, we look to an exciting future where FIB processing will grow in importance and open new avenues in many areas of science and technology, spanning quantum technology to the life sciences.

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AUTHOR DECLARATIONS

Conflict of Interest

The authors have no conflicts to disclose.

Author Contributions

Katja Höflich: Conceptualization (equal); Data curation (equal); Visualization (lead); Writing – original draft (lead); Writing – review & editing (equal). **Gerhard Hobler:** Conceptualization (equal); Writing – original draft (lead); Writing – review & editing (equal). **Frances Allen:** Writing – original draft (lead), Writing – review & editing (lead). **Tom Wirtz:** Conceptualization (equal) Writing – original draft (lead). **Gemma Rius:** Conceptualization (equal); Data curation (lead); Visualization (equal); Writing – original draft (equal). **Lisa McElwee-White:** Writing – original draft (equal); Writing – review & editing (lead). **Arkady V. Krasheninnikov:** Writing – original draft (equal). **Matthias Schmidt:** Writing – original draft (equal). **Ivo Utke:** Writing – original draft (equal). **Nico Klingner:** Writing – original draft (equal). **Markus Osenberg:** Writing – original draft (equal). **Rosa Córdoba:** Writing – original draft (equal). **Flyura Djurabekova:** Writing – original draft (equal). **Ingo Manke:** Writing – original draft (equal). **Philip Moll:** Writing – original draft (equal). **Mariachiara Manocchio:** Writing – original draft (equal). **José María De Teresa:** Writing – original draft (equal). **Lothar Bischoff:** Writing – original draft (equal). **Johann Michler:** Writing – original draft (equal). **Olivier De Castro:** Writing – original draft (equal). **Anne Delobbe:** Writing – original draft (equal). **Peter Dunne:** Writing – original draft (equal). **Oleksandr V. Dobrovolskiy:** Writing – original draft (equal). **Natalie Frese:** Writing – original draft (equal). **Armin Götzhäuser:** Writing – original draft (equal). **Paul Mazarov:** Writing – original draft (equal). **Dieter Koelle:** Writing – original draft (equal). **Wolfgang Möller:** Writing – original draft (equal). **Francesc Pérez-Murano:** Writing – original draft (equal). **Patrick Philipp:** Writing – original draft (equal). **Florian Vollnhals:** Writing – original draft (equal). **Gregor Hlawacek:** Conceptualization (lead); Funding acquisition (lead); Visualization (equal); Writing – original draft (equal); Writing – review & editing (equal).

DATA AVAILABILITY

Data sharing is not applicable to this article as no new data were created or analyzed in this study.

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